

RMCProfile: Big Box modelling of crystalline to amorphous materials

Matt Tucker Wojciech Slawinski
(ORNL, USA) (Warsaw)

ORNL is managed by UT-Battelle, LLC for the US Department of Energy

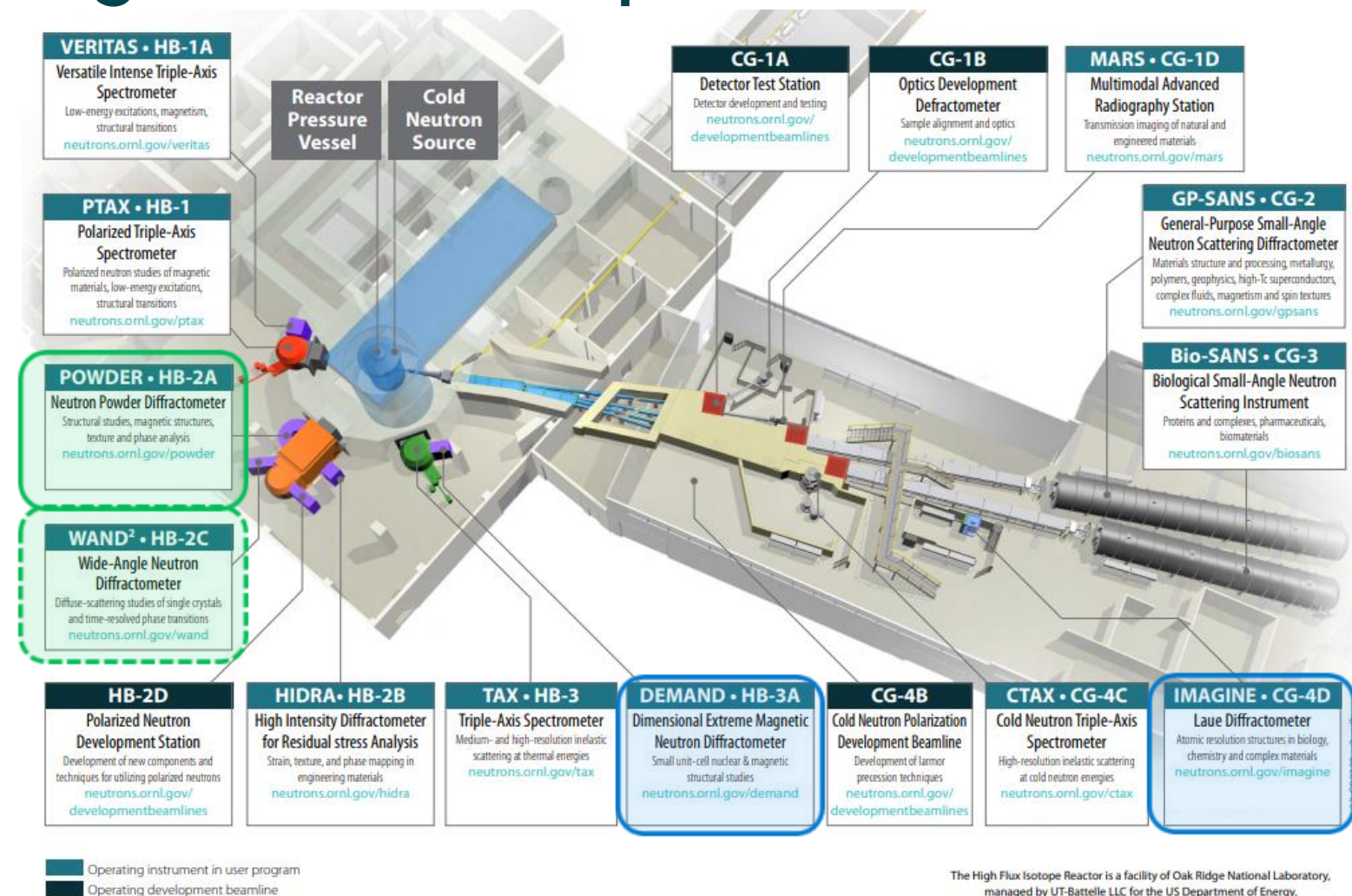


Diffraction Section @ ORNL



Diffraction section includes 2 groups with 9 instruments

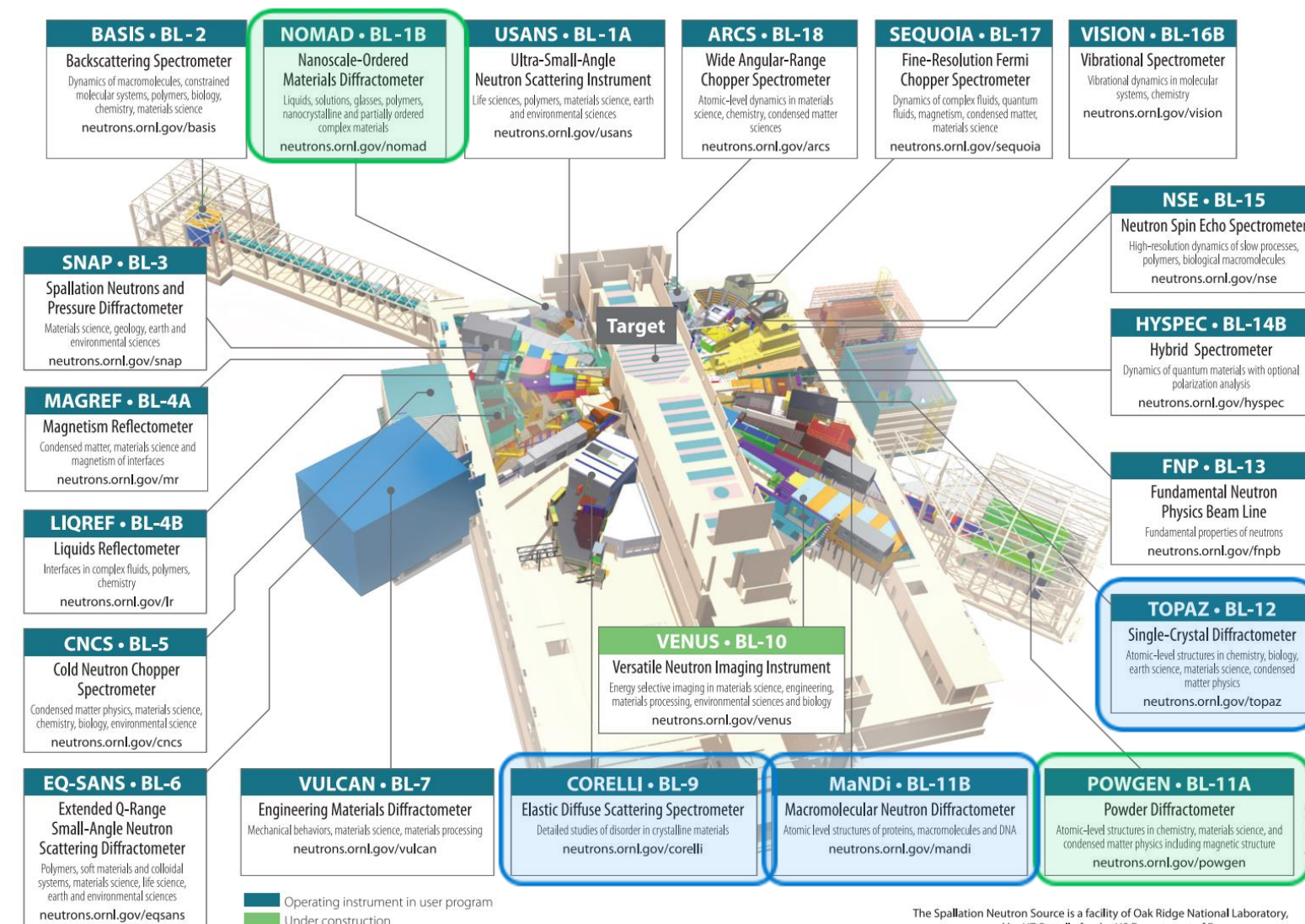
High Flux Isotope Reactor



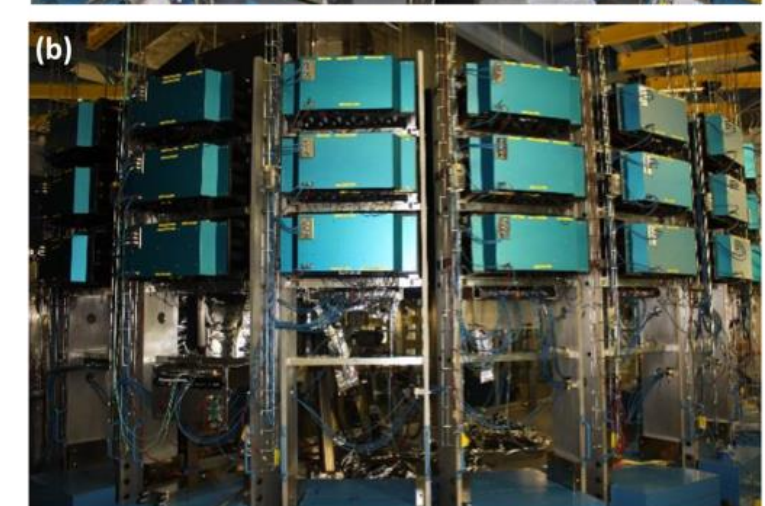
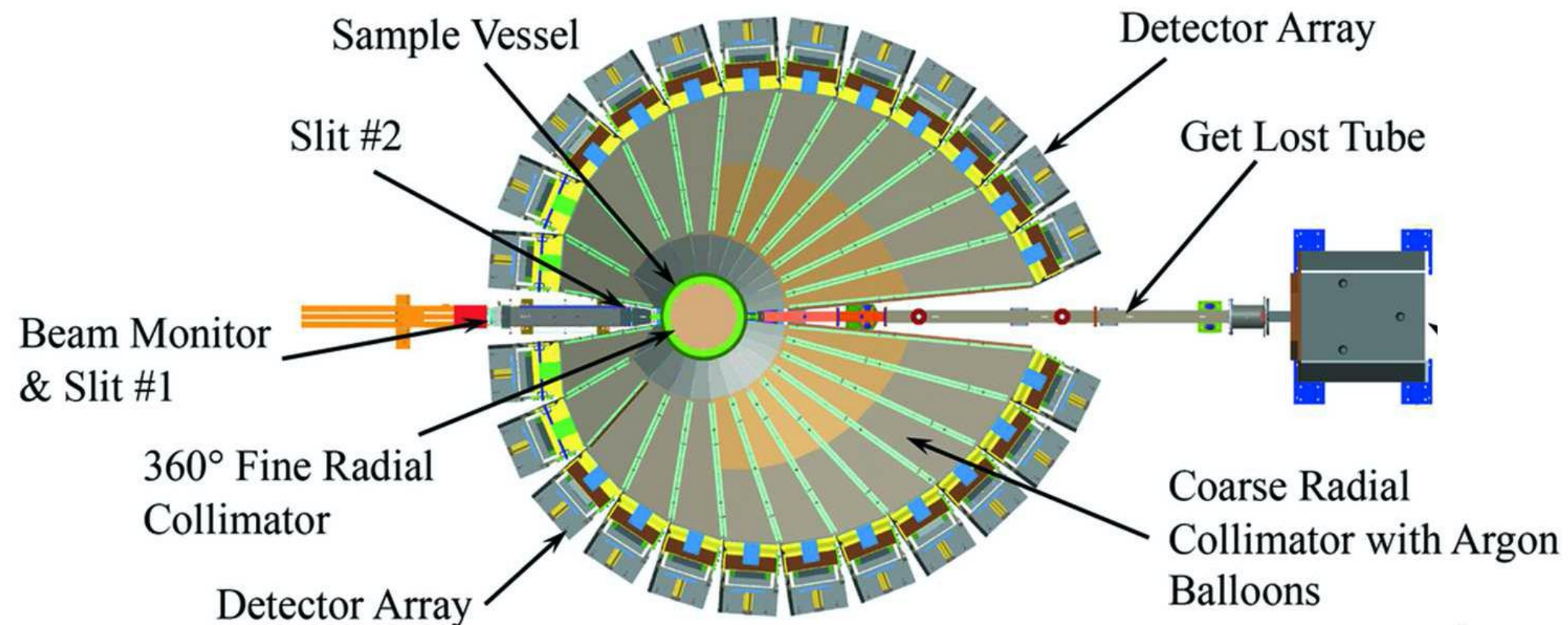
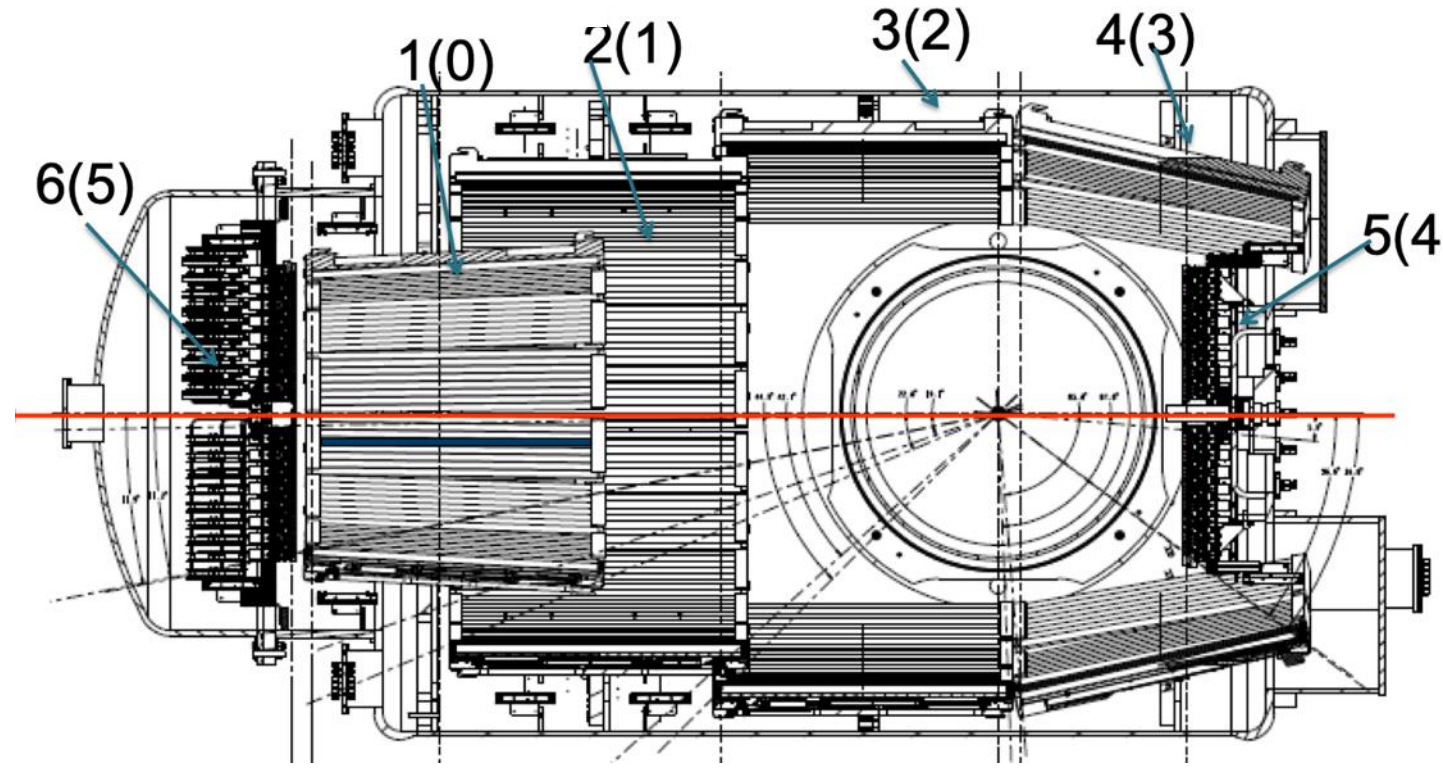
- 2 Groups
 - Powder Diffraction Group
 - Single Crystal Diffraction Group

- 9 Instruments
 - 4 at the HFIR
 - 5 at the SNS

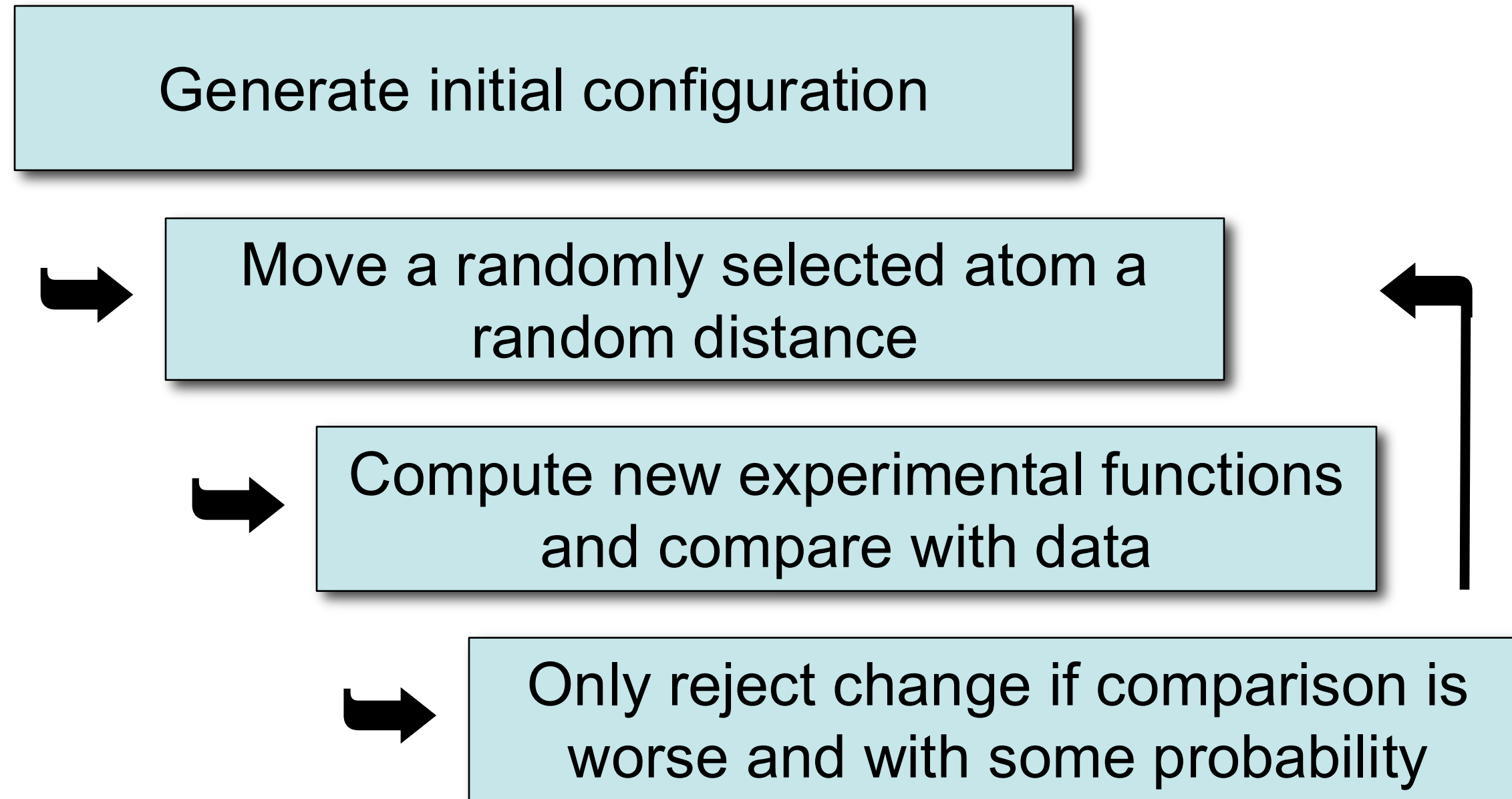
Spallation Neutron Source



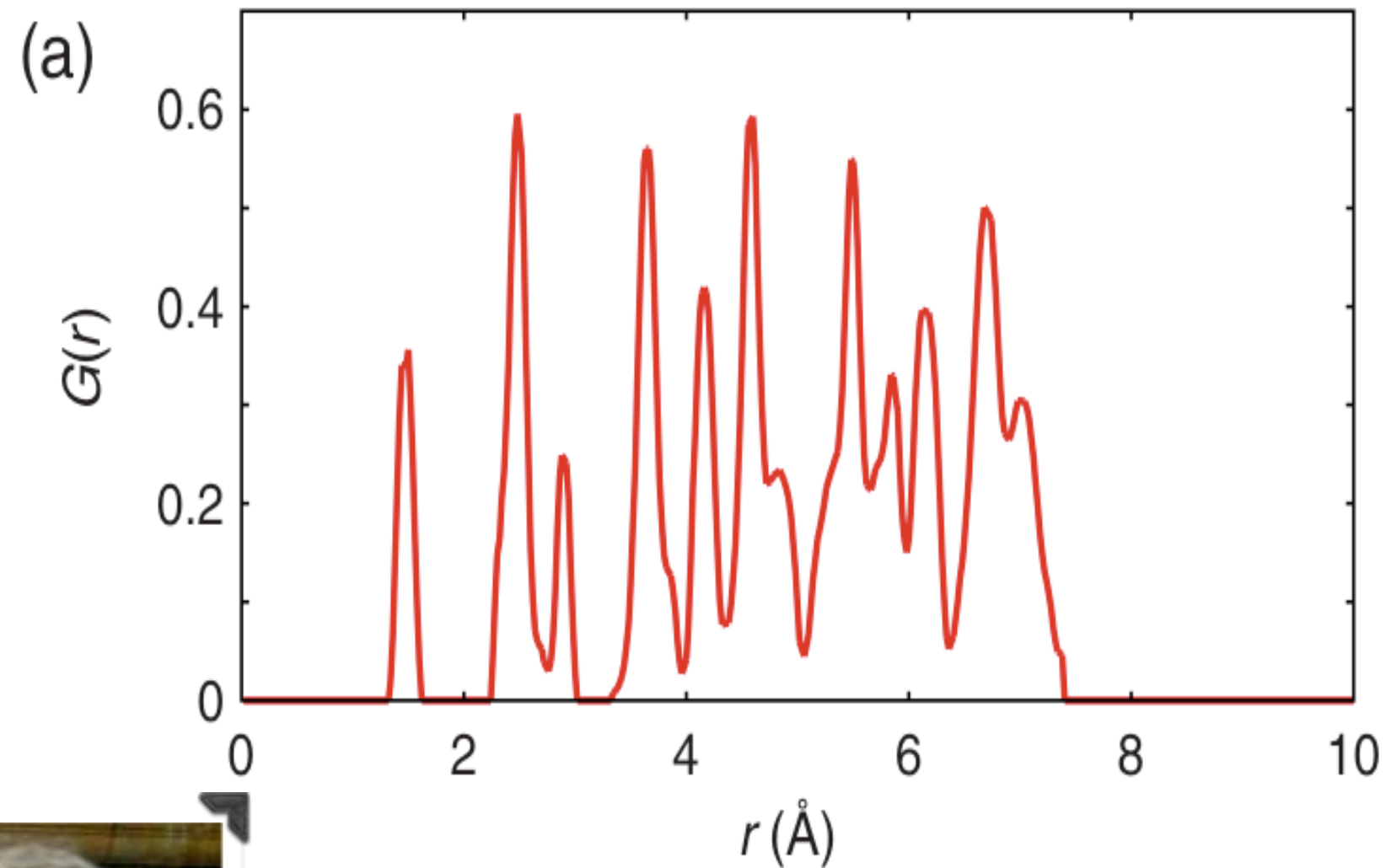
NOMAD and POWGEN



The Reverse Monte Carlo algorithm



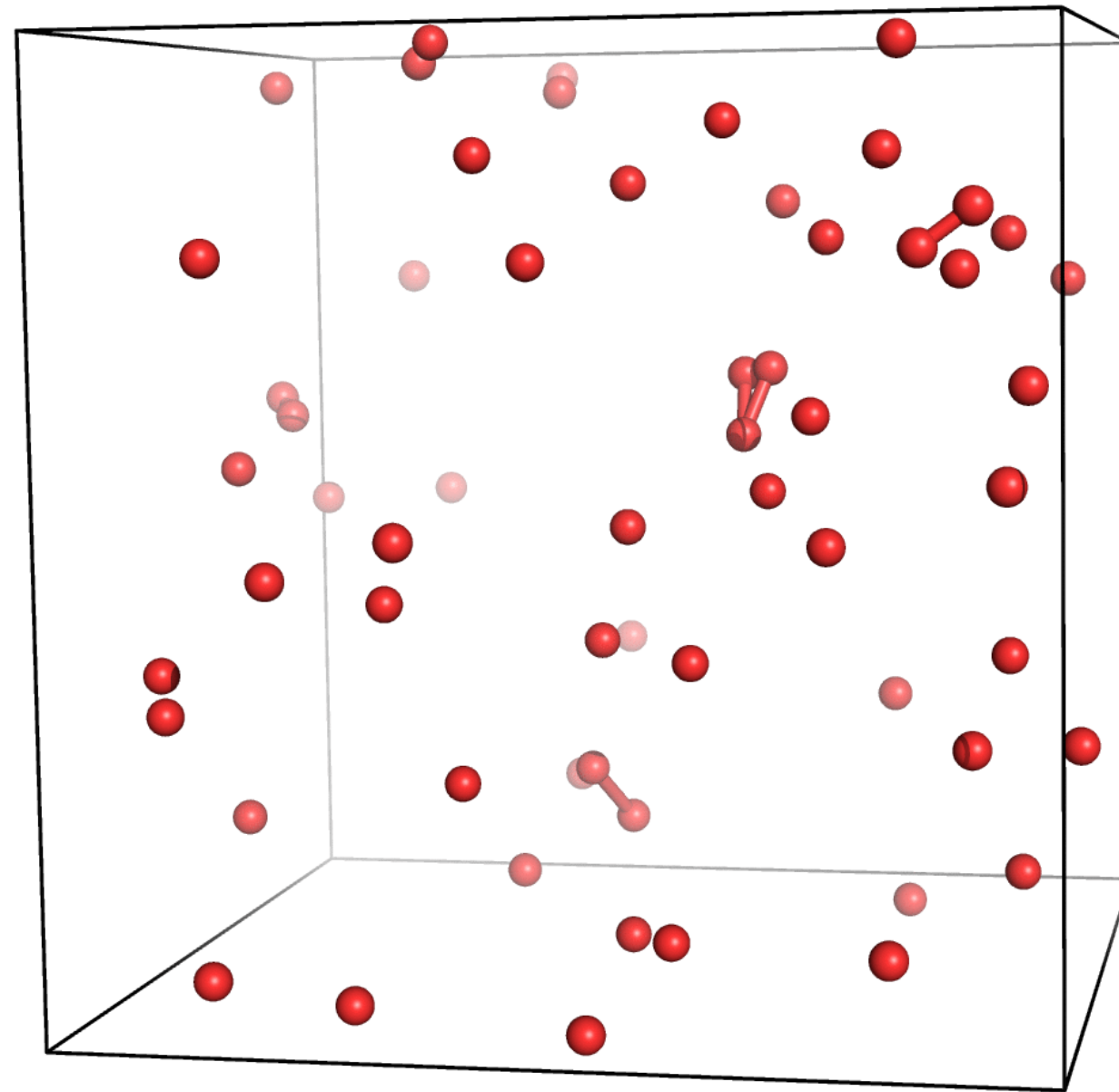
RMC in action: C60



Cliffe M J *et al*
PRL **104** (2010) 1255013

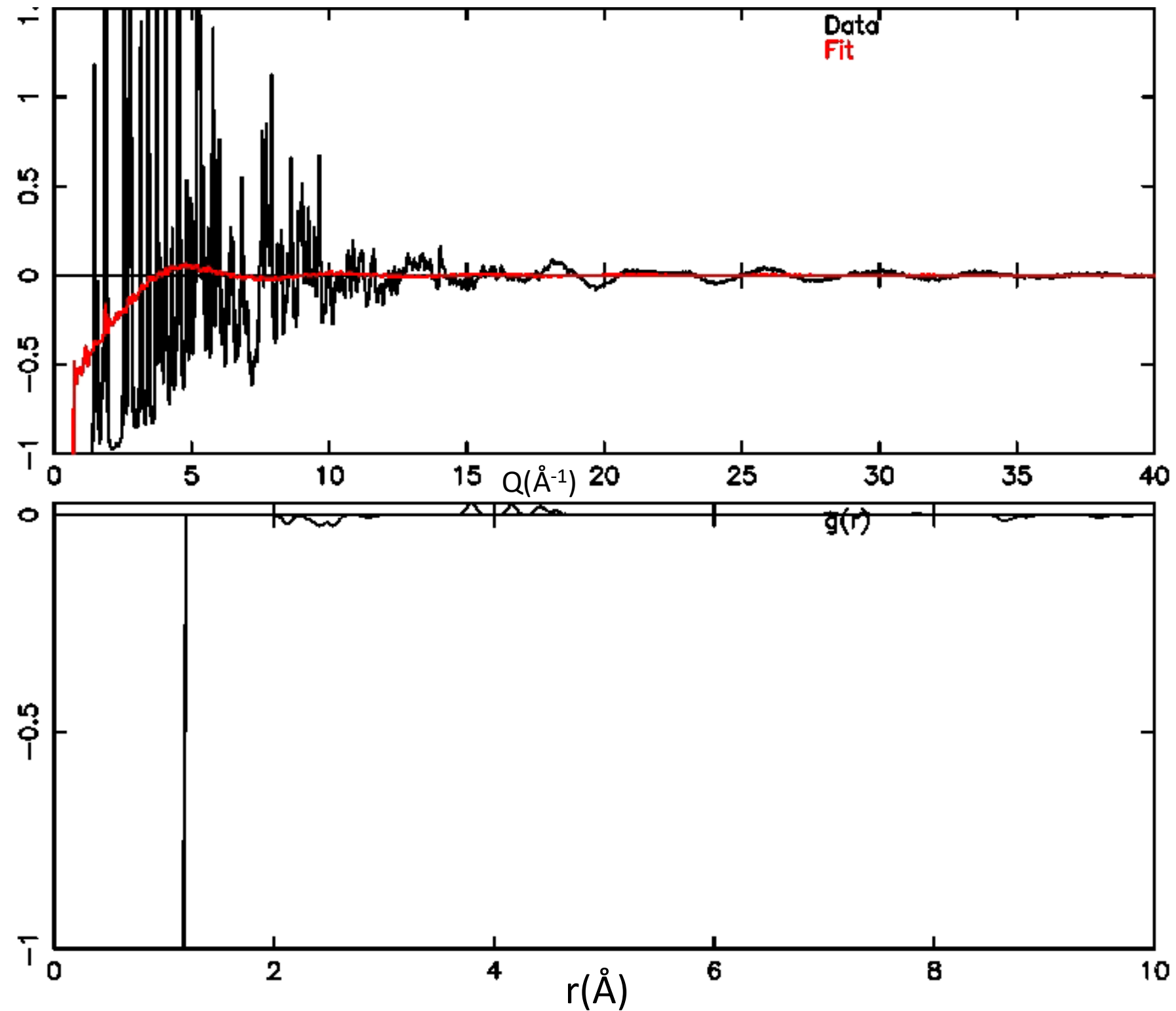
INVERT

RMC in action: C60

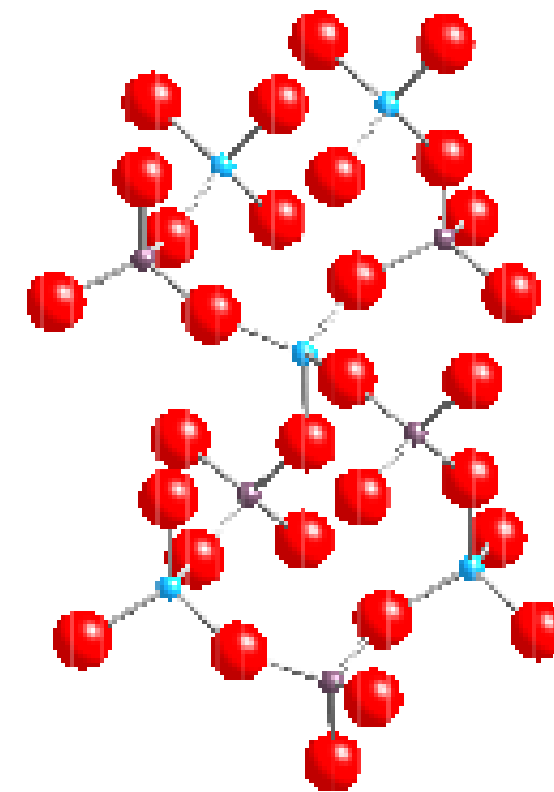
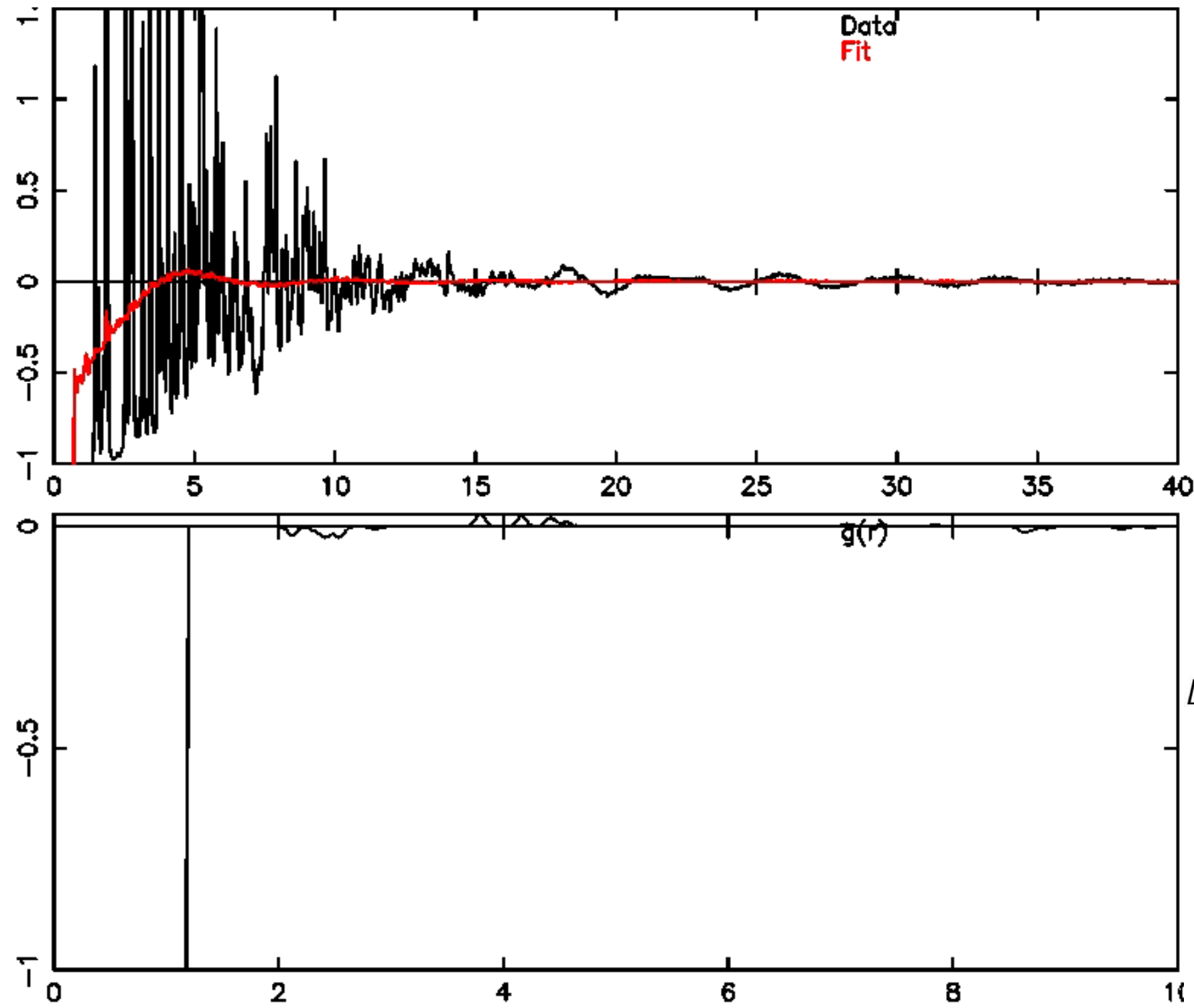


Cliffe M J *et al*, *PRL* **104** (2010) 1255013

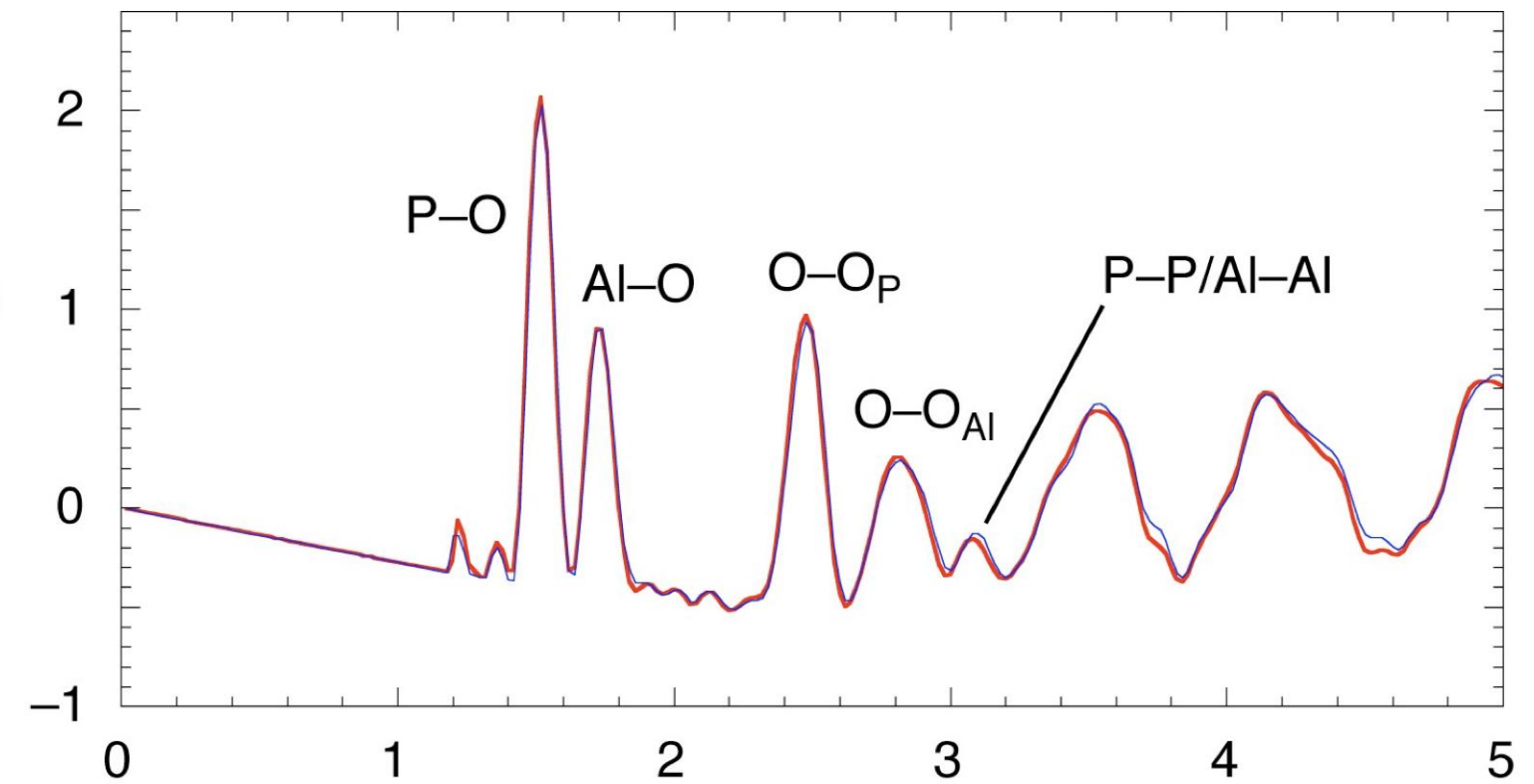
RMC in action



RMC in action



AlPO_4



RMC programs

← → ↻

rmcprofile.pages.ornl.gov

☆ ⚙ □ M ⋮

RMC
PROFILE



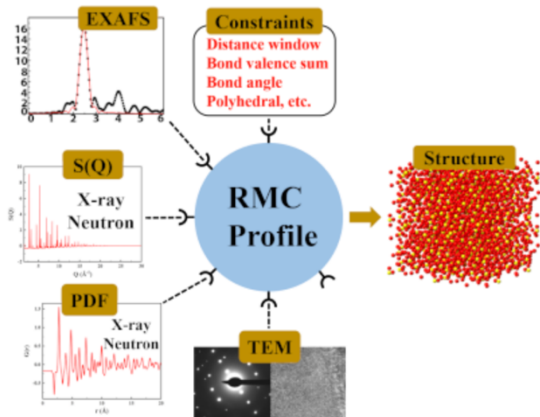
SEARCH ABOUT POSTS PACKAGE COMMUNITY OTHERS

RMC
PROFILE

RMCProfile

Reverse Monte Carlo for crystalline and disordered materials

Welcome to the RMCProfile website hosted at Oak Ridge National Laboratory (ORNL) in US. Here you can download the RMCProfile software, view documentation and examples, join community for discussion and learn about updates of the package, etc.



This version of RMC was built from the original RMCA code of McGreevy & Pusztai to determine the local structure of crystalline materials while still being capable of analyzing disordered systems. The current version of RMCProfile results from a collaboration between scientists at ISIS facility (UK), Spallation Neutron Source (SNS at Oak Ridge National Laboratory, US), University of Cambridge (UK), University of Oxford (UK), Queen Mary University of London

Yuanpeng
Zhang
(ORNL)



rmcprofile.ornl.gov

External Links

Monte Carlo

DISCUS

RMC++

HRMC

fullrmc

RMC for EXAFS

EPSR

RMC at ISIS facility

RMCProfile website - legacy

Wikipedia for RMC

← → ↻

rmcprofile.ornl.gov

☆ ⚙ □ M ⋮

UT POSTS PACKAGE COMMUNITY OTHERS

ssion. Discussions are hosted as
ssion, one does need a GitHub

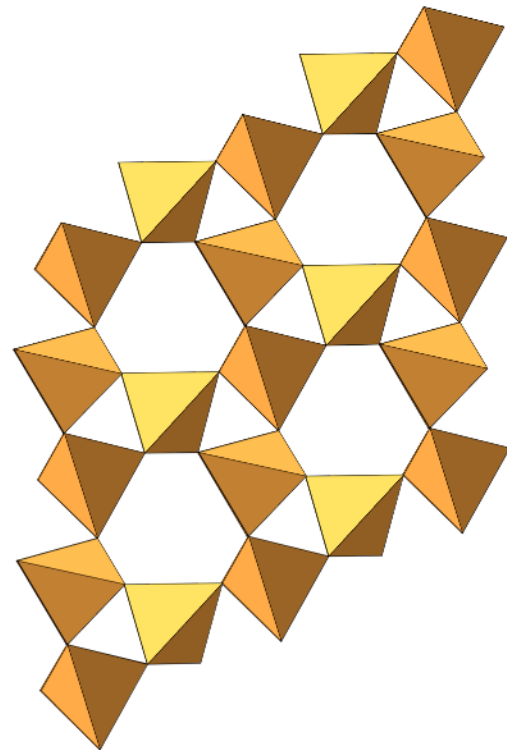
logy

UNIVERSITY OF
OXFORD

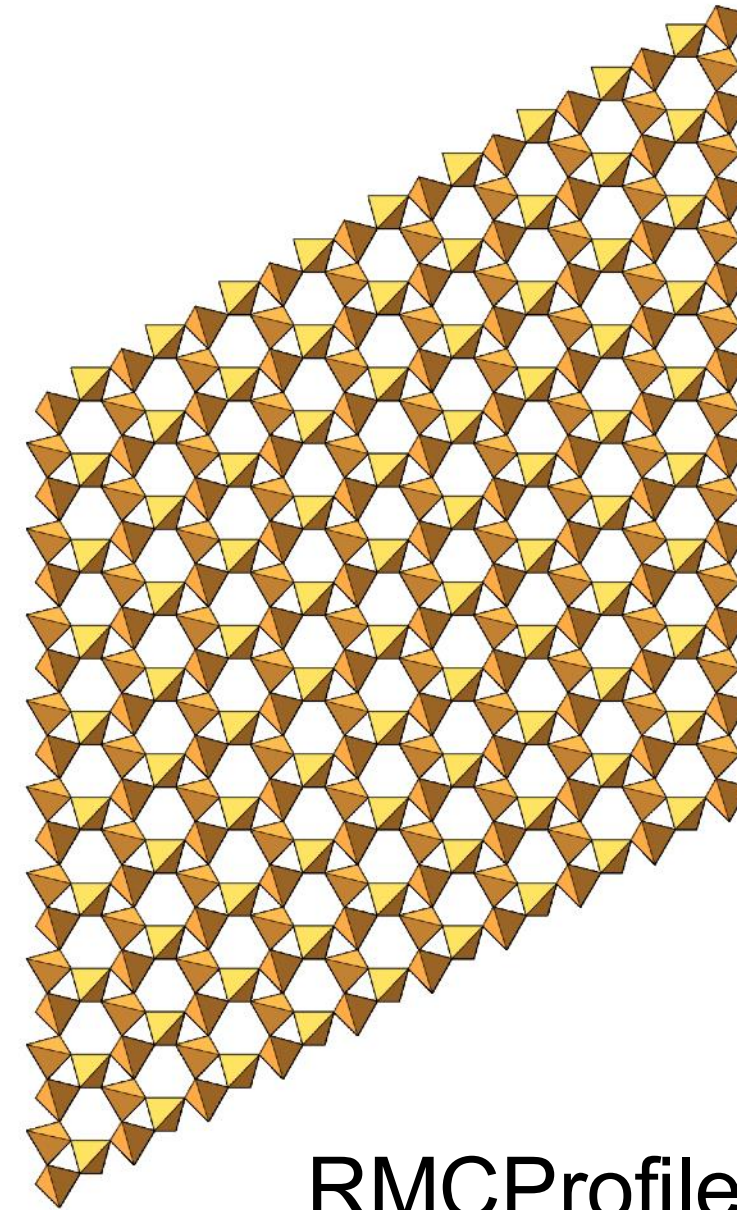
Facilities

US - ORNL - NOMAD
US - ORNL - POWGEN
US - APS - 11-ID-B
US - BNL - 28-ID-1
UK - ISIS - GEM
UK - ISIS - Polaris
UK - Diamond - I15-1
Japan - J-PARC - NOVA
Japan - SPRING8 - BL14B1
Japan - SPRING8 - BL22XU
Europe - ESS - HEIMDAL
China - SSRF - BL13W1

Big box vs small box models

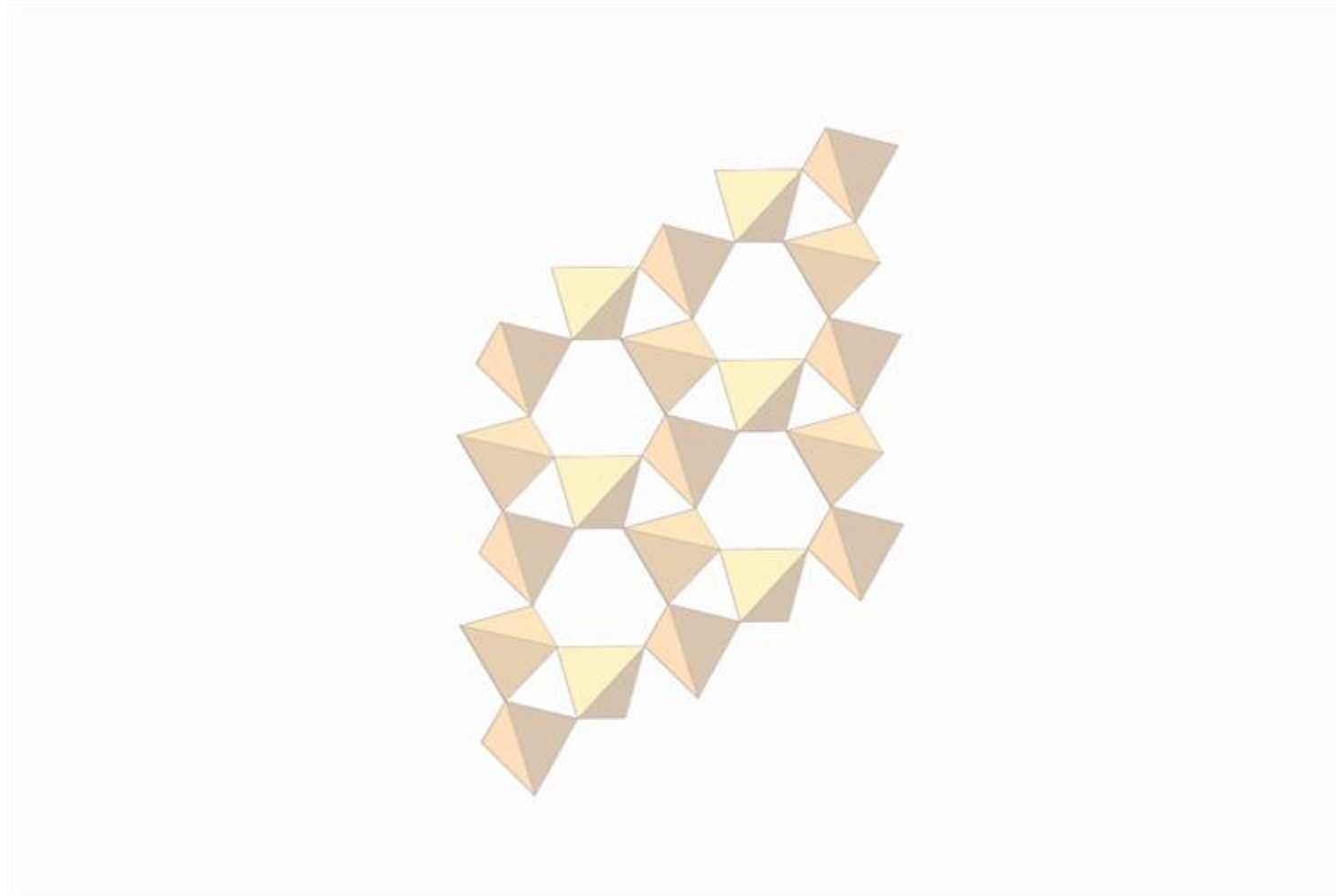


PDFgui
(r-space Reitveld)

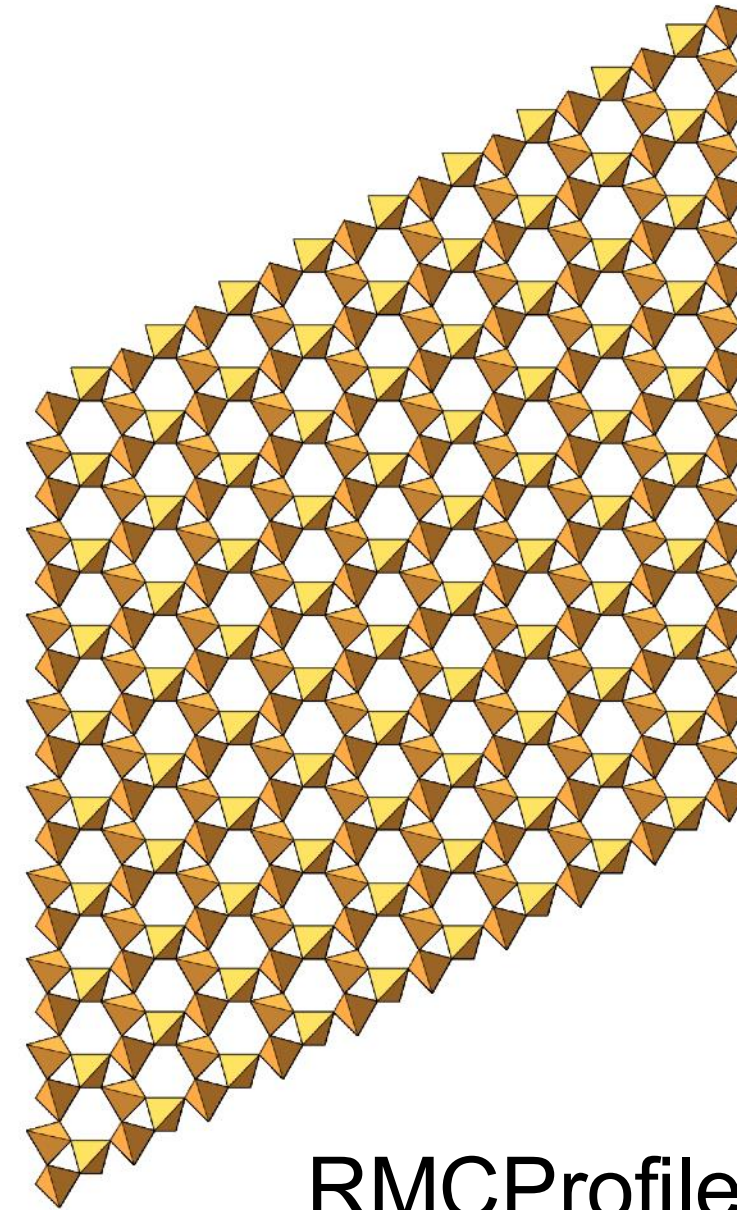


RMCPProfile
(Reverse Monte Carlo)

Big box models



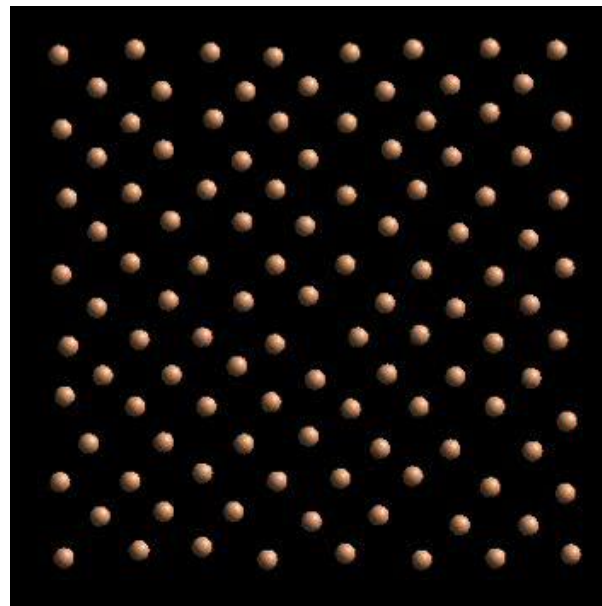
PDFgui
(r-space Reitveld)



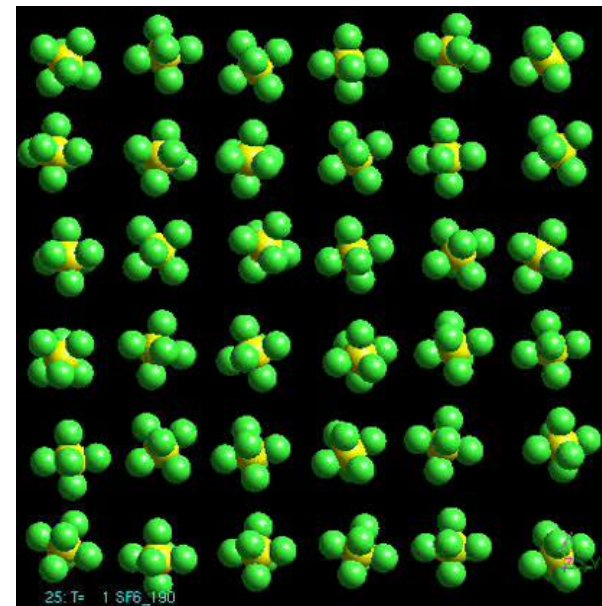
RMCProfile
(Reverse Monte Carlo)

Disordered materials

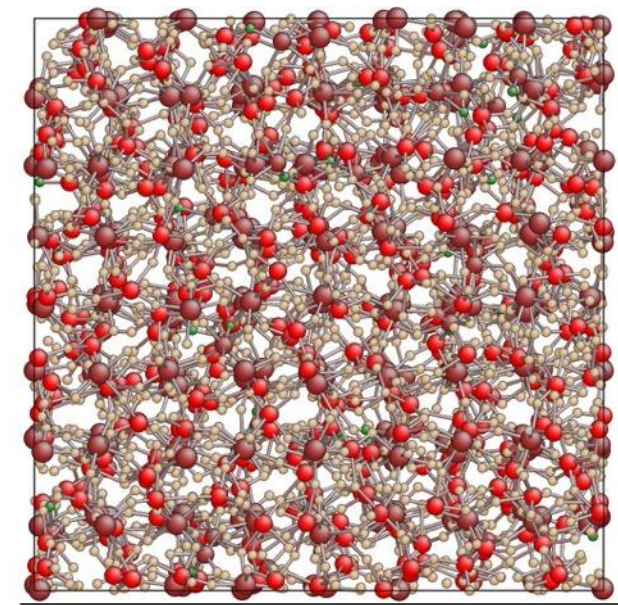
Simple
crystals



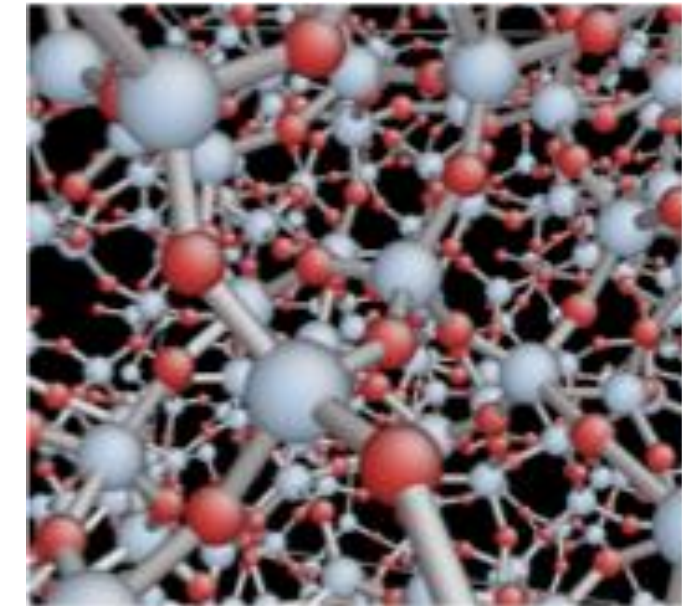
Disordered
crystals



Amorphisation



Amorphous



RMCPProfile

The RMC Method

Reverse Monte Carlo Simulation: a new technique for the determination of disordered structures

McGreevy R L and Pusztai L, *Molecular Simulation* 1(1988) 359



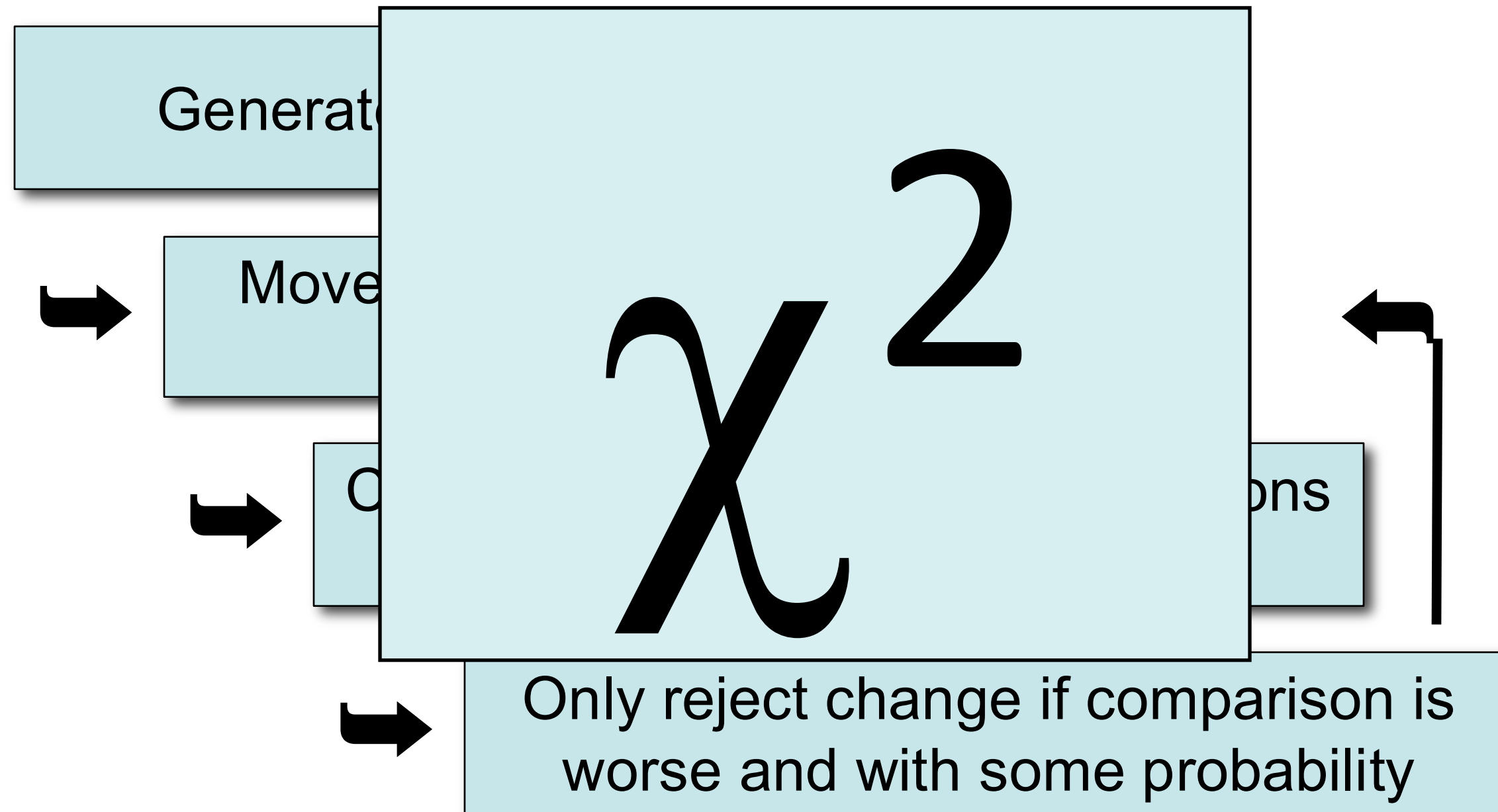
We have developed a new technique, based on the standard Monte Carlo simulation method with Markov chain sampling, where a set of three dimensional particle configurations are generated that are consistent with the experimentally measured structure factor, $A(Q)$, and radial distribution function, $g(r)$, of a liquid or other disordered system. Consistency is determined by a standard χ^2 test using the experimental errors. No input potential is required. We present initial results for liquid argon. Since the technique can work directly from the structure factor it promises to be extremely powerful for modelling the structures of glasses or amorphous materials. It also has many other advantages in multicomponent systems and as a tool for experimental data analysis.

Key words: Monte Carlo, structure factor, radial distribution function, liquid, glass.

PACS numbers: 02.50, 61.25, 61.40.



The Reverse Monte Carlo algorithm



RMC χ^2

$$\chi_{\text{RMC}}^2 = \sum_j \chi_j^2$$

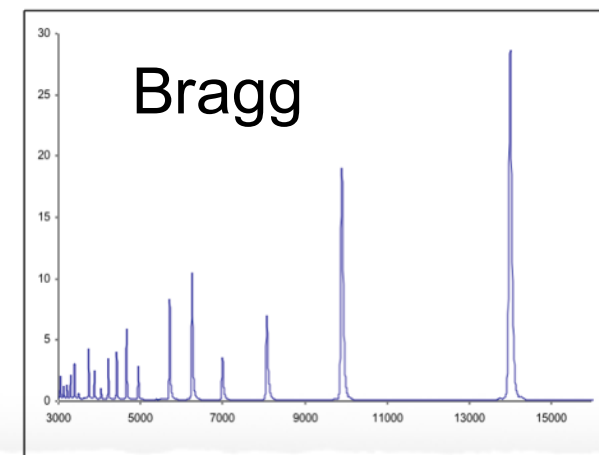
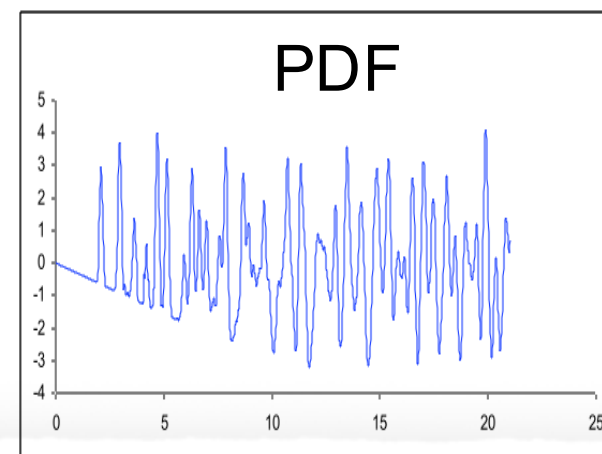
$$\chi_{F(Q)}^2 = \sum_j [F_{\text{calc}}(Q_j) - F_{\text{box}}(Q_j)]^2 / \sigma_{F(Q)}^2(Q_j) \quad \text{Total scattering}$$

$$\chi_{G(r)}^2 = \sum_j [G_{\text{calc}}(r_j) - G_{\text{expt}}(r_j)]^2 / \sigma_{G(r)}^2(r_j) \quad \text{PDF}$$

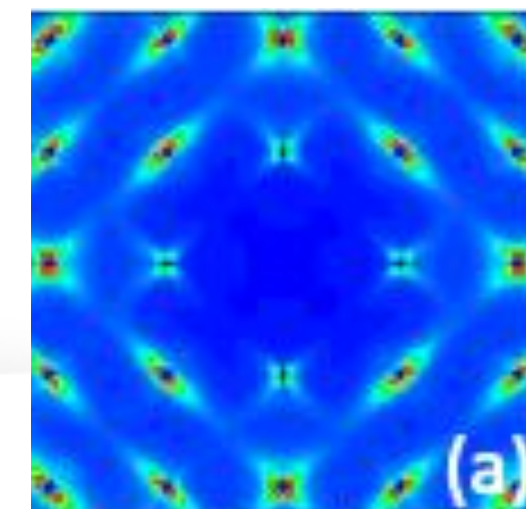
$$\chi_{\text{profile}}^2 = \sum_j [I_{\text{calc}}(t_j) - s' I_{\text{expt}}(t_j)]^2 / \sigma_{I(i)}^2(t_j) \quad \text{Bragg profile}$$

$$\chi_f^2 = \sum_k w_k [f_k^{\text{calc}} - f_k^{\text{req}}]^2 \quad \text{Constraints/restraints}$$

RMC PROFILE



Single Crystal Diffuse



F(Q)
PRL 96, 047209 (2006)

F(Q)

PHYSICAL REVIEW LETTERS

week ending
3 FEBRUARY 2006

Magnetic Structure of MnO at 10 K from Total Neutron Scattering Data

Andrew L. Goodwin,¹ Matthew G. Tucker,¹ Martin T. Dove,^{1,*} and David A. Keen^{2,3}

¹Department of Earth Sciences, Cambridge University, Downing Street, Cambridge CB2 3EQ, United Kingdom

²Department of Physics, Oxford University, Clarendon Laboratory, Parks Road, Oxford OX1 3PU, United Kingdom

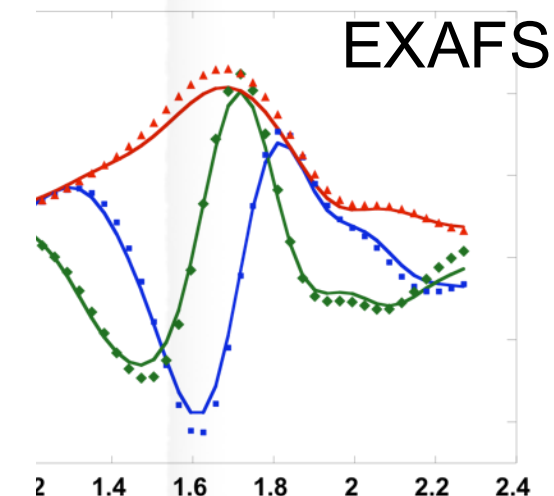
³ISIS Facility, Rutherford Appleton Laboratory, Chilton, Didcot, Oxfordshire OX11 0QX, United Kingdom

(Received 24 October 2005; published 2 February 2006)

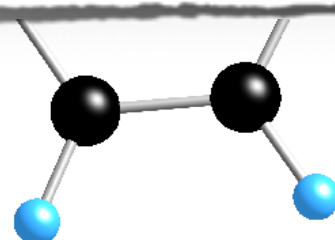
Total neutron scattering data from a powdered sample of MnO collected at 10 K have been analyzed using the reverse Monte Carlo method to refine the nuclear and magnetic structure. The results give the first unambiguous assignment of the average magnetic structure. The magnetic moments are aligned ferromagnetically within (111) sheets with the magnetization vectors of alternate sheets along axes parallel and antiparallel to the $\langle 11\bar{2} \rangle$ directions, albeit with a small modulated out-of-plane component. Small displacements of Mn and O (modulated with the same periodicity) accompany the magnetic ordering and both atomic and magnetic structures may be described in the monoclinic space group $C2$.

DOI: [10.1103/PhysRevLett.96.047209](https://doi.org/10.1103/PhysRevLett.96.047209)

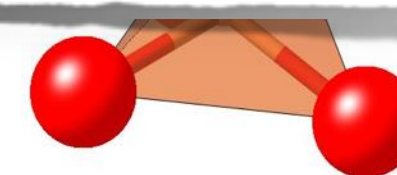
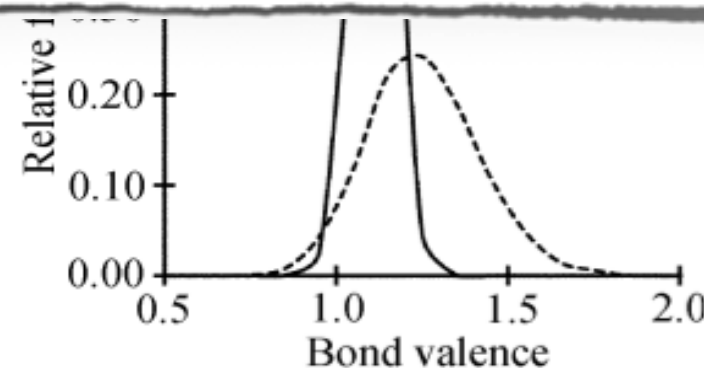
PACS numbers: 75.25.+z, 02.70.Uu, 61.12.Ex



Polyhedral
restraints



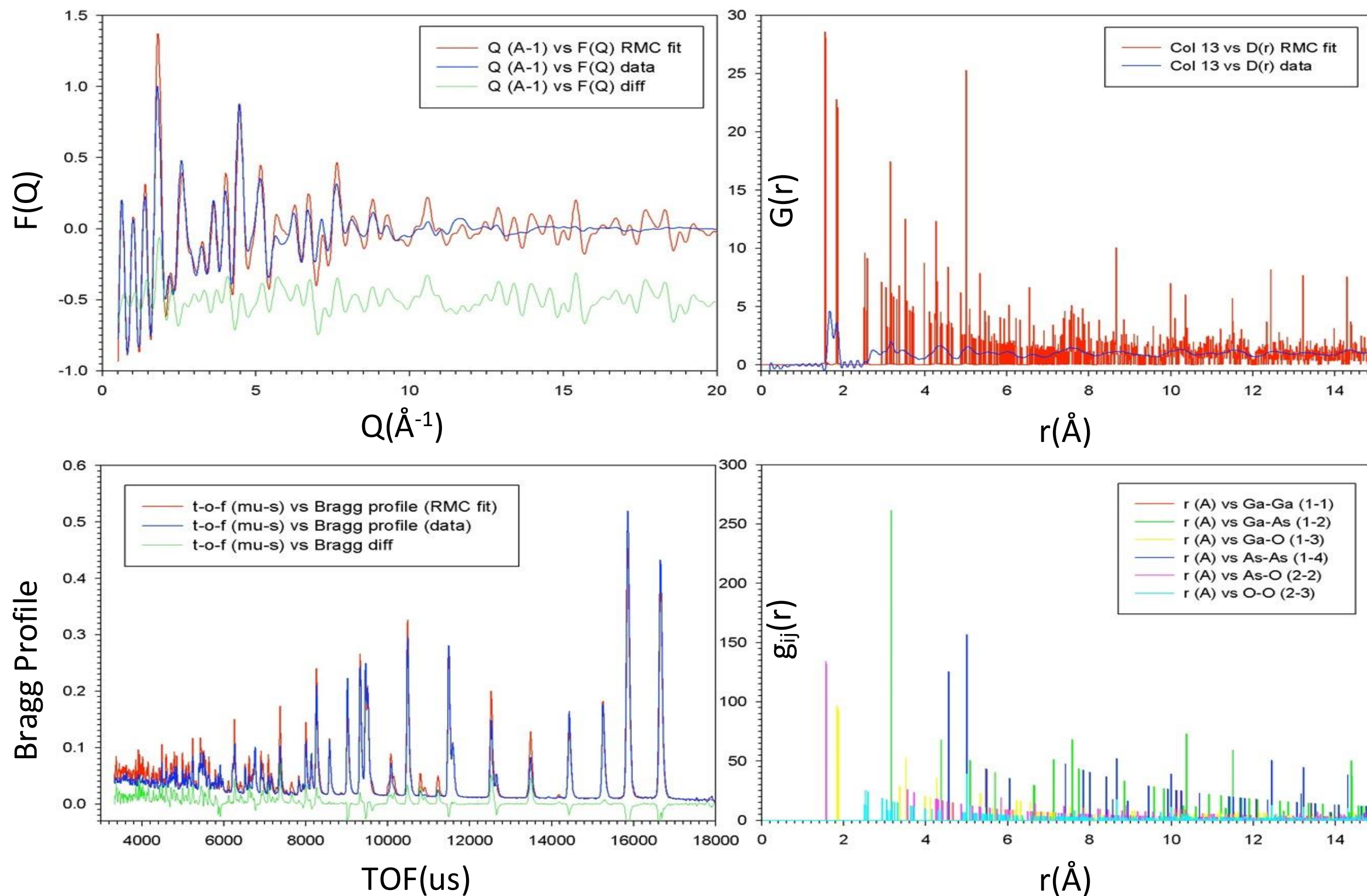
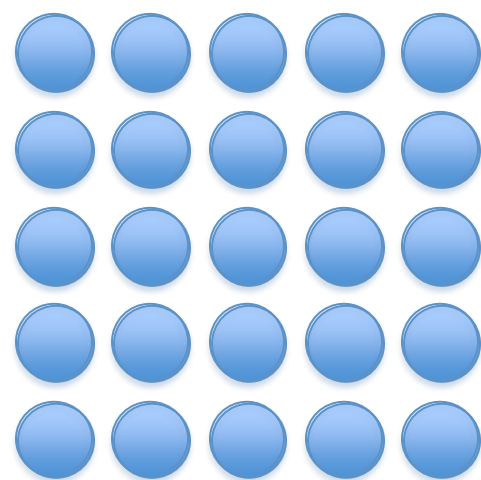
Molecular
potentials



M G Tucker et al, *J. Phys.: Condens. Matter* **19**, 335218 (2007).

RMCProfile refinement

GaAsO₄ RT.files



4608 atoms

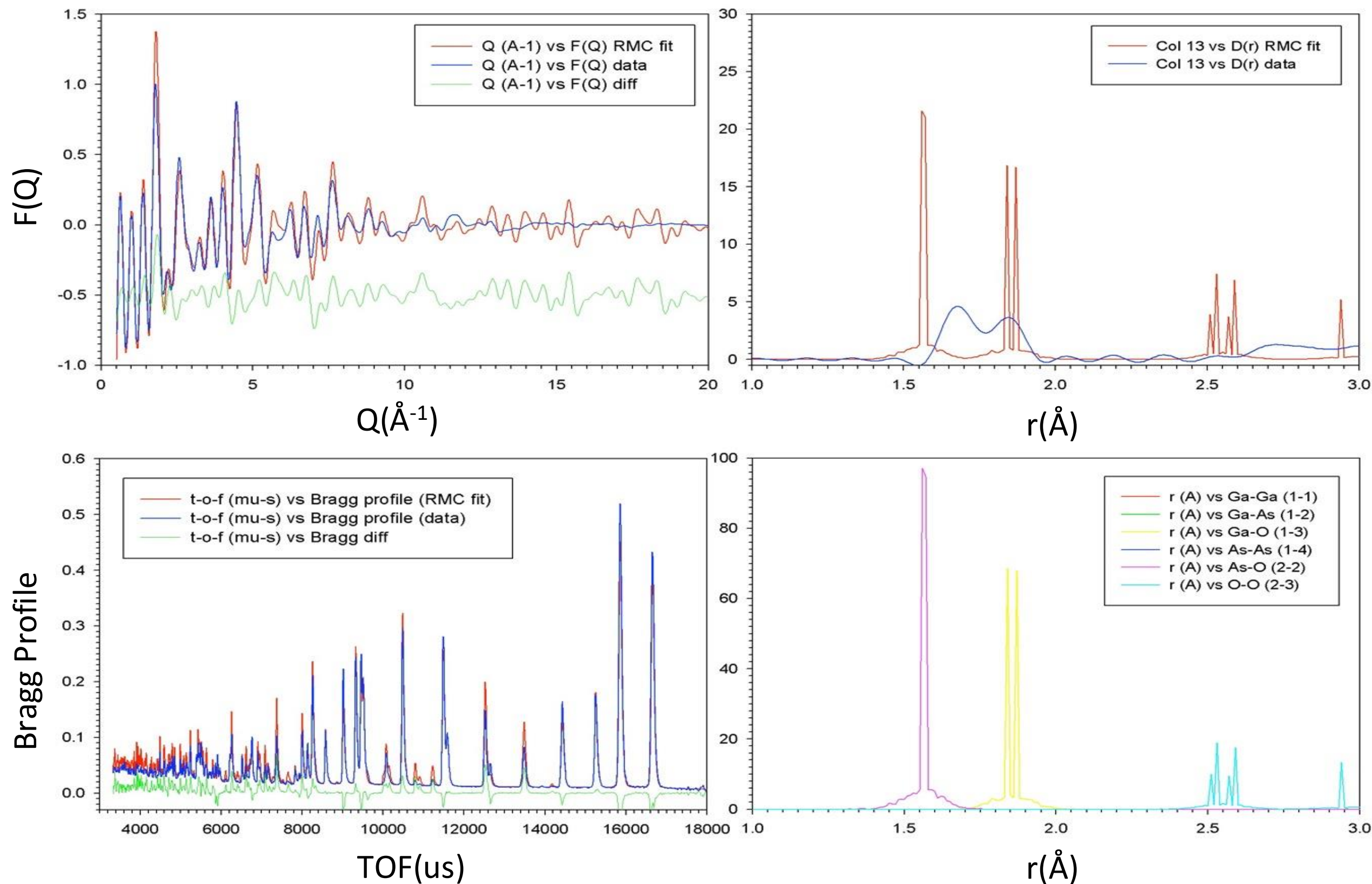
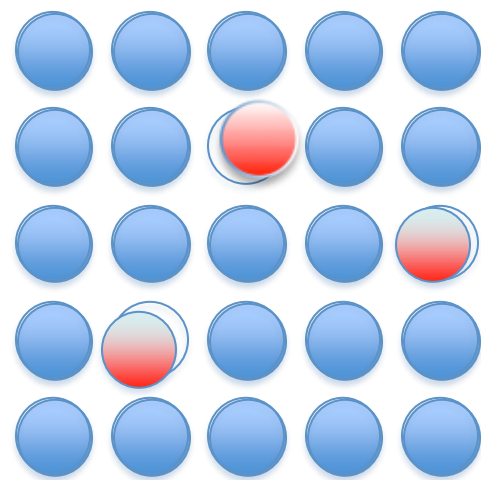
0 moves

$$\chi^2 = 3023$$

Slides from
Dave Keen

RMCProfile refinement

GaAsO₄ RT.files



4608 atoms

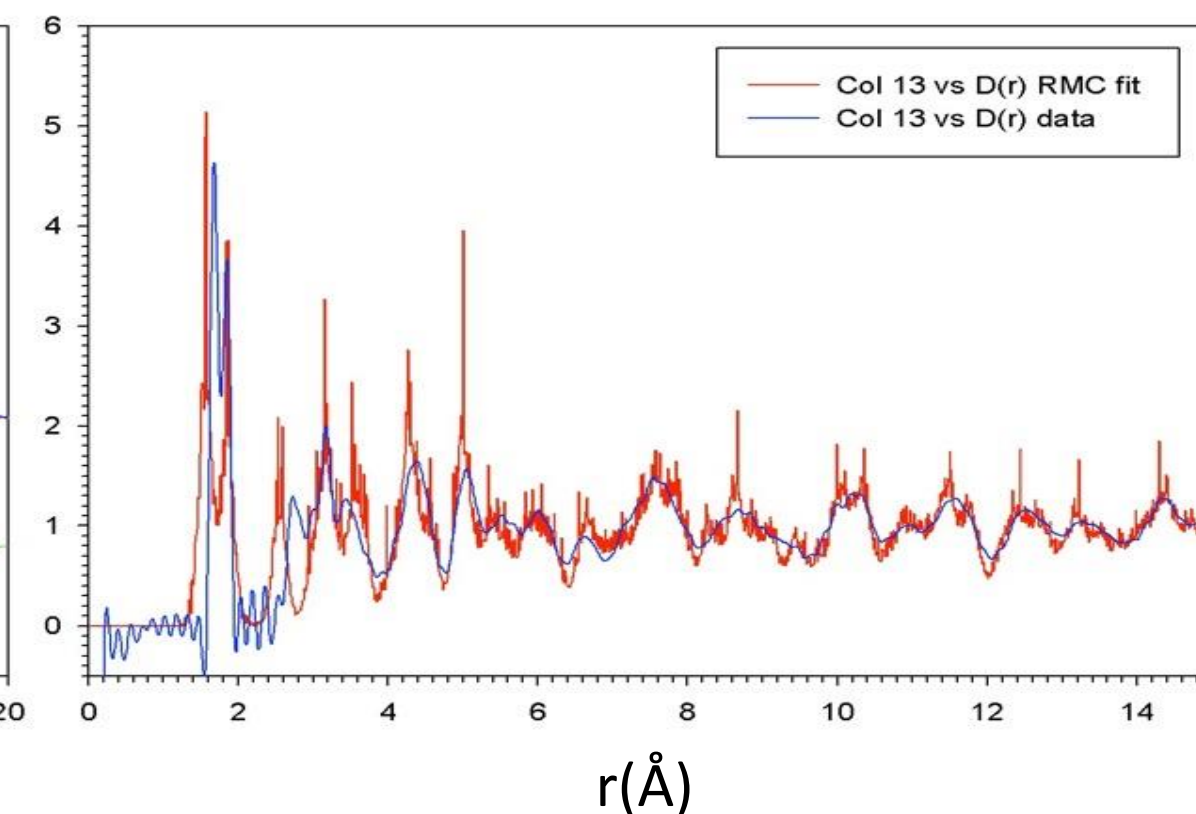
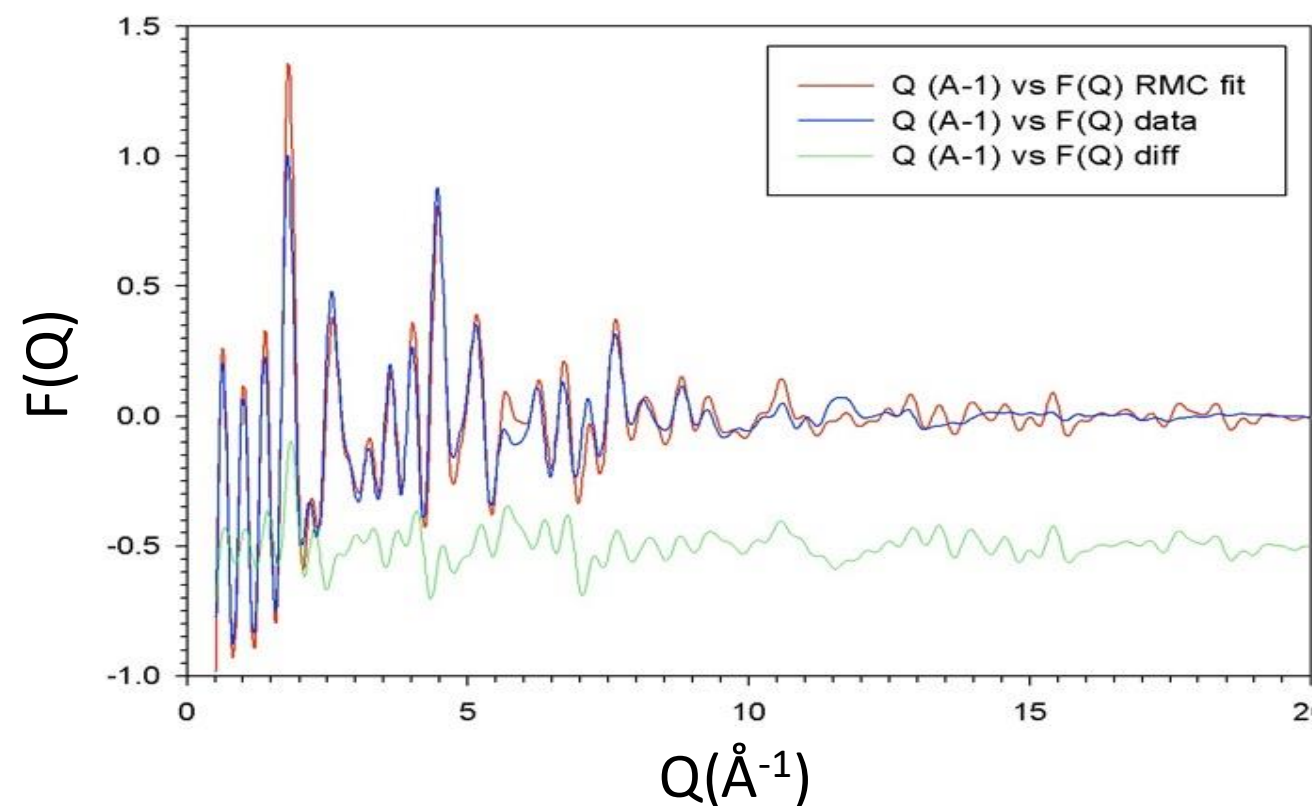
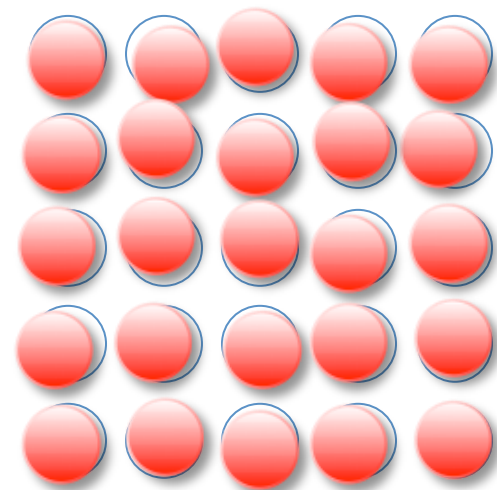
834 moves

$$\chi^2 = 1629$$

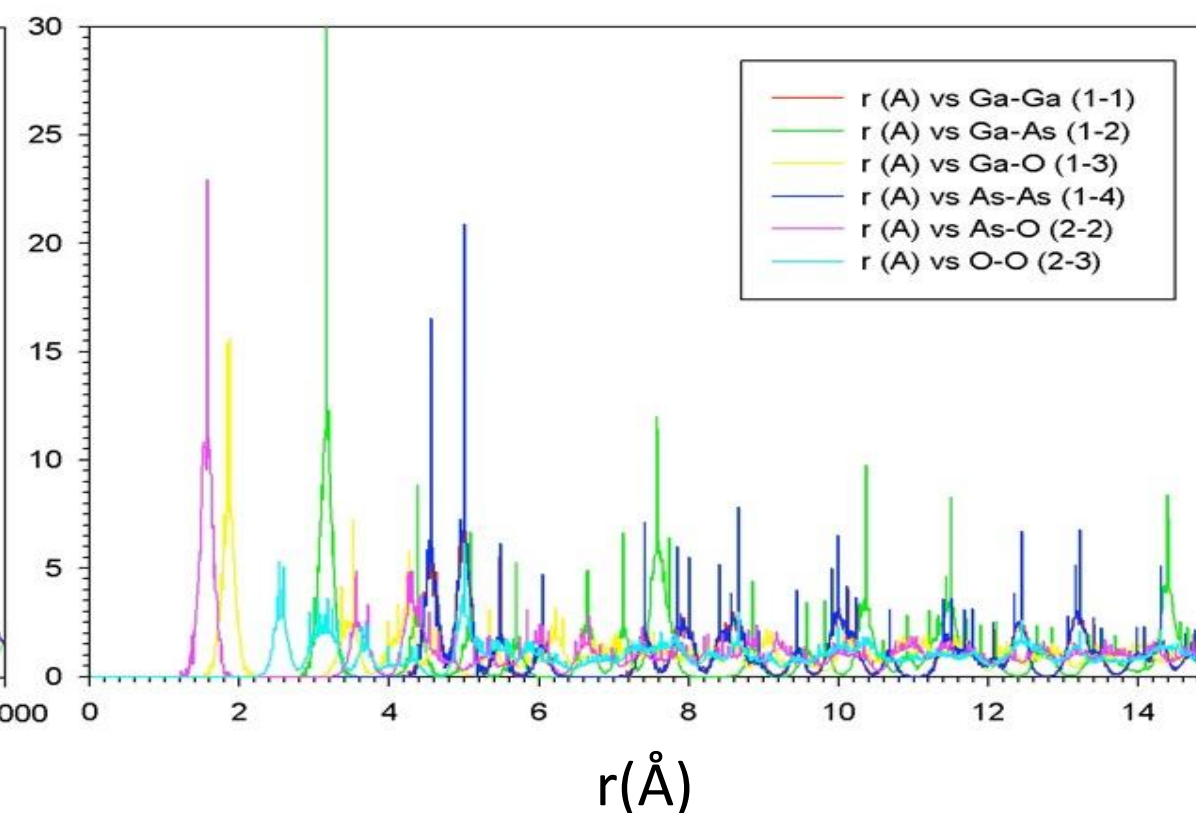
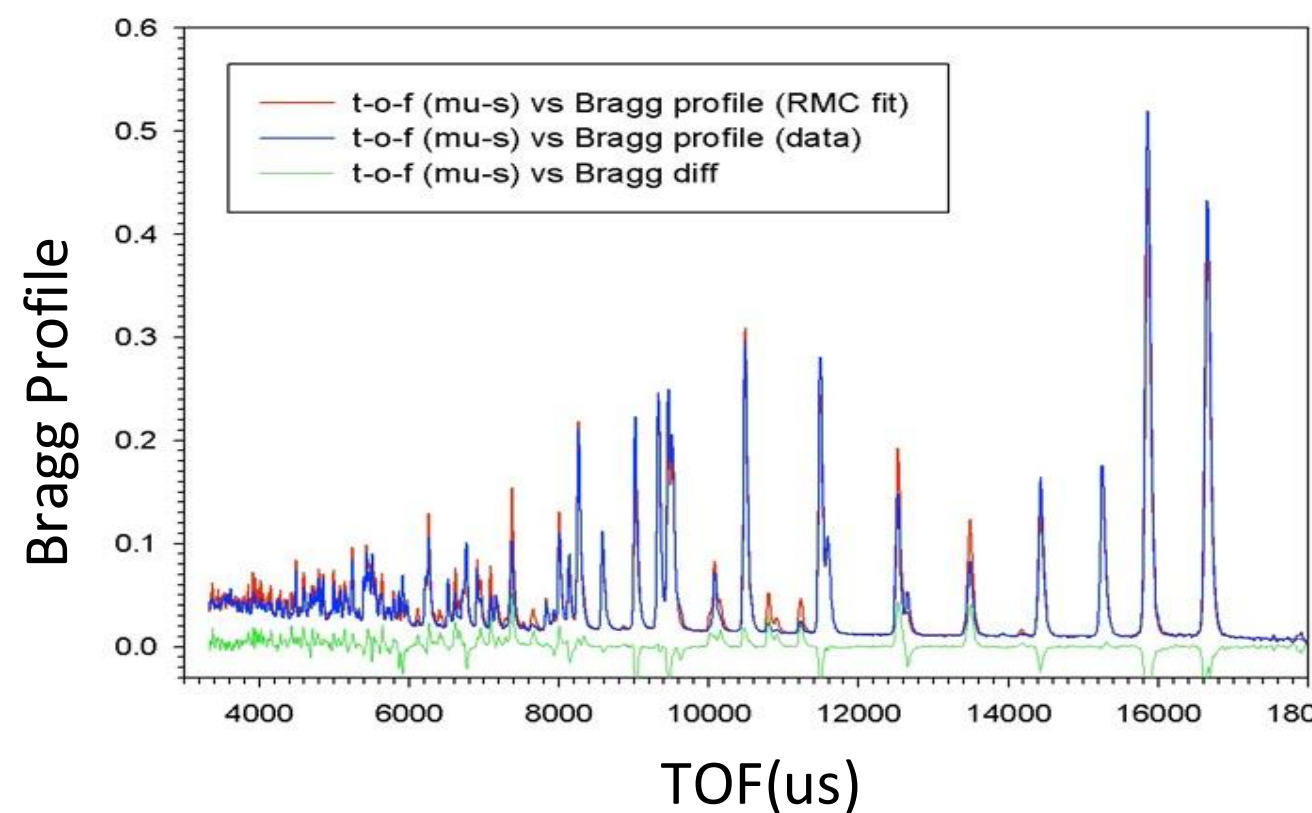
Slides from
Dave Keen

RMCProfile refinement

GaAsO4 RT.files



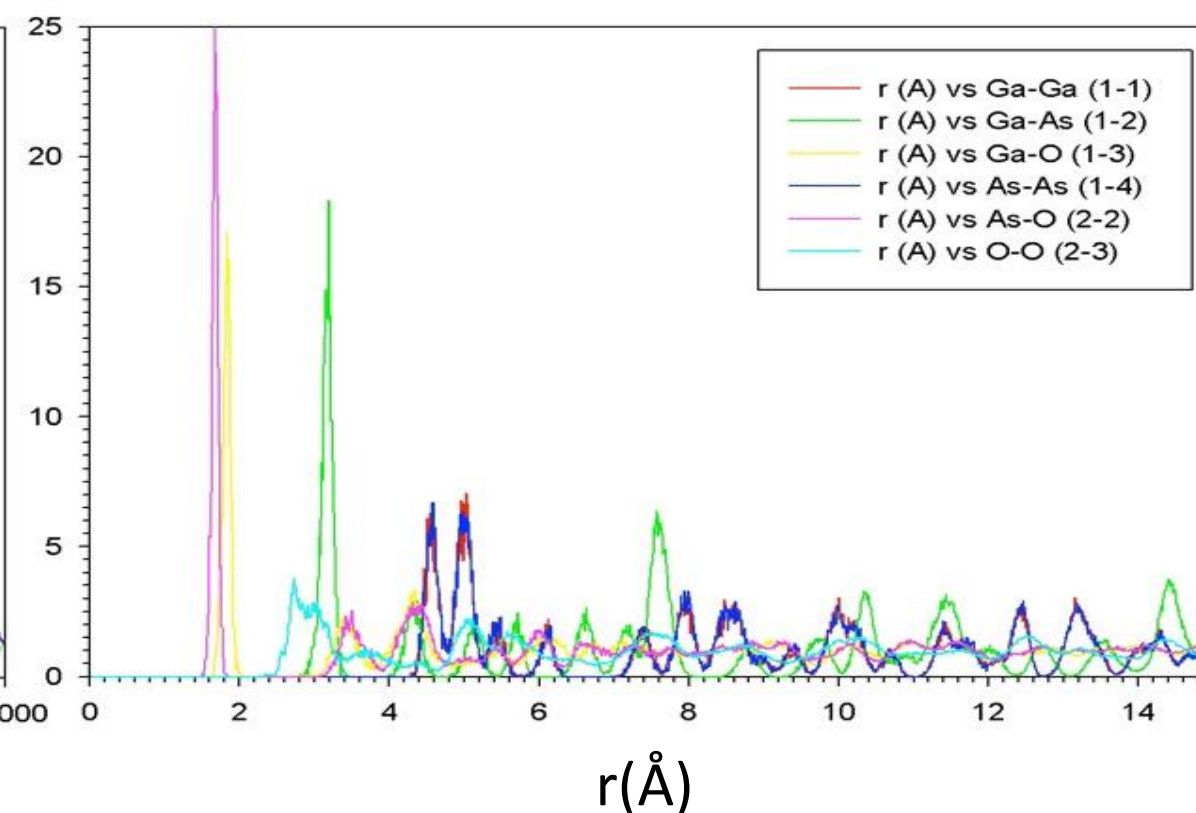
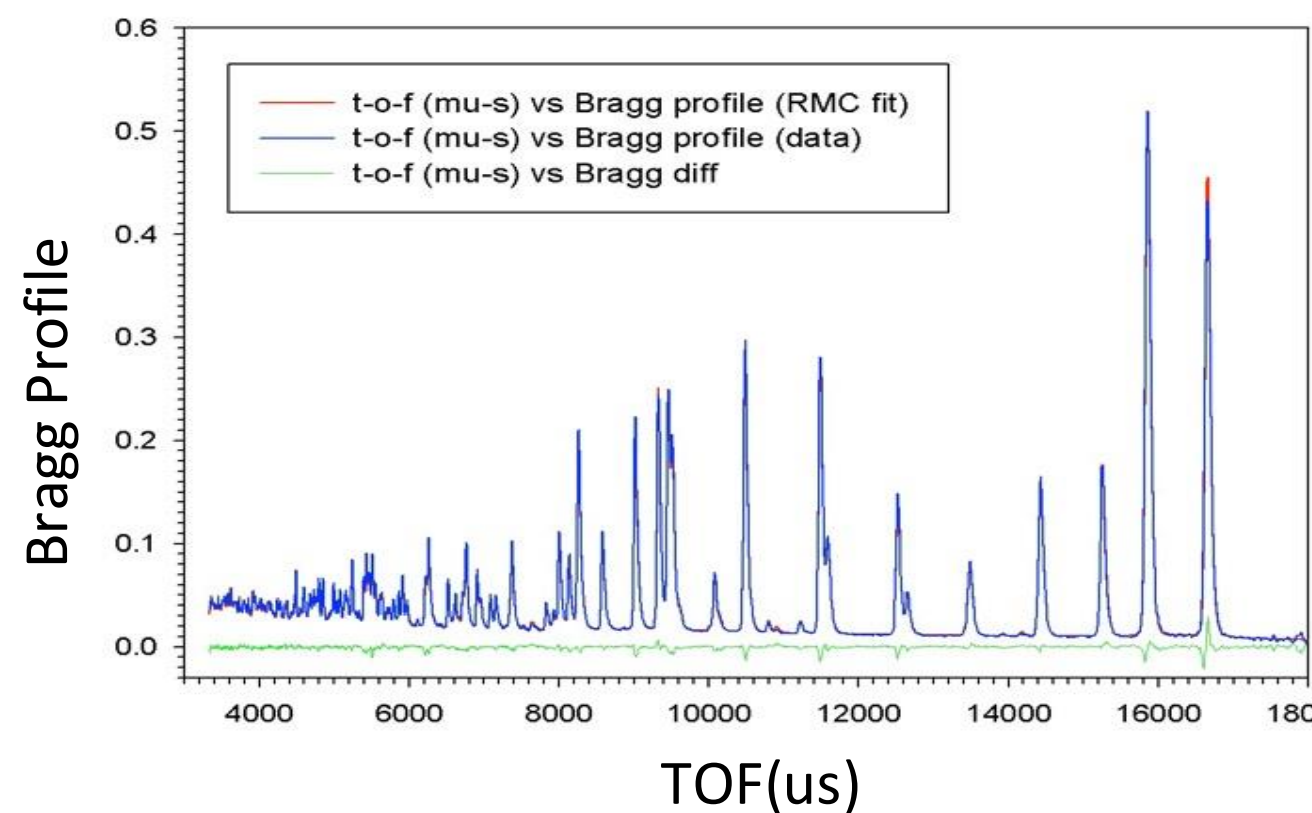
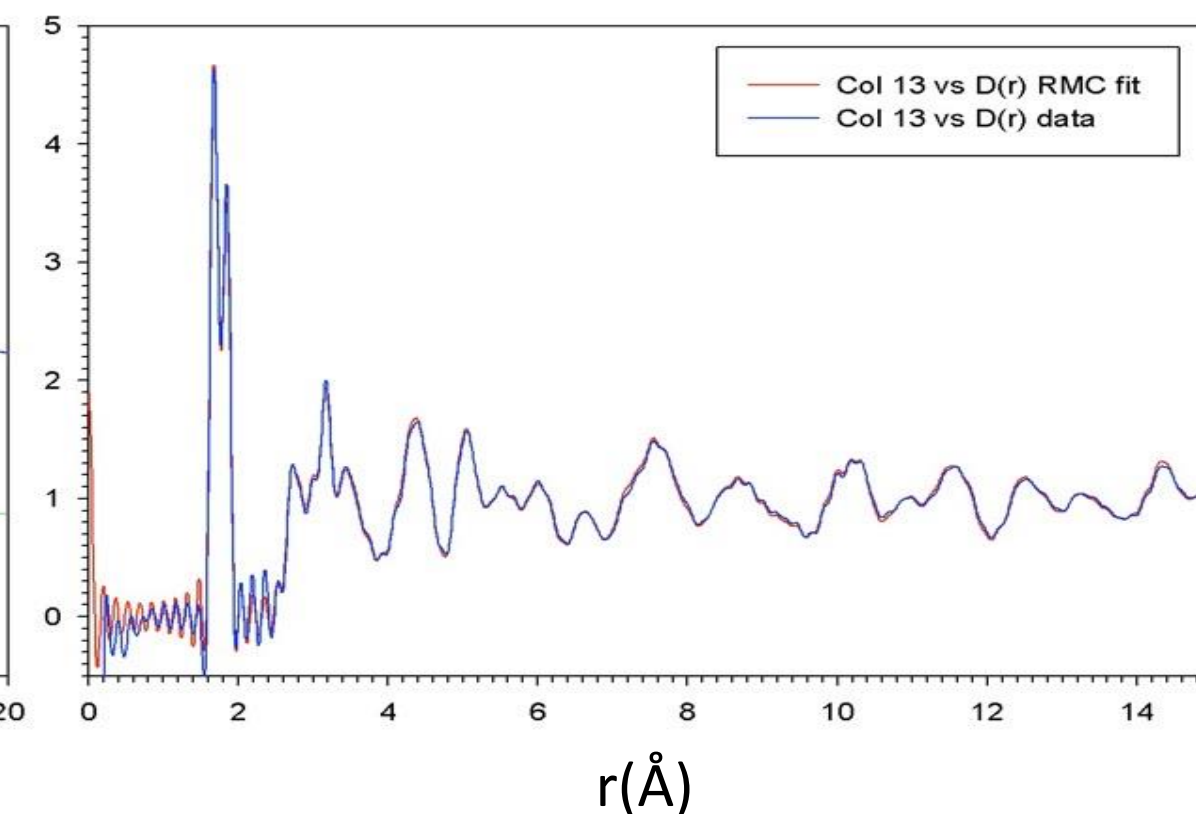
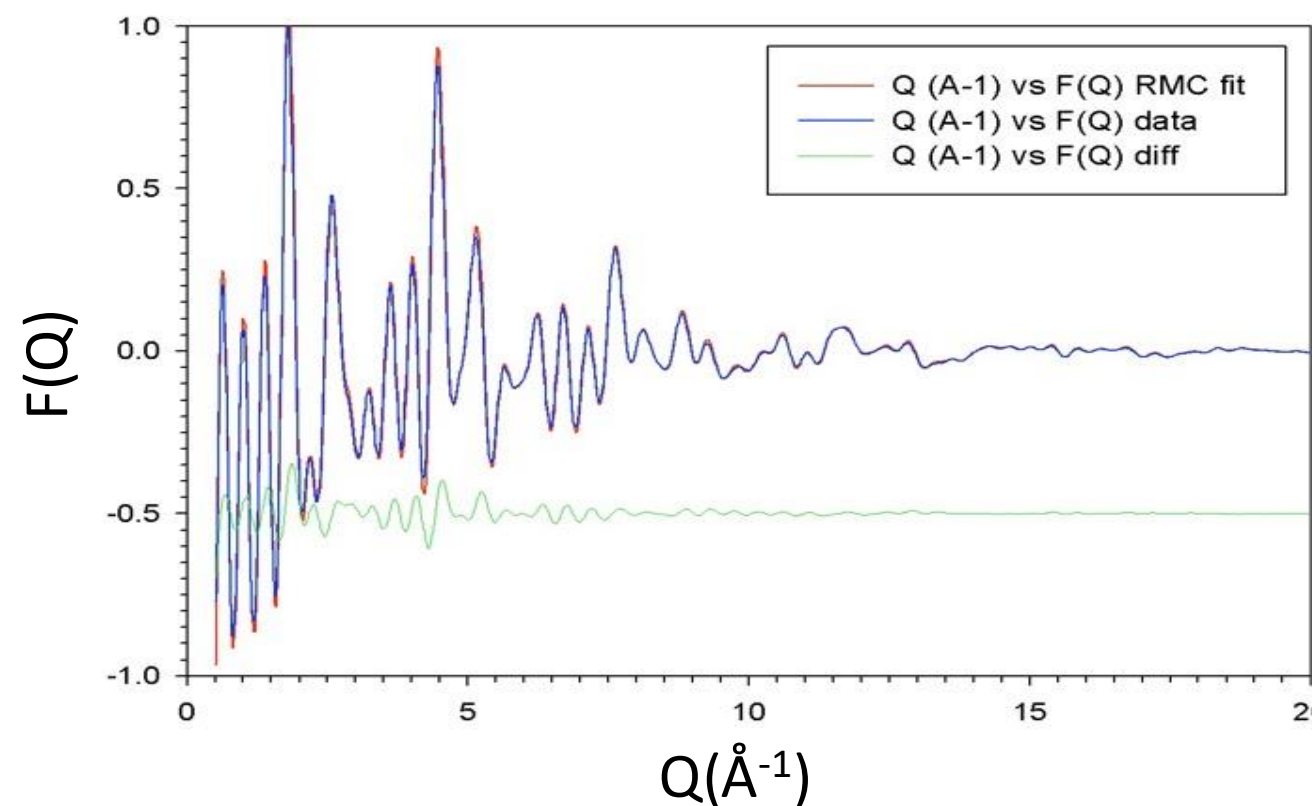
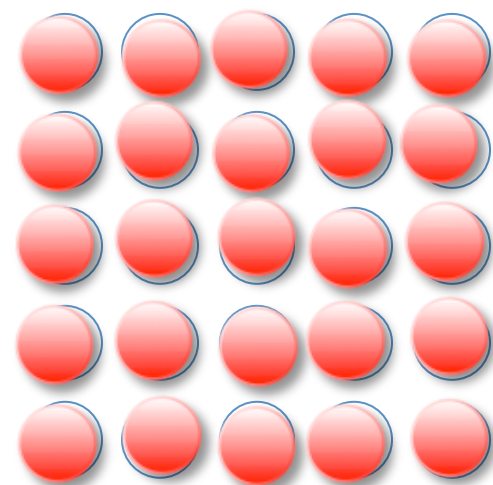
4608 atoms
4699 moves
 $\chi^2 = 92.3$



Slides from
Dave Keen

RMCProfile refinement

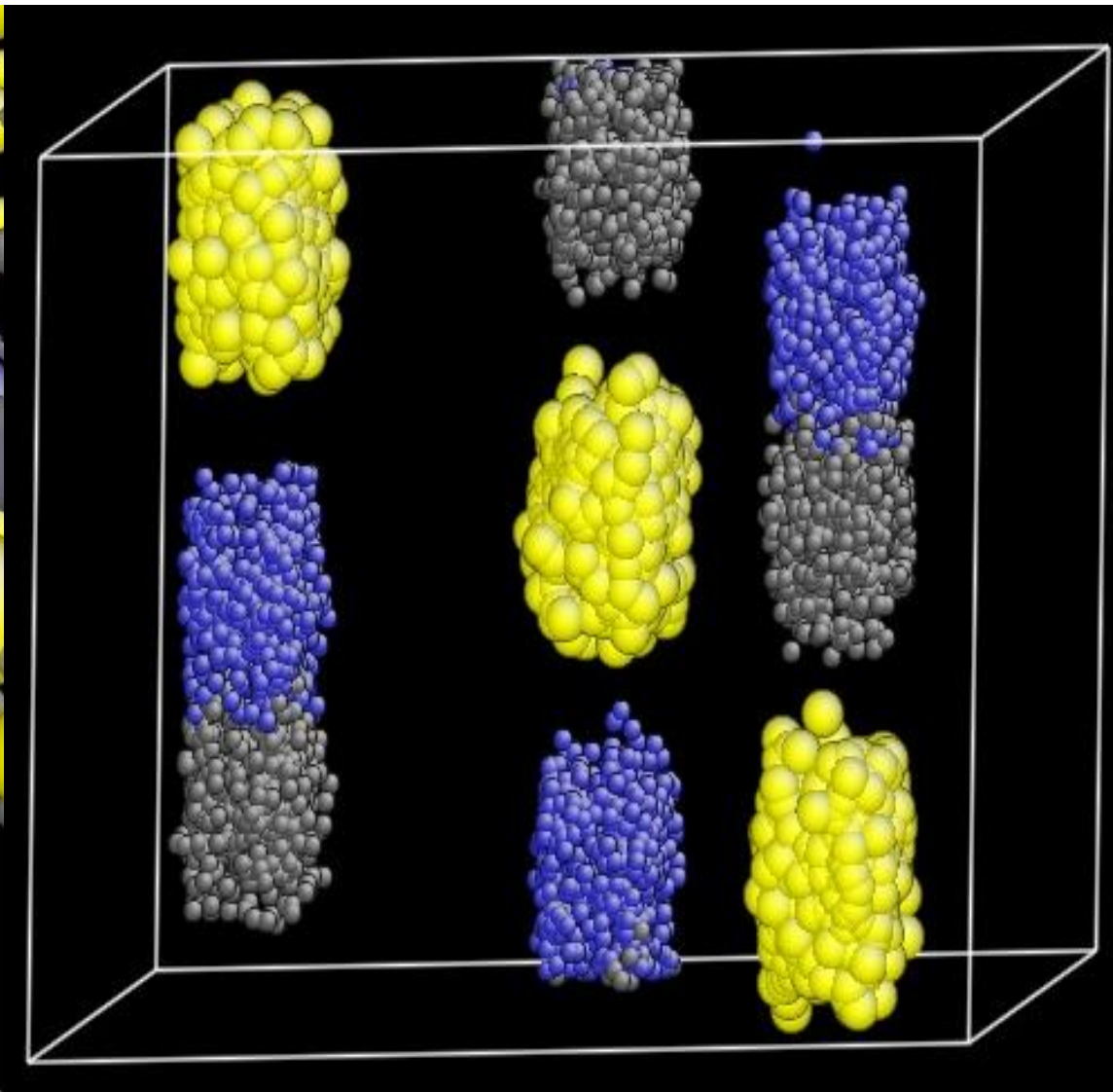
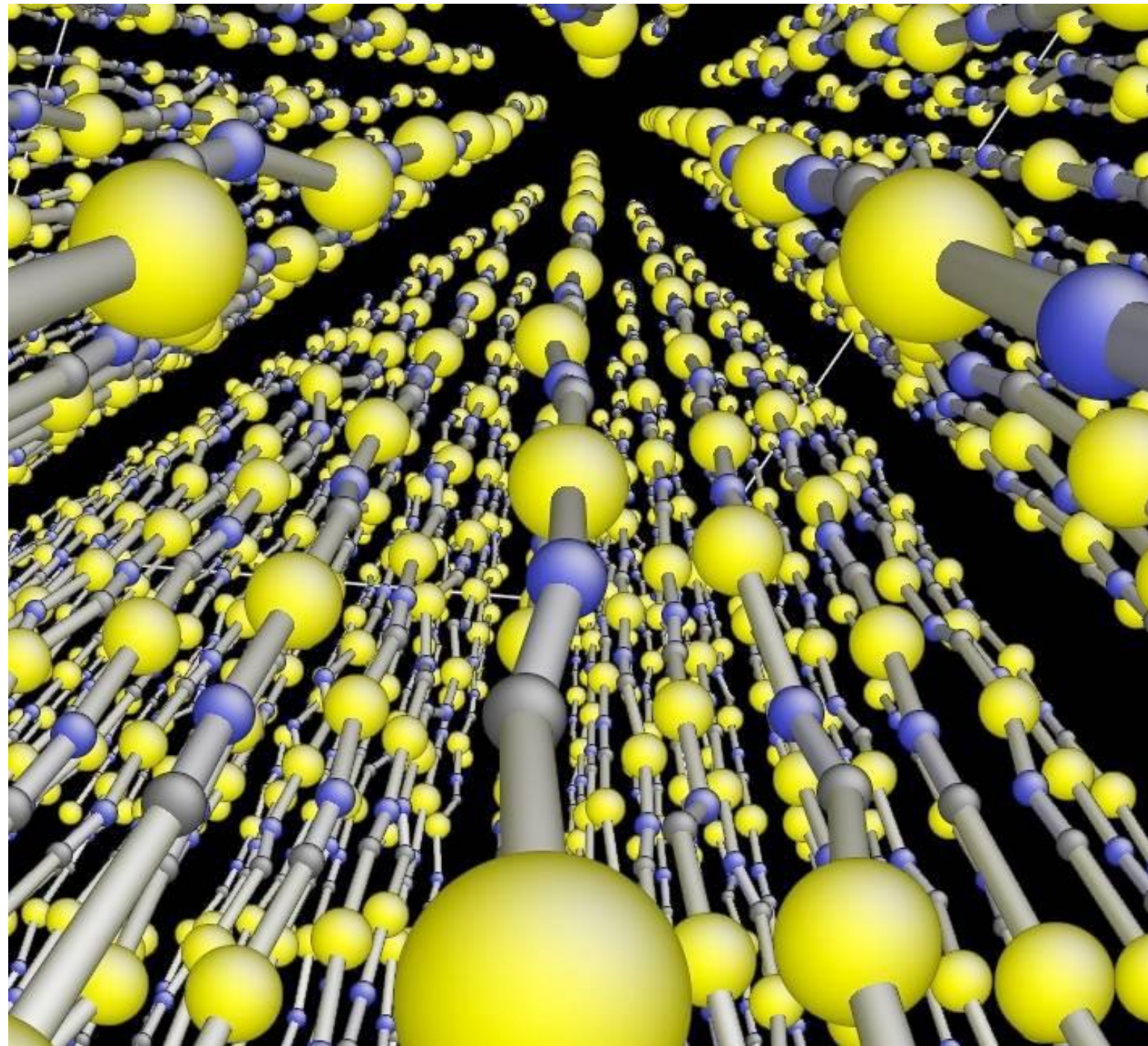
GaAsO₄ RT.files



4608 atoms
RMC converged
 $\chi^2 = 3.43$

Slides from
Dave Keen

Big box models



Collapsed to unit cell

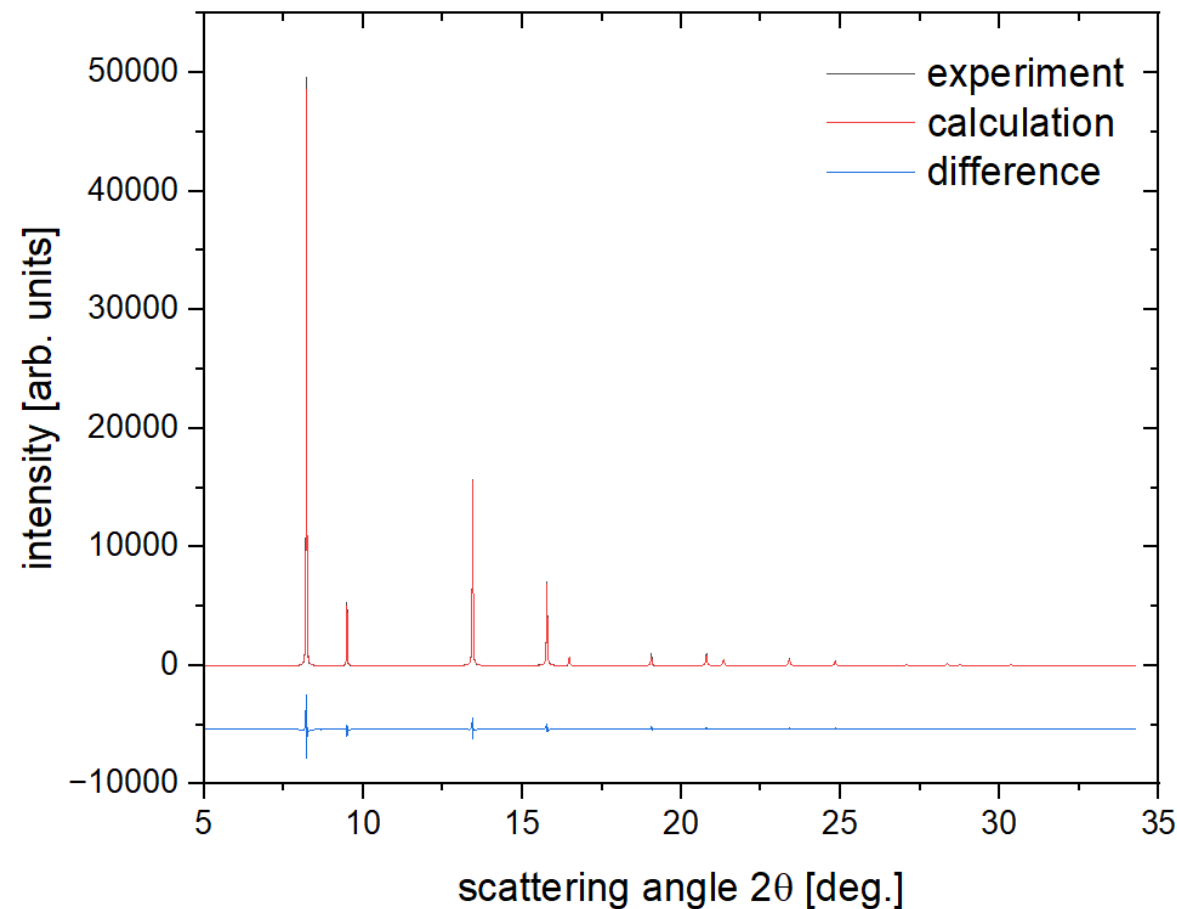
RMCProfile7

- **Multiphase refinement**
 - calculation of all dataset types can be done for multiple phases*
- **RMCProfile7 can fit**
 - real space dataset (X-Ray and neutron) can be calculated as a back Fourier Transform of reciprocal space dataset
 - full GSAS-II compatibility (easy Bragg data extraction)
- **Variety of additional constraints**
 - molecular potentials (distances, angles, torsion angle, inversion angle, planarity + variants)
 - potentials and swaps are now compatible
- **Molecule (rigid body) move type**
 - molecule (rigid body) type move
 - swap between atoms and atoms, atoms to molecules and molecules to molecules

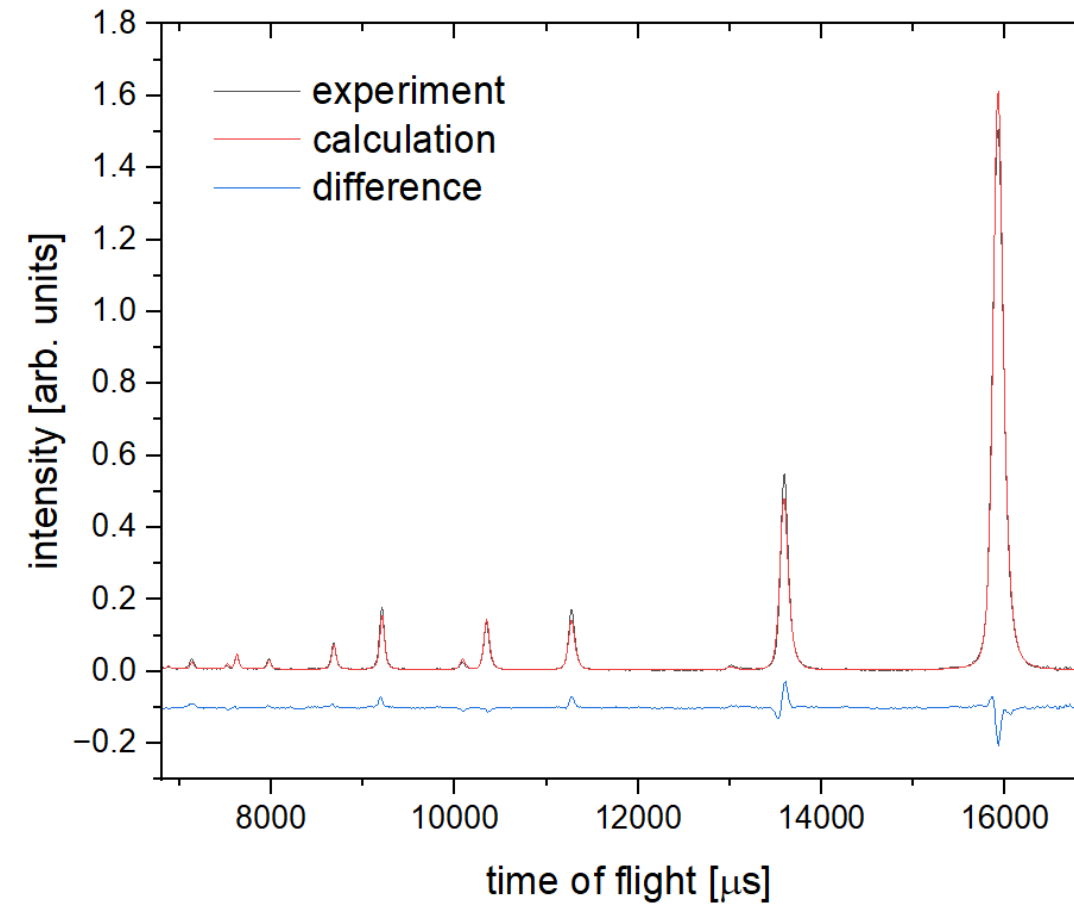
Example 1.

$\text{Sc}_x\text{V}_{1-x}\text{O}_{2-\delta}$ X-Ray and neutron complementarity

x-ray diffraction

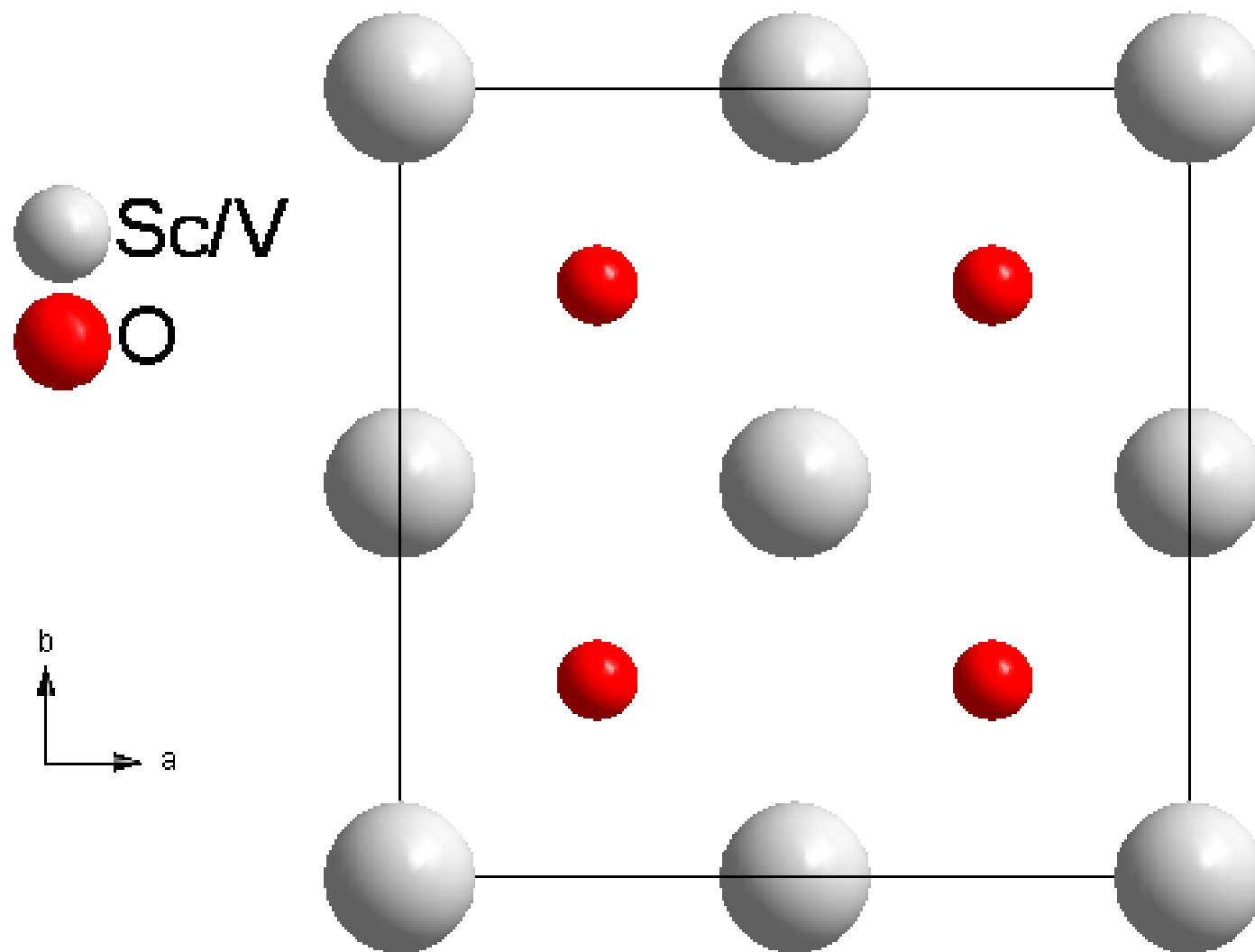


neutron diffraction



$\text{Sc}_x\text{V}_{1-x}\text{O}_{2-\delta}$ X-Ray and neutron complementarity

Average structure with $\delta = 0.25$



General

Origin ScVO_{2.5}

Bibliographic data

Phase data

Space-group F m -3 m (225) - cubic

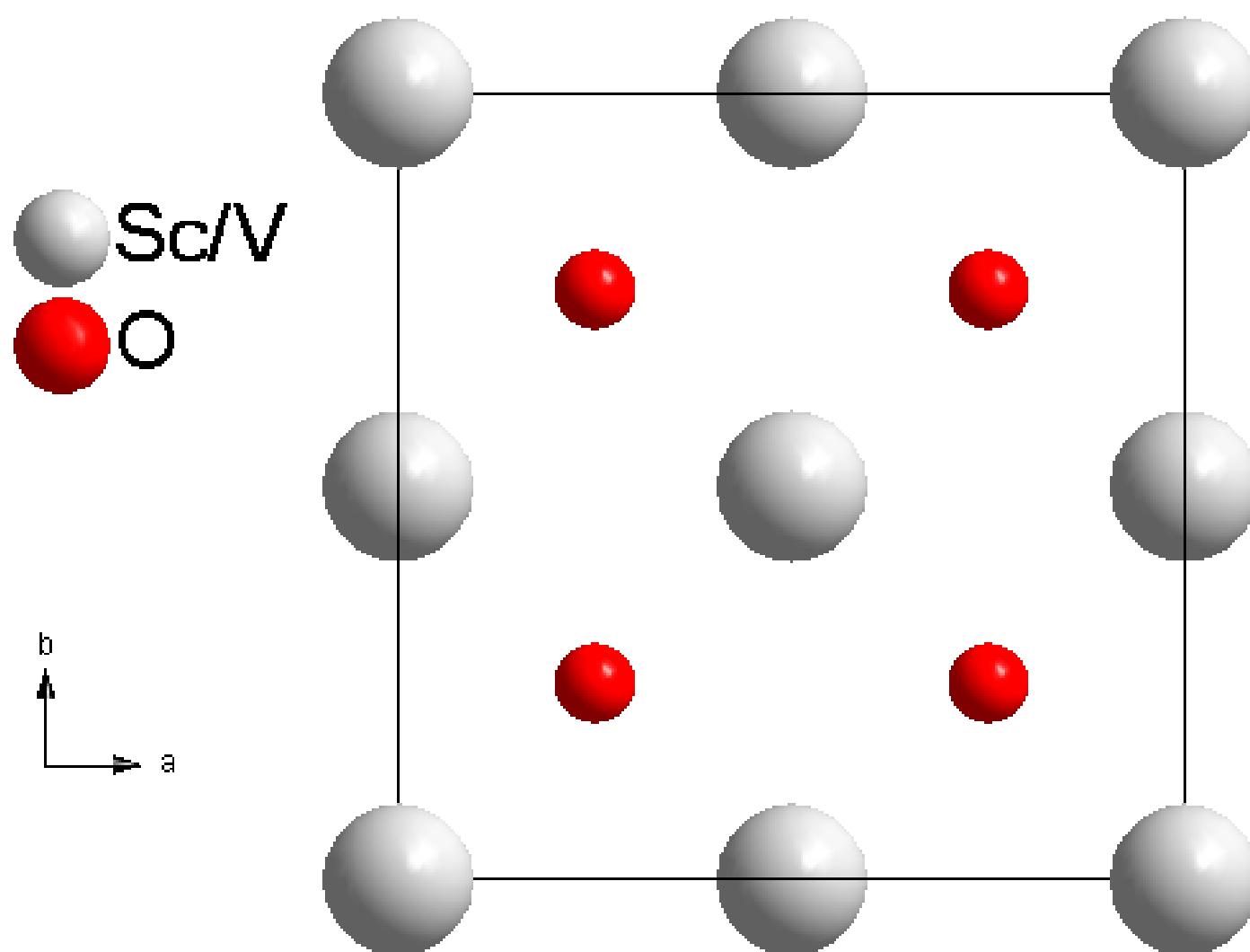
Cell $a=4.9883 \text{ \AA}$
 $V=124.13 \text{ \AA}^3$ $Z=1$

Atomic parameters

Atom	Wyck.	Site	S.O.F.	x/a	y/b	z/c	U [$\text{\AA}^2$]
V	4a	m-3m	0.3333	0	0	0	0.0620
Sc	4a	m-3m	0.6667	0	0	0	0.0435
O1	8c	-43m	0.875	1/4	1/4	1/4	0.1003

$\text{Sc}_x\text{V}_{1-x}\text{O}_{2-\delta}$ X-Ray and neutron complementarity

Average structure



General	
Origin	ScVO2.5
Bibliographic data	
Phase data	
Space-group	F m -3 m (225) - cubic
Cell	a=4.9883 Å V=124.13 Å ³ Z=1

Atomic parameters							
Atom	Wyck.	Site	S.O.F.	x/a	y/b	z/c	U [Å ²]
V	4a	m-3m	0.3333	0	0	0	0.0620
Sc	4a	m-3m	0.6667	0	0	0	0.0435
O1	8c	-43m	0.875	1/4	1/4	1/4	0.1003

$\text{Sc}_x\text{V}_{1-x}\text{O}_{2-\delta}$ X-Ray and neutron complementarity

from Bragg diffraction

- unit cell size
- crystallographic lattice
- average atom positions and average distances

$$d_{ij} = \langle r_i \rangle - \langle r_j \rangle$$

- structure composition

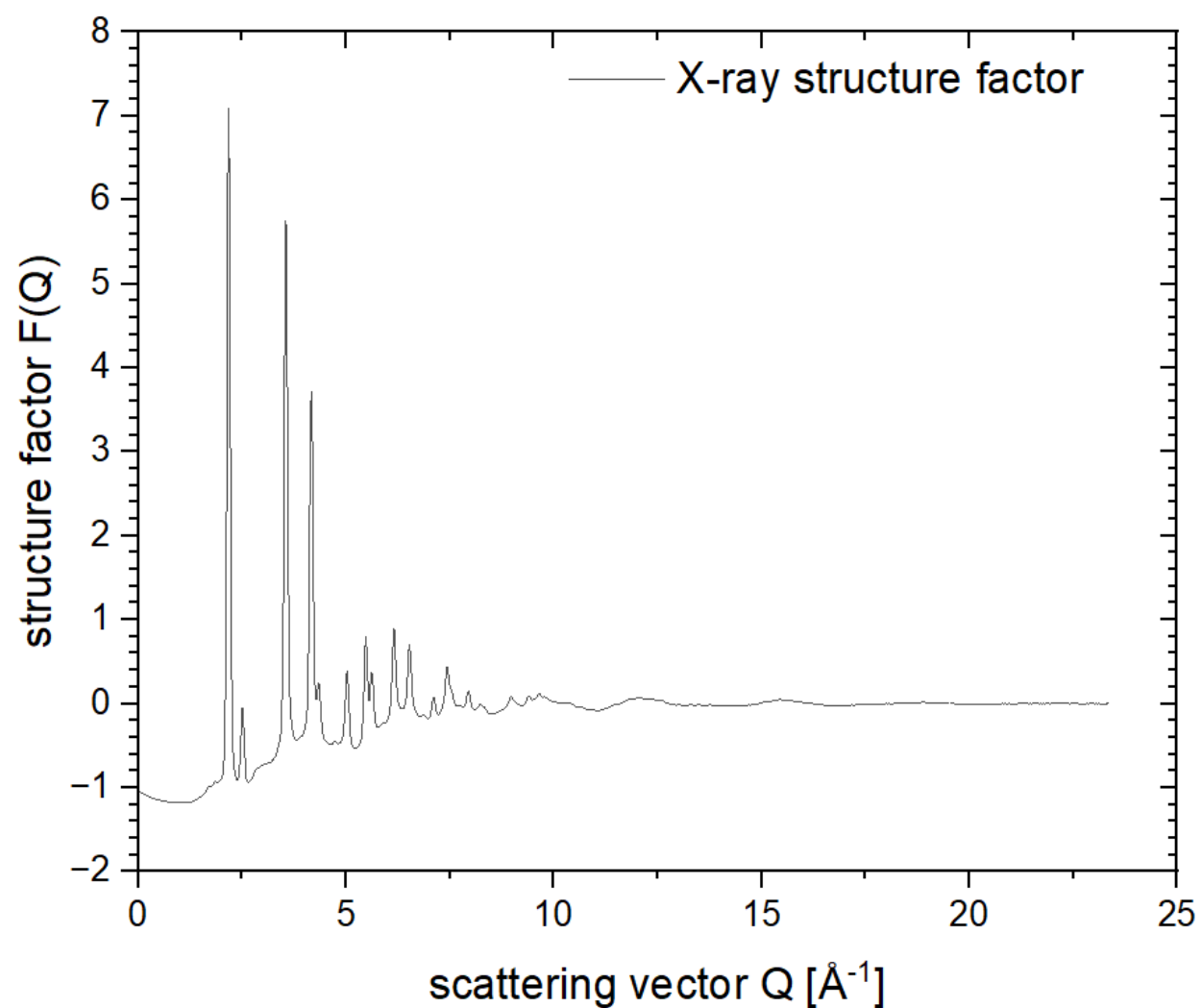
from Total Scattering

- local crystal structure
- local distortions and dislocations
- individual atom positions and distances

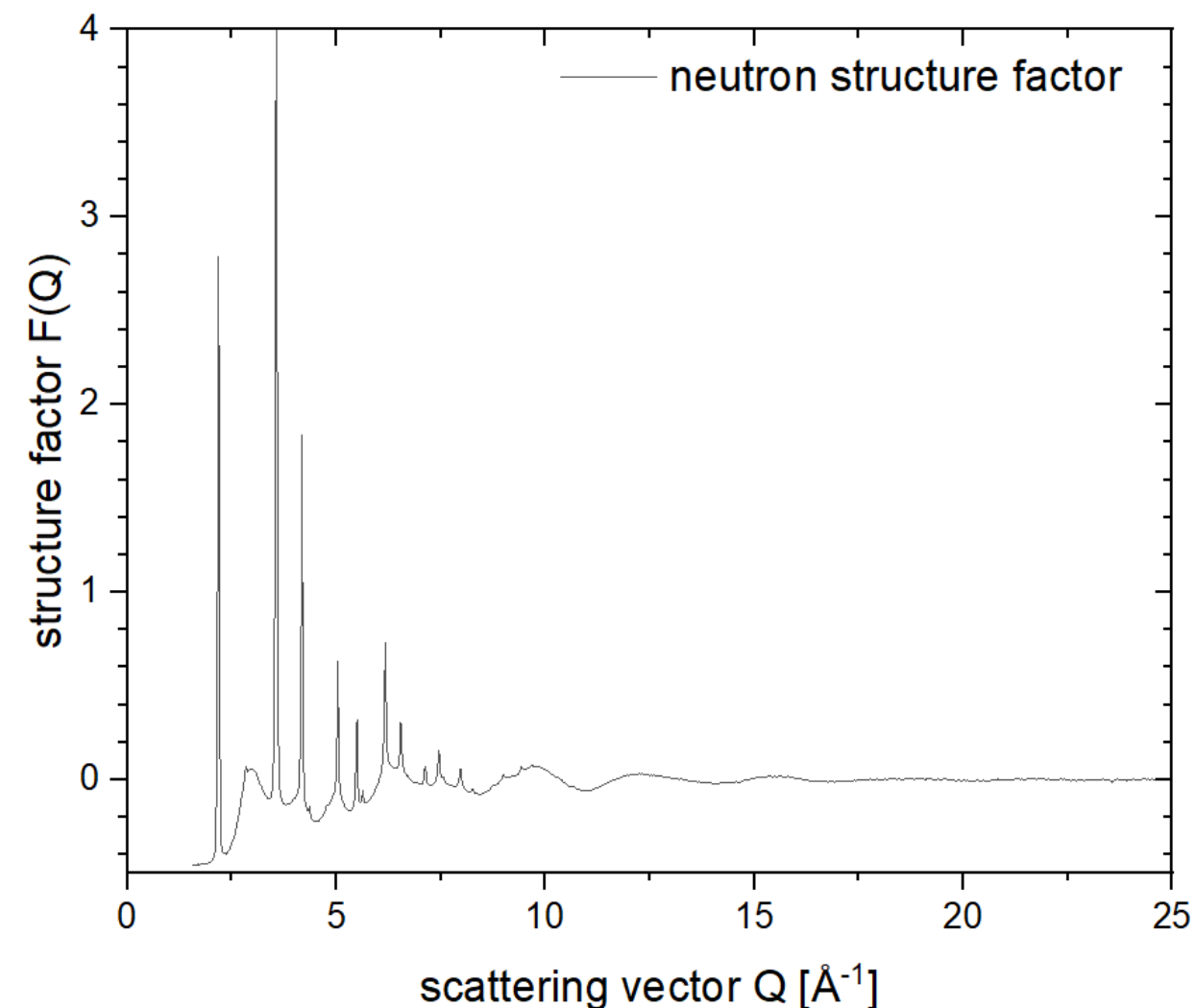
$$d_{ij} = \langle r_i - r_j \rangle$$

$\text{Sc}_x\text{V}_{1-x}\text{O}_{2-\delta}$ X-Ray and neutron complementarity

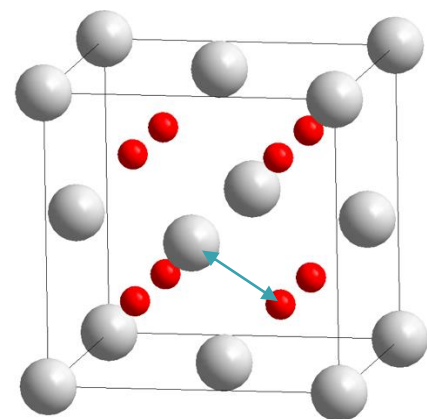
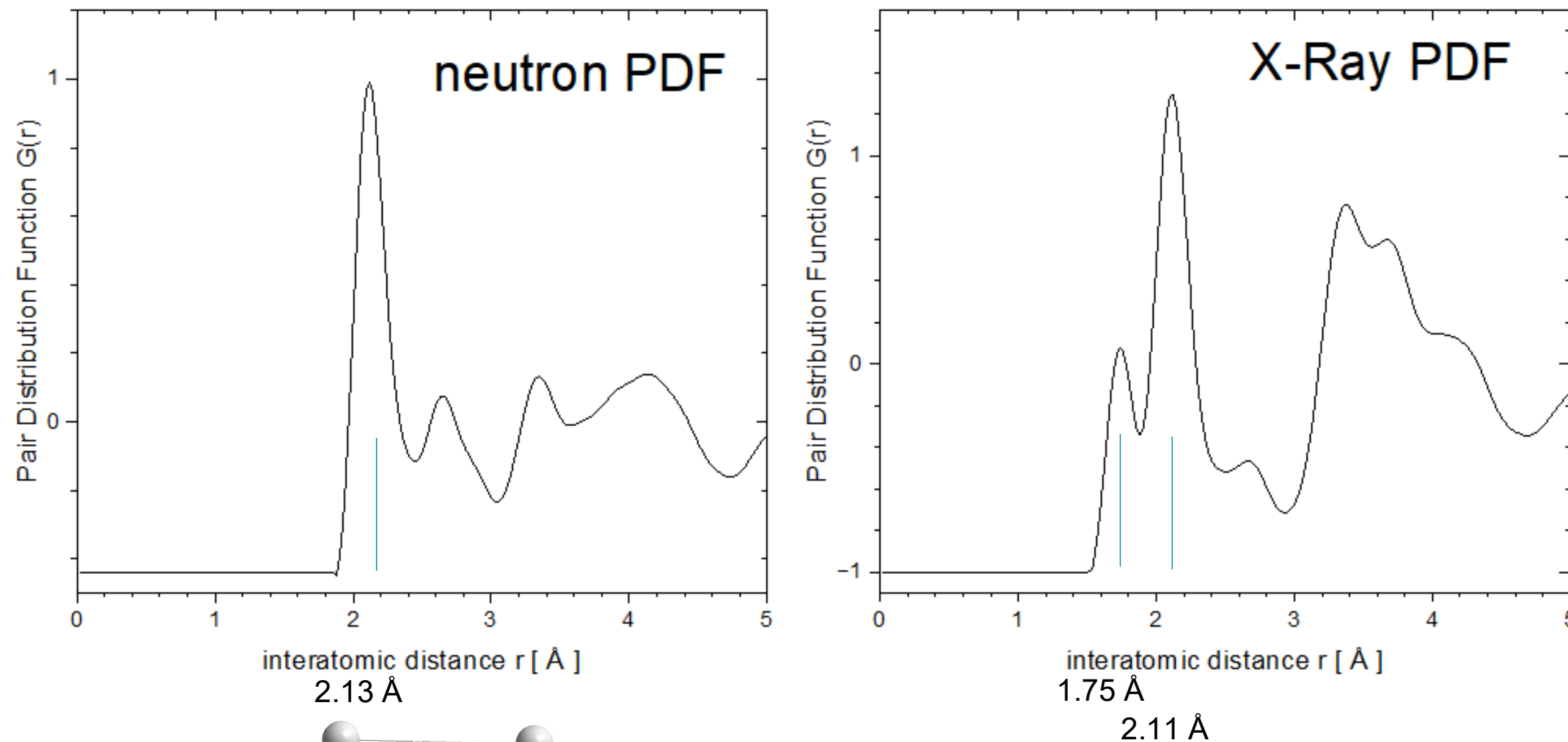
x-ray diffraction



neutron diffraction

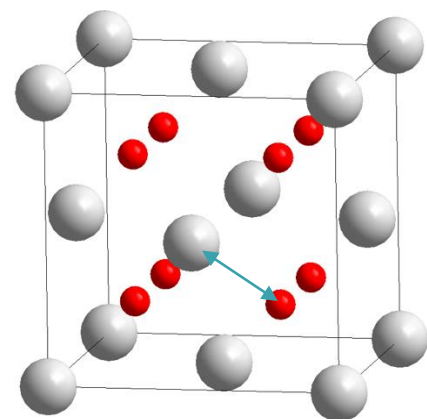
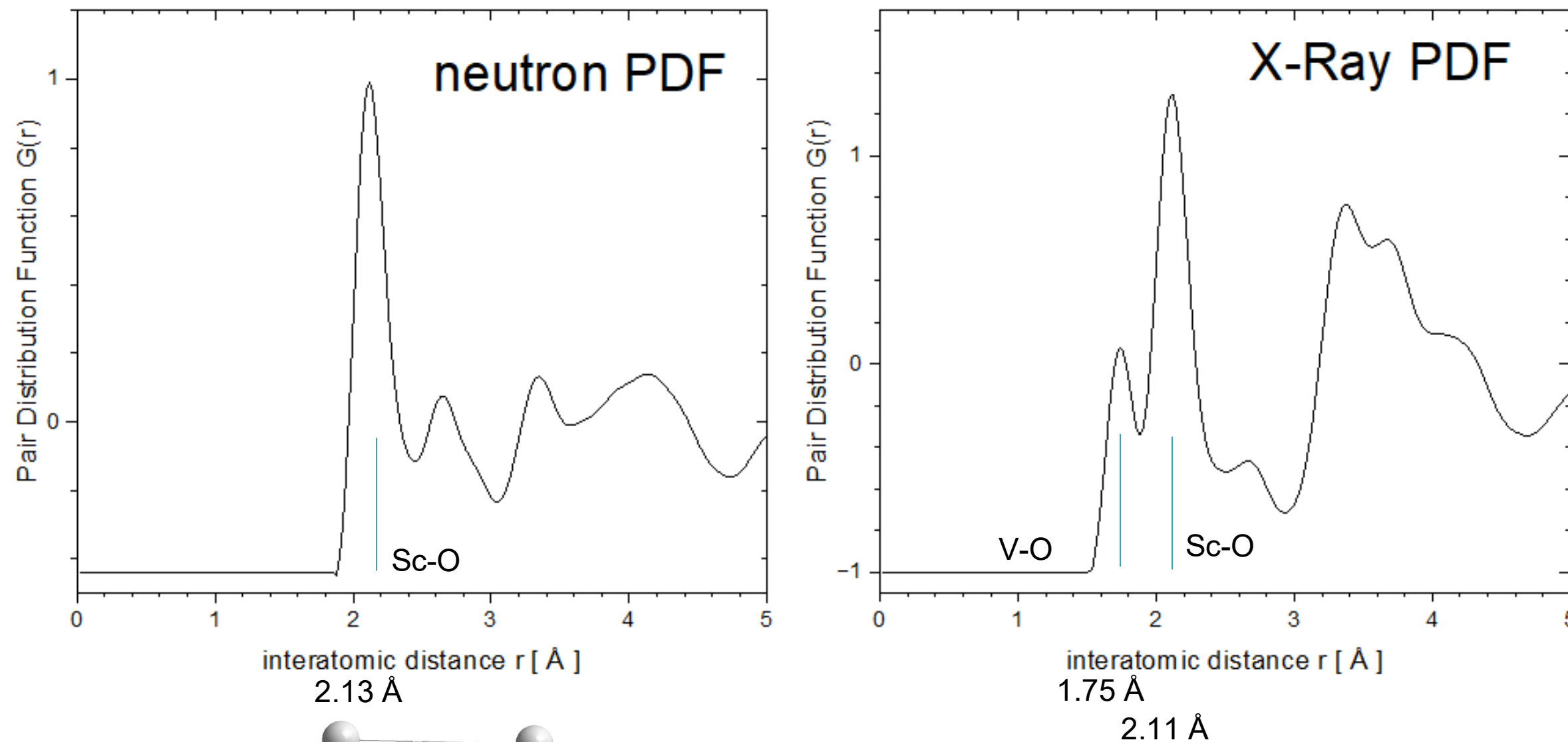


$\text{Sc}_x\text{V}_{1-x}\text{O}_{2-\delta}$ X-Ray and neutron complementarity



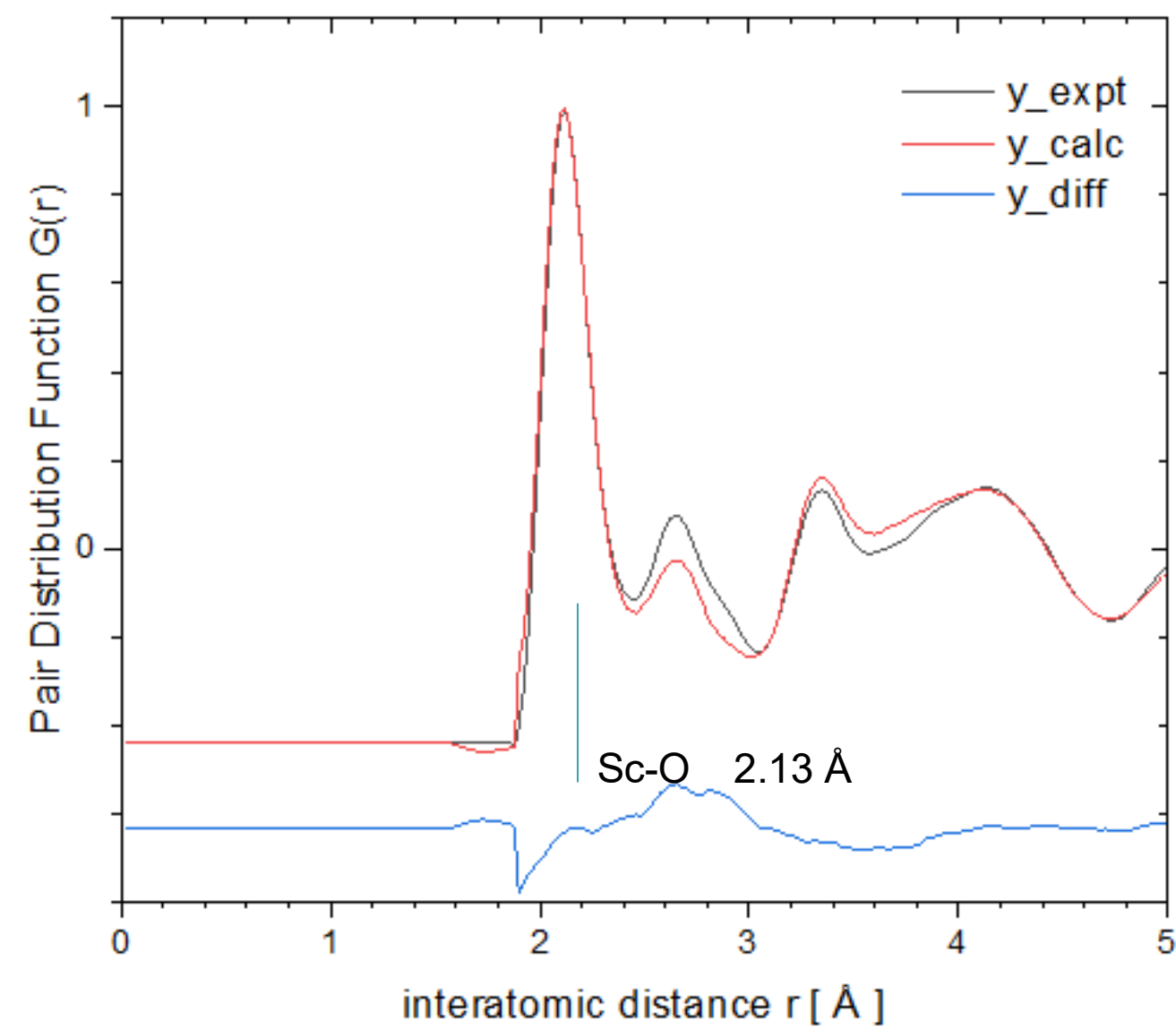
$d_{\text{Sc/V-O}} = 2.16 \text{ Å}$ (from average structure)

$\text{Sc}_x\text{V}_{1-x}\text{O}_{2-\delta}$ X-Ray and neutron complementarity

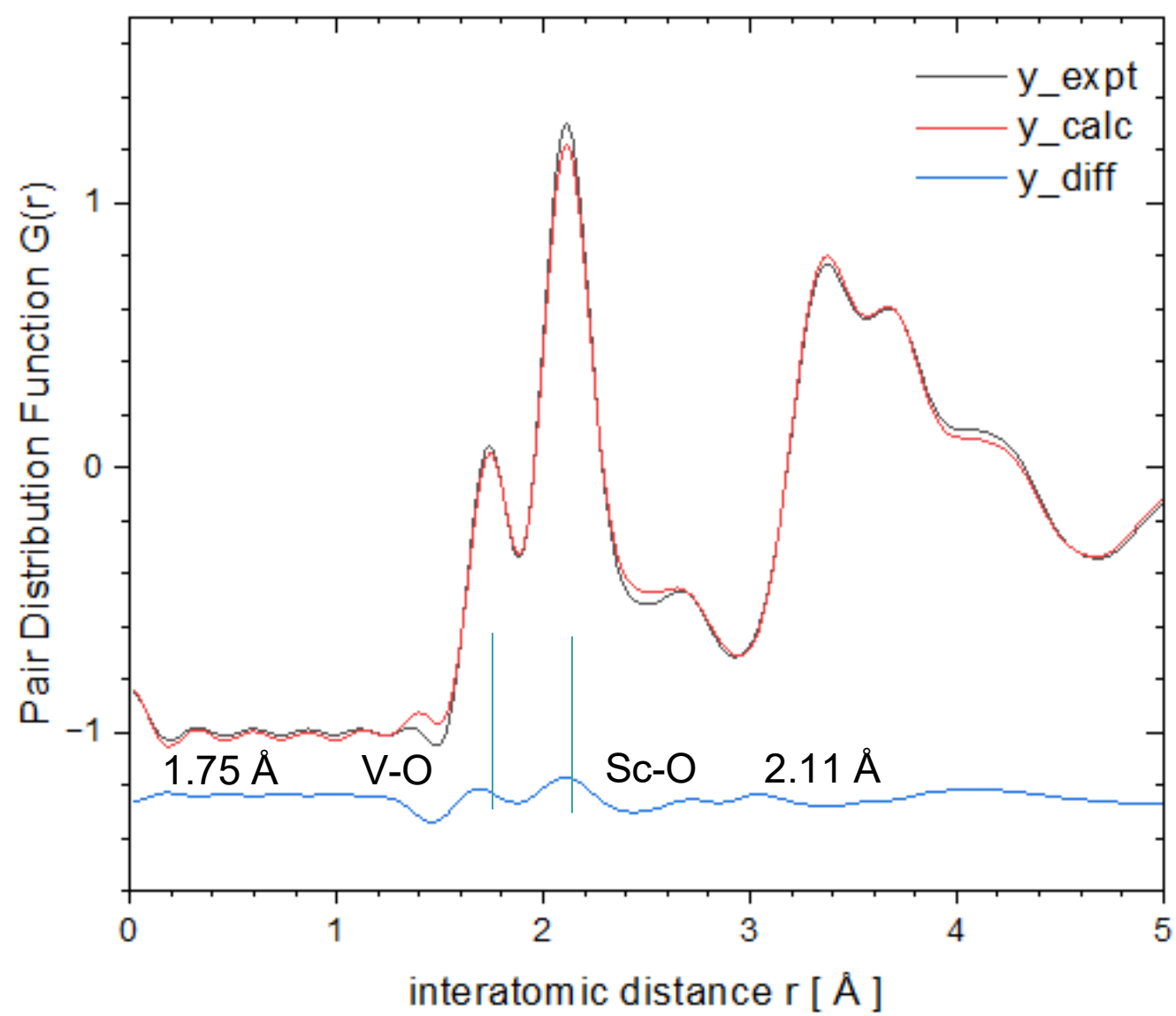


$d_{\text{Sc/V}-\text{O}} = 2.16 \text{ Å}$ (from average structure)

Sc_xV_{1-x}O_{2-δ} X-Ray and neutron complementarity



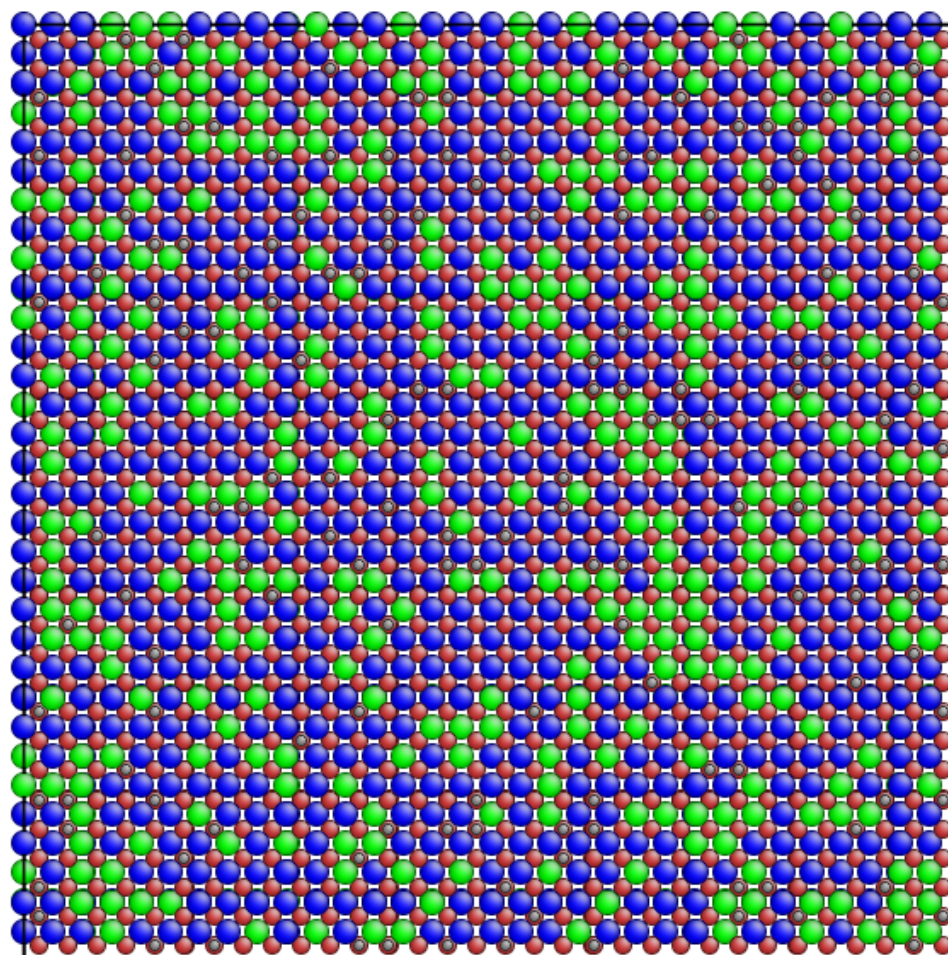
neutrons	Sc	V	O
Sc	0.202	-0.006	0.501
V		0.000	-0.008
O			0.311



X-Rays	Sc	V	O
Sc	0.154	0.169	0.308
V		0.046	0.169
O			0.154

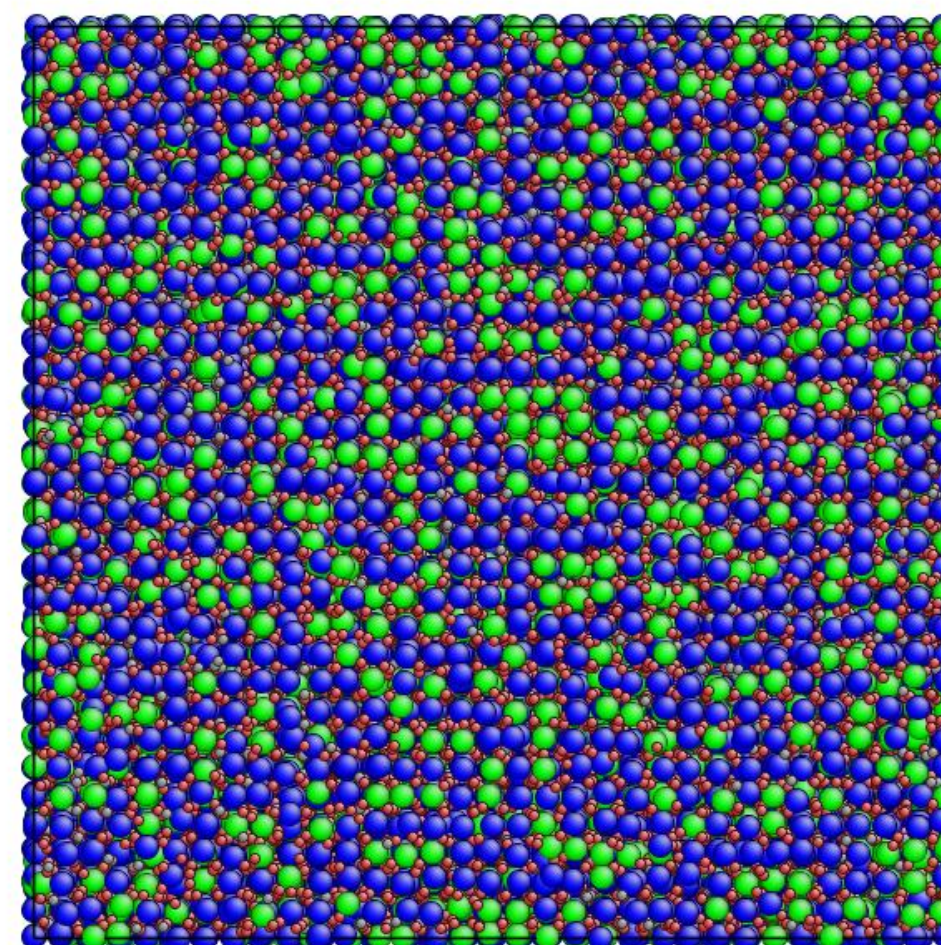
$\text{Sc}_x\text{V}_{1-x}\text{O}_{2-\delta}$ initial vs final configuration

initial configuration

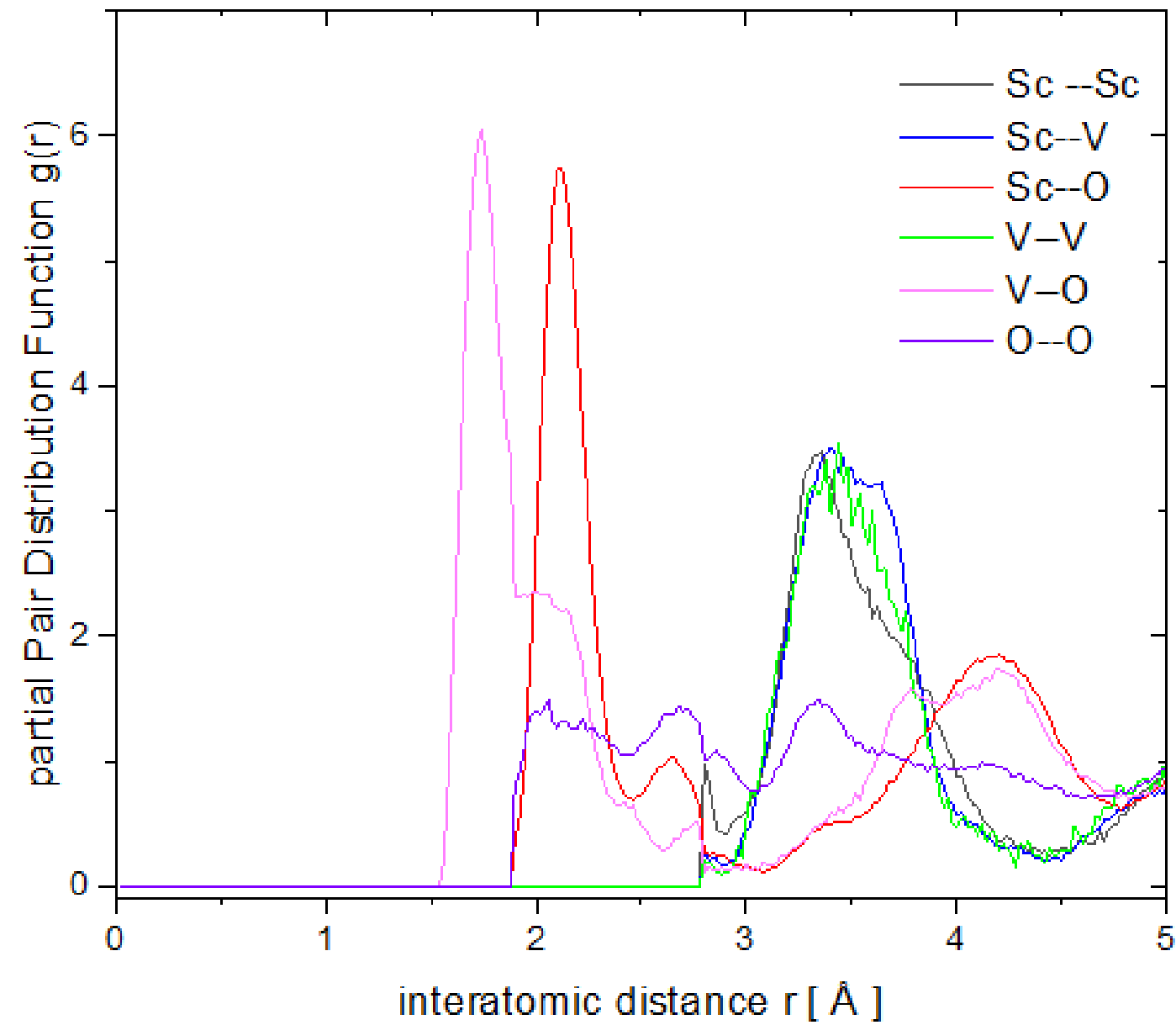
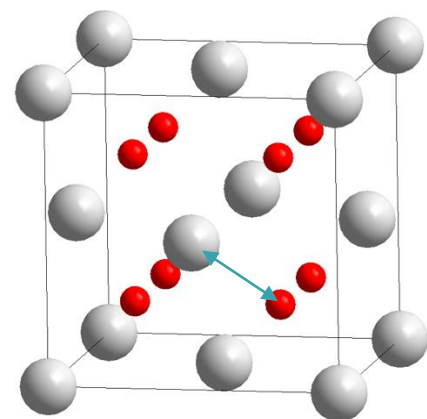


16x16x16 supercell

final configuration

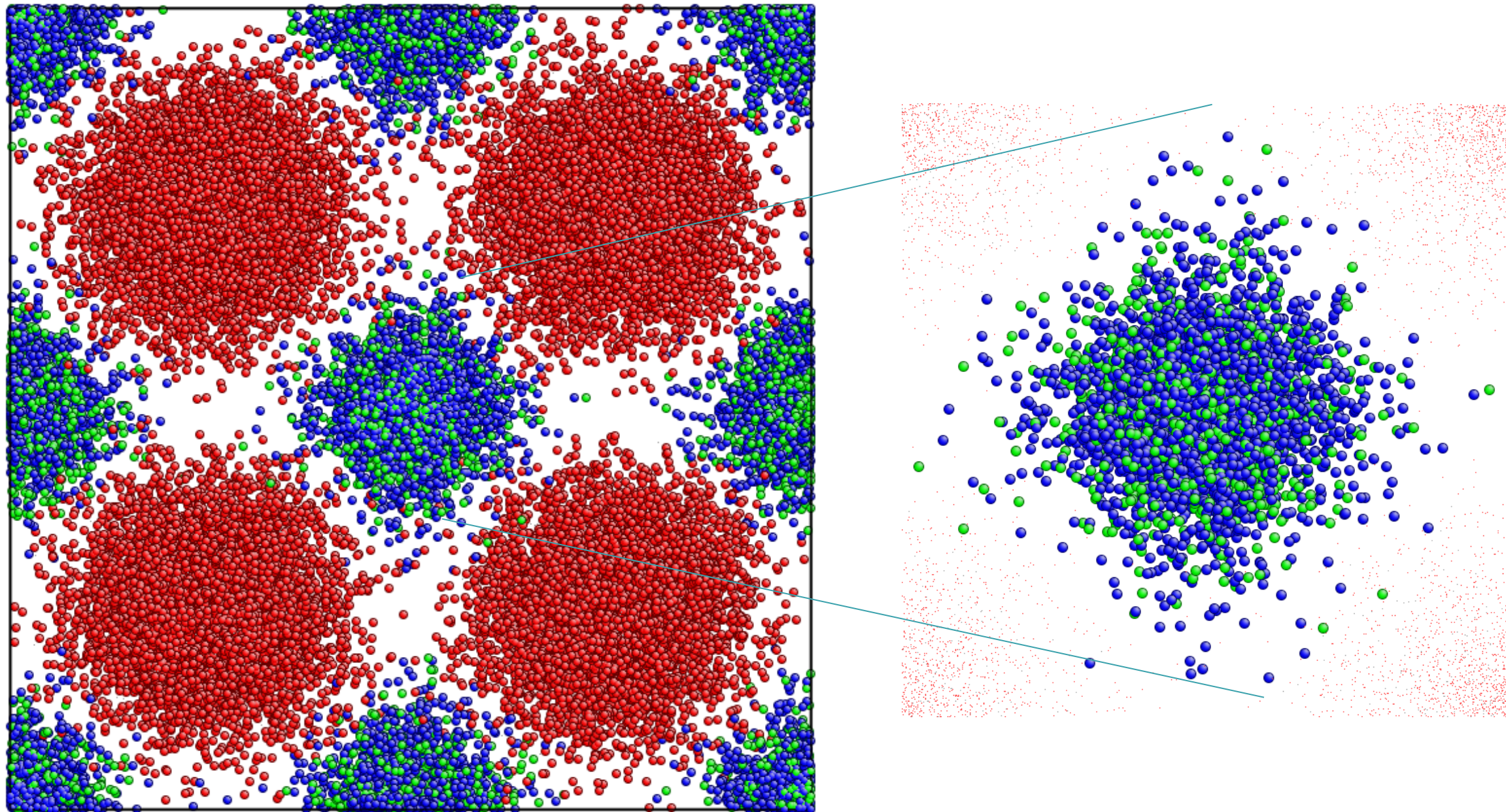


$\text{Sc}_x\text{V}_{1-x}\text{O}_{2-\delta}$ partials

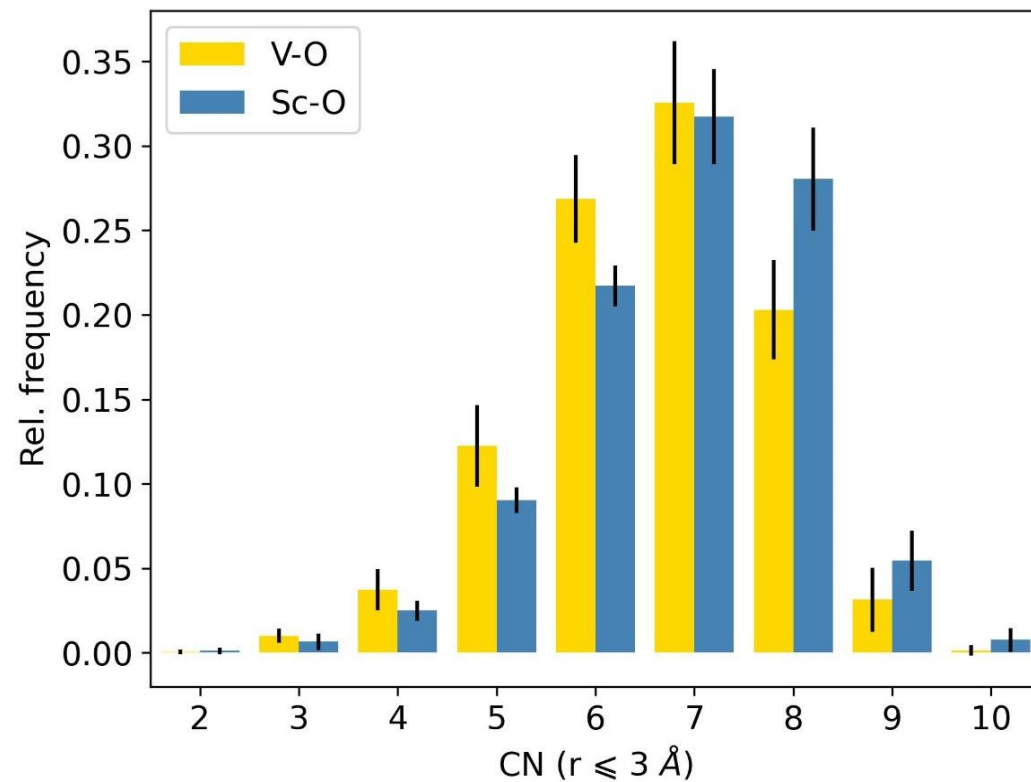


$d_{\text{Sc/V}-\text{O}} = 2.16 \text{ \AA}$ (from average structure)

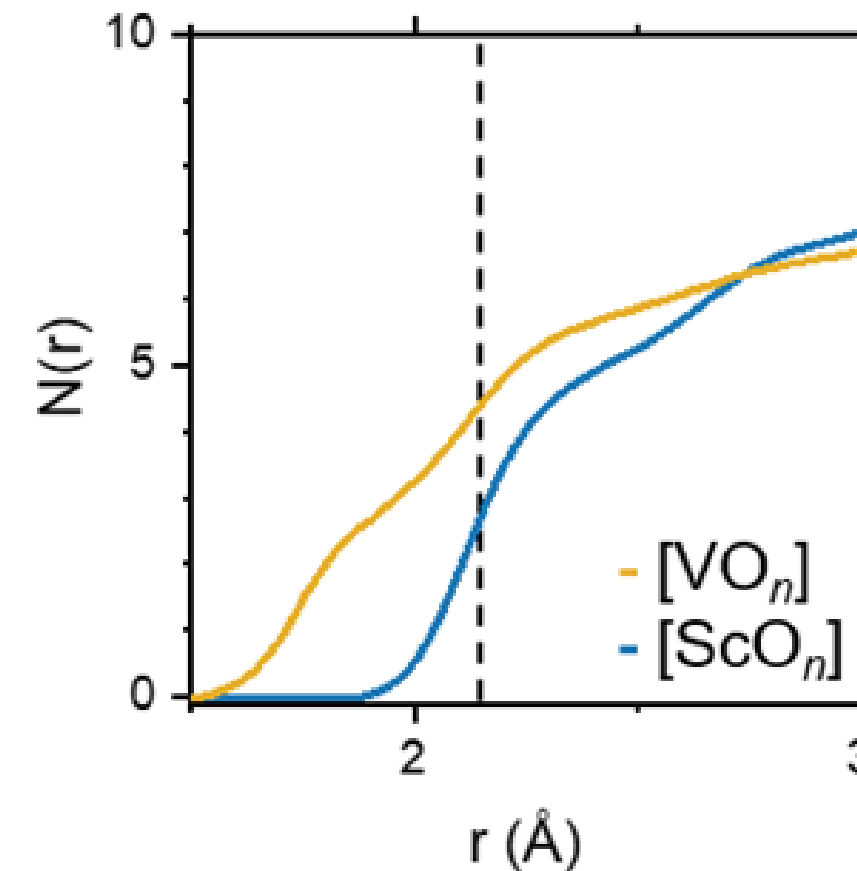
$\text{Sc}_x\text{V}_{1-x}\text{O}_{2-\delta}$ post refinement analysis



$\text{Sc}_x\text{V}_{1-x}\text{O}_{2-\delta}$ post refinement analysis



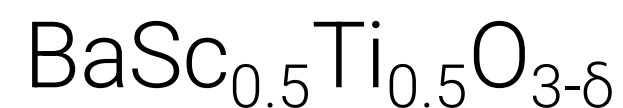
Distribution of coordination numbers for $[\text{VO}_n]$ (yellow) and $[\text{ScO}_n]$ (blue) polyhedra in $\text{c-Sc}_2\text{VO}_{5.25}$ obtained from 10 independent RMC refinements. Bond lengths are limited to $r_{\text{max}} = 3.0 \text{ \AA}$.



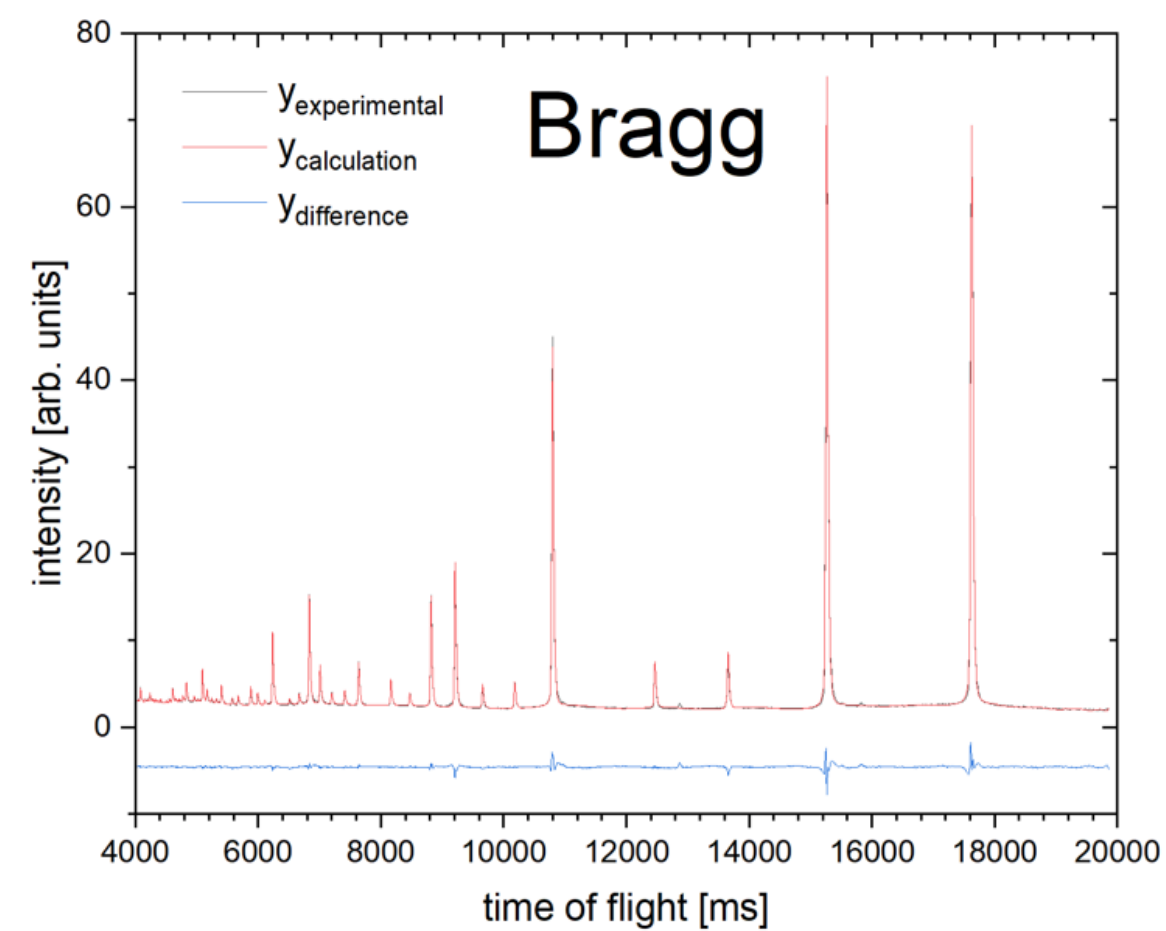
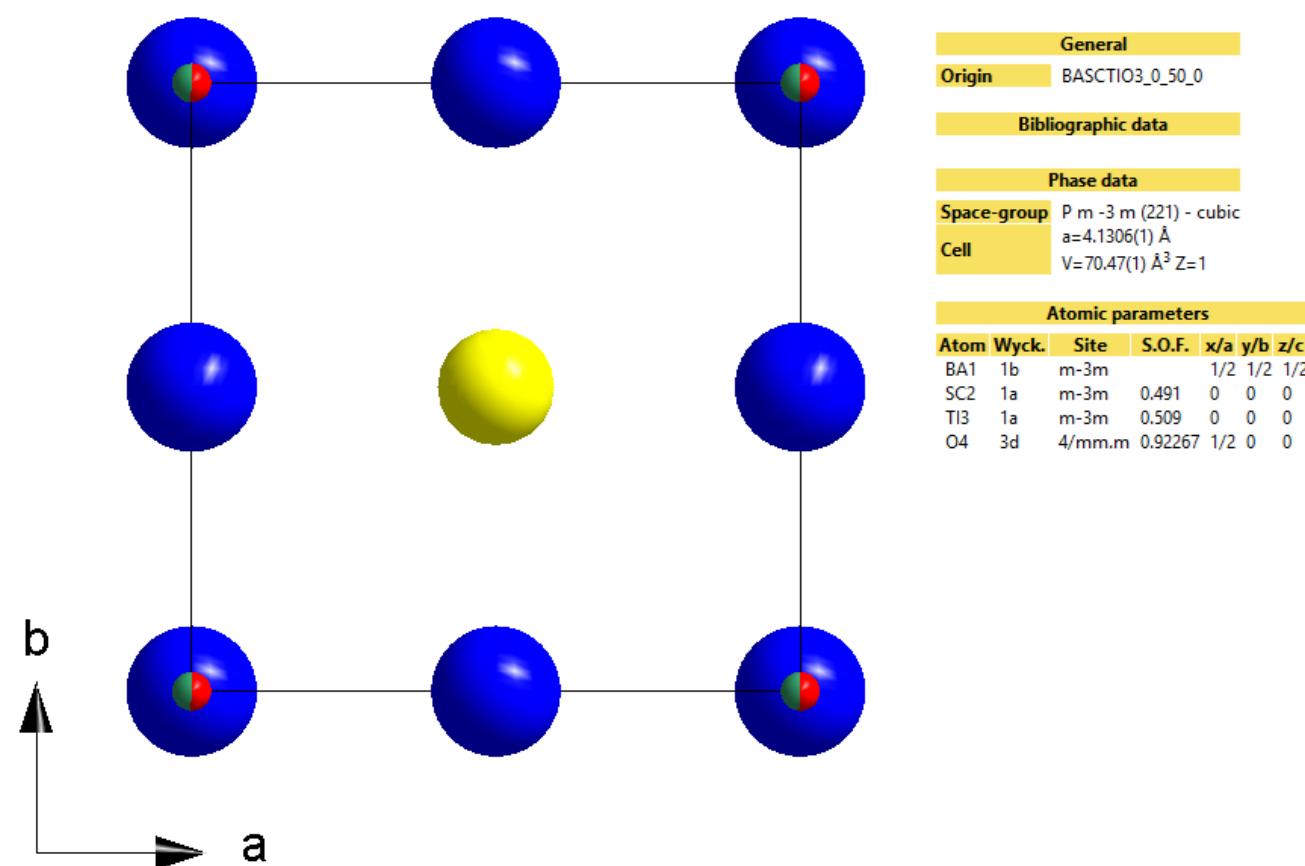
Coordination numbers for $[\text{VO}_n]$ (yellow) and $[\text{ScO}_n]$ (blue) polyhedra in $\text{c-Sc}_2\text{VO}_{5.25}$. Bond lengths are limited to $r_{\text{max}} = 3.0 \text{ \AA}$.

Example 2.

$\text{BaSc}_x\text{Ti}_{1-x}\text{O}_{3-\delta}$ neutron elemental contrast

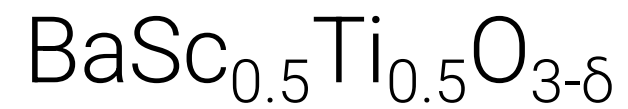


neutron diffraction

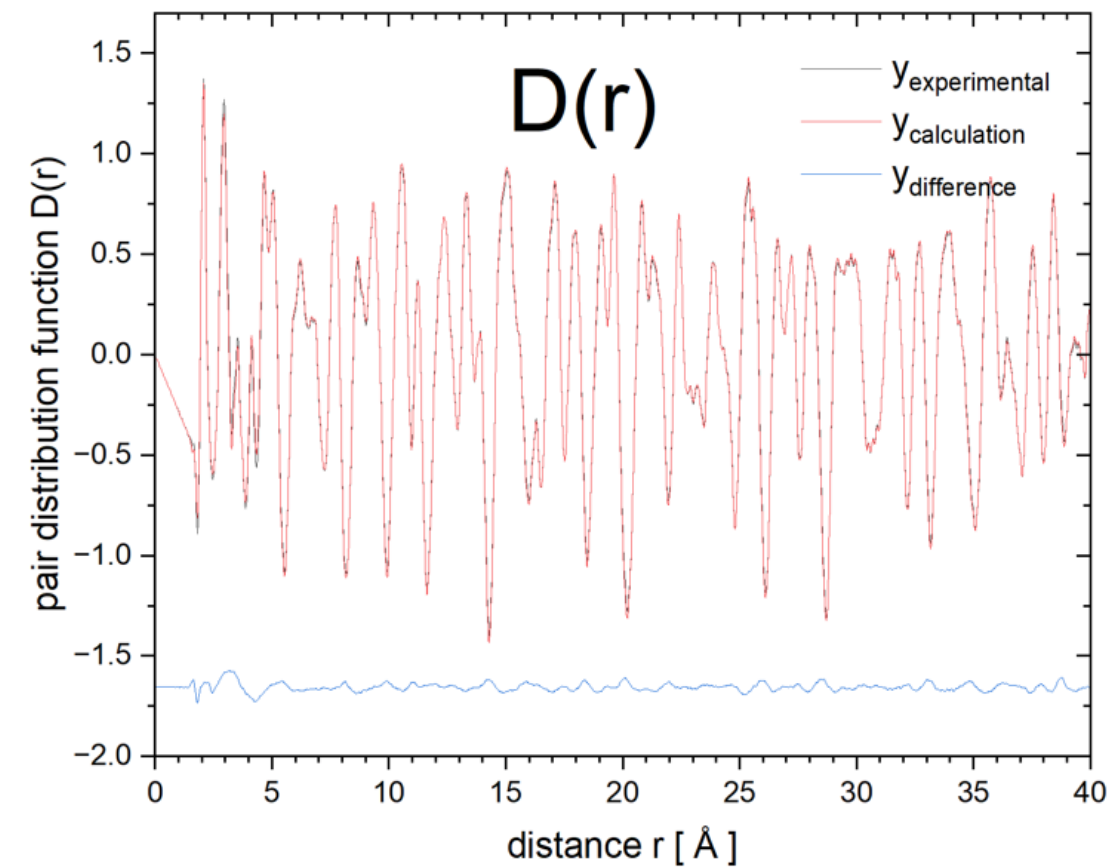
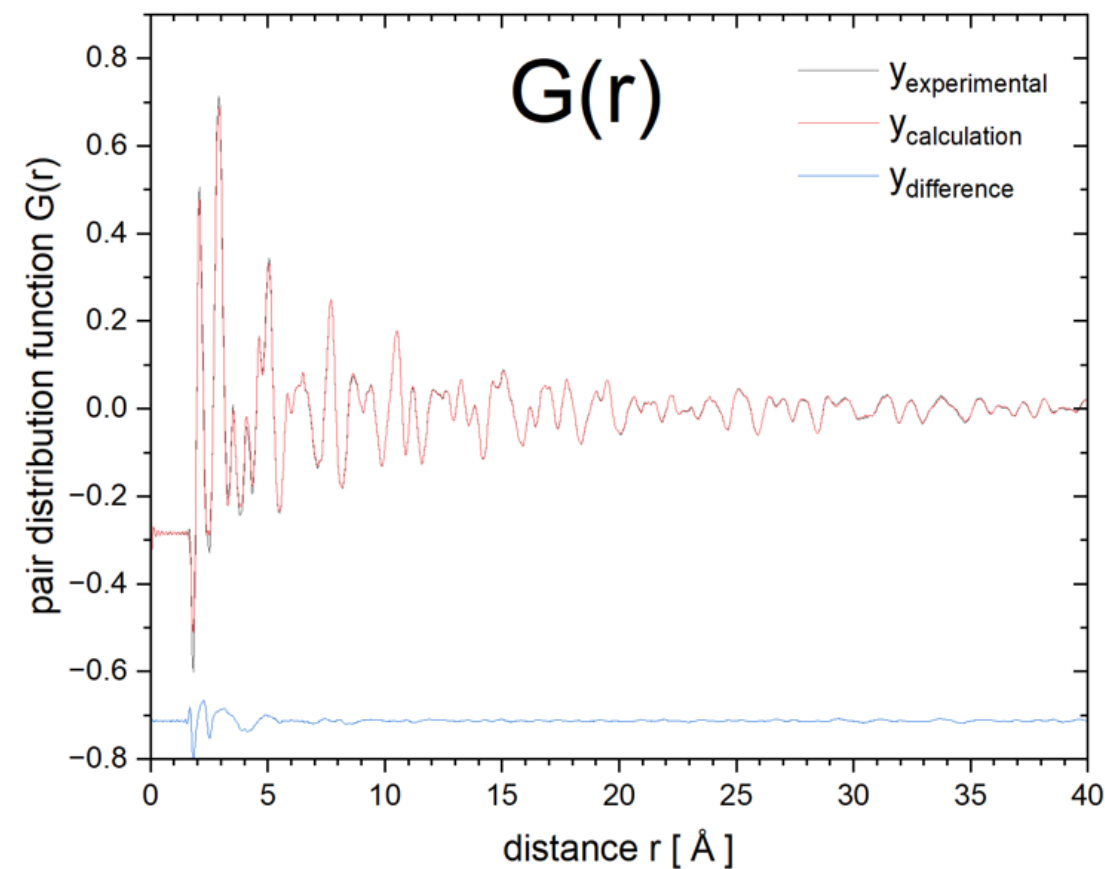


Example 2.

$\text{BaSc}_x\text{Ti}_{1-x}\text{O}_{3-\delta}$ neutron elemental contrast

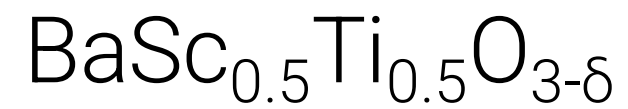


neutron diffraction

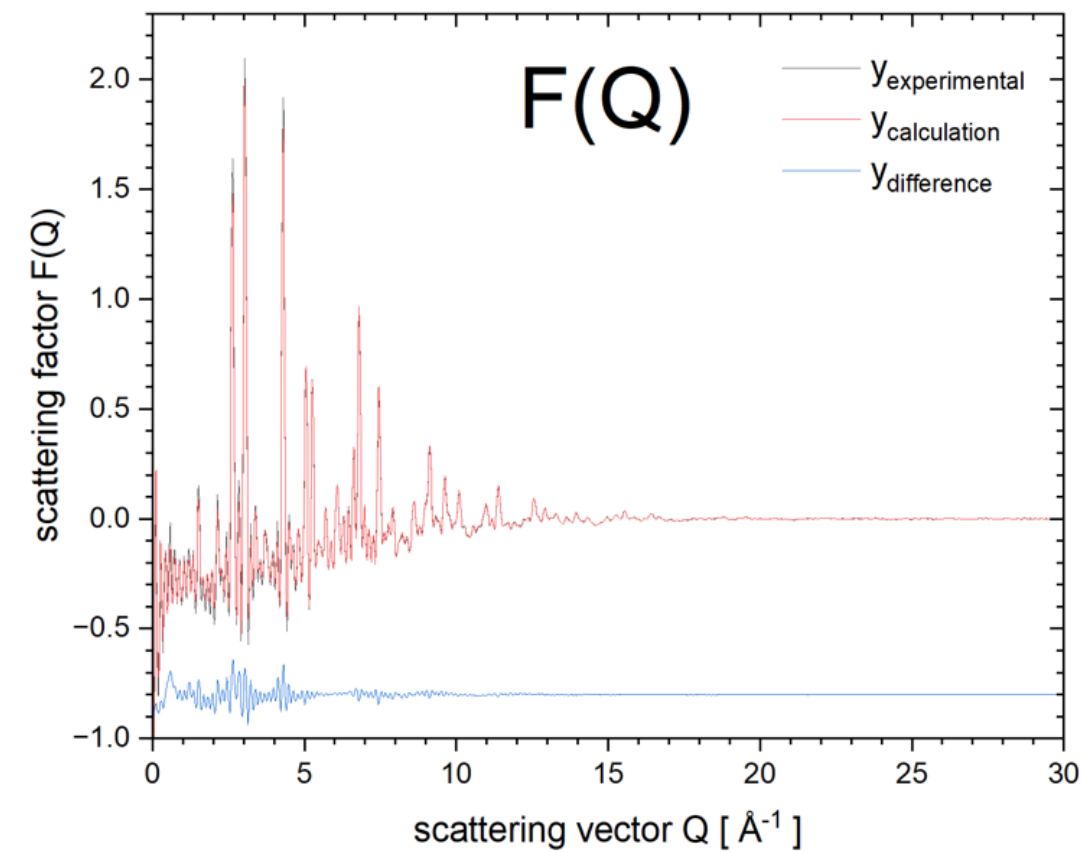
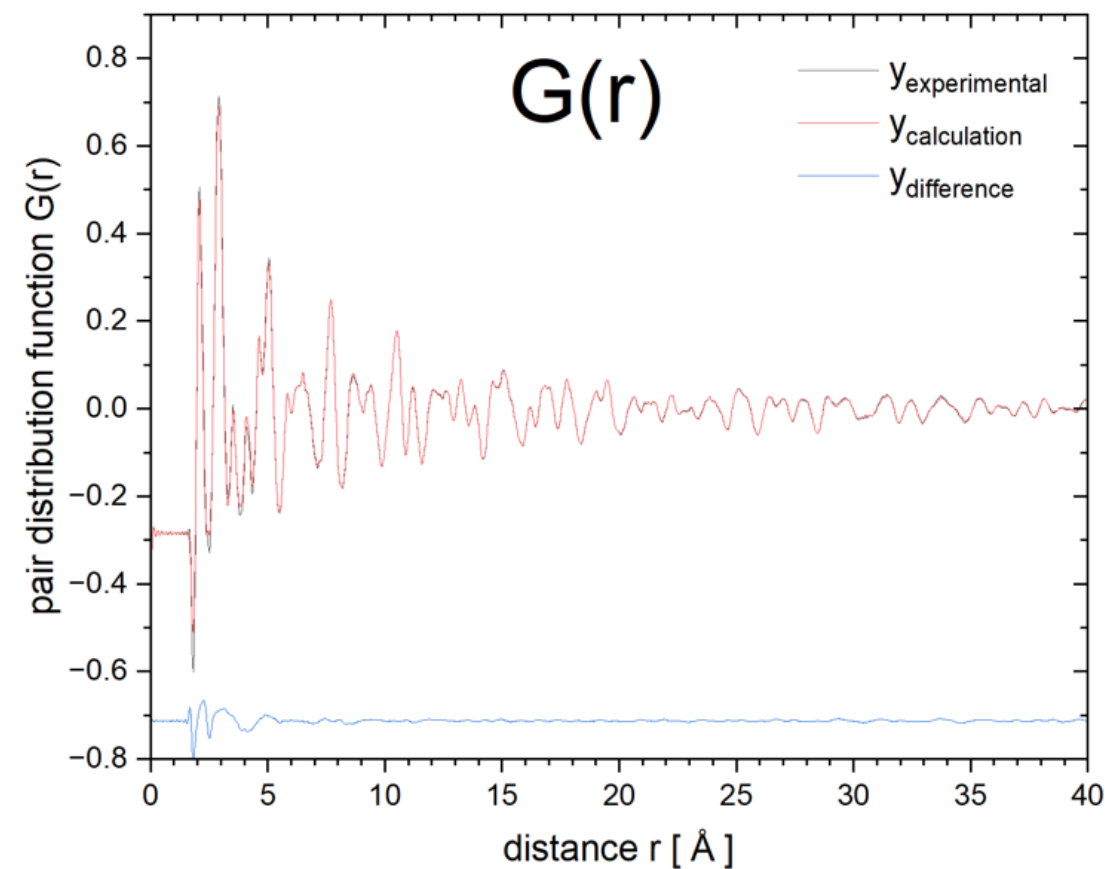


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$\text{BaSc}_x\text{Ti}_{1-x}\text{O}_{3-\delta}$ neutron elemental contrast

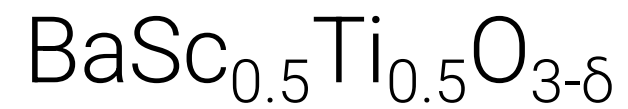


neutron diffraction

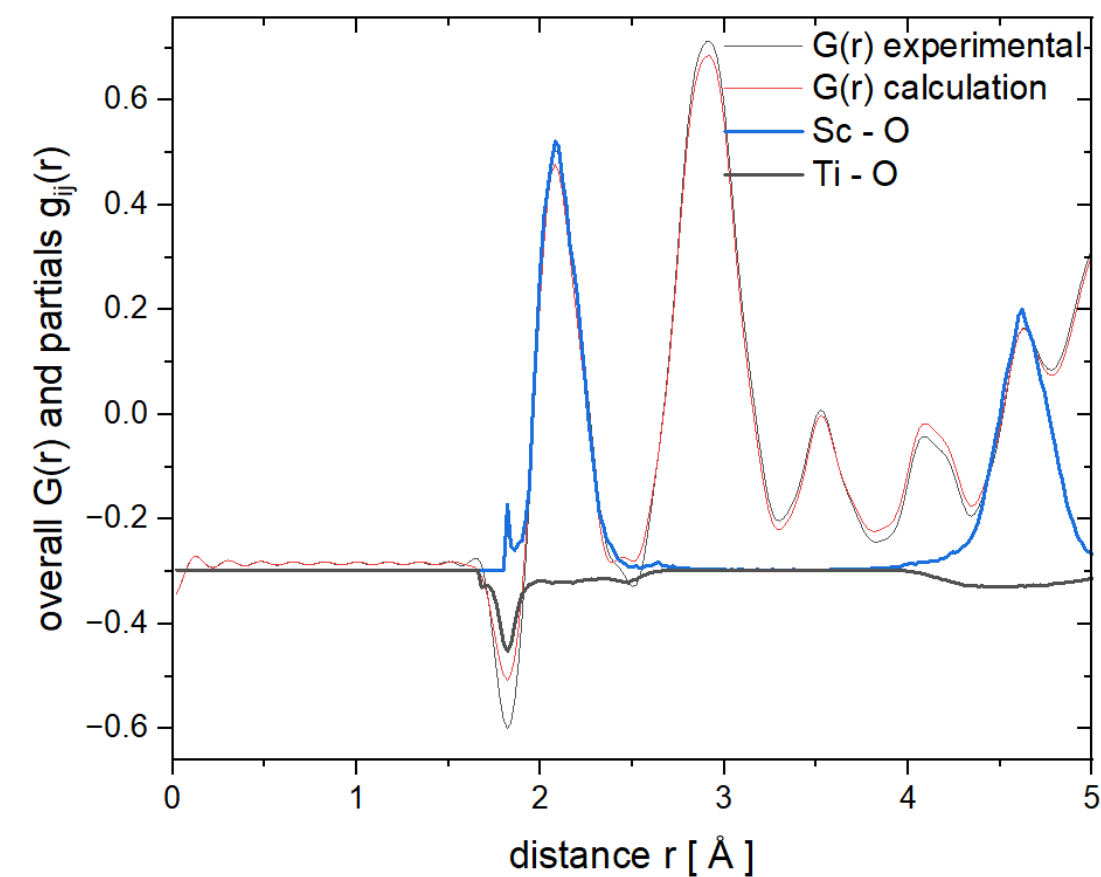
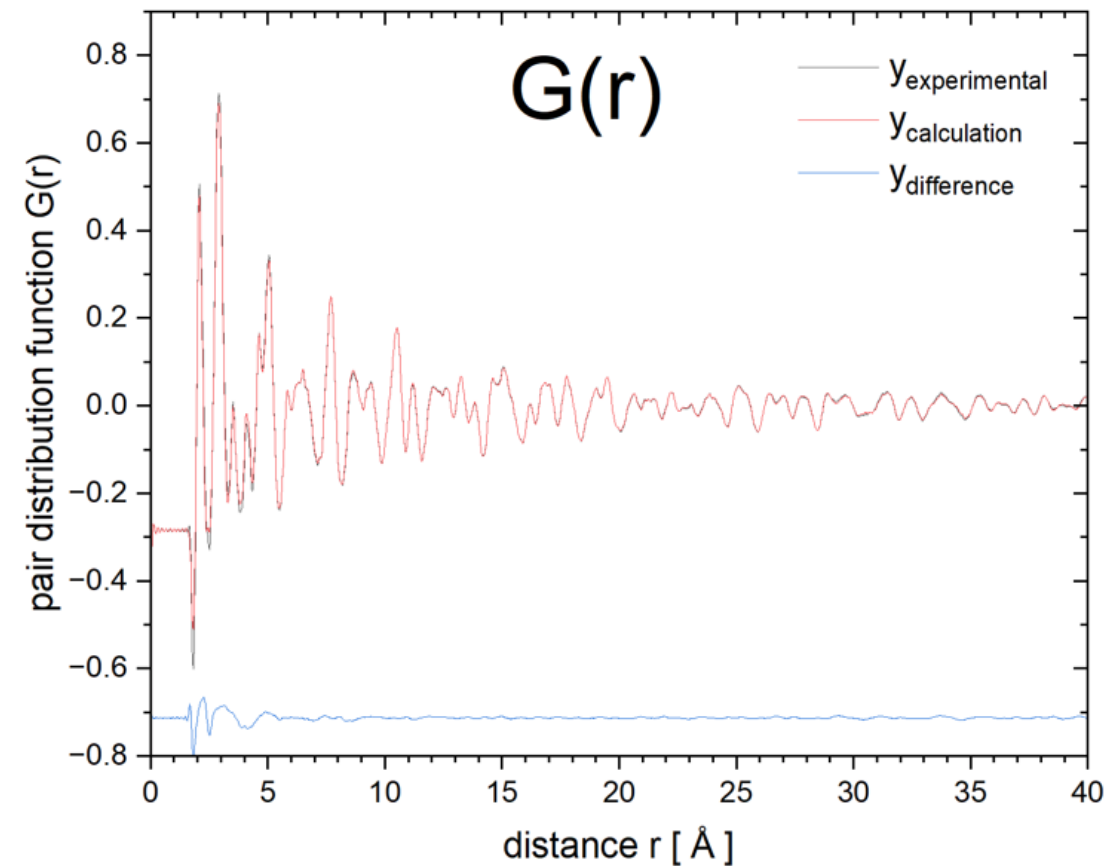


Example 2.

$\text{BaSc}_x\text{Ti}_{1-x}\text{O}_{3-\delta}$ neutron elemental contrast

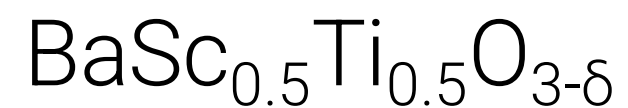


neutron diffraction



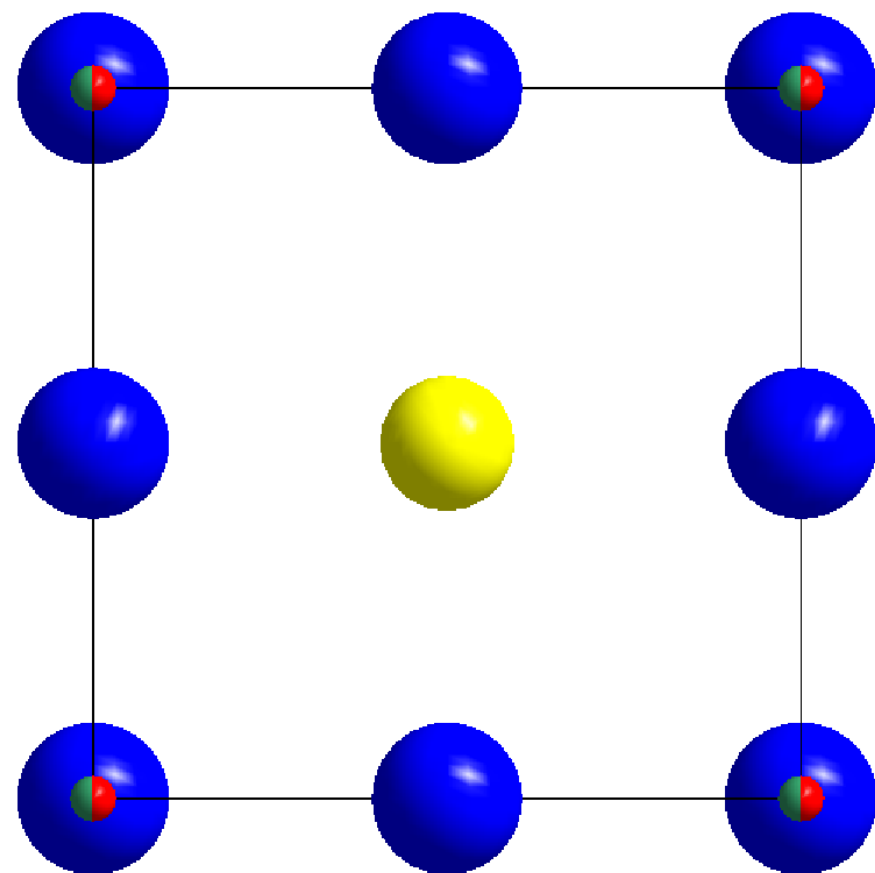
Example 2.

$\text{BaSc}_x\text{Ti}_{1-x}\text{O}_{3-\delta}$ neutron elemental contrast

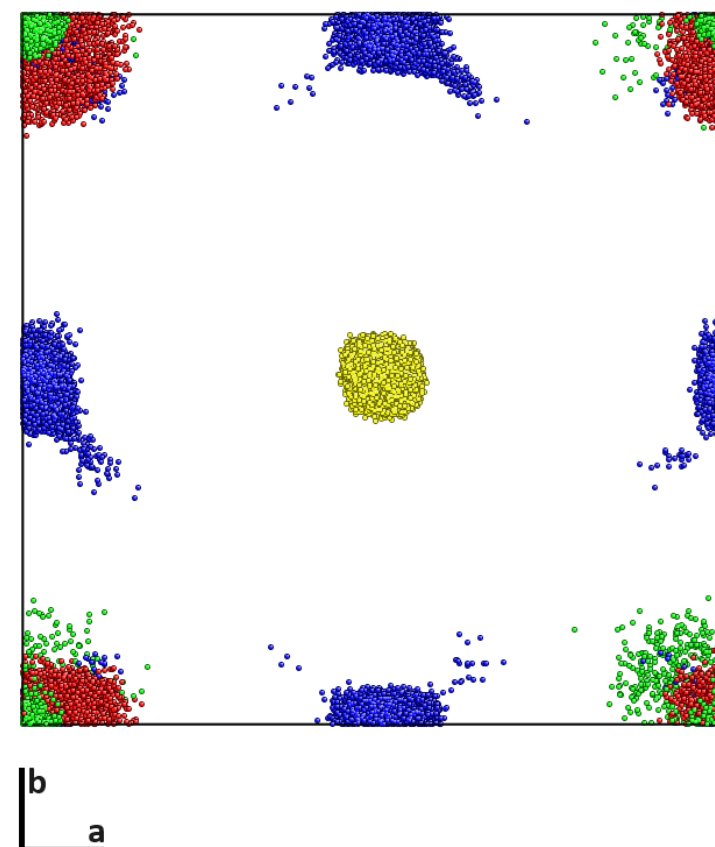


neutron diffraction

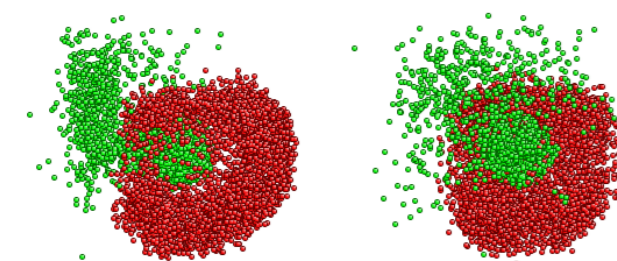
average



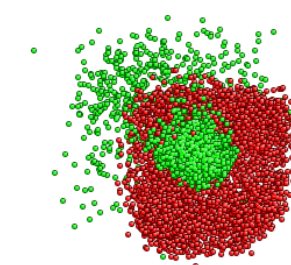
big box local structure modelling



top view

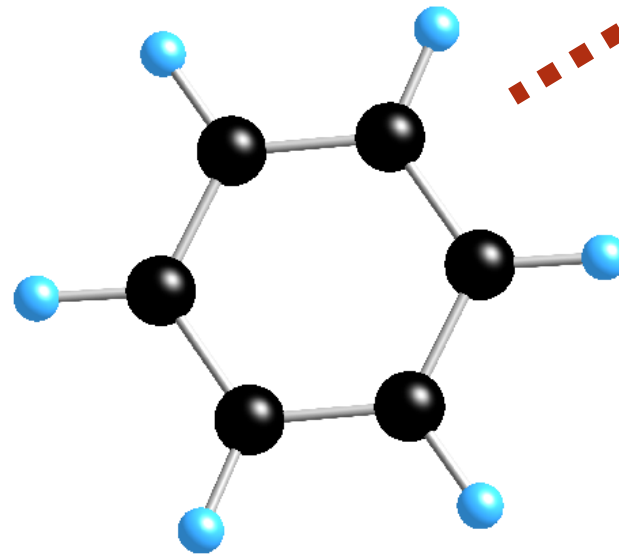
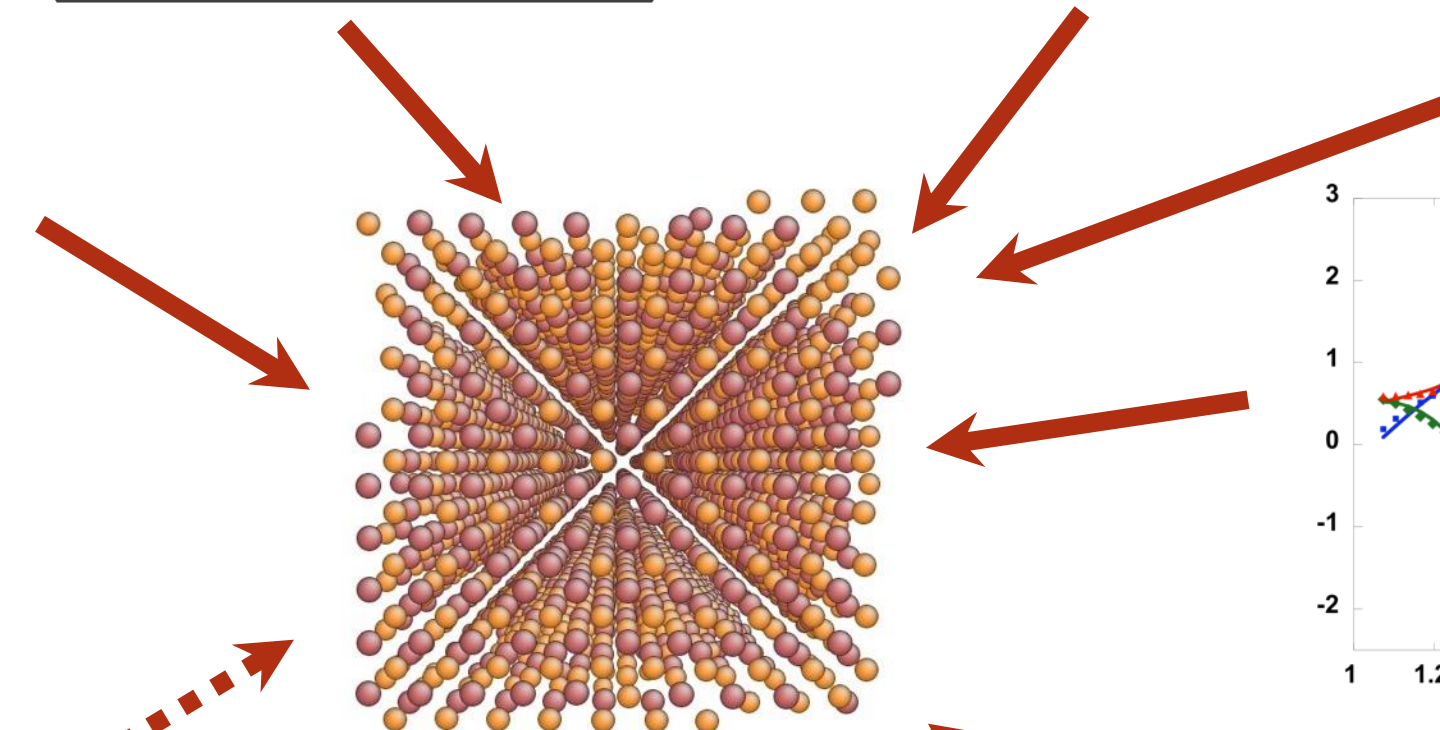
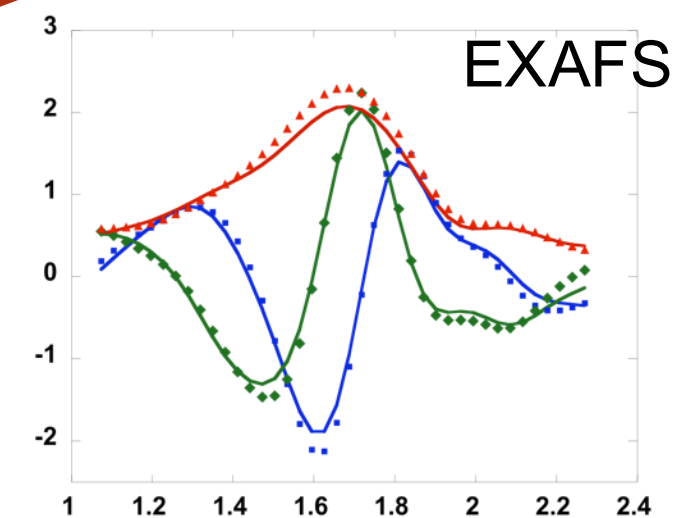
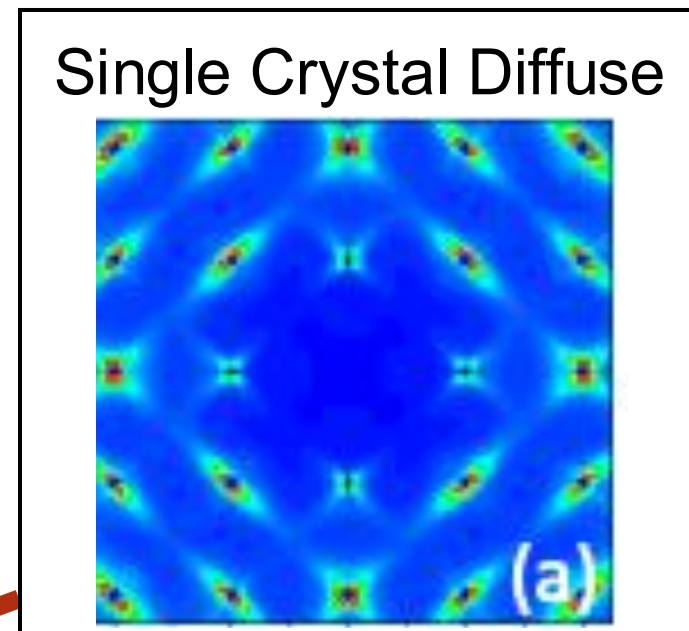
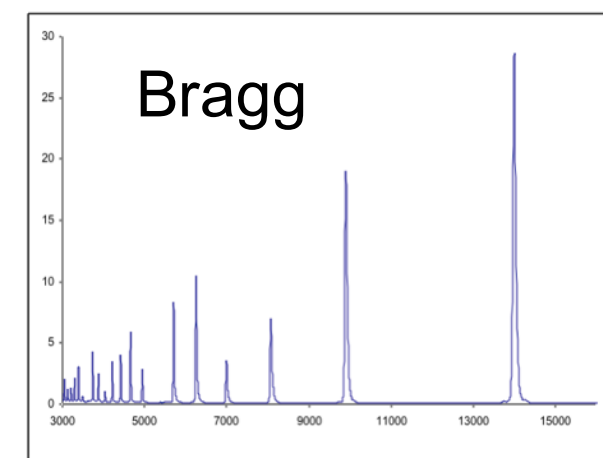
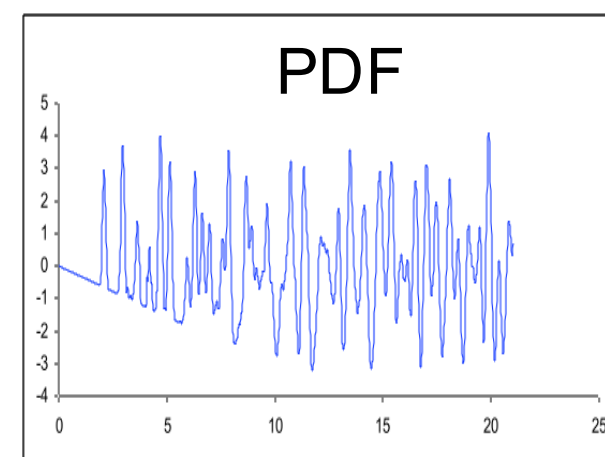
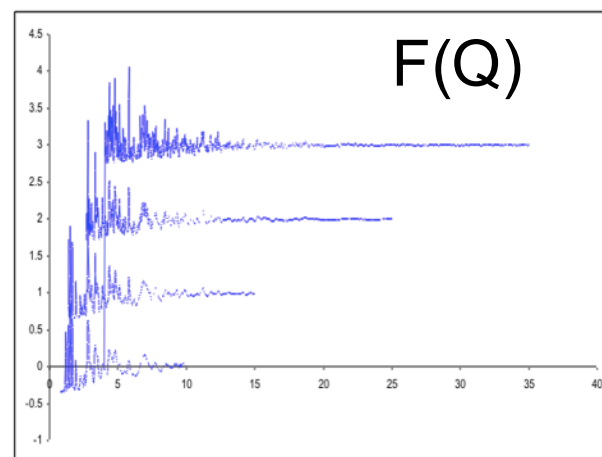


cross section

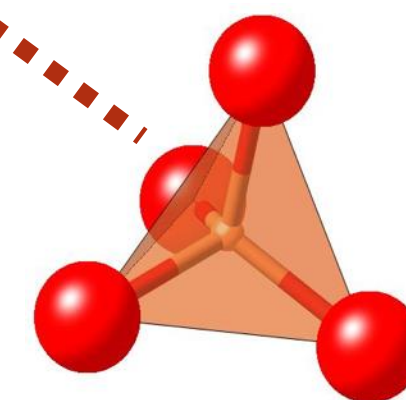
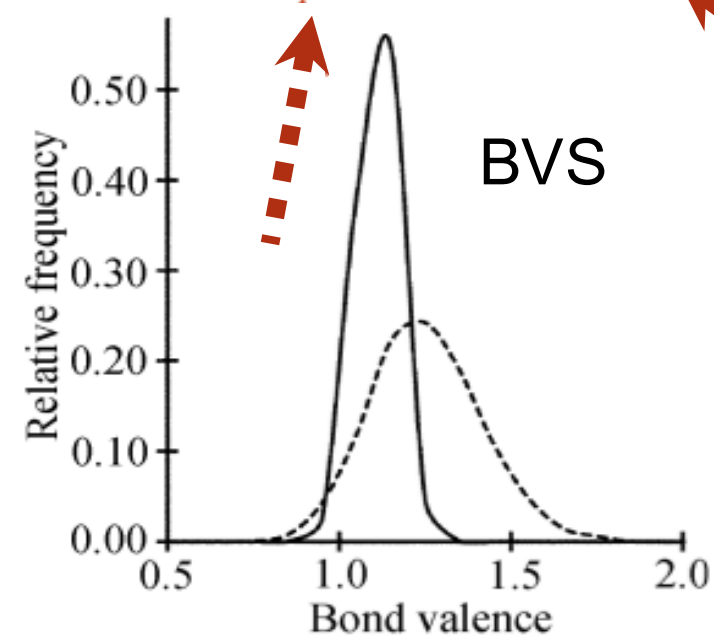


Ba
Sc
Ti
O

RMC PROFILE



Molecular
potentials



Polyhedral
restraints

M G Tucker et al, *J. Phys.: Condens. Matter* **19**, 335218 (2007).

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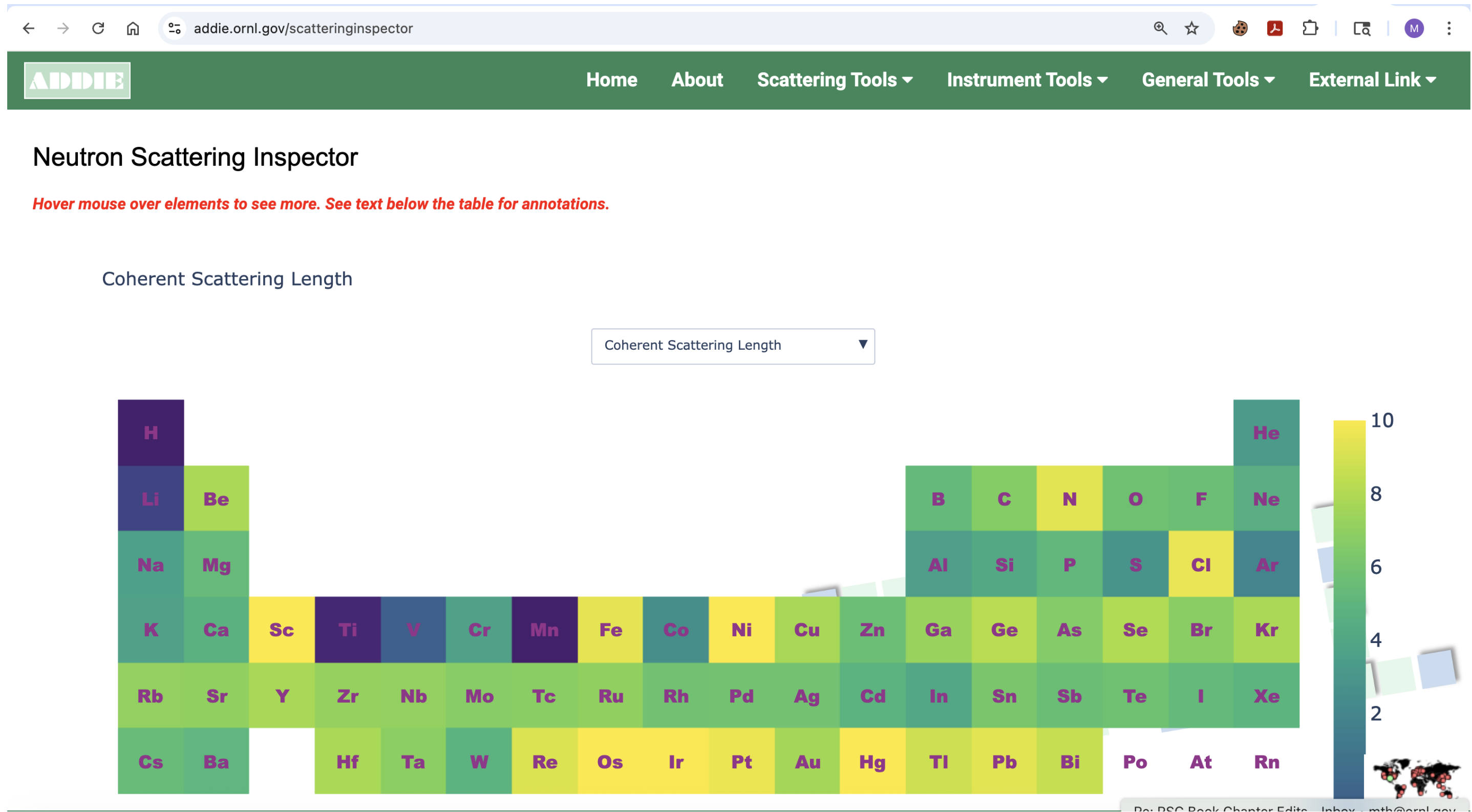
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(ORNL)

Yuanpeng Zhang
(ORNL)

Maksim Eremenko
(ORNL)

rmcprofile.ornl.gov

addie.ornl.gov



Thank you!

- ▶ Please get in touch tuckermg@ornl.gov
- ▶ wslawinski@chem.uw.edu.pl

rmcprofile.ornl.gov
addie.ornl.gov