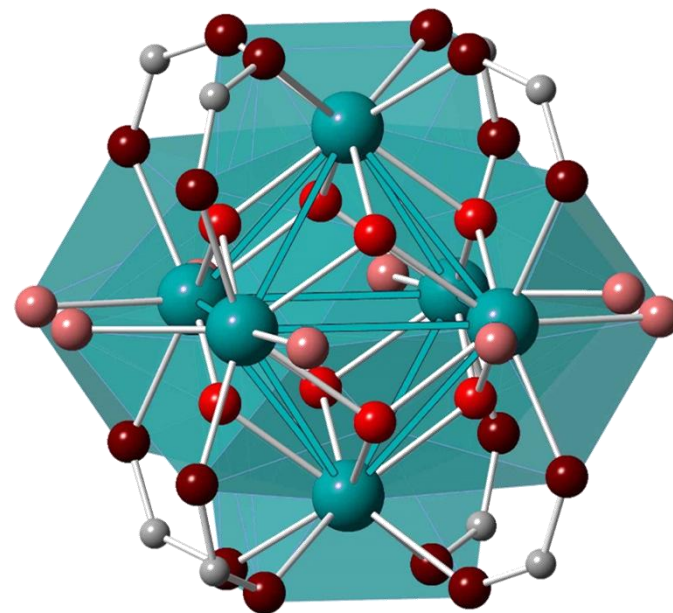


# Exploring materials function and reactivity *in situ*

Karena W. Chapman

*Endowed Chair in Materials Chemistry  
Department of Chemistry, Stony Brook University*

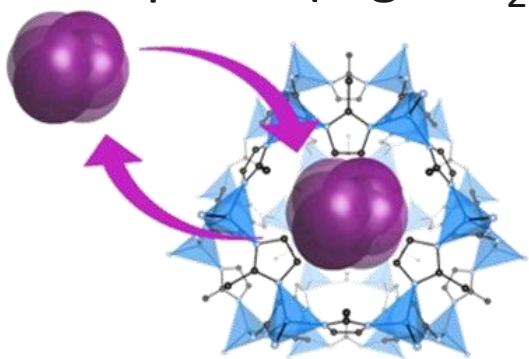


Stony Brook University

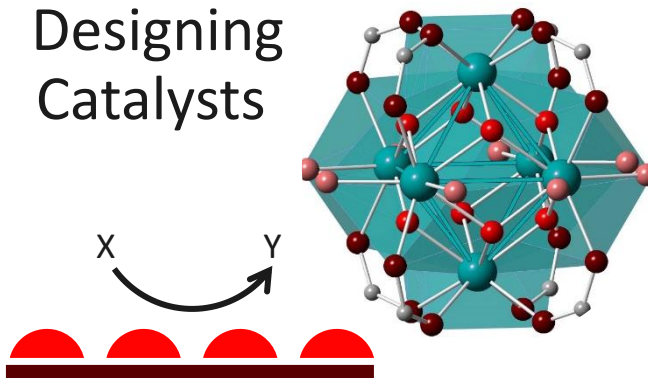
# Clean energy research spanning many applications

*Understanding function & reactivity from atoms to applications*

Gas capture (e.g. CO<sub>2</sub>)



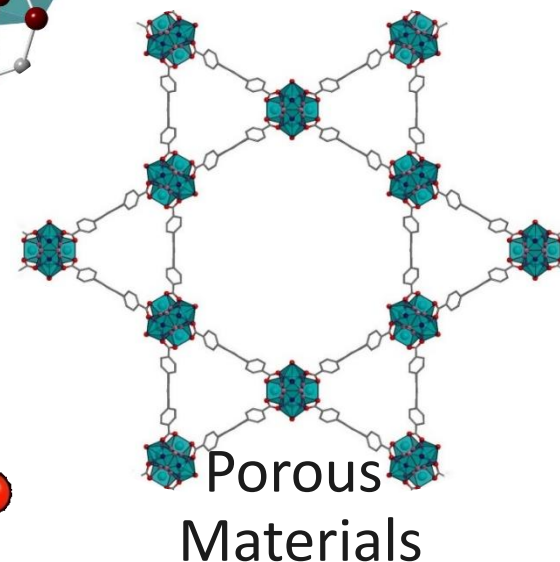
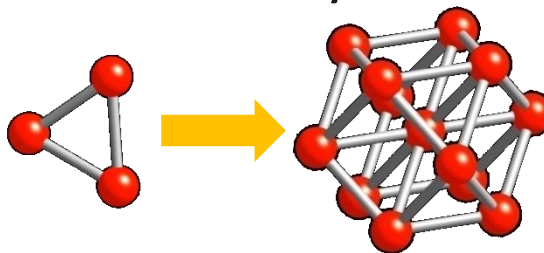
Designing Catalysts



Energy Storage

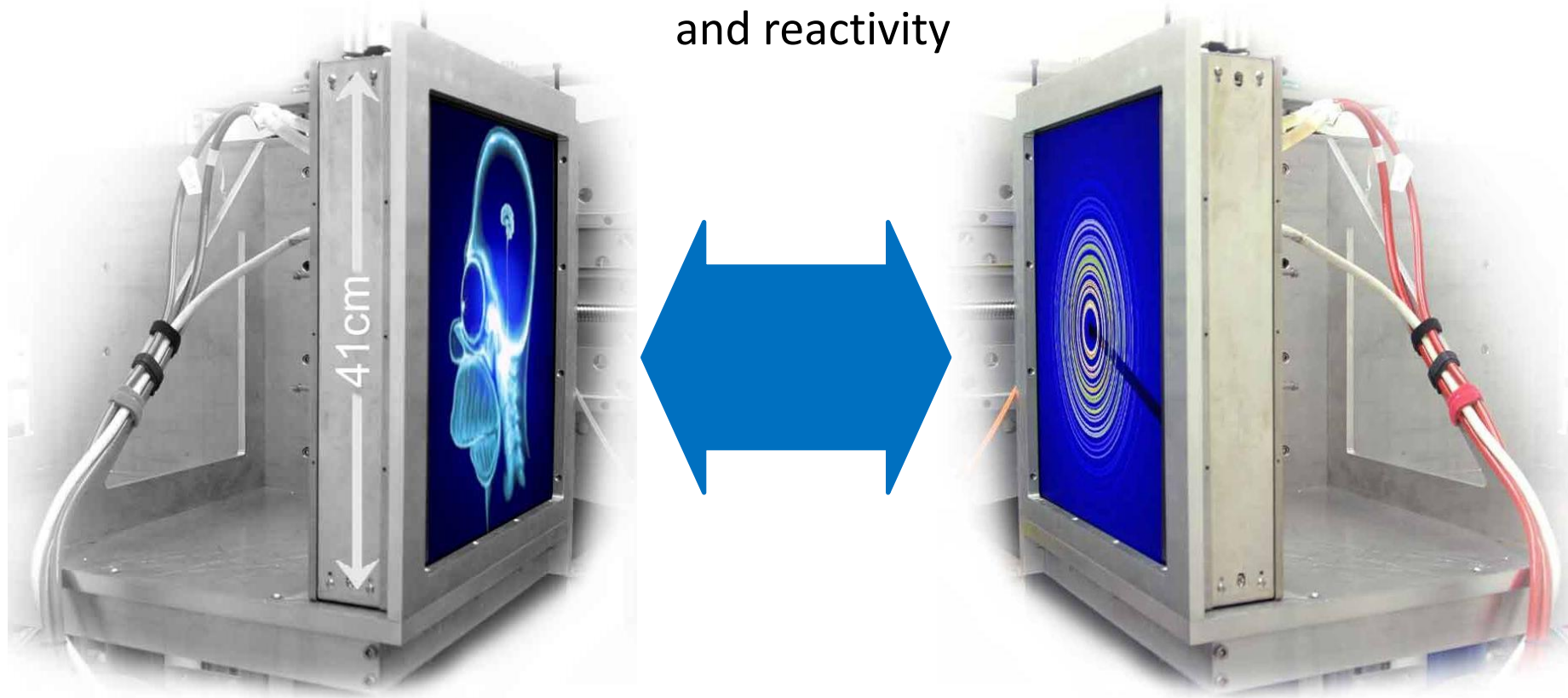


Science of Synthesis



# Illuminating structure, function & reactivity

X-rays can penetrate working systems without damage, to allow us to probe their function and reactivity

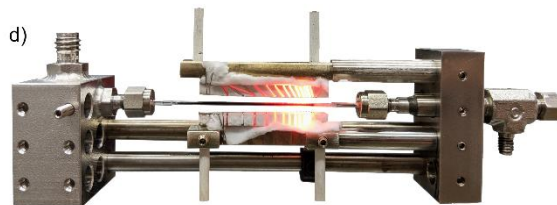
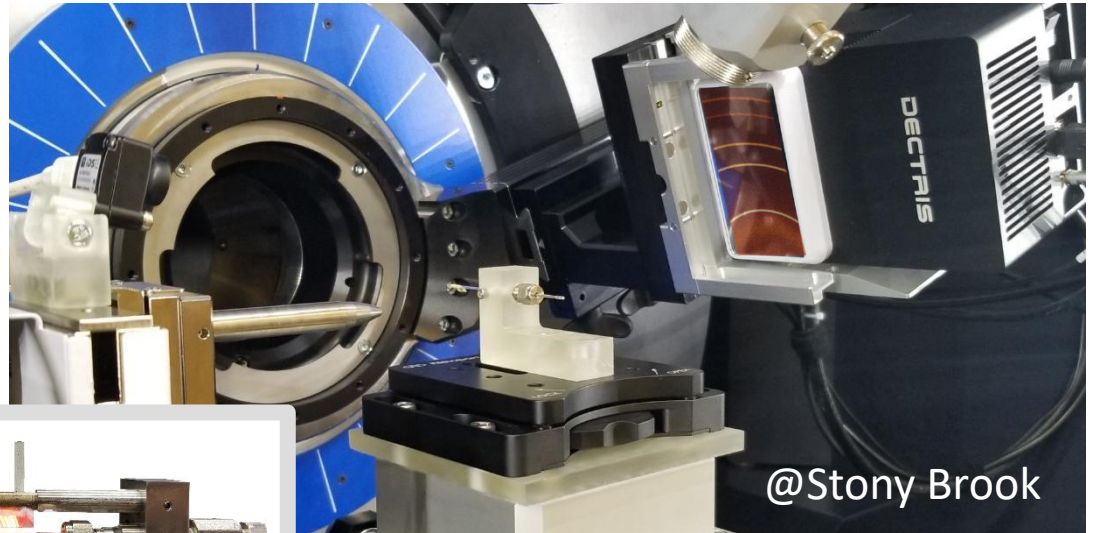


# Exploring materials function with advanced tools

We probe the atomic structure of functional materials

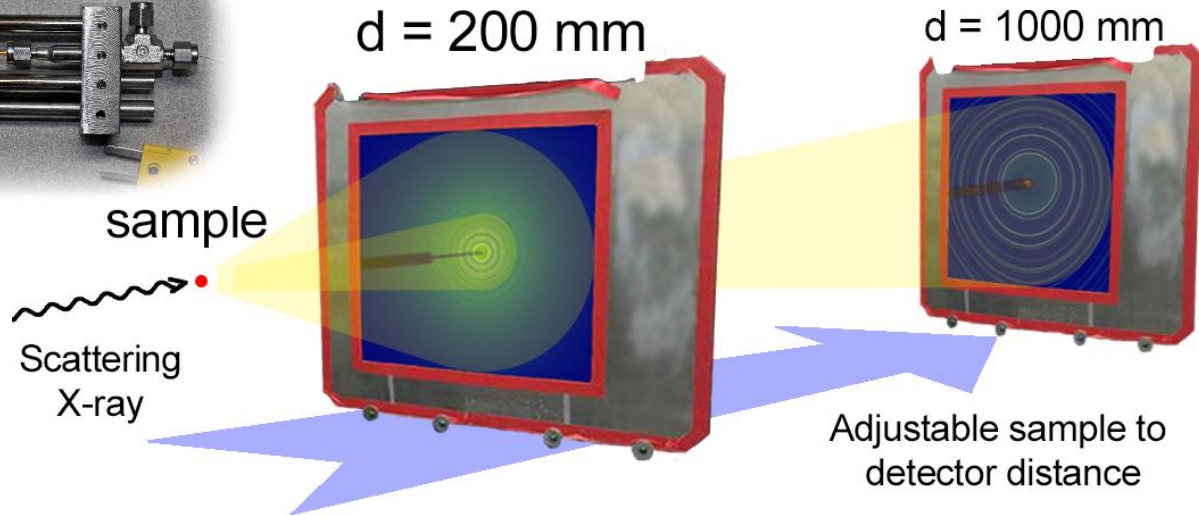
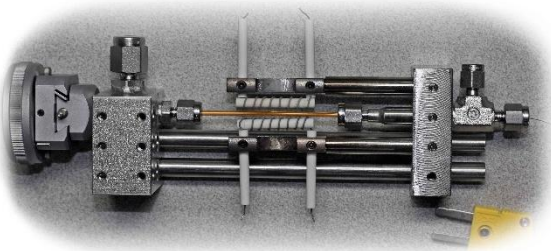
During operation, reaction or changing environmental variable

To develop a predictive understanding of functional properties



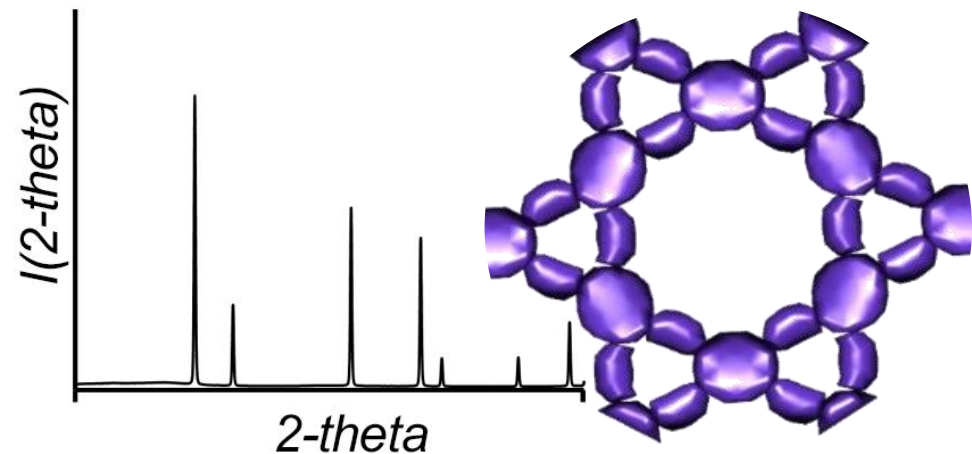
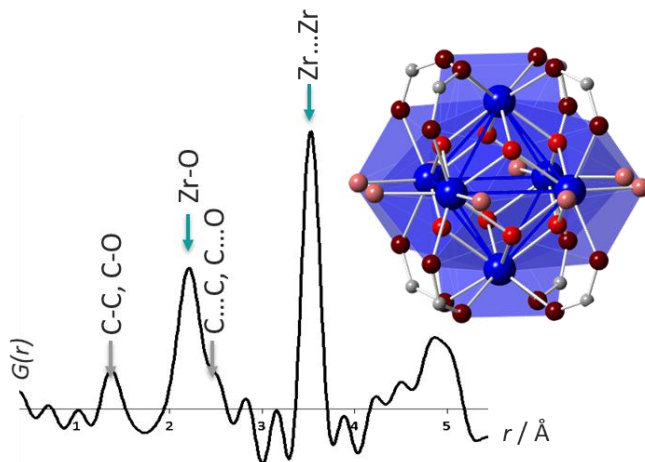


# Multimodal correlation of local & long range structure

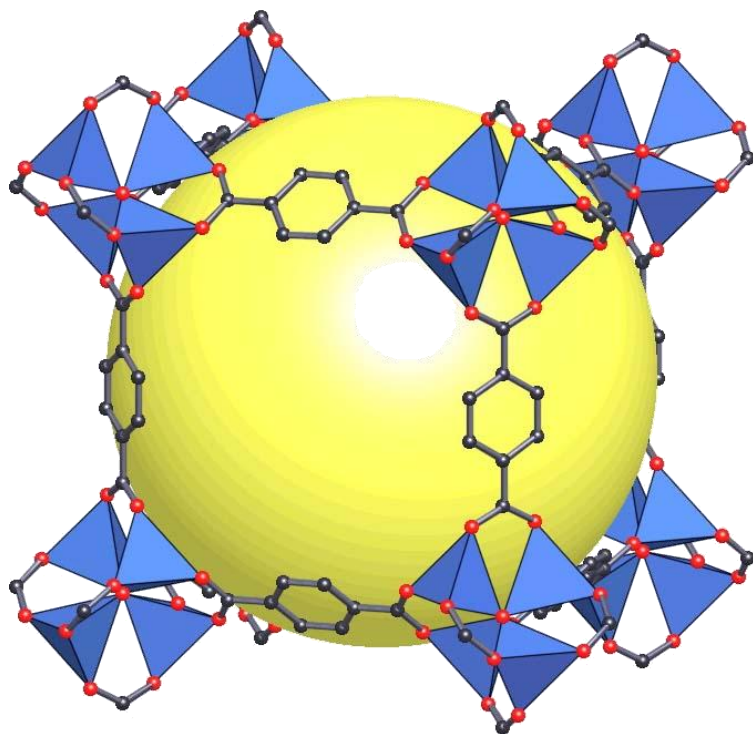


High-resolution  
atom position for PDF

High-resolution  
peak profile for XRD



# MOFs as a versatile platform for exploring catalysis



The crystallinity of the MOF lattice favors chemically uniform active sites.

This facilitates characterization and understanding of the catalyst structure-activity relationship

The MOF structure and chemistry can be tuned, and the surface functionalized to systematically explore catalysis



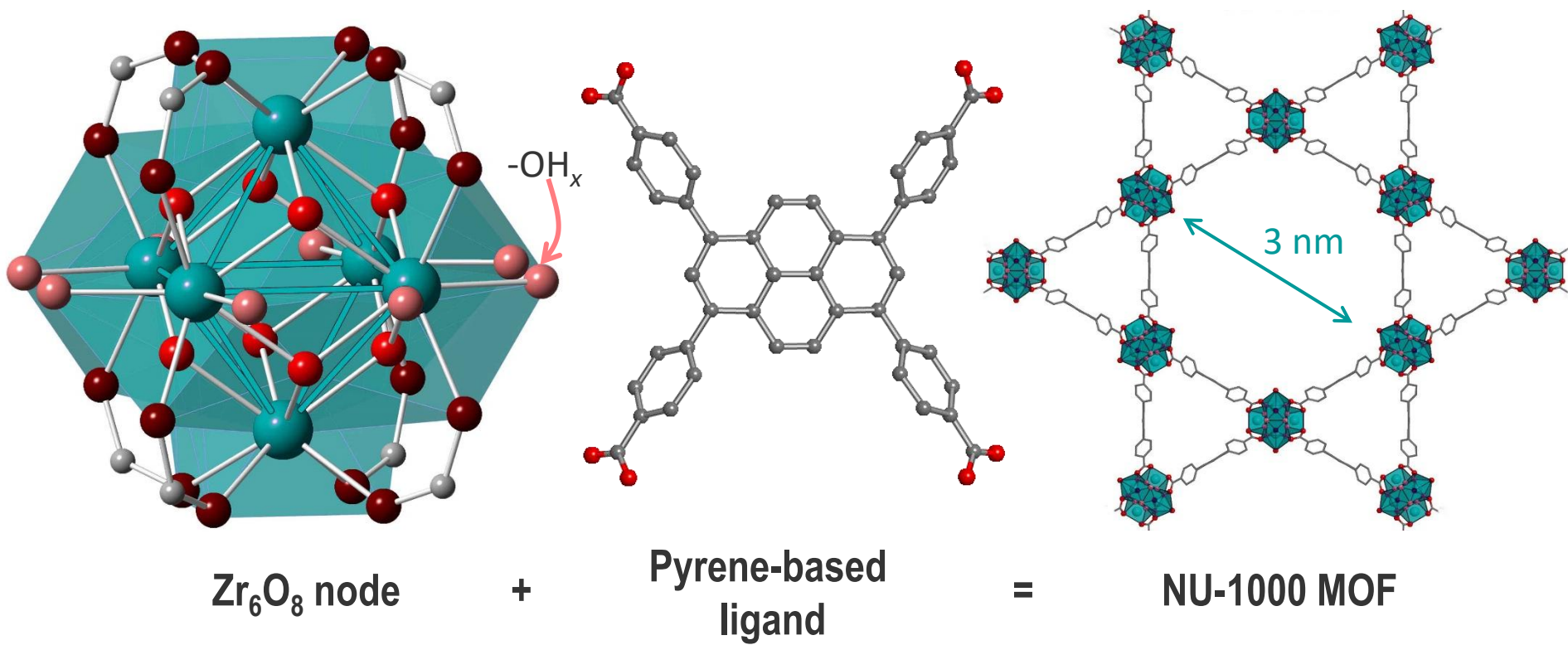
INORGANOMETALLIC  
CATALYST DESIGN CENTER



# NU-1000 MOF: A highly porous framework

A Northwestern University MOF

The metal-organic framework NU1000 is of interest as a catalyst and catalyst support

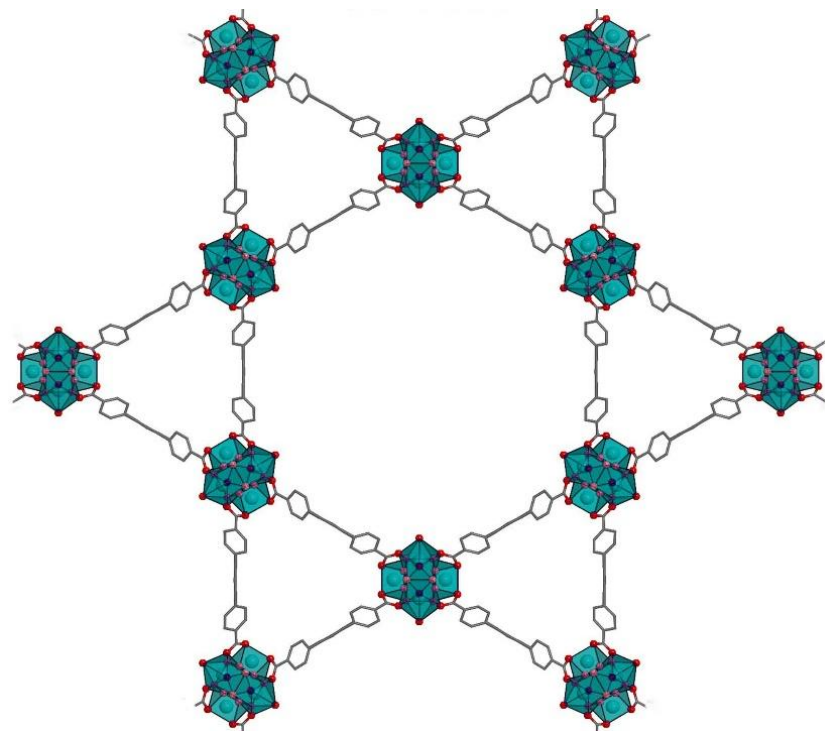
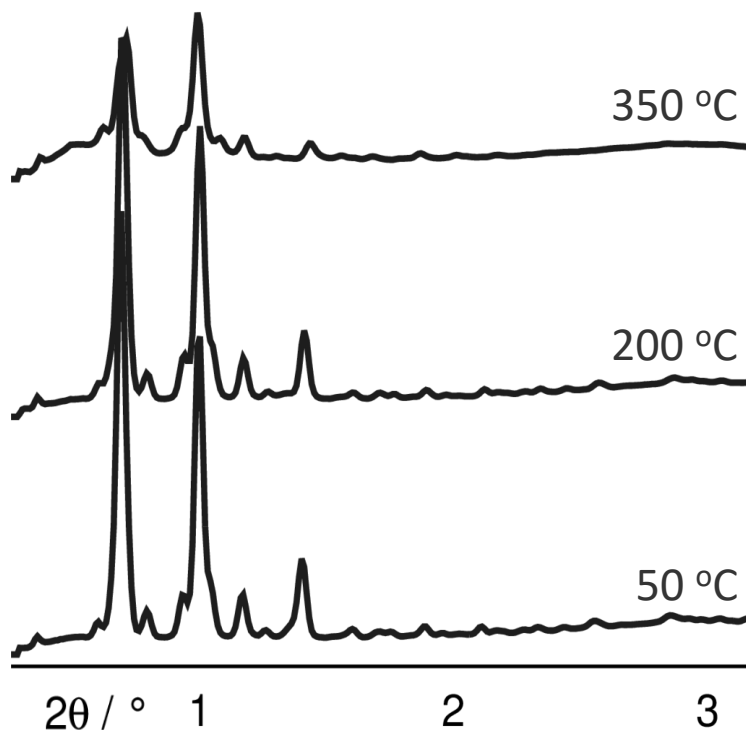


Farha, Hupp *et al.* *J. Am. Chem. Soc.*, **2013**, 135, 10294–10297

# Subtle changes to powder diffraction

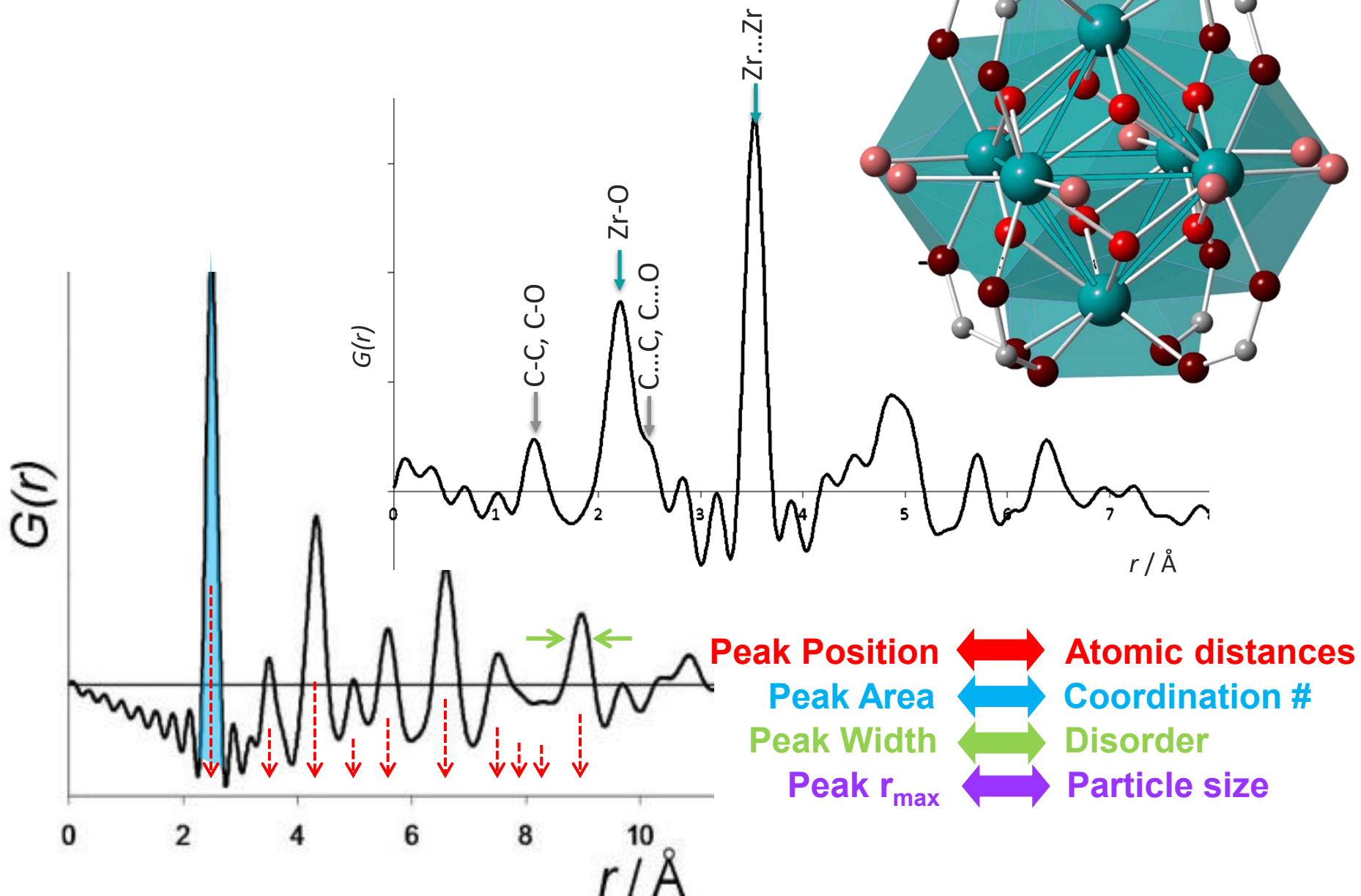
The strength of  $\text{Zr}^{\text{IV}}$ -O bonds within zirconia nodes imparts high stability to NU-1000, allowing it to tolerate reaction conditions.

There are no obvious changes to crystal symmetry/structure.





# A local structure picture from PDF

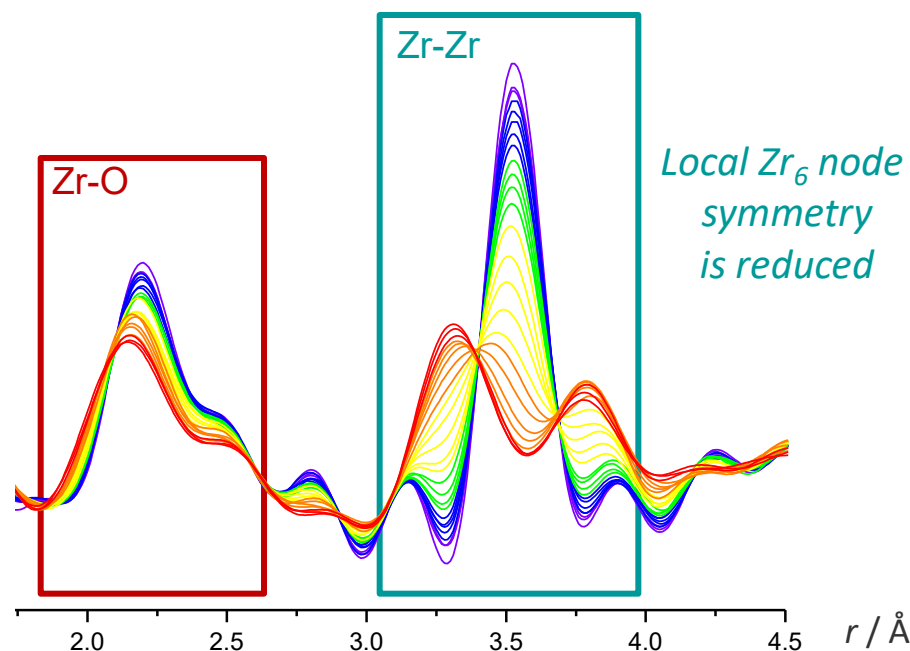
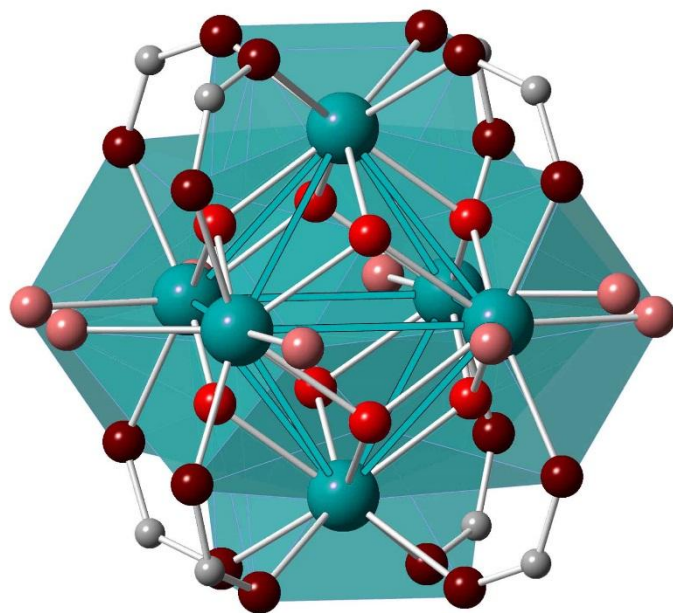


# PDF identifies a large irreversible transition



Variable temperature PDF studies reveal an unexpected and irreversible change in the local atomic structure – reducing the symmetry of the  $\text{Zr}_6$ -based nodes.

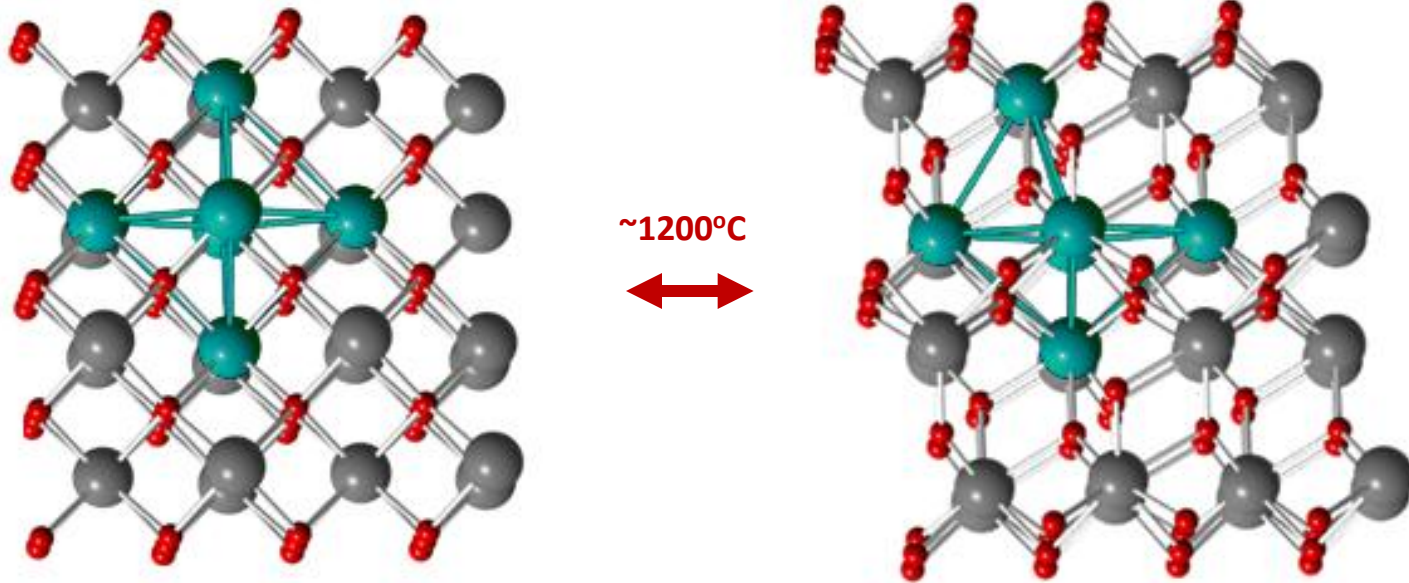
$\text{Zr}_6\text{O}_8$  nodes, while chemically robust are not static



Experimental PDF data collected at 11-ID-B  
at the Department of Energy's Advanced Photon Source  
heating from 50 °C to 350 °C

# Consider the phase transitions of zirconia

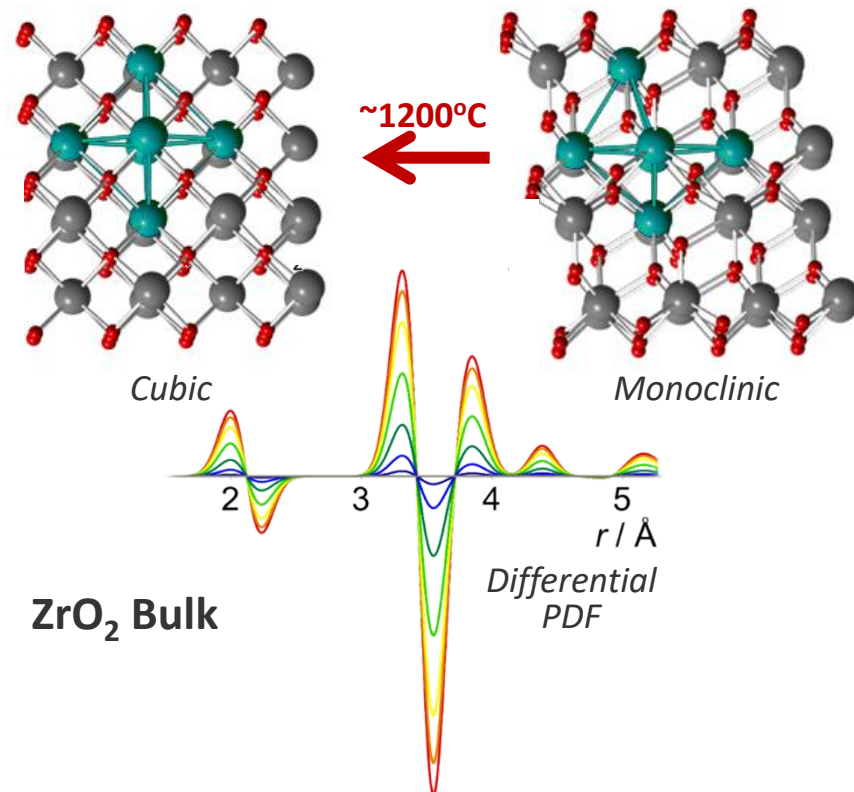
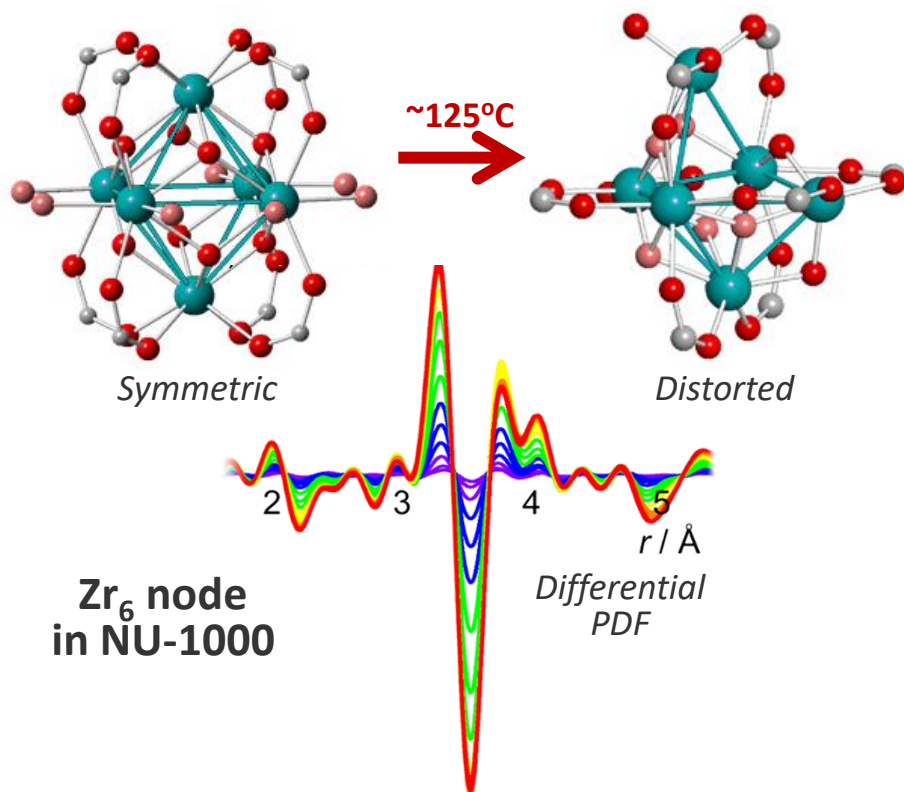
Transitions between cubic, tetragonal and monoclinic forms of bulk zirconia occur at high temperature



# MOF nodes are the ultimate nanoparticles

Local structure changes within node  
matches those in **bulk zirconia**,  
but occur at much **lower temperatures**  
and in the **reverse direction**.

Distorted node is more catalytically active

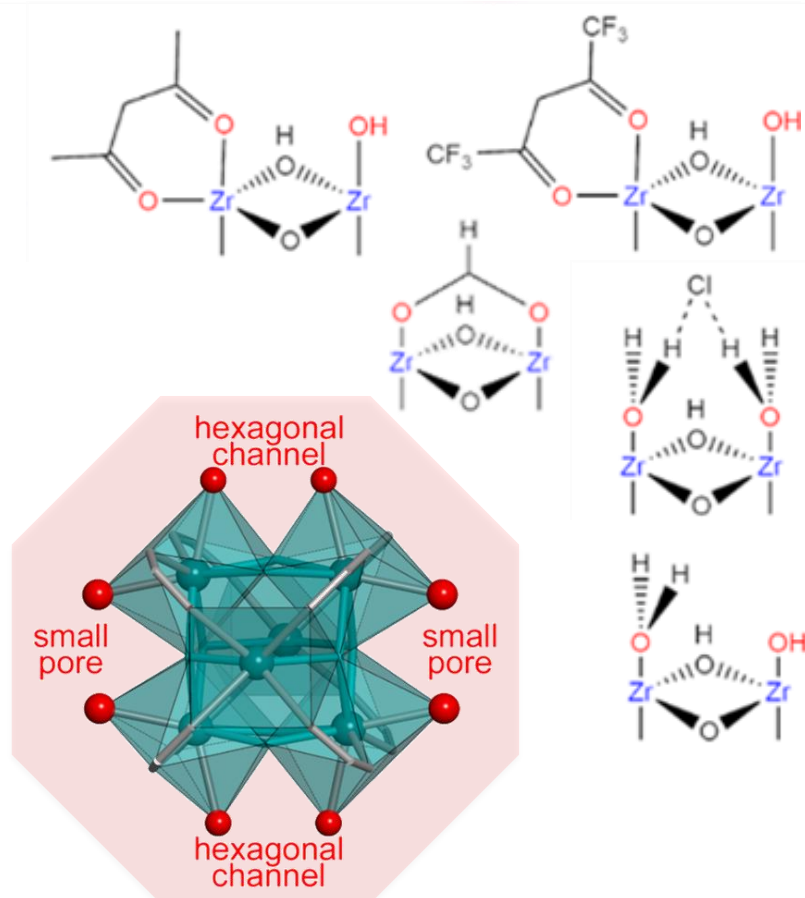
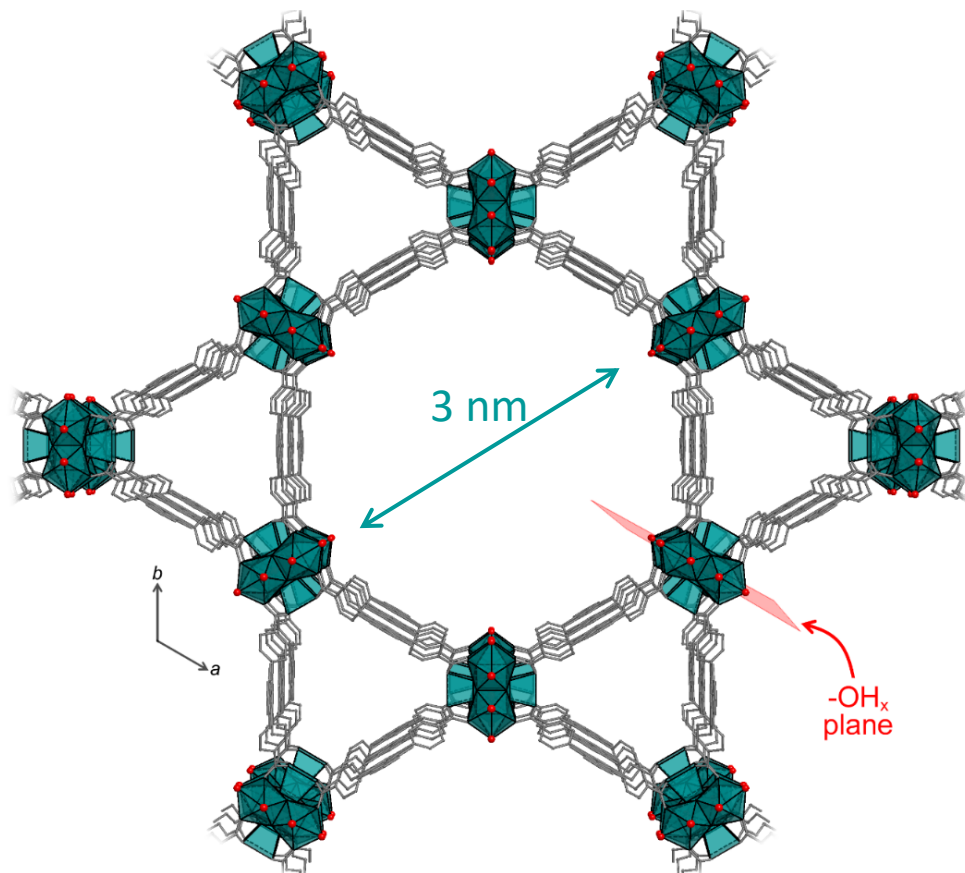




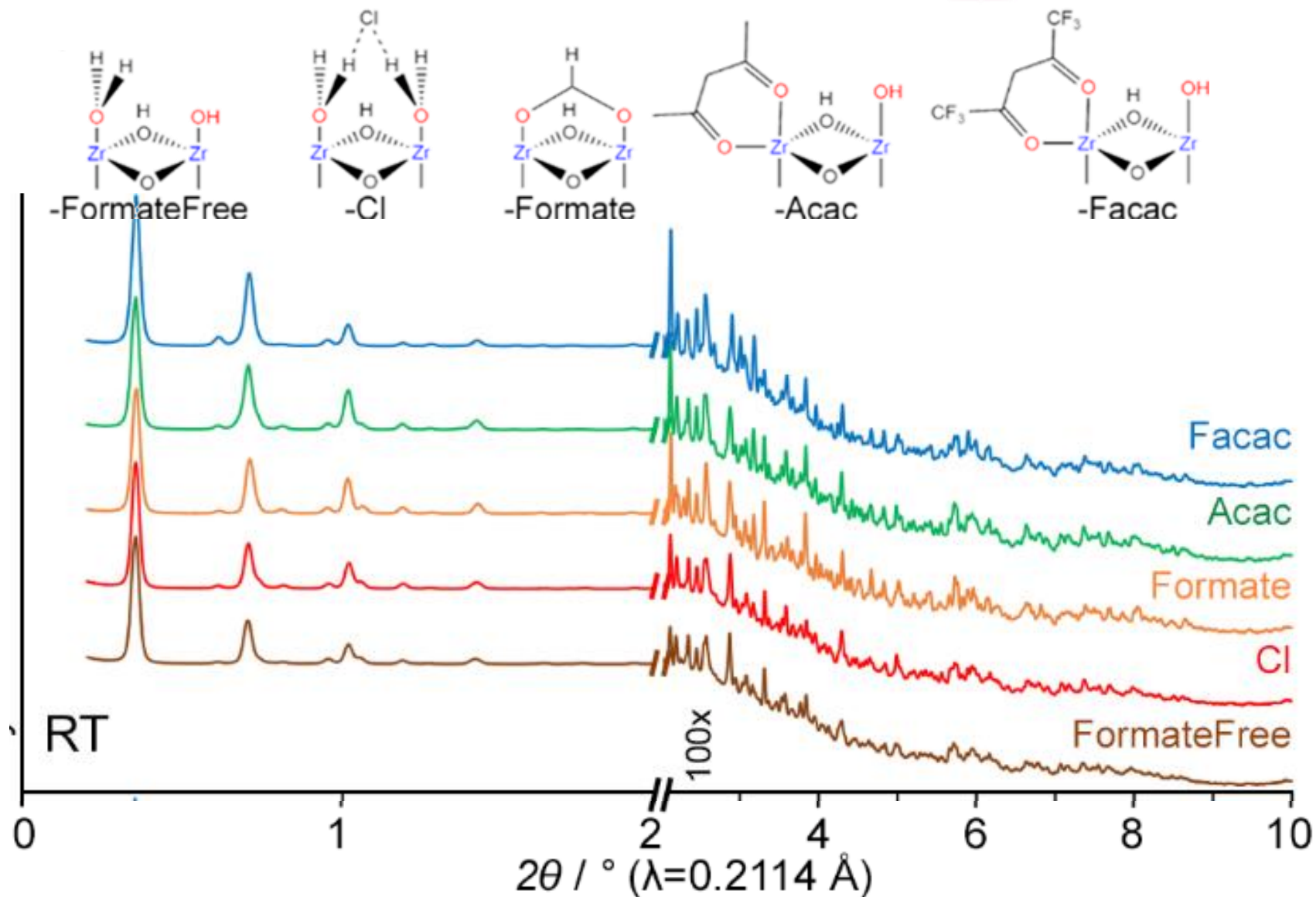
# Chemical functionalization of the node



Joe Hupp



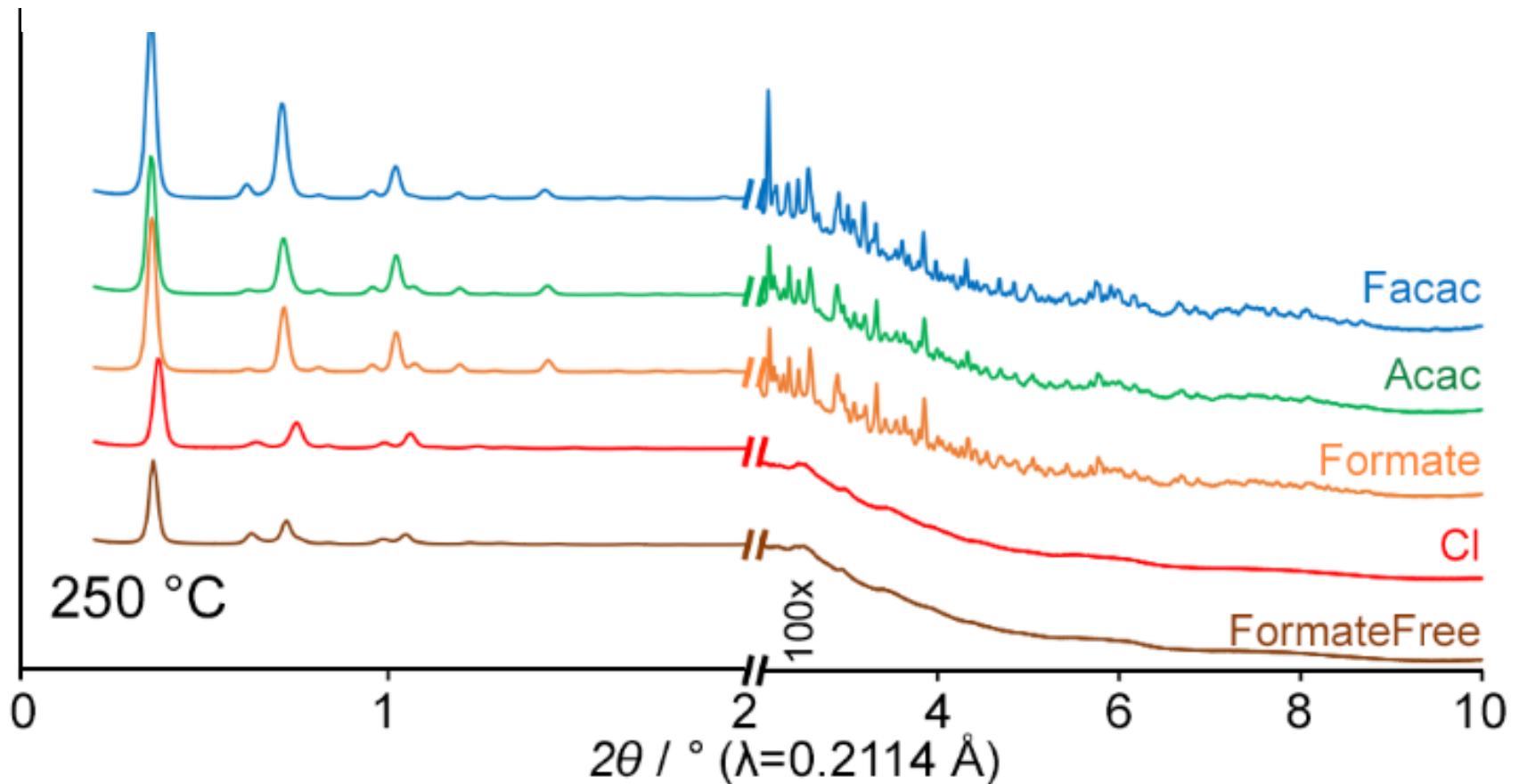
# Decorated NU-1000s have same structure



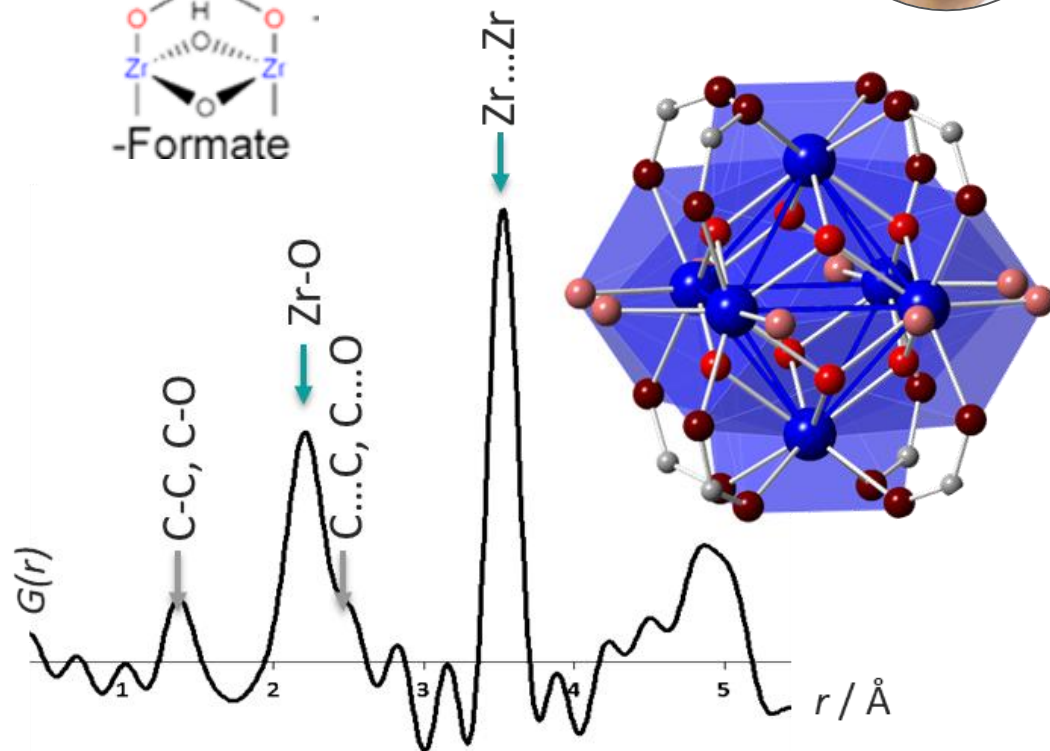
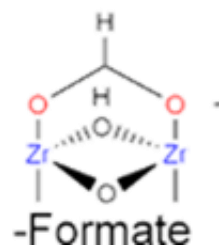
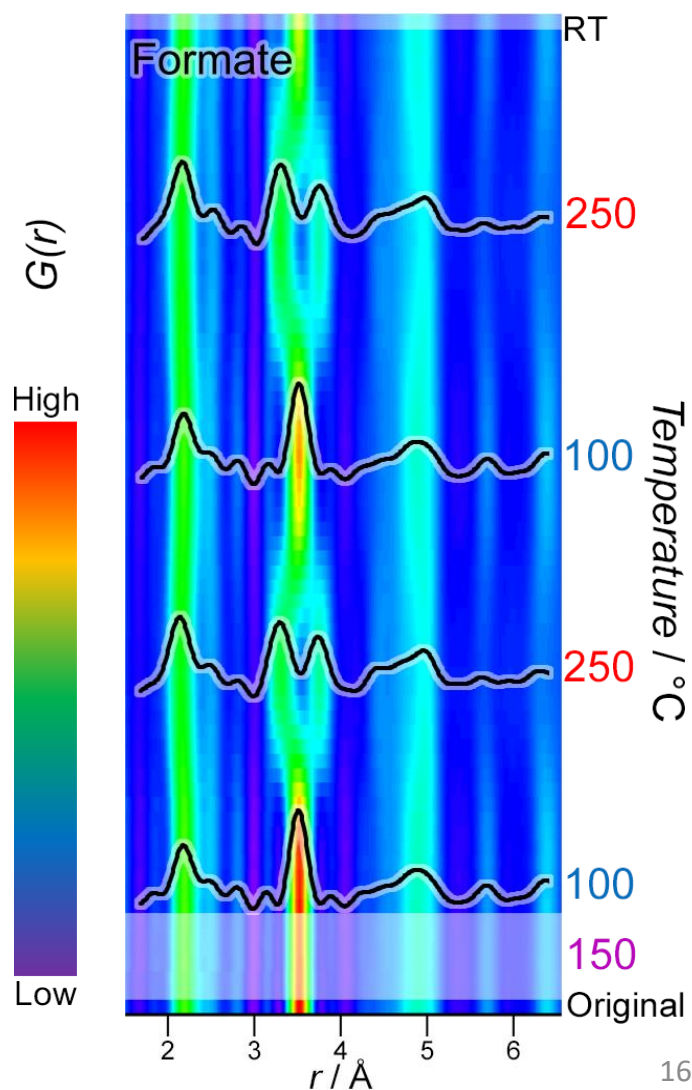
# Decorated NU-1000s have same structure

*But different temperature dependence*

-Cl and Formate free NU-1000 lose crystallinity at 250 °C



# Zr<sub>6</sub> node distorts reversibly

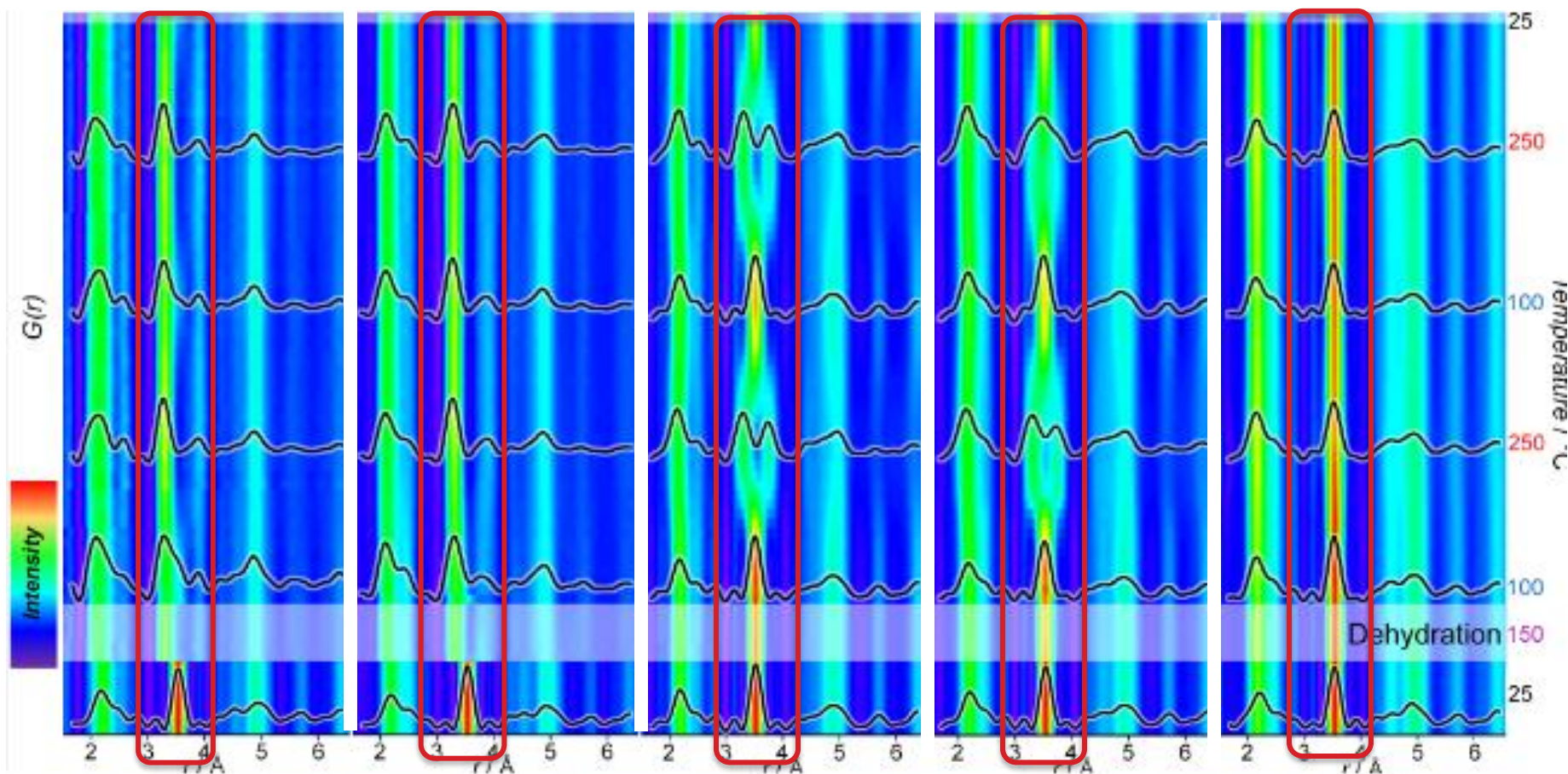
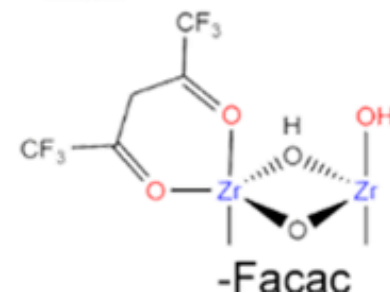
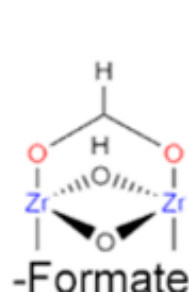
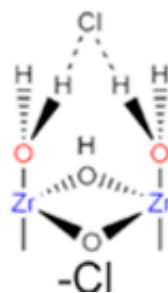
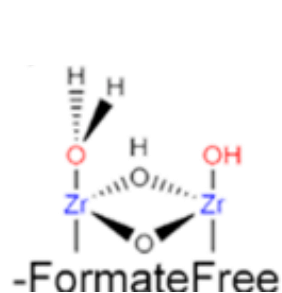


PDF - Local structure independent of crystallinity

Peak Position  $\longleftrightarrow$  Atomic distances  
 Peak Area  $\longleftrightarrow$  Coordination #  
 Peak Width  $\longleftrightarrow$  Disorder



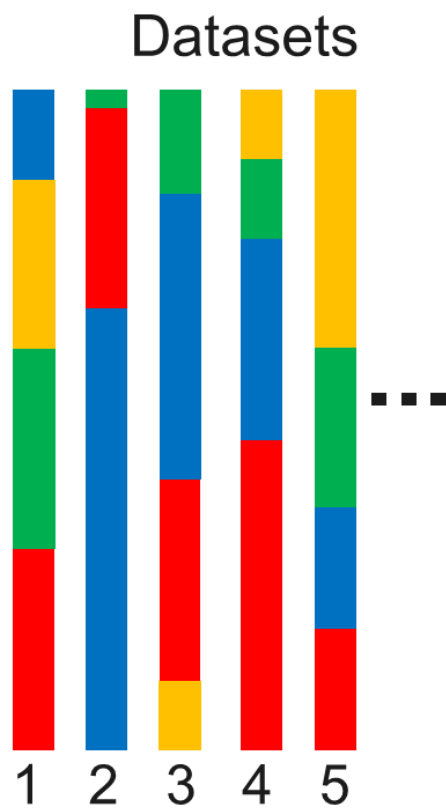
# PDF shows different node distortions



# Multivariate analysis separates distinct local states

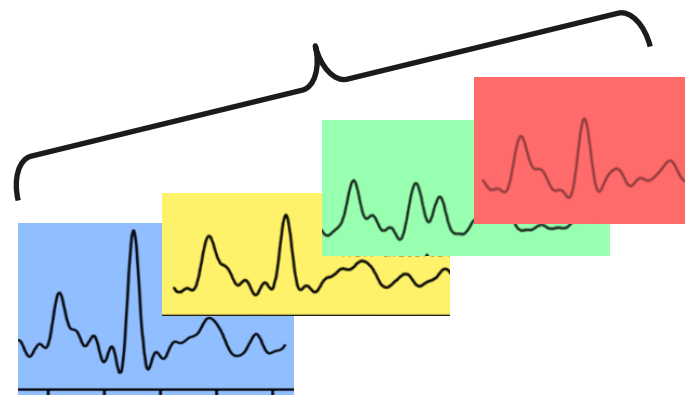
Dimensional reduction algorithms such as Non-Negative Matrix Factorization (NMF) cluster features that change together

This provides a model-free approach to separate and quantify distinct “states” or phases of the system.

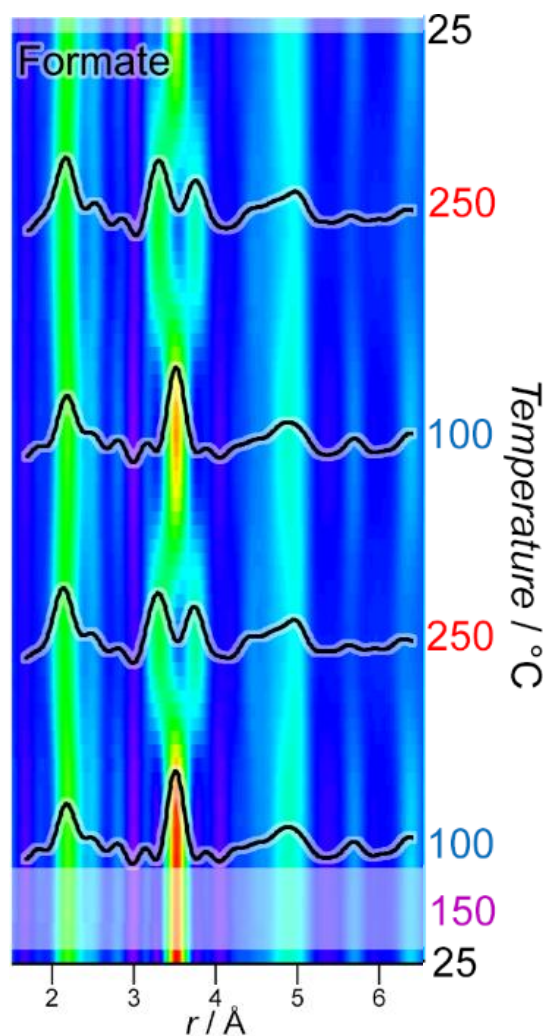


$$\begin{bmatrix} W \\ W \\ W \\ W \end{bmatrix} \times \begin{bmatrix} H \\ H \\ H \\ H \\ H \\ H \\ H \\ H \end{bmatrix} \approx \begin{bmatrix} V \\ V \\ V \\ V \\ V \\ V \\ V \\ V \end{bmatrix}$$

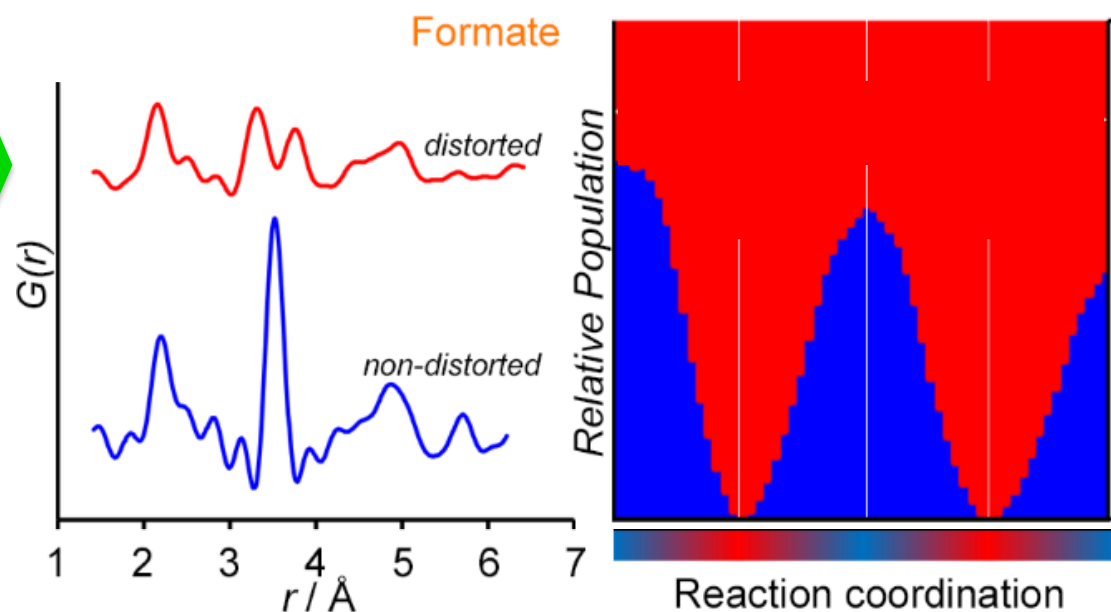
Weighting  $\times$  Components



# Multivariate analysis decouples distinct states

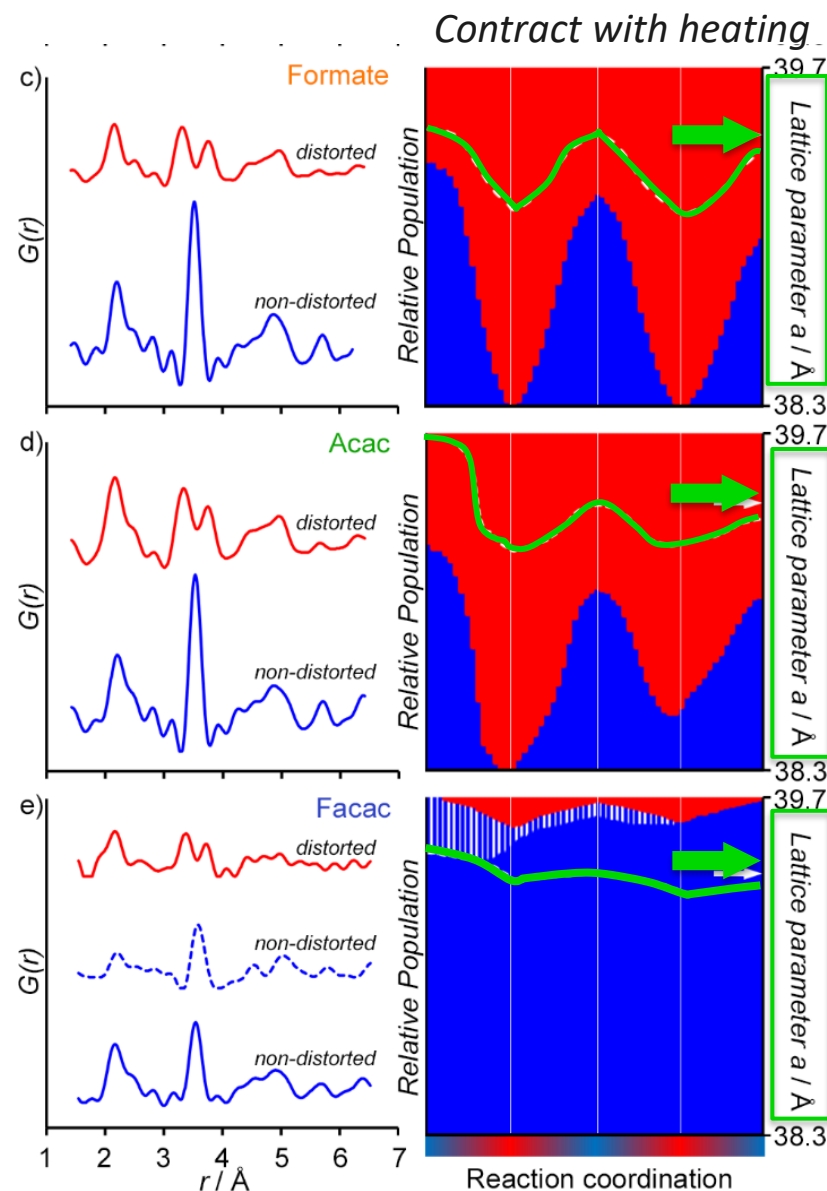
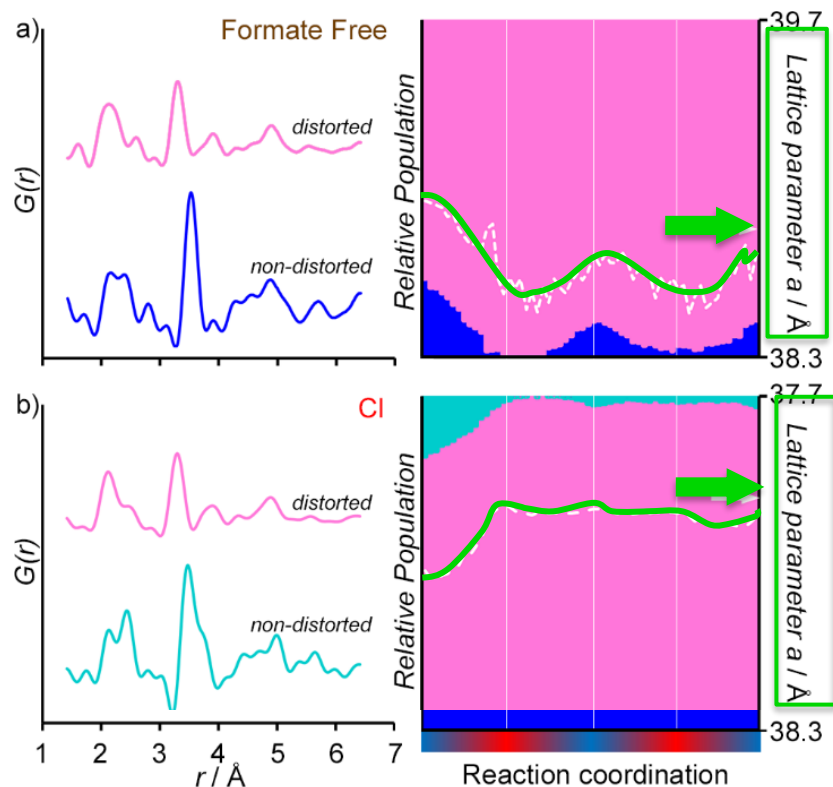


Simplifies data by isolating the components of the data that change together,  
and re-factorizes the data as those discrete components and their time-dependent weightings



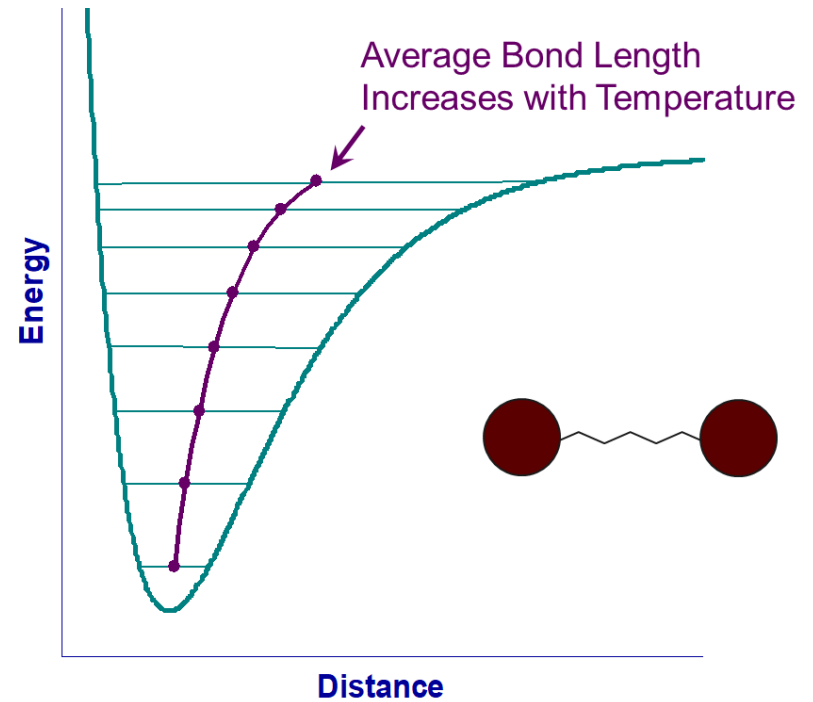
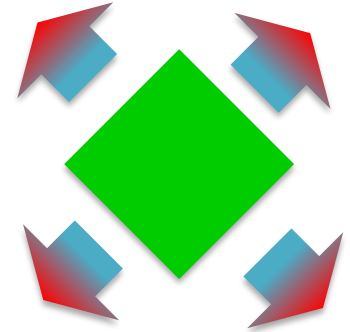
# Local distortion linked to long-range lattice flexing

Thermal expansion  
along the  $a$  axis is  
directly correlated with  
the distorted node population



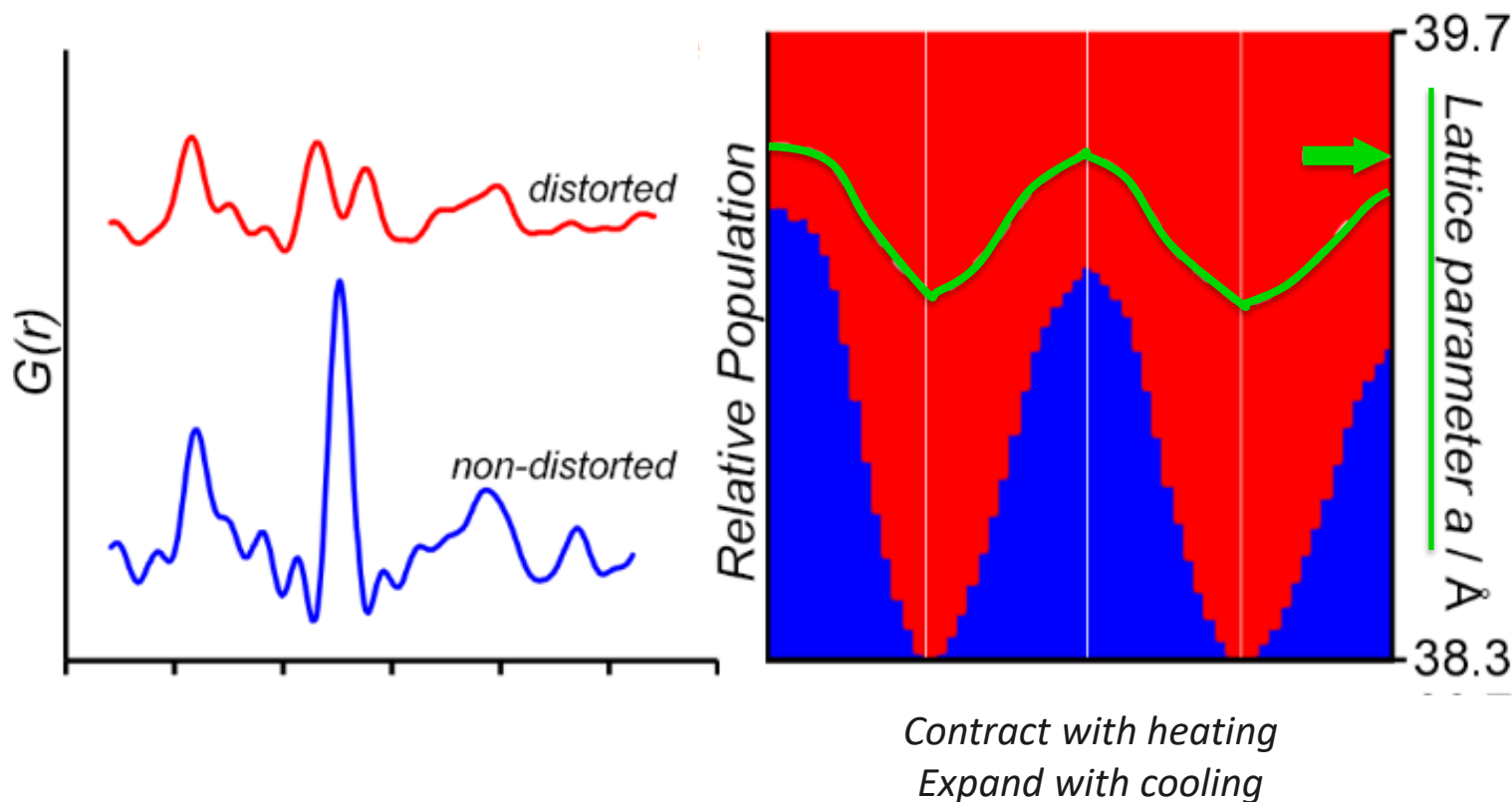


# Thermal Expansion - What is normal?



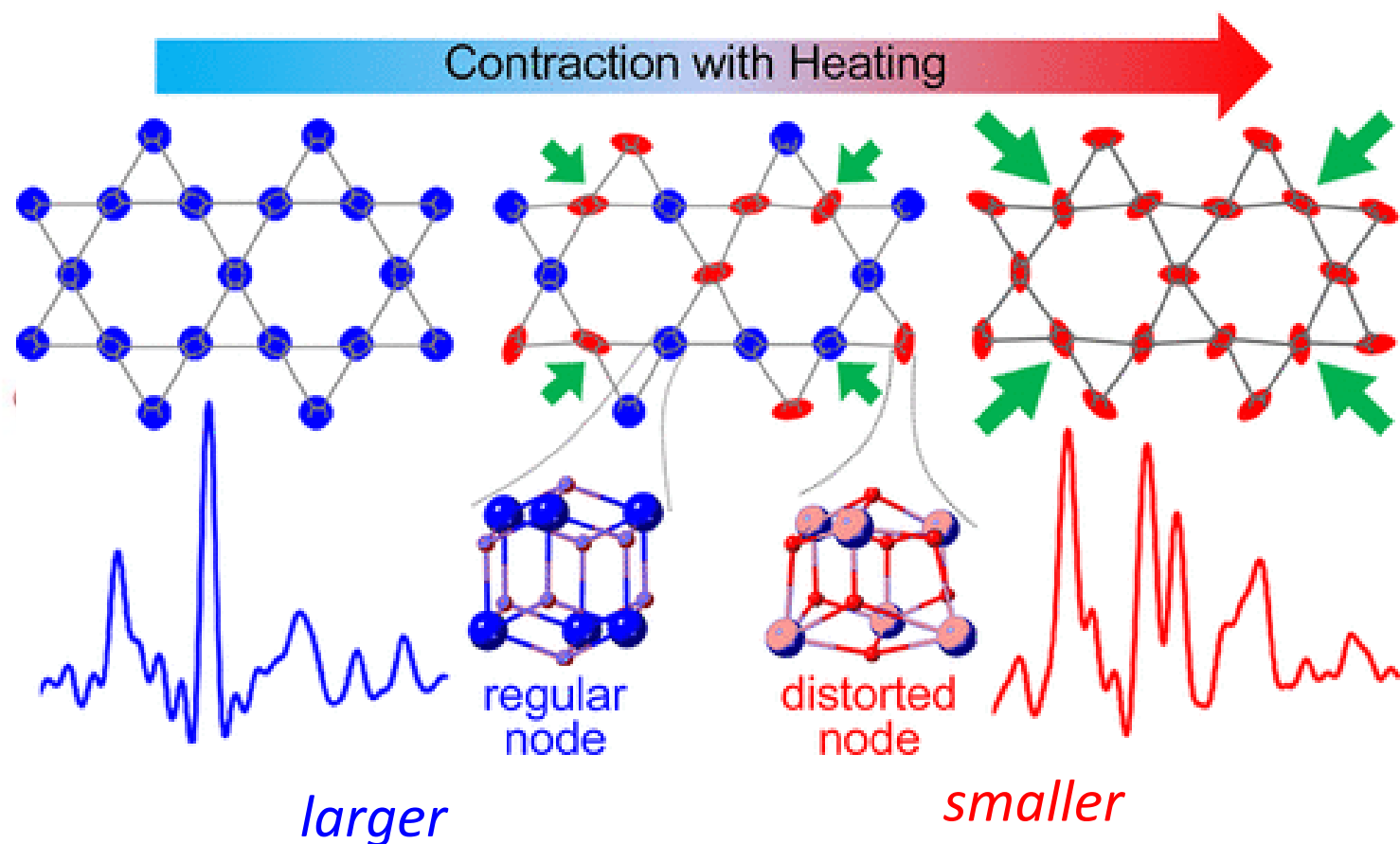
# Node distortion as a novel mechanism for NTE

Negative thermal expansion along the  $a$  axis is directly correlated with the distorted node population



# Node distortion as a novel mechanism for NTE

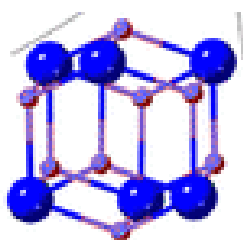
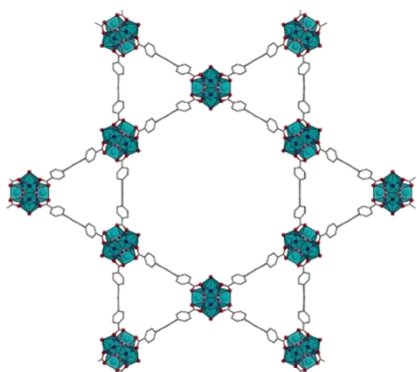
Increasing the relative population of distorted (smaller) nodes with increasing temperature leads to a contraction of the MOF lattice



# Node distortion as a novel mechanism for NTE

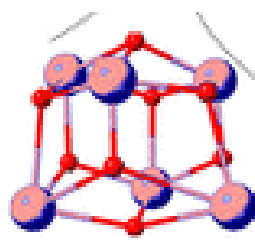
Increasing the relative population of distorted (smaller) nodes with increasing temperature leads to a contraction of the MOF lattice.

Distinct from established NTE mechanisms



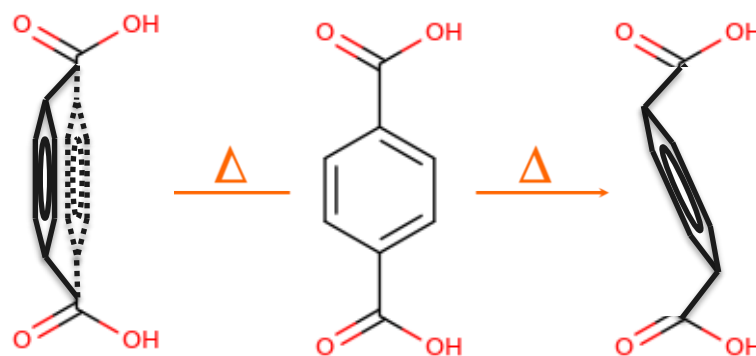
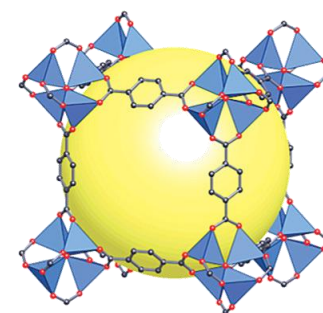
regular  
node

*larger*



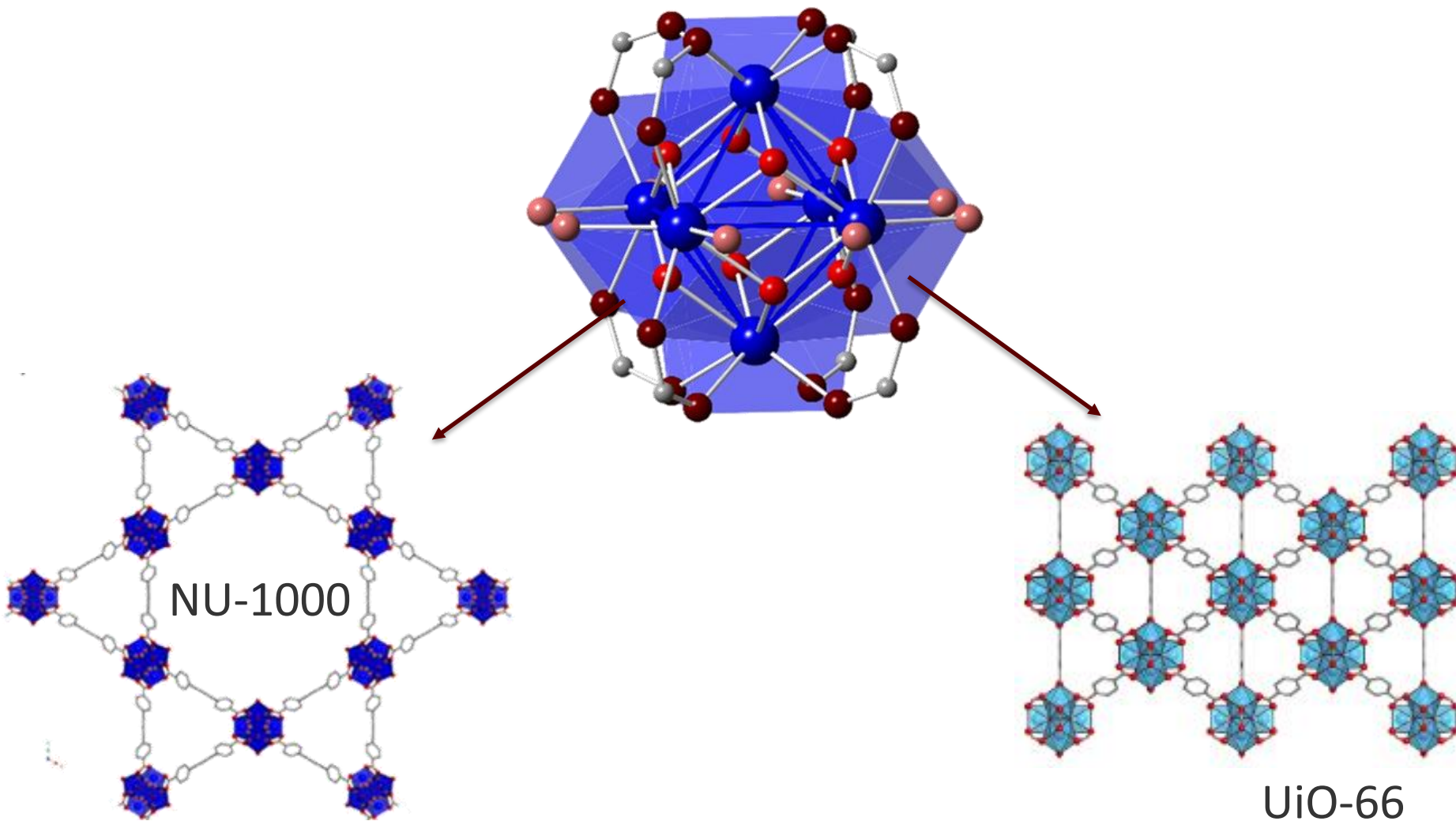
distorted  
node

*smaller*





## Zr<sub>6</sub> nodes in other MOFs

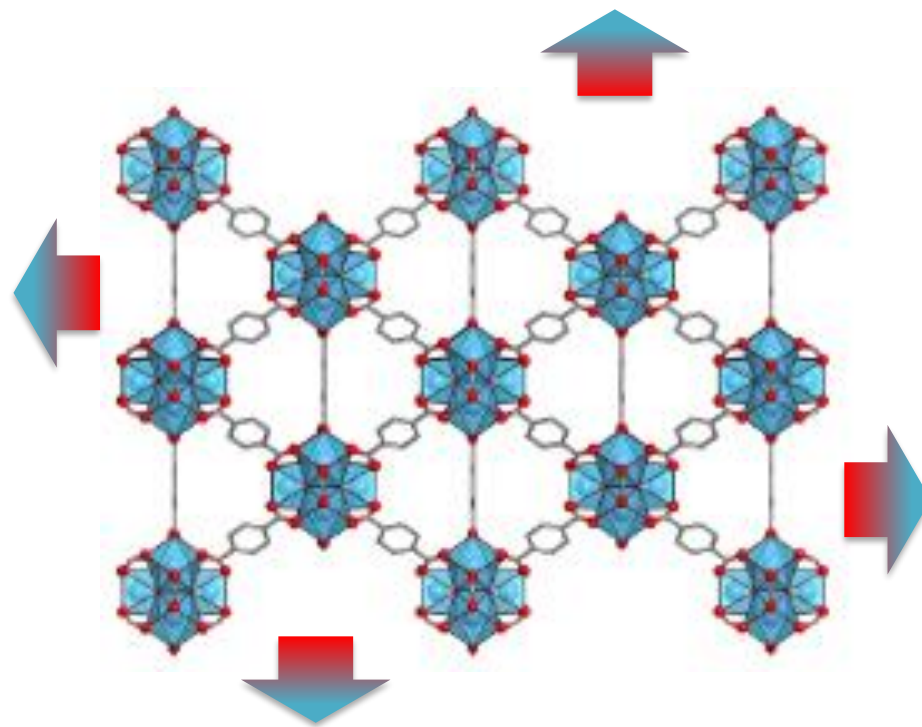
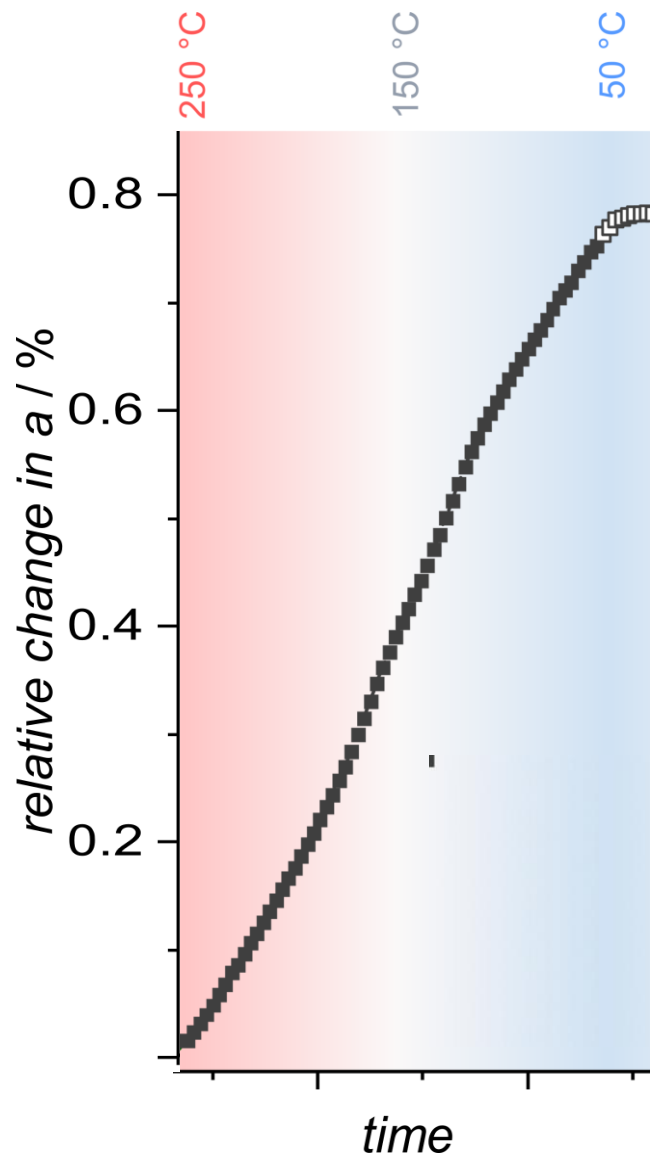


# Evaluating NTE in UiO66

Simon Vornholt

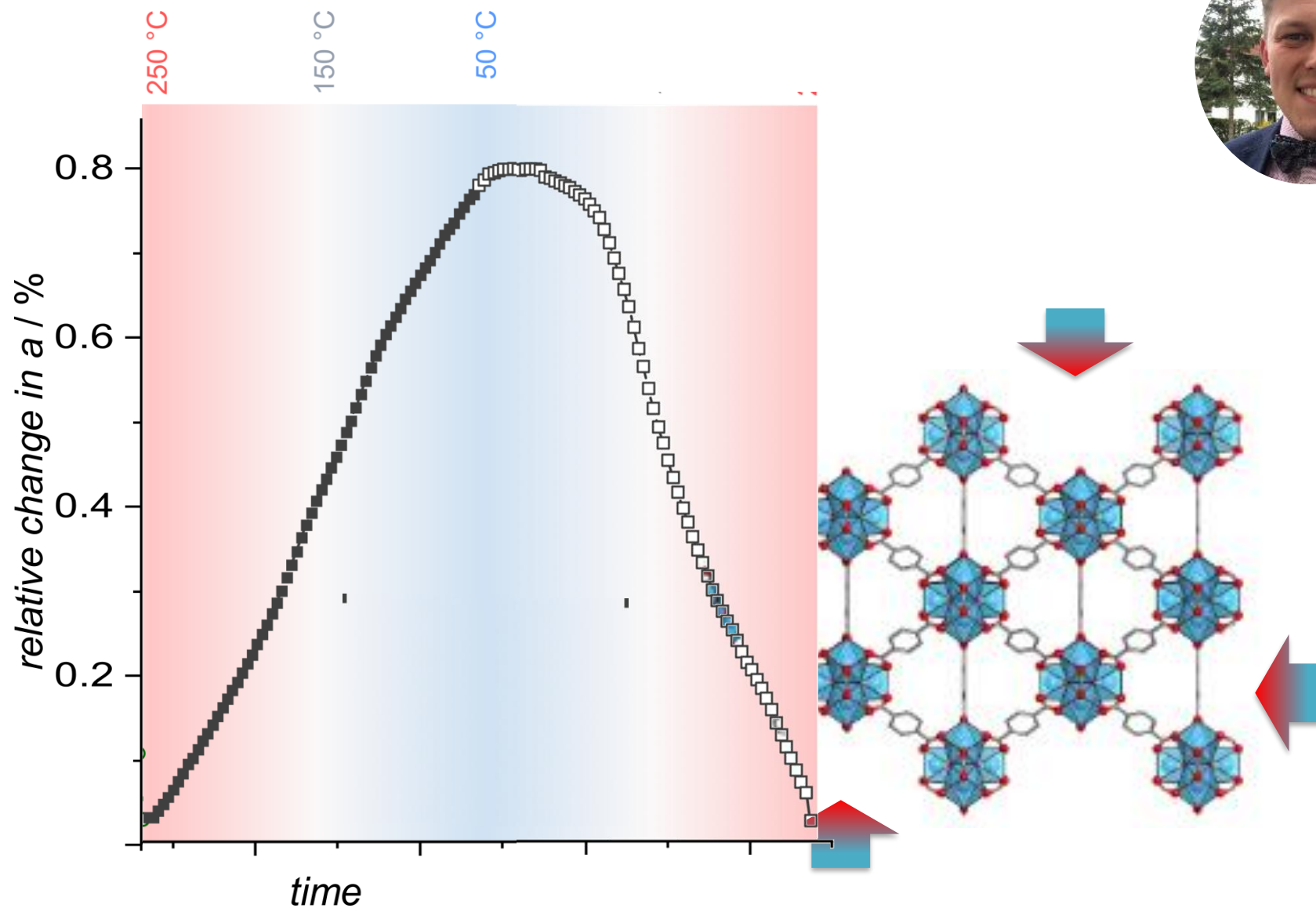


UiO-66 shows isotropic NTE.  
The lattice expands as the  
temperature decreases



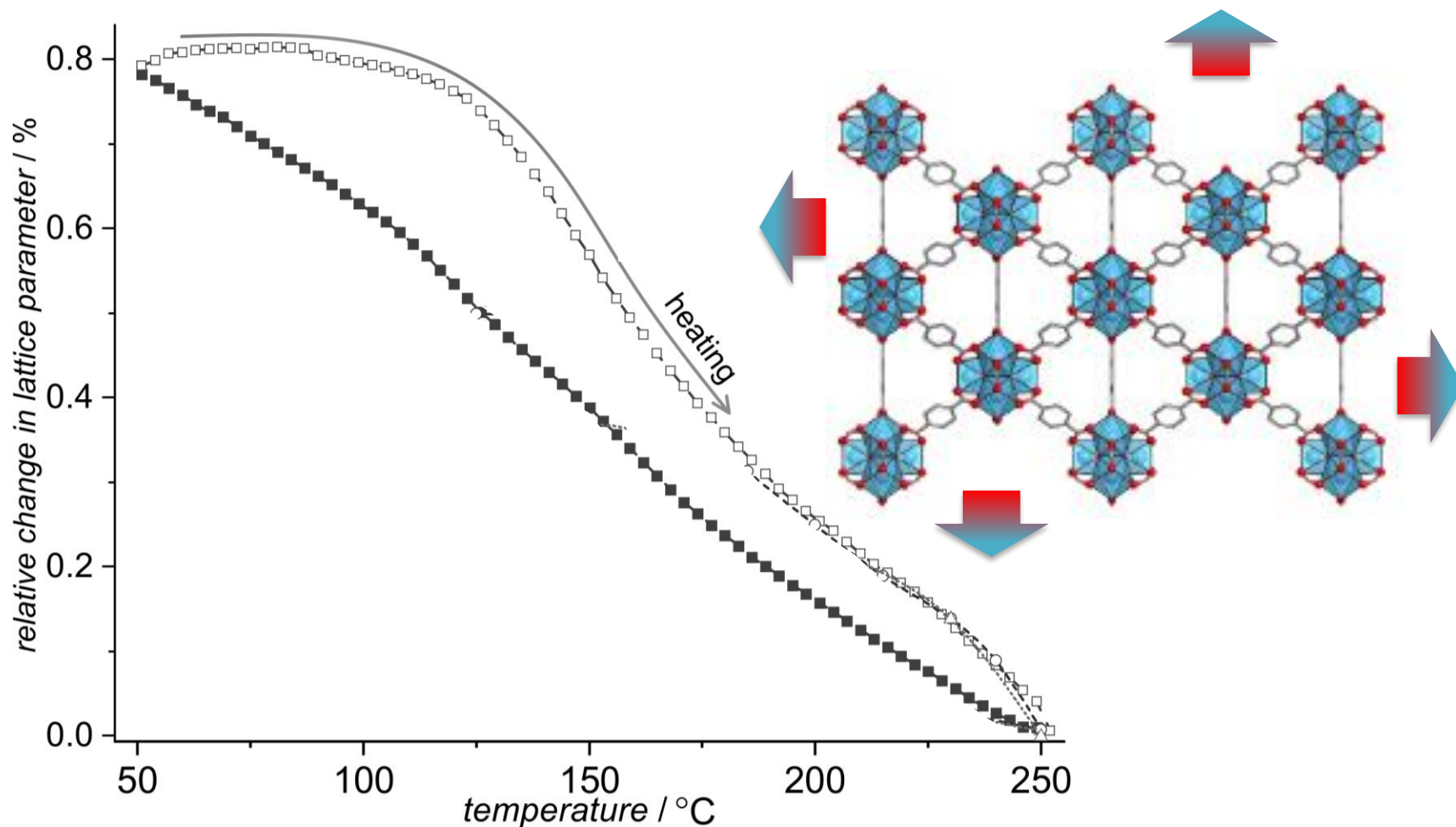
# Evaluating NTE in UiO66, the original Zr<sub>6</sub>-MOF

Simon Vornholt



# Unexpected hysteresis

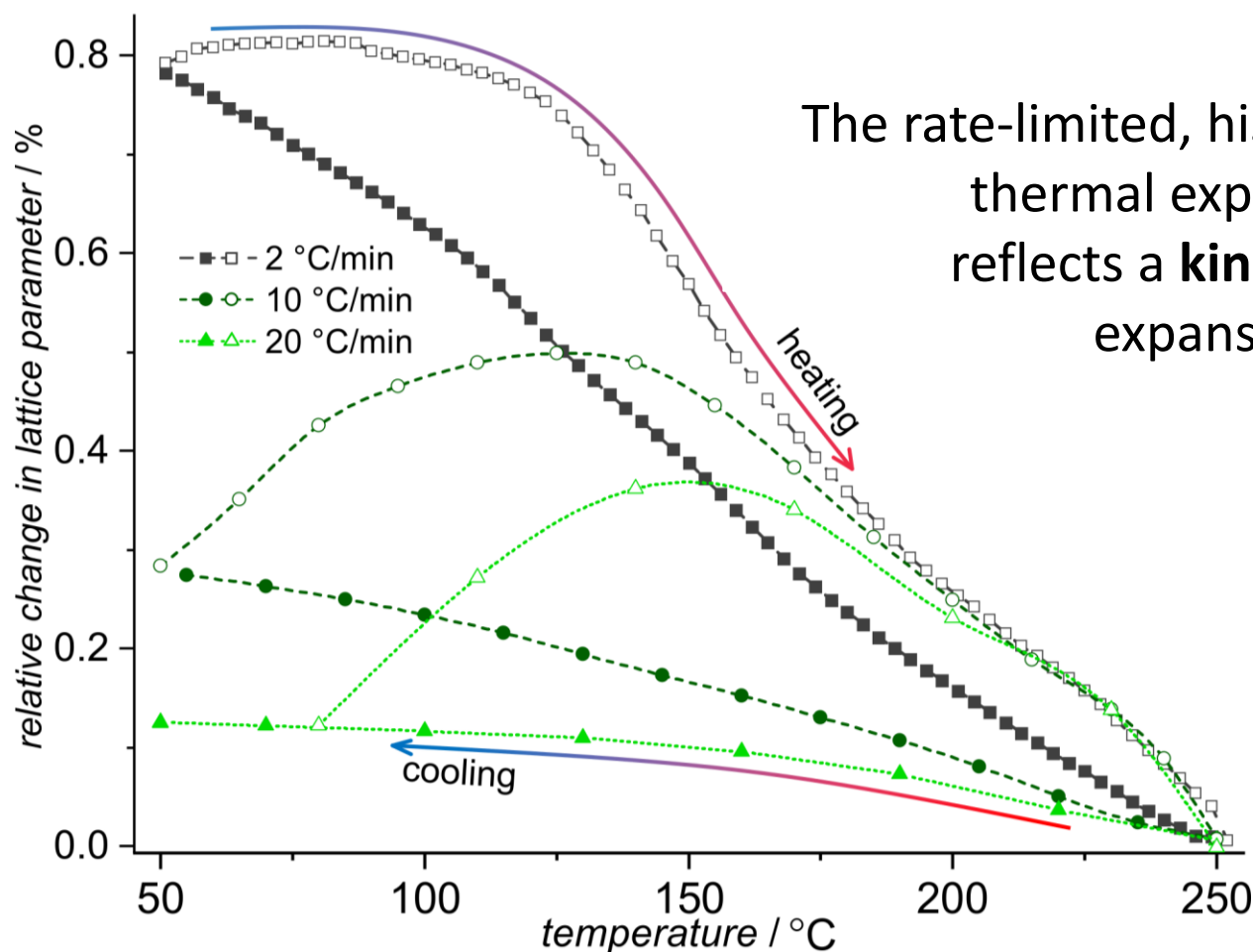
Pronounced hysteresis in the cubic UiO-66 lattice dimension is evident between cooling and heating steps





# Unexpected hysteresis & rate-dependence

The hysteresis and apparent thermal expansion depend on both the thermal history and ramp rates



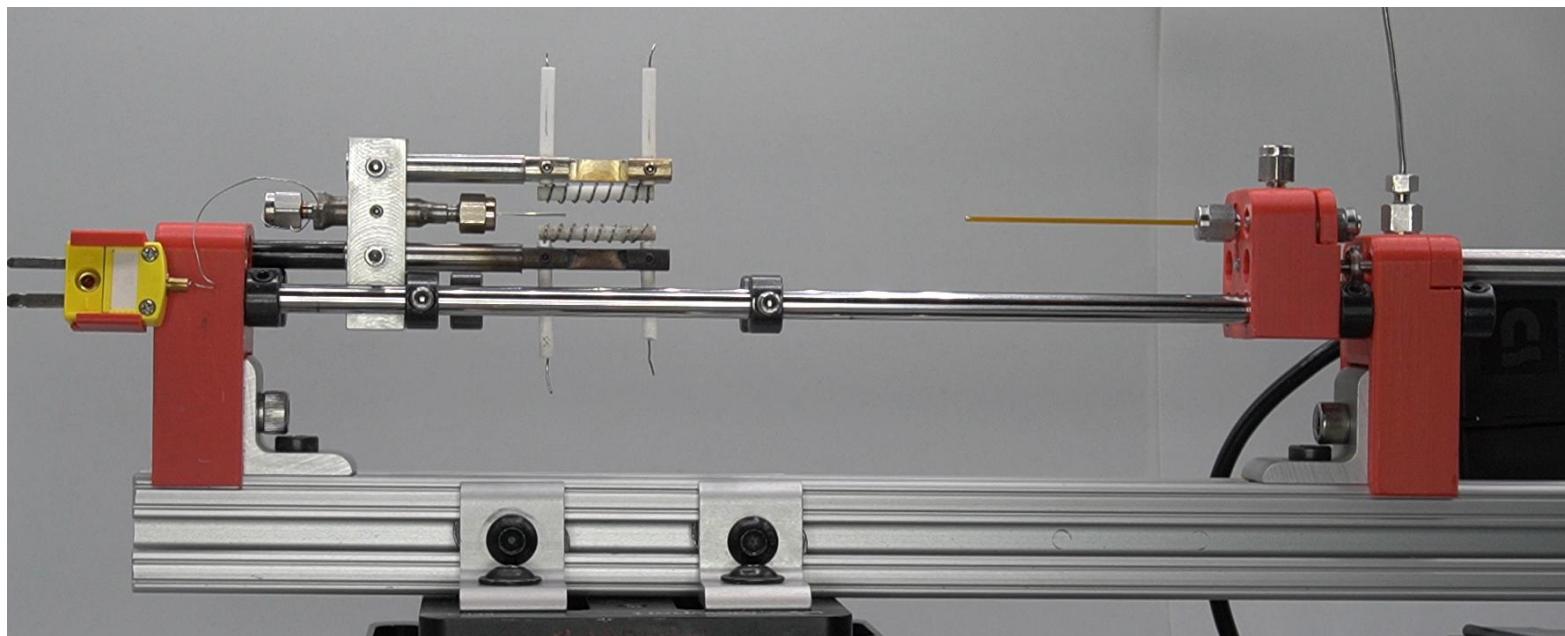
The rate-limited, history-dependent thermal expansion in UiO-66 reflects a **kinetic limitation** in expansion of the lattice with cooling.

# Rapid-Actuating Pneumatic Thermal Reactor - RAPTR

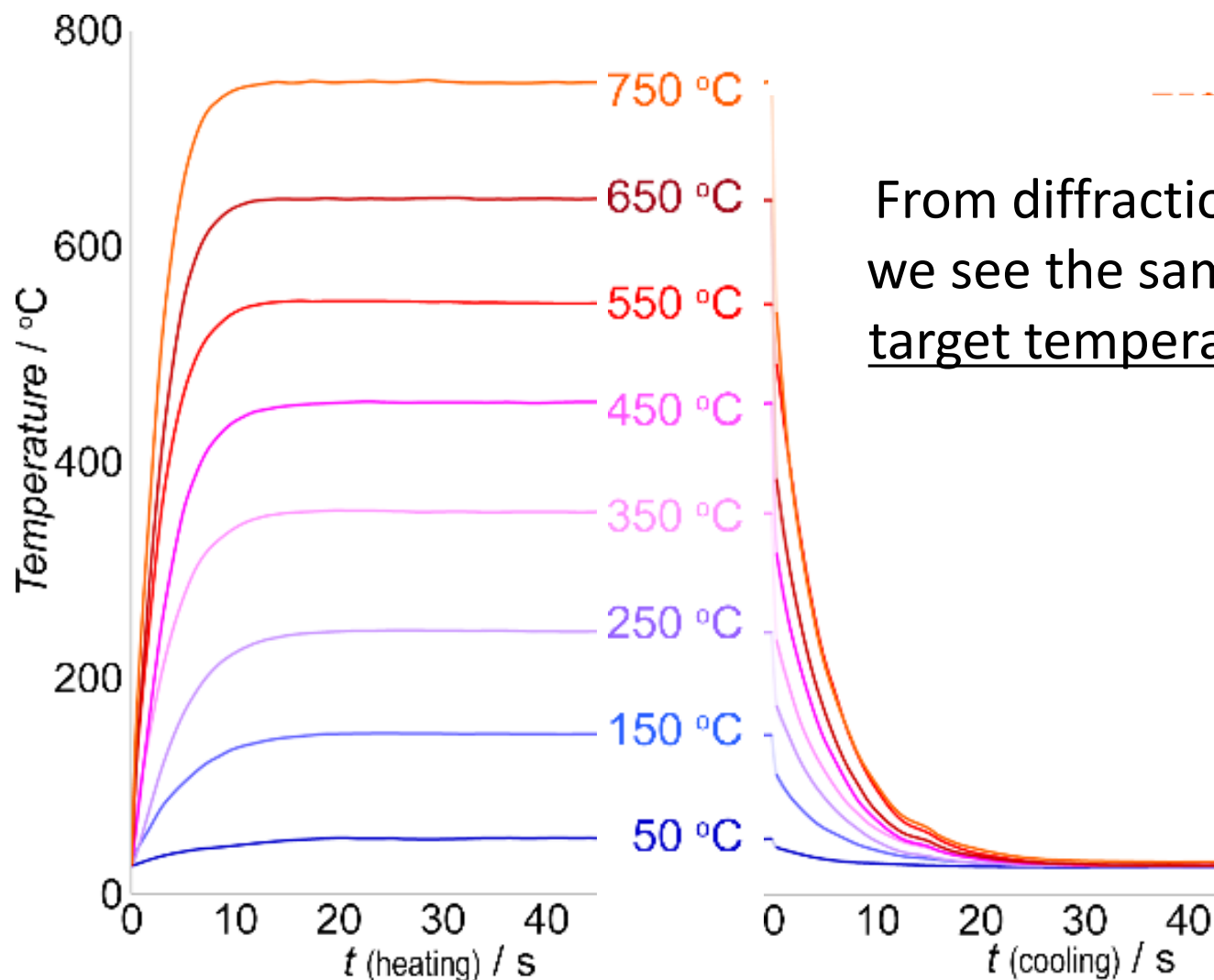
The RAPTR quickly heats and cools samples by translating them into and out of a pre-heated hot zone.

So the heating is only limited by the thermal mass of the sample itself

It adapts the resistive heating elements from the original or thermal gradient heater.



# Fast heating & cooling performance

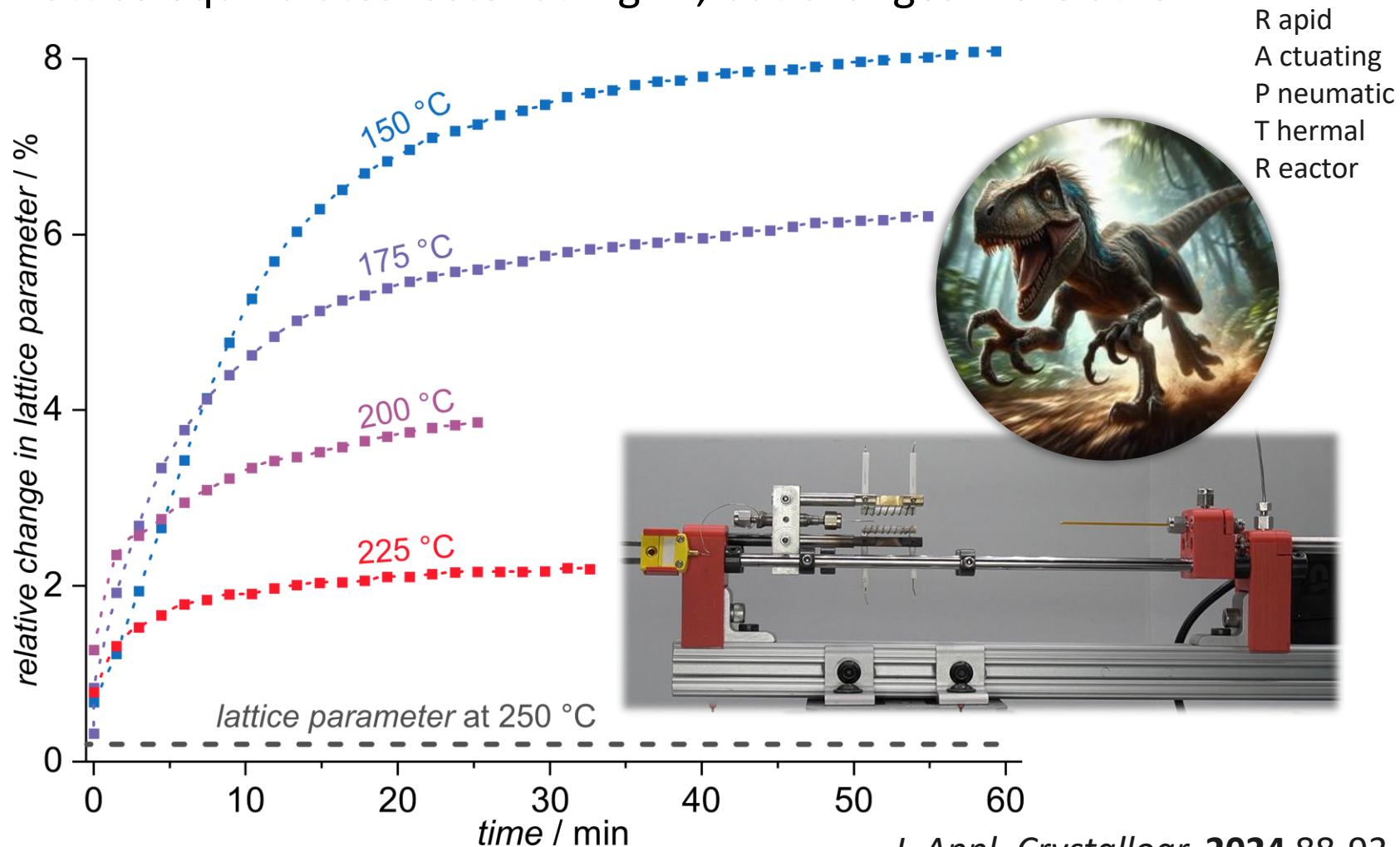


From diffraction thermometry we see the sample reaches the target temperature within 10s.

# Decoupling time- & temperature-dependence

Use the RAPTR to quantify isothermal relaxation.

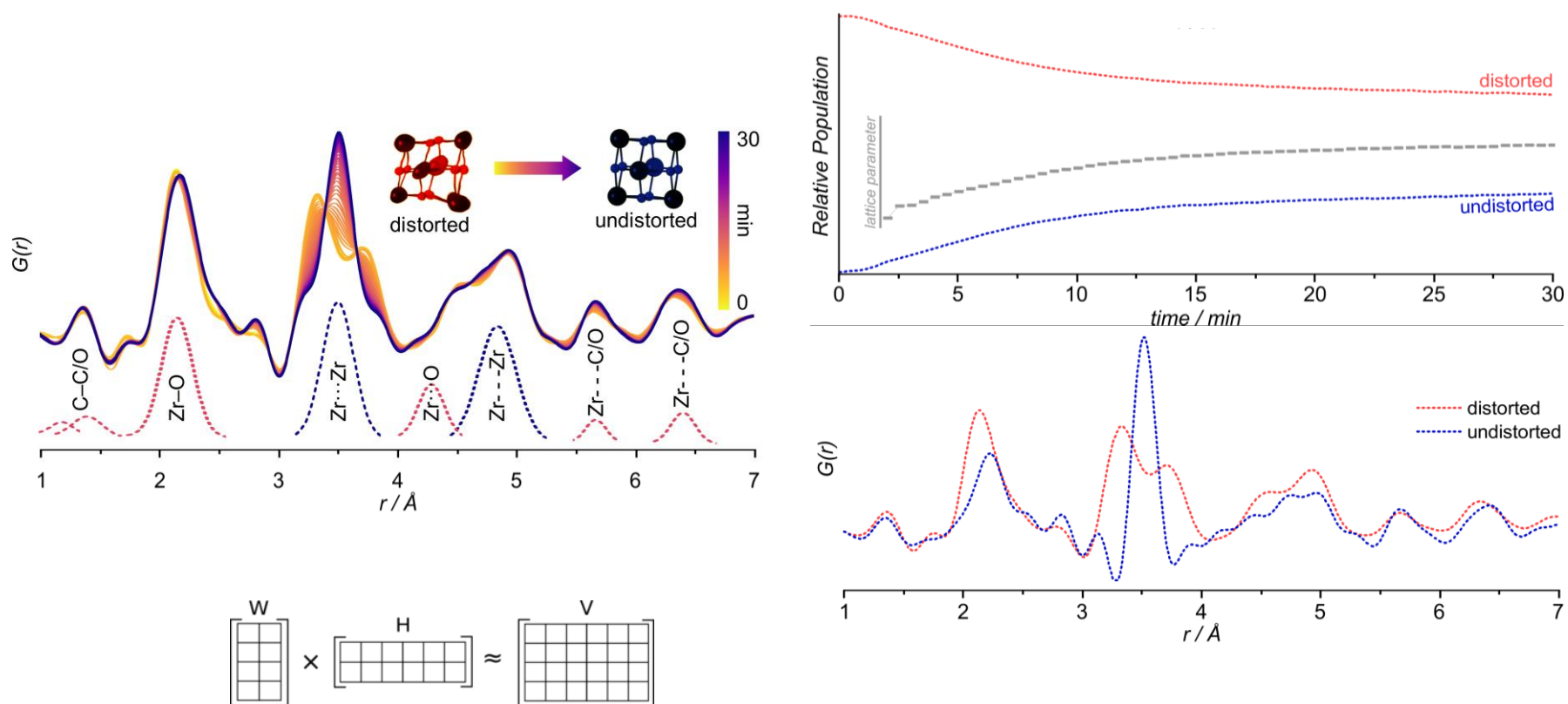
Lattice equilibrates faster at high T, but changes more at low T





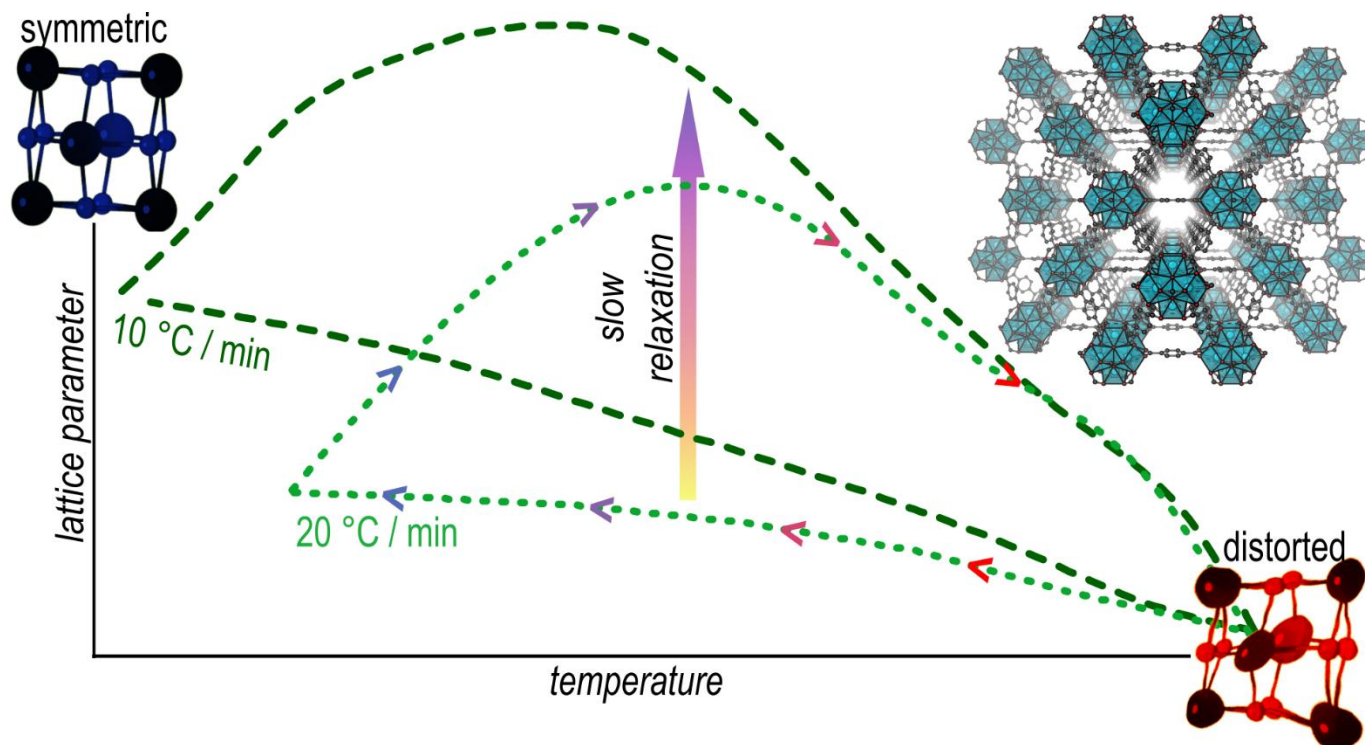
# Correlation to local node transformation

Rate-dependences and lattice hysteresis is a direct consequence of kinetic trapping of the node-distorted state.



# Node transformation frustrated by framework links

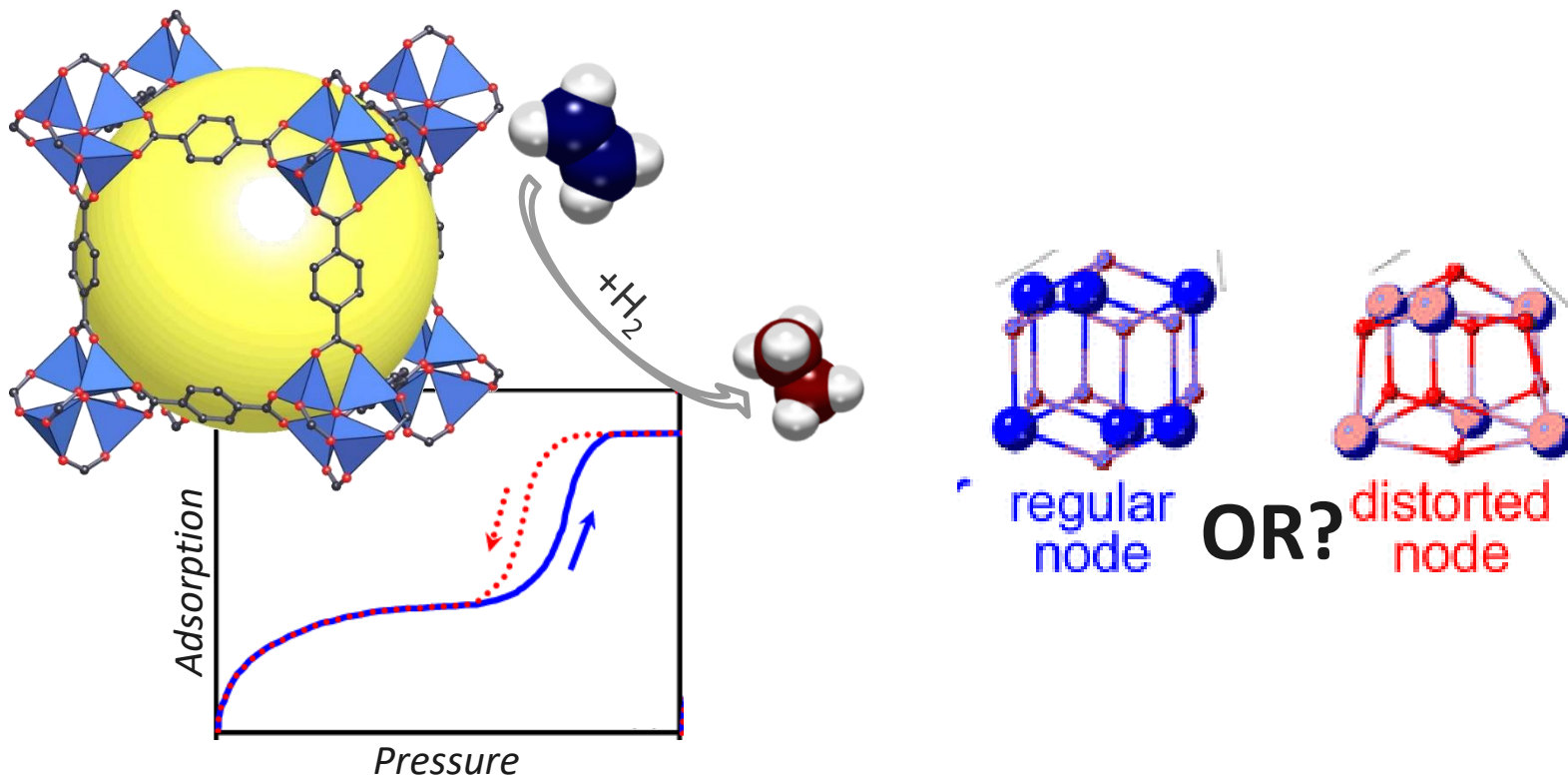
While node distortion is a local phenomenon, the physical connection between nodes via the organic linker ligand means that distortion at connected nodes are coupled



# Implications for catalysis and gas capture

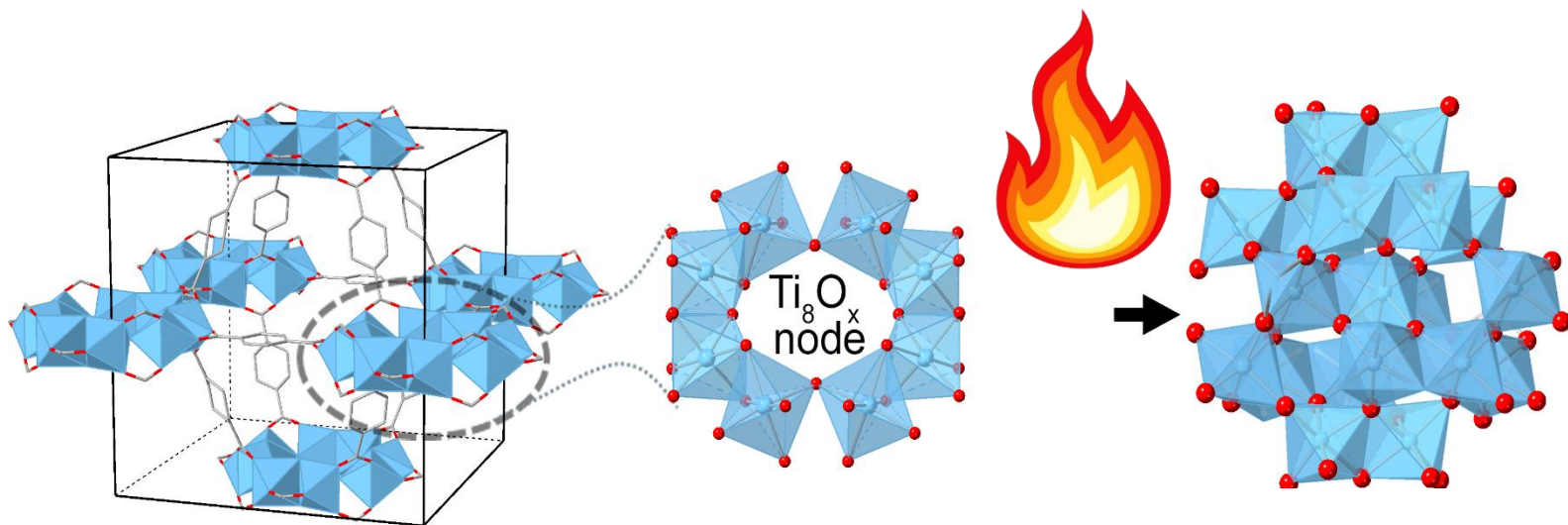
Kinetic trapping of the node-distorted state has broad implications for characterizing and applying these Zr-MOFs.

The cooling following activation can alter the node and lattice critically impacting gas binding, pore volume, & accessible catalytic sites



# MOFs at the edge of stability

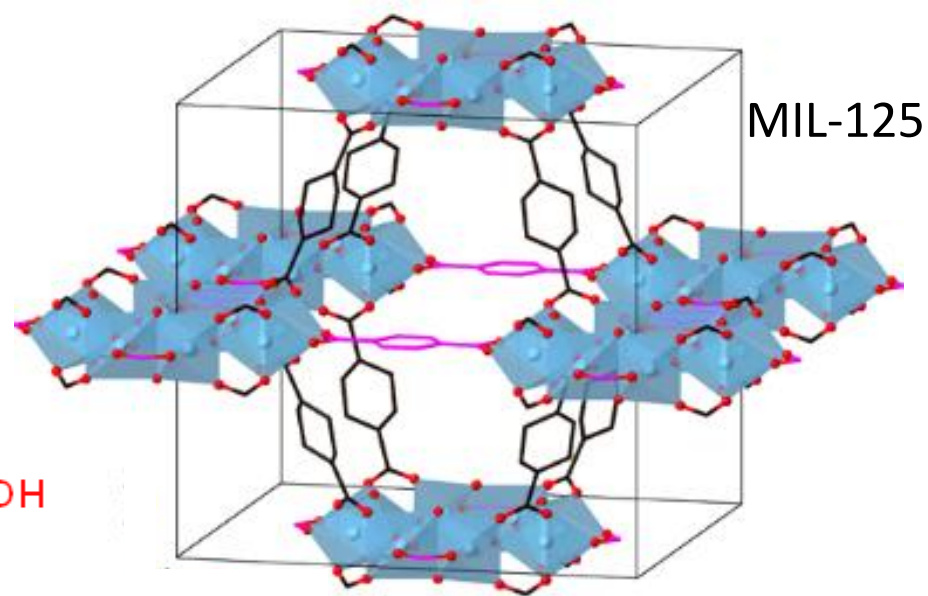
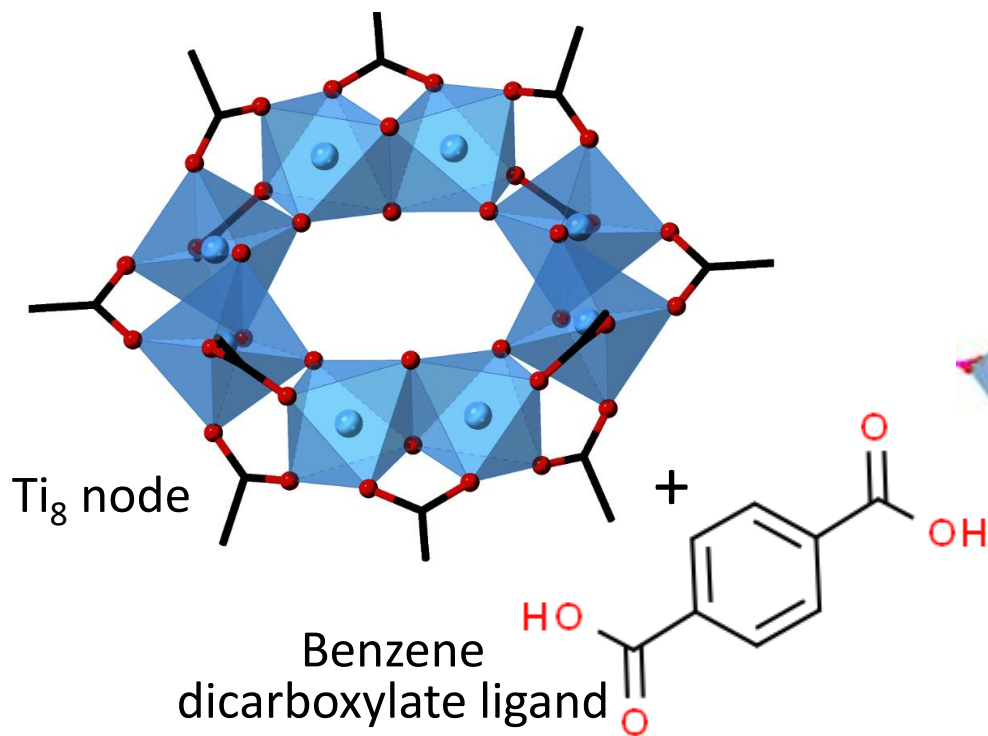
Hypothesis: The node structure and distortions impact nanoparticle formation





# Ti<sub>8</sub> nodes in MIL-125

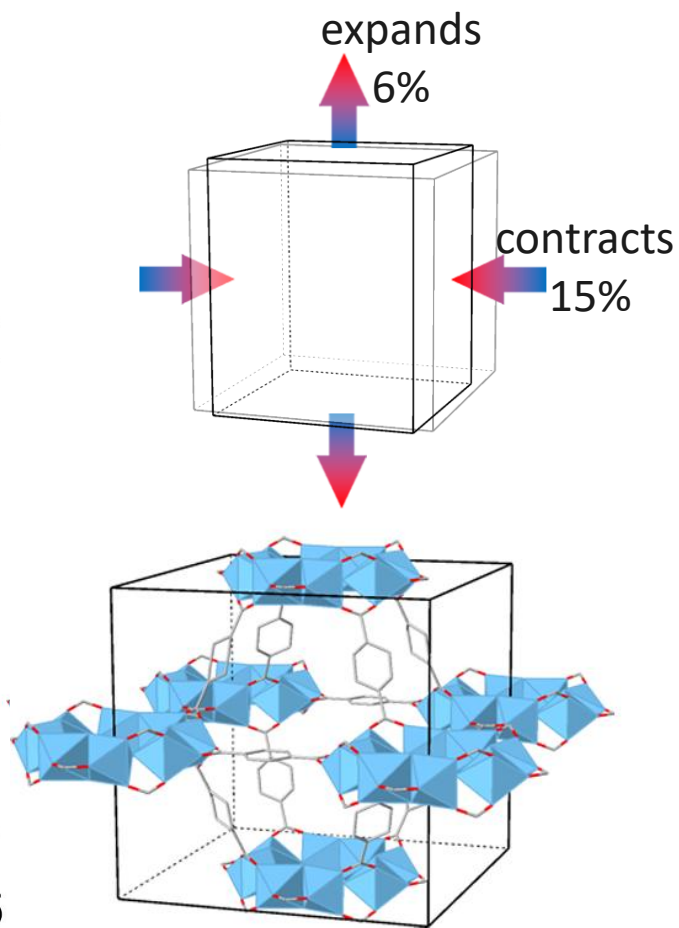
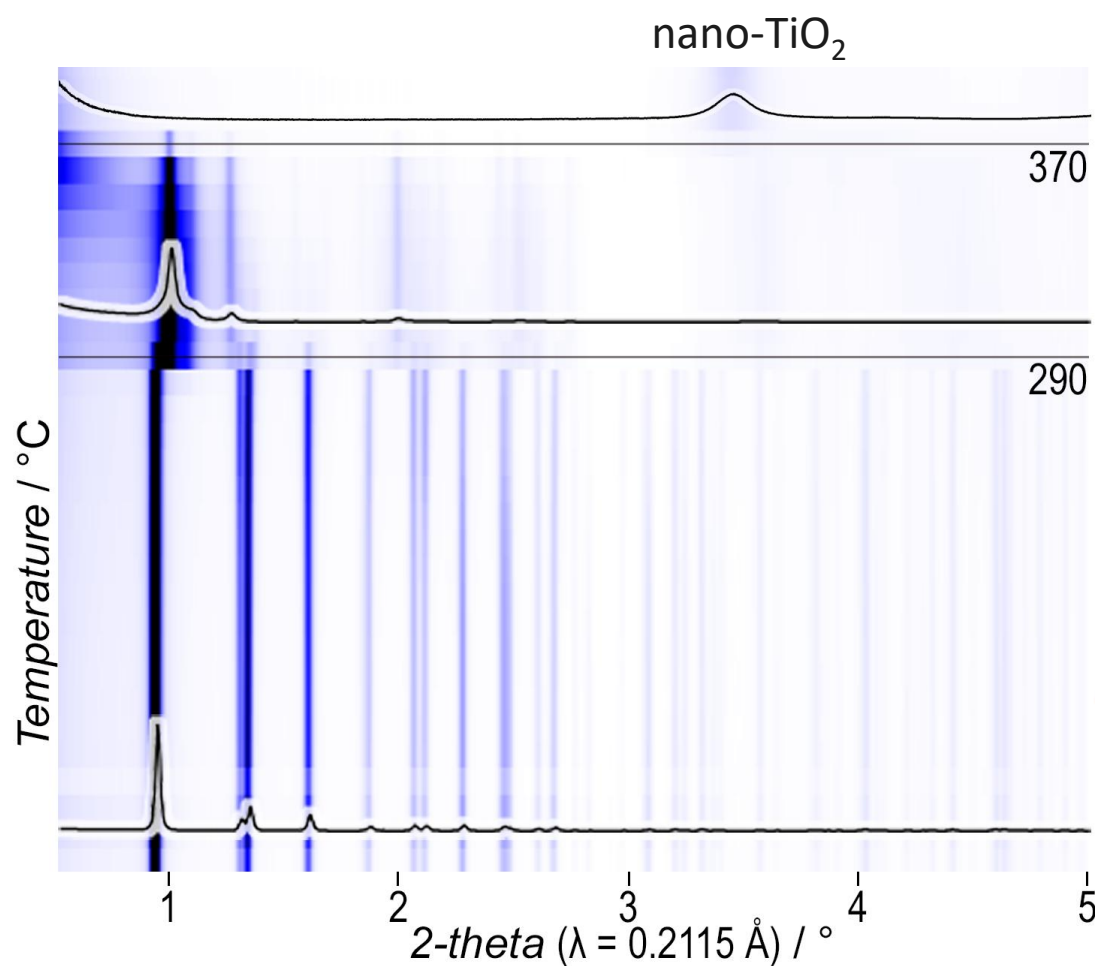
The cyclic Ti<sub>8</sub> node contains alternating edge- and corner-shared TiO<sub>6</sub> octahedra. These are connected by 12 BDC ligands.



# Lattice distortion precedes disassembly

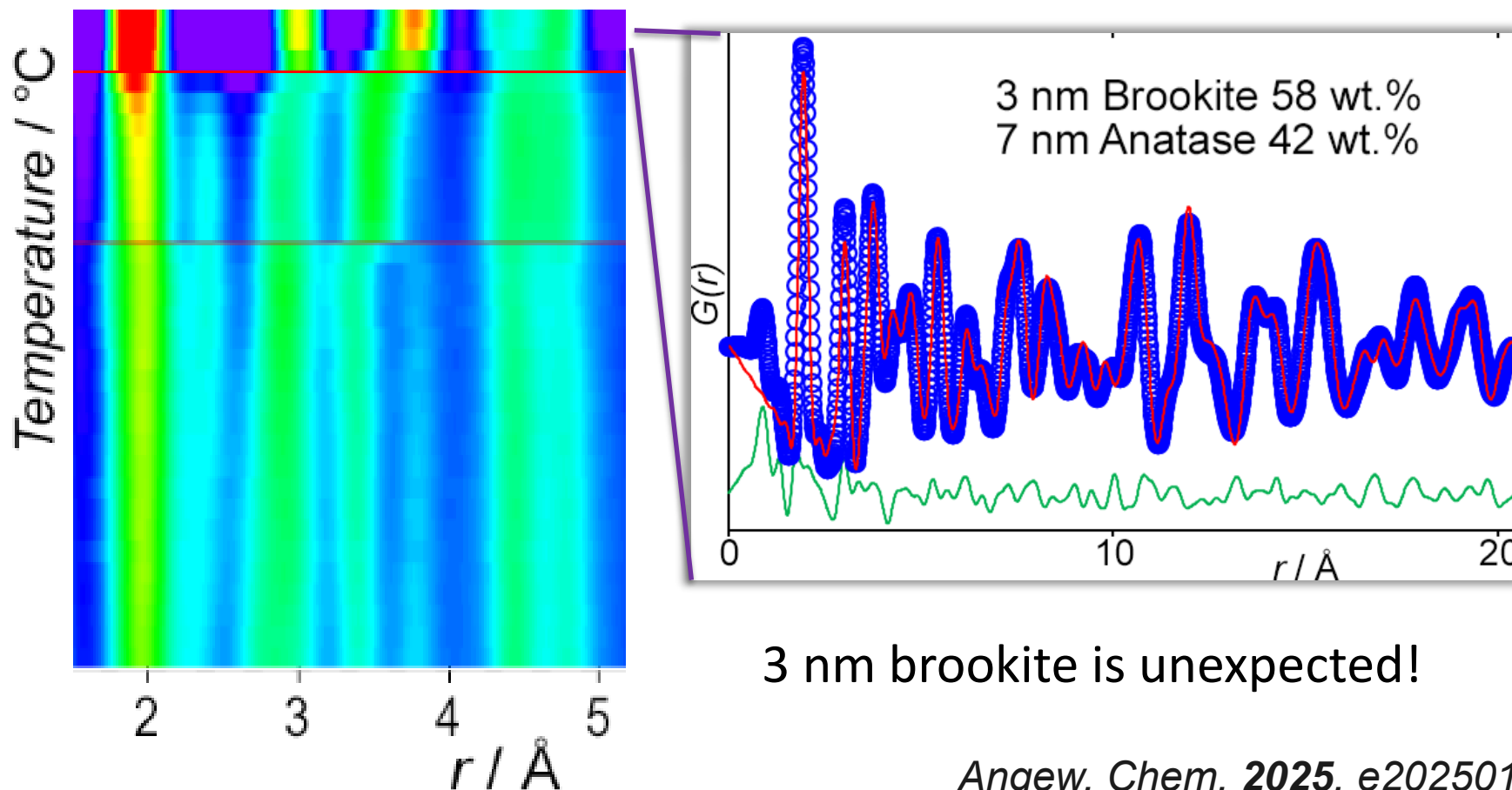


The large contraction in the  $a$ -direction  
(& small expansion in  $c$ -)

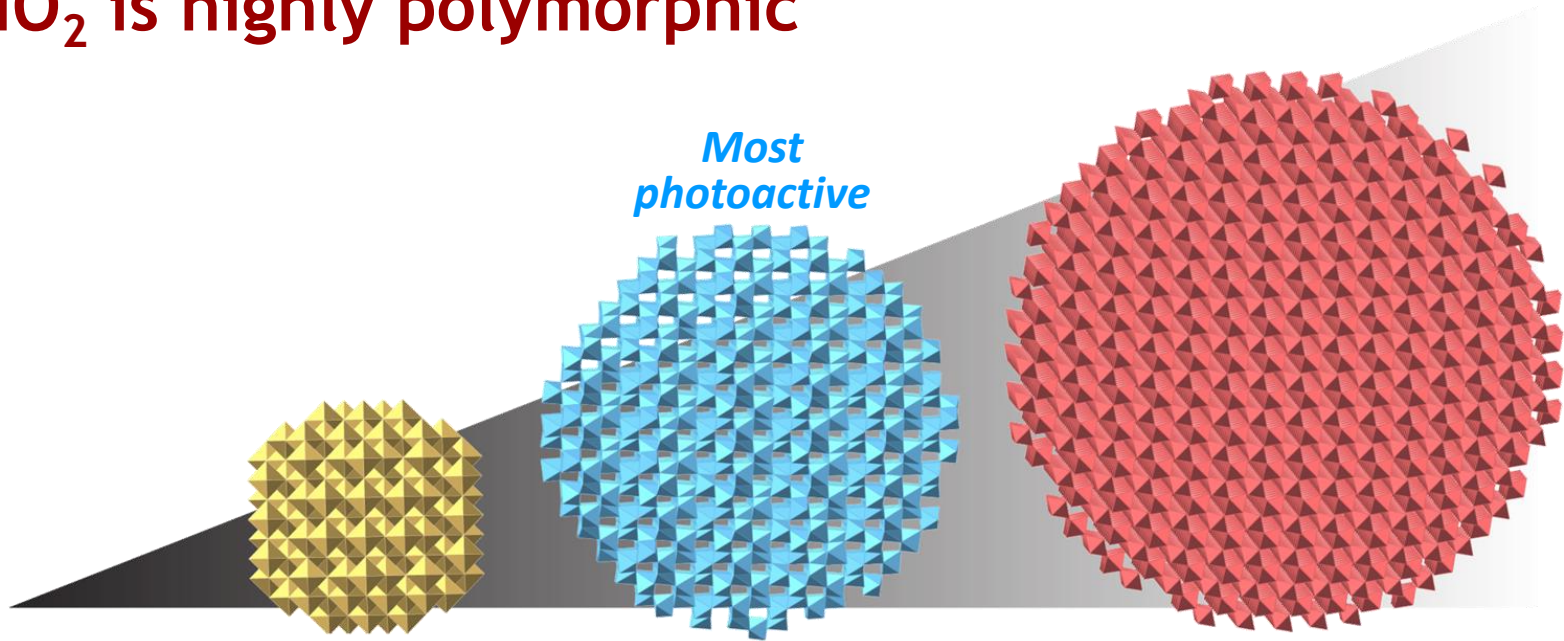


# Unusual metastable $\text{TiO}_2$ polymorph observed

Previously, only anatase and rutile have been seen.



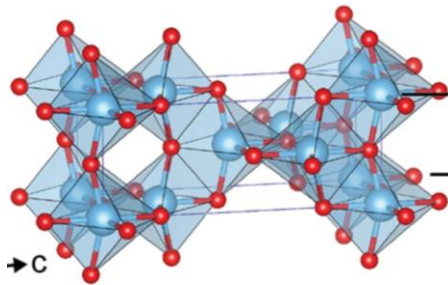
# TiO<sub>2</sub> is highly polymorphic



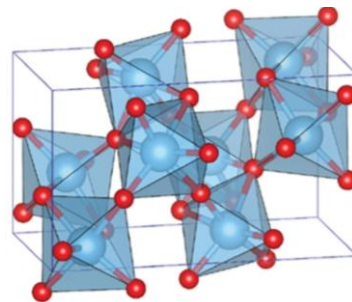
Anatase  
(Metastable)

Brookite  
(Metastable)

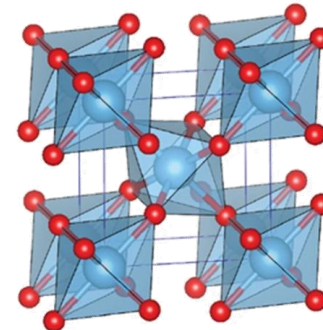
Rutile



4 edge sharing  
4 corner sharing  
(all straight)



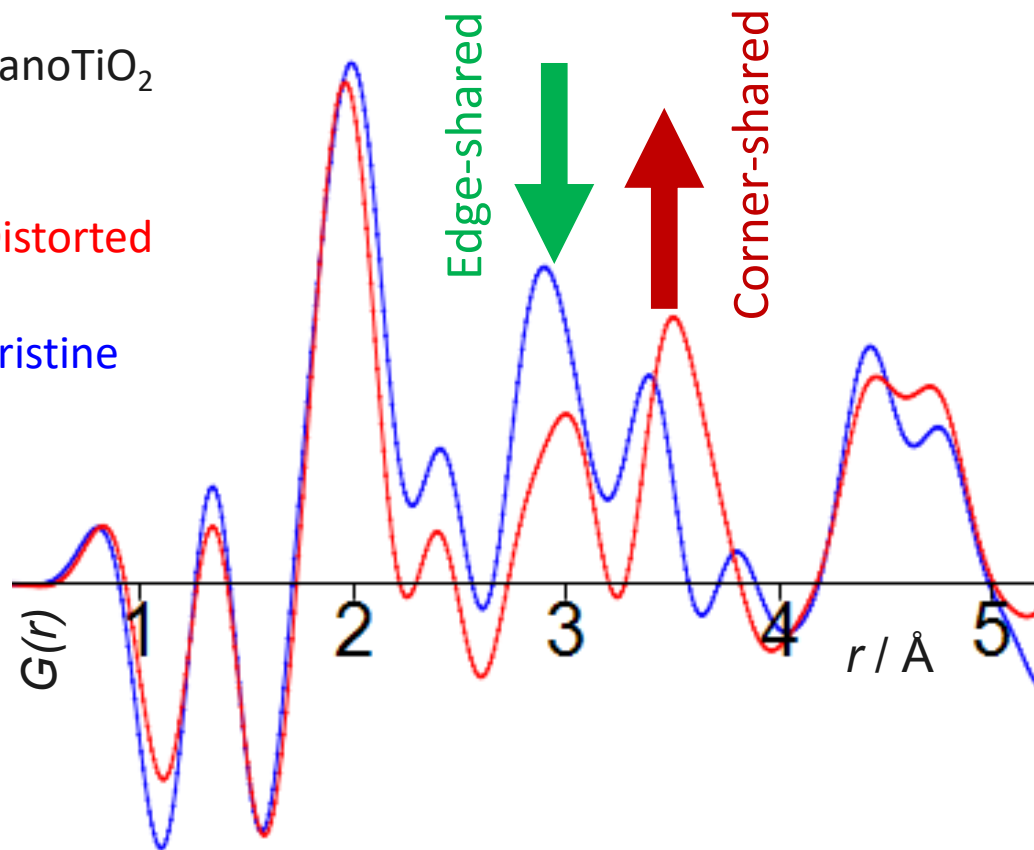
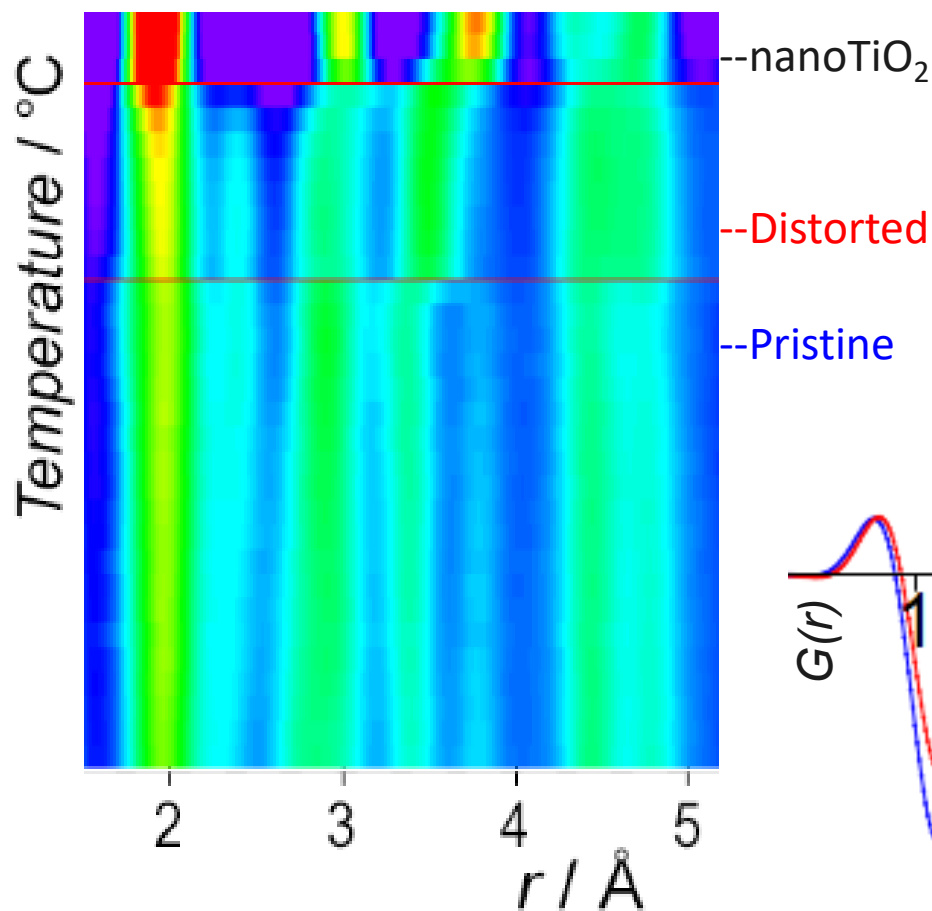
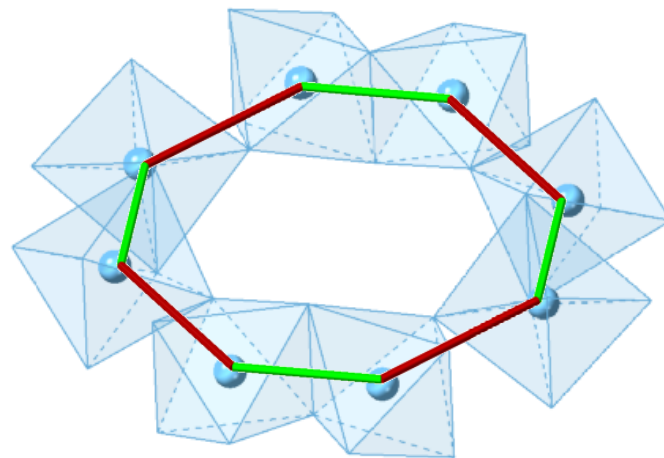
3 edge sharing  
6 corner sharing  
(4 bent, 2 straight)



2 edge sharing  
8 corner sharing  
(all bent)

# Ti<sub>8</sub> node distorts

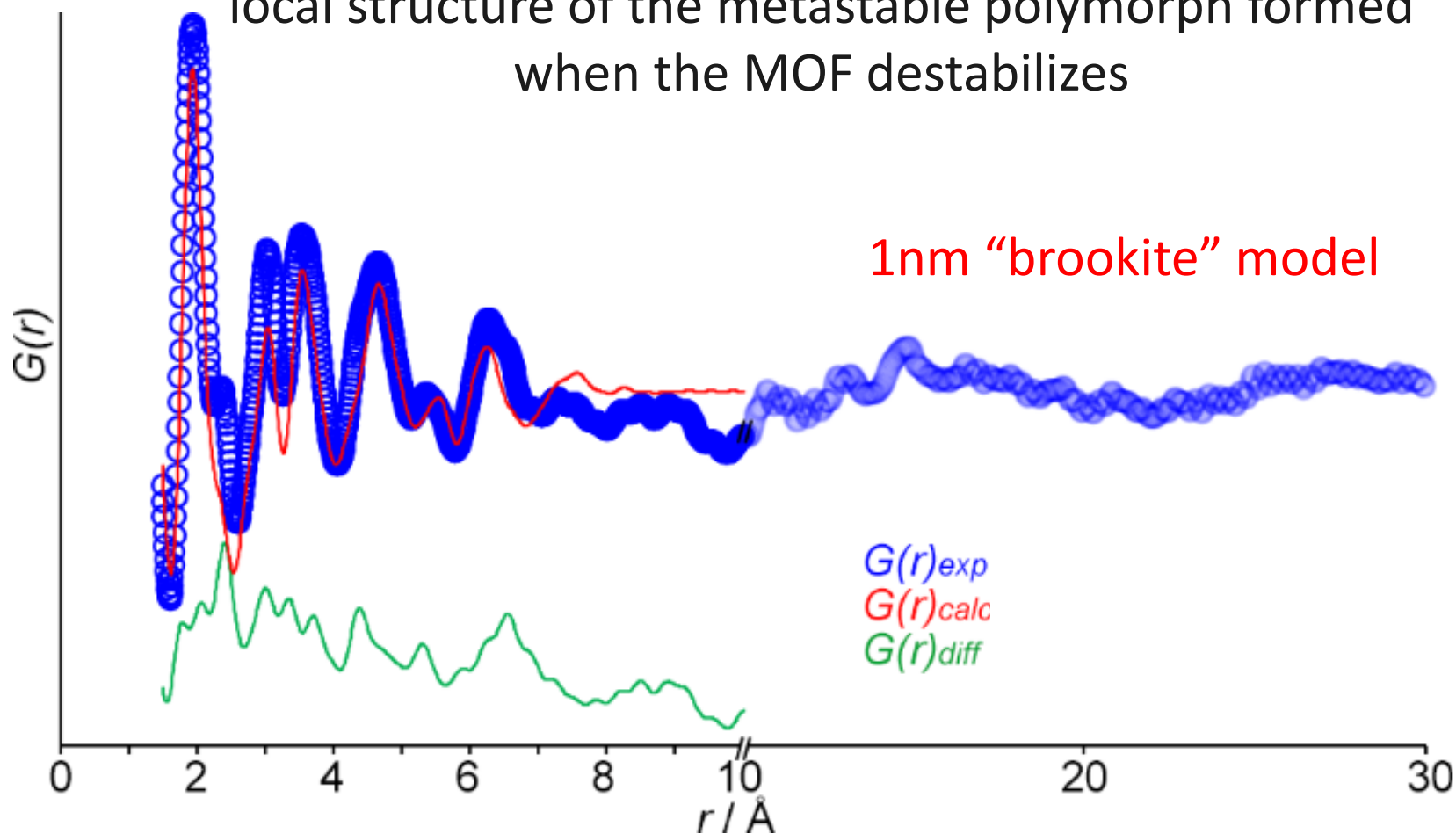
Change from equal number of **edge-** and **corner-**shared TiO<sub>6</sub> (like anatase) to more **corner-**shared TiO<sub>6</sub> (like brookite)





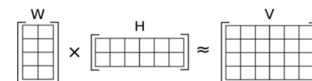
# Distorted $\text{Ti}_8$ node resembles brookite polymorph

The structure of the distorted node matches the local structure of the metastable polymorph formed when the MOF destabilizes

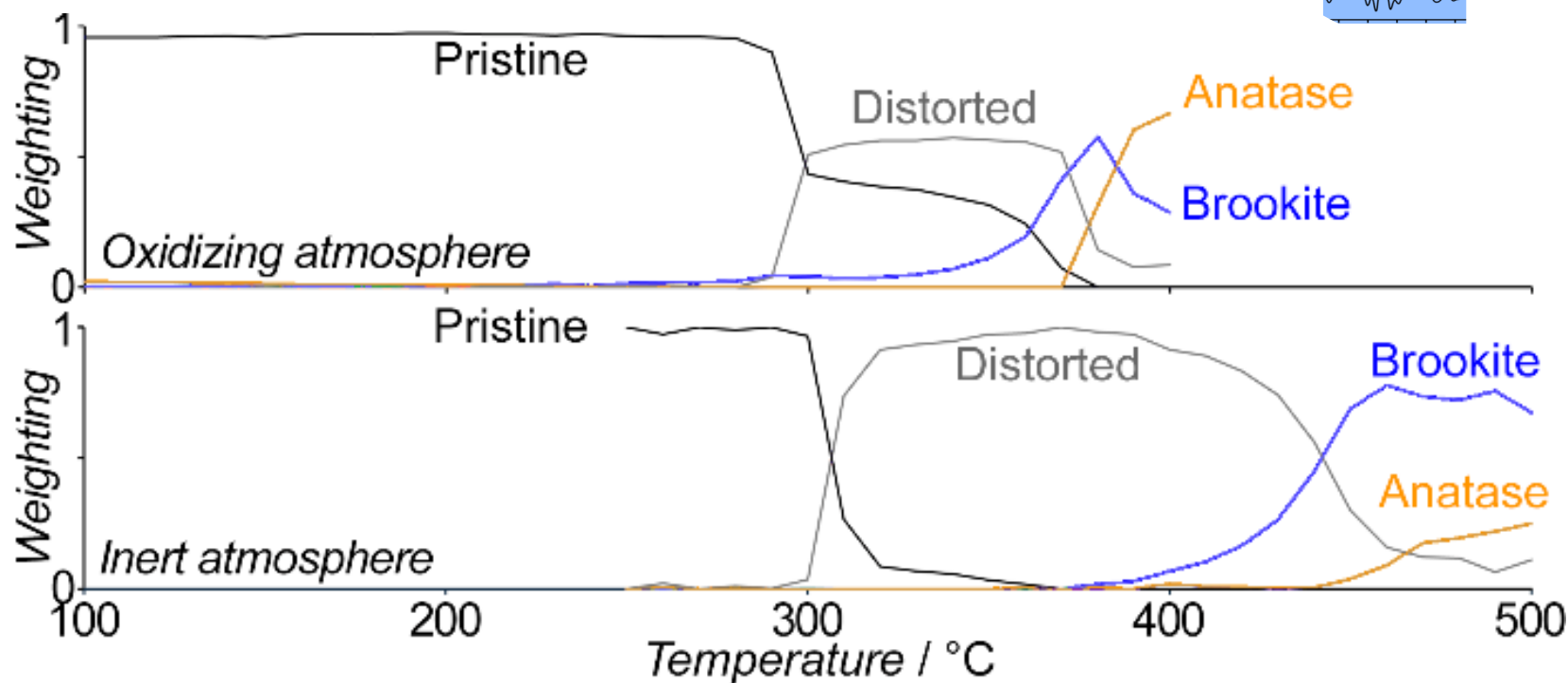
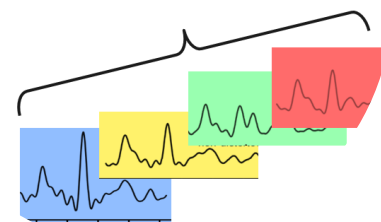


# Atmosphere enhances distortion AND brookite

Changing to inert atmosphere shifts  
MOF decomposition  
to higher T, favors a more complete distortion  
and increases brookite product.



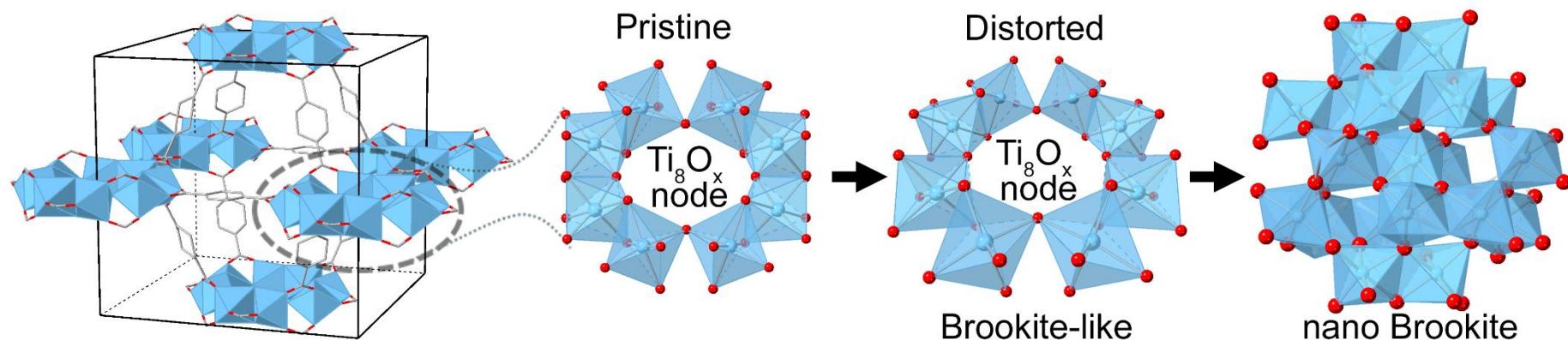
Weighting  $\times$  Components  
■ ■ ■ ■



# MOFs at the edge of stability

MOFs can serve as unconventional precursors for nanoparticles.

The preorganization of species within the metal-oxo nodes can guide the transformation to unusual nanoparticle polymorphs under mild conditions

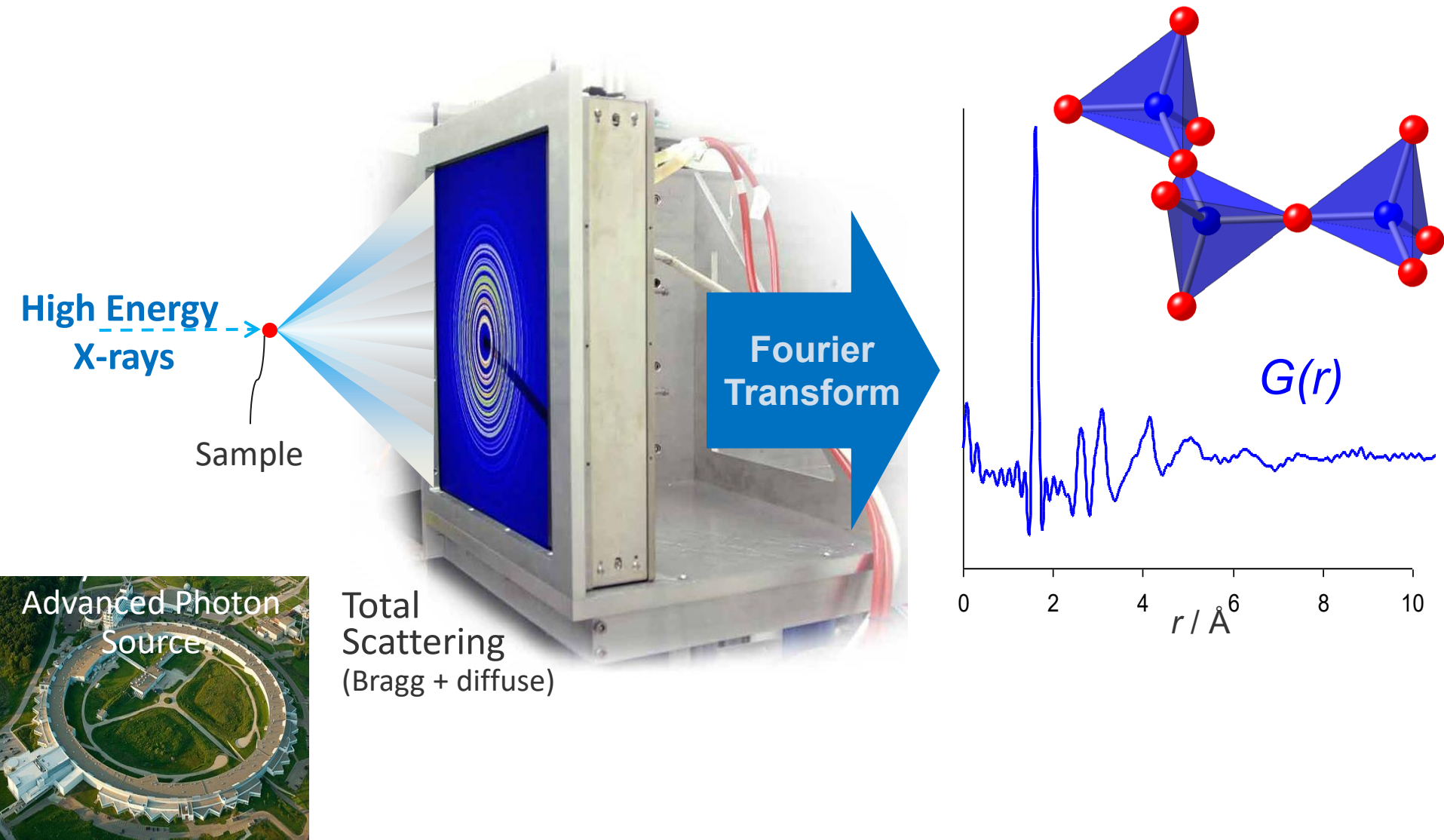


## Modern in situ PDF measurements



# Modern X-ray Pair Distribution Function measurements

A specialized synchrotron powder diffraction experiment





# A primer on integrating area detectors

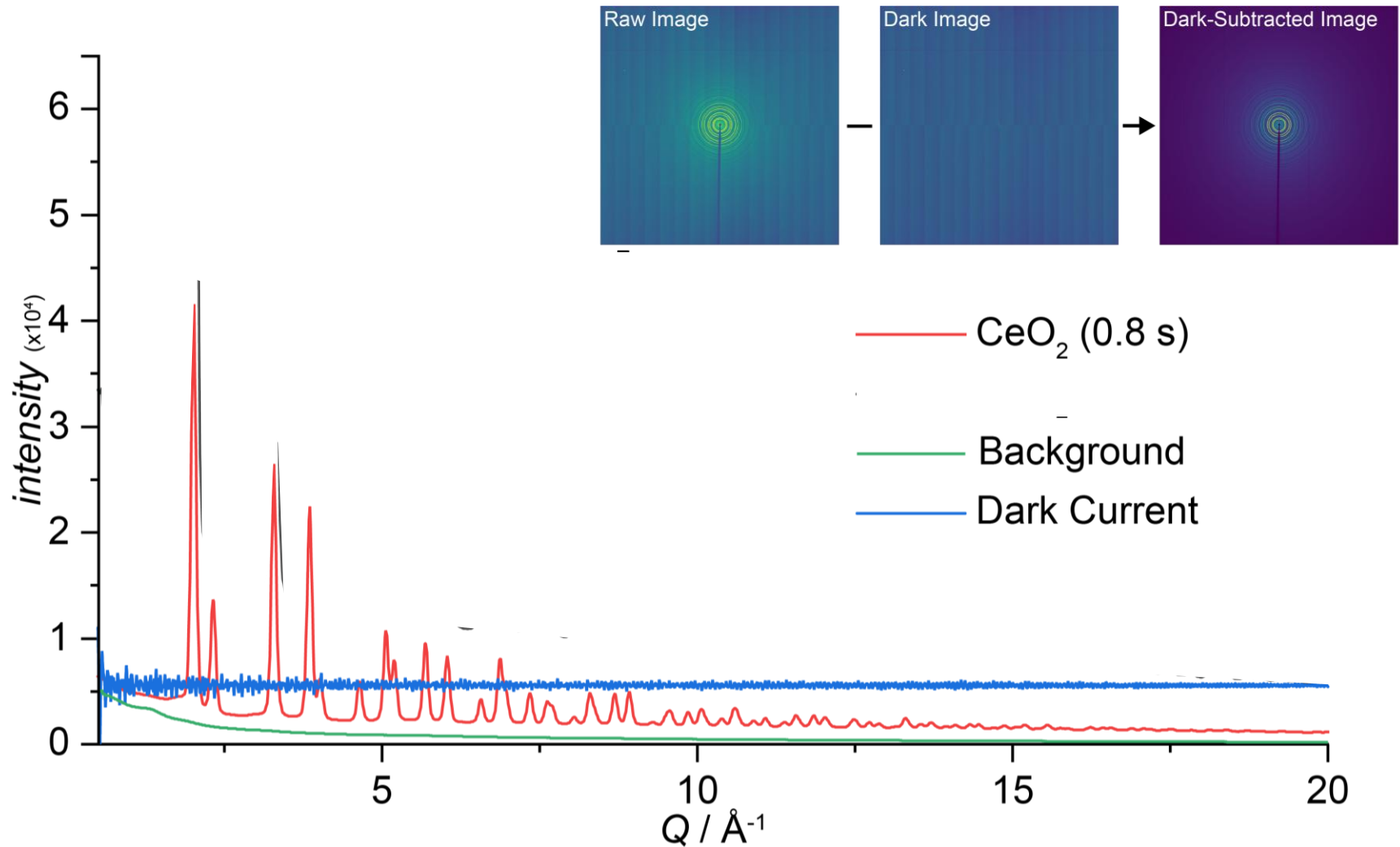
Amorphous Si-based area detectors such as the Varex/PerkinElmer can measure up to 64,000 counts per pixel.

This includes the *sample scattering* and a *dark baseline signal* originating from detector electronics.

Multiple detector read outs are summed to improve signal-to-noise

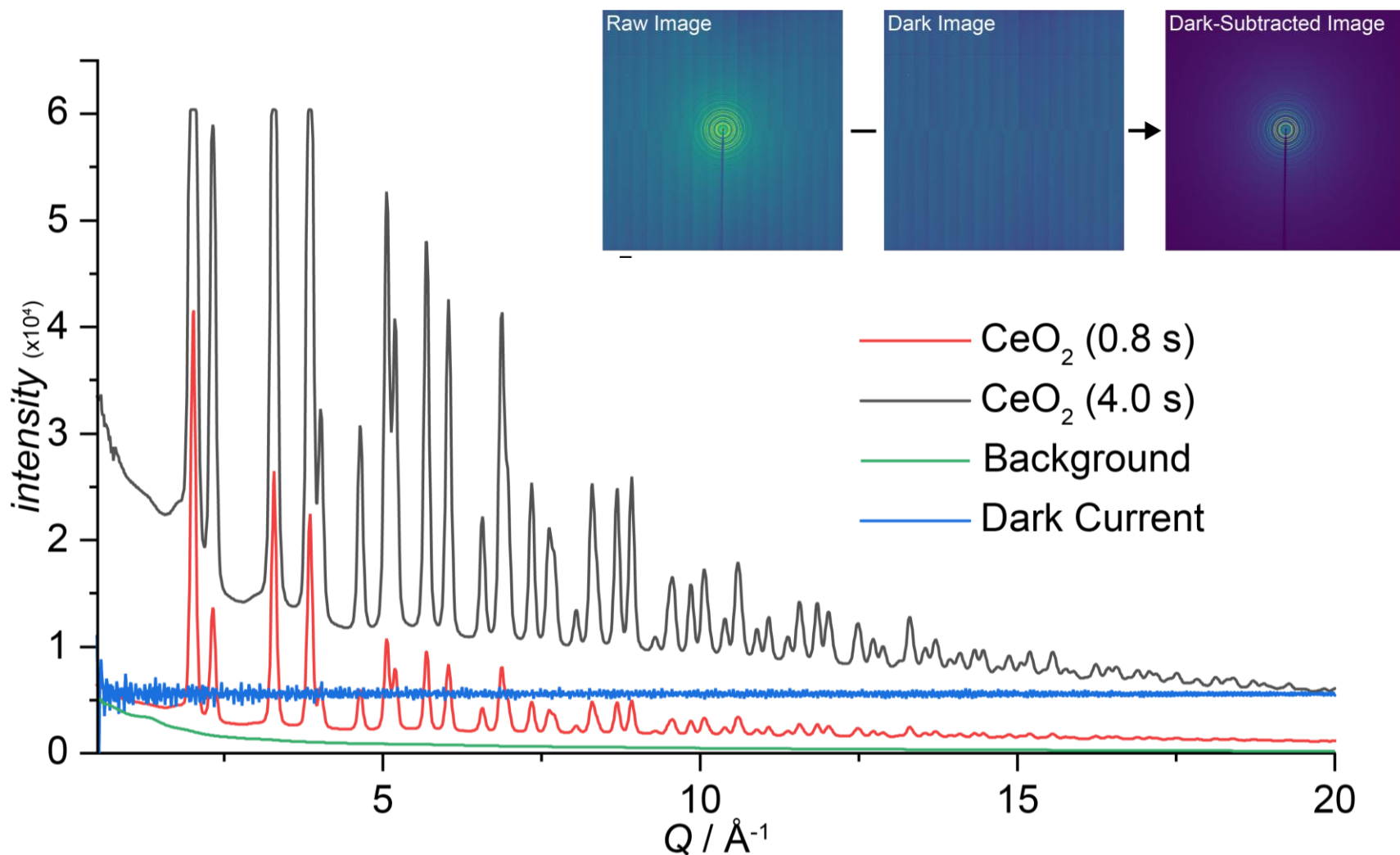


# Limited dynamic range of area detectors

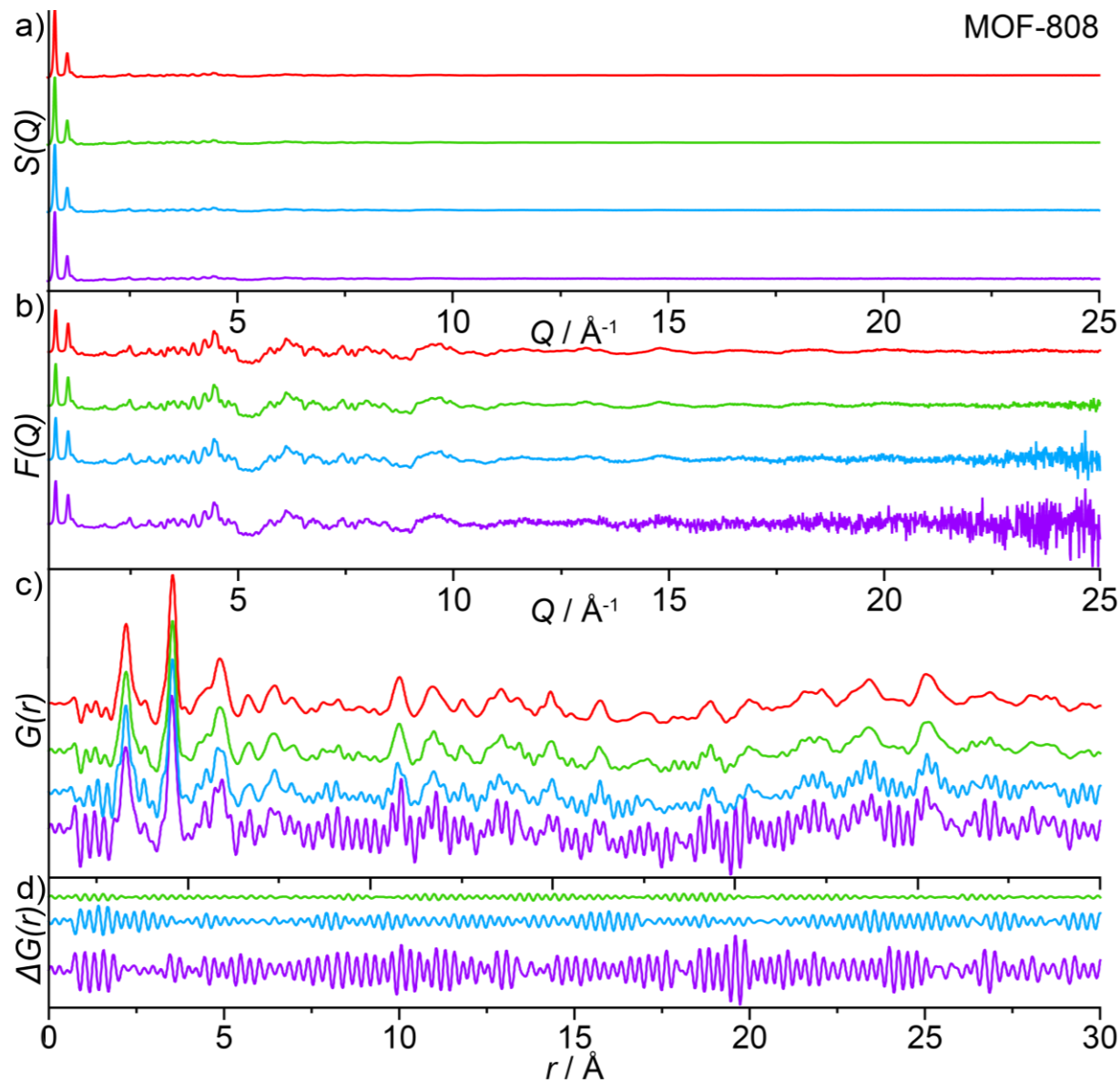


# Limited dynamic range of area detectors

Intensities above 64,000 counts are truncated

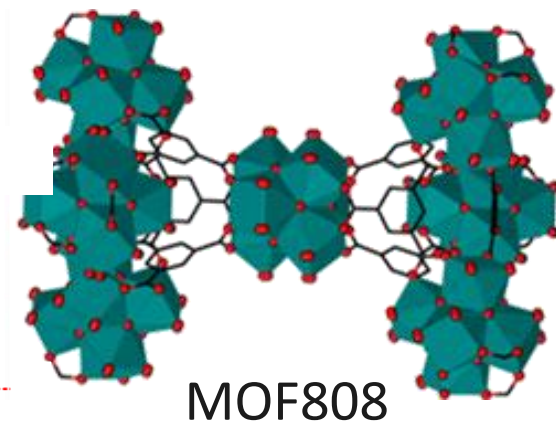
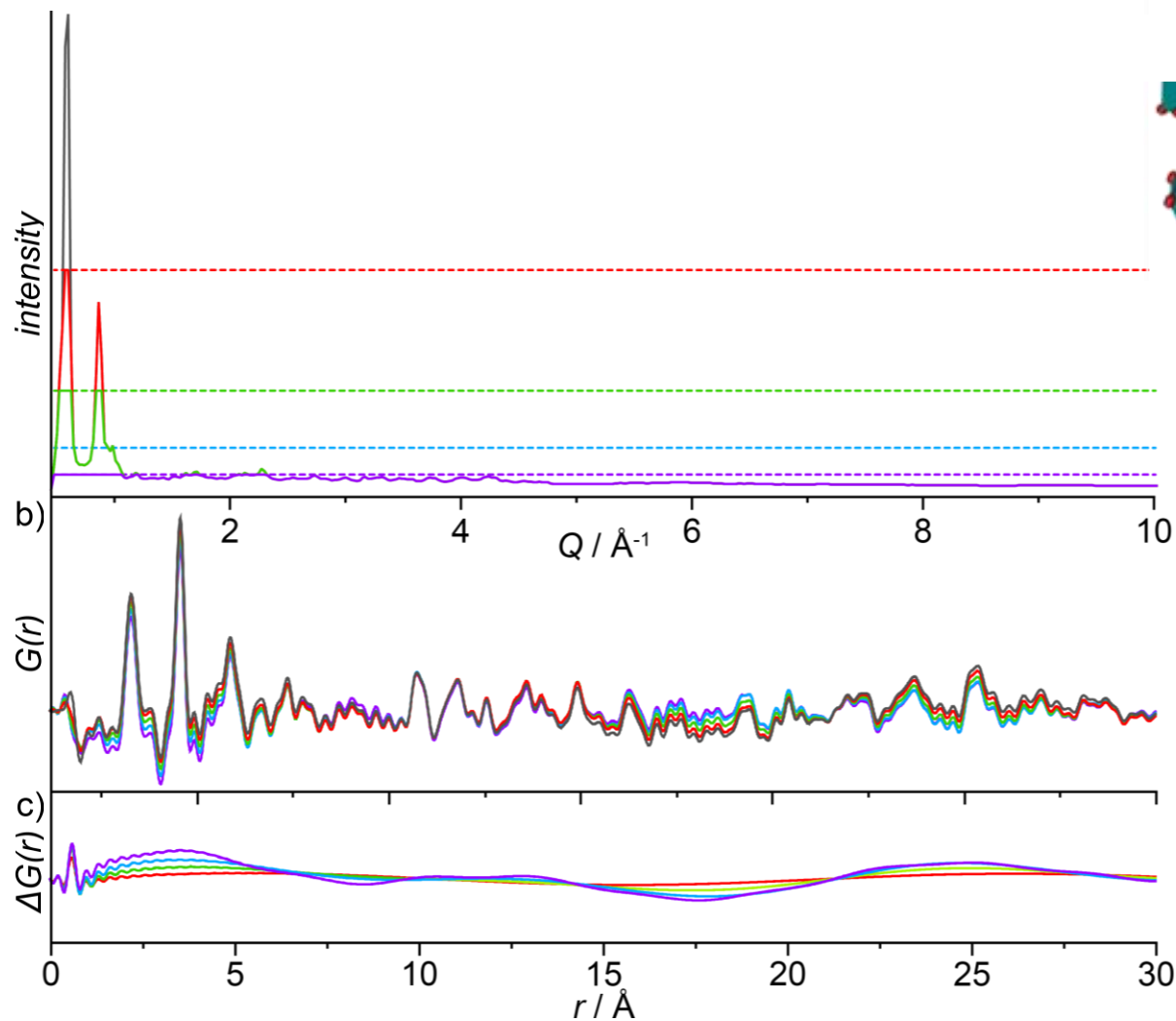


# Noise in the image is amplified in $S(Q)/G(r)$



# Peaks may become overexposed during *in situ* study

E.g. during crystallization

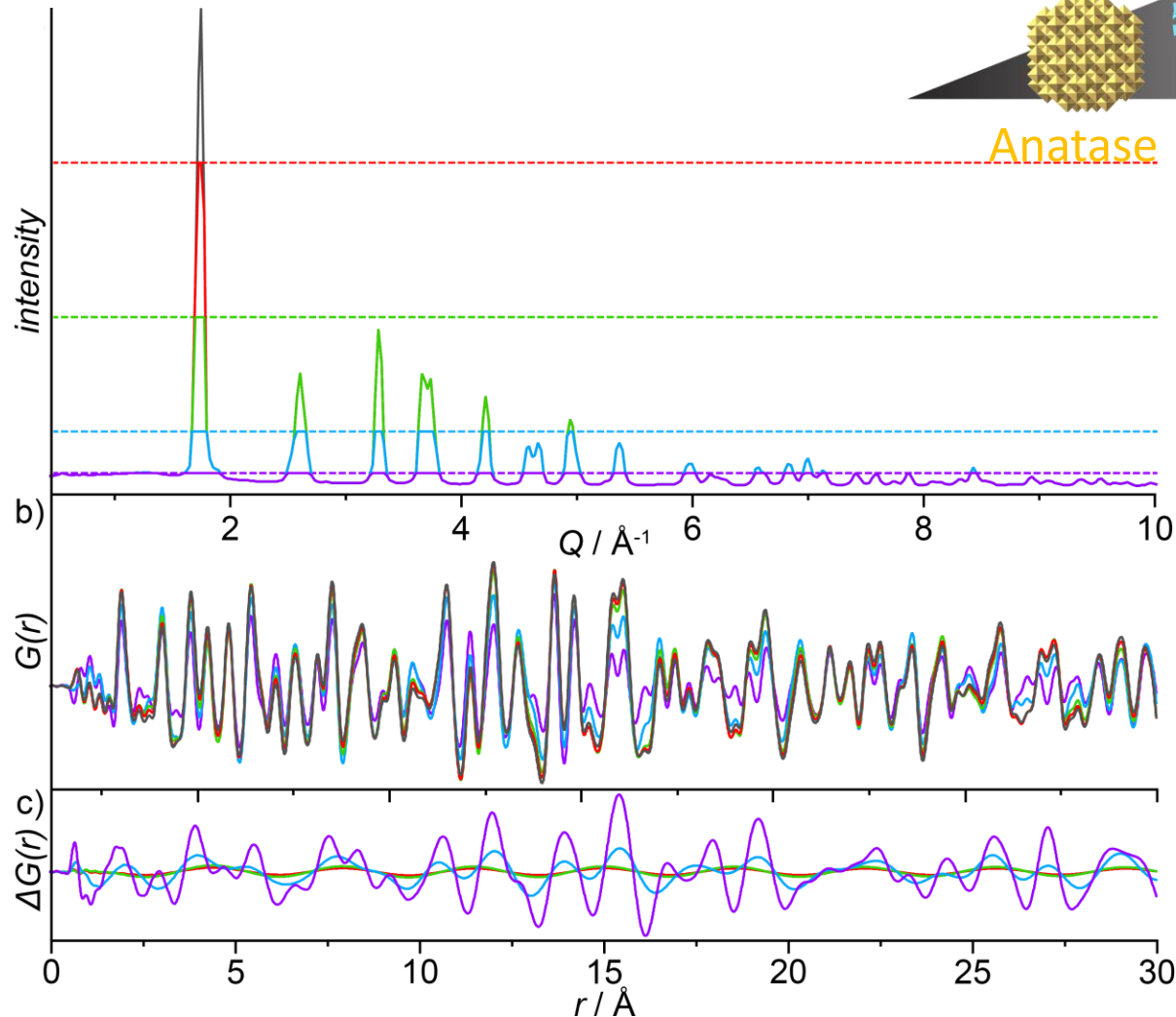
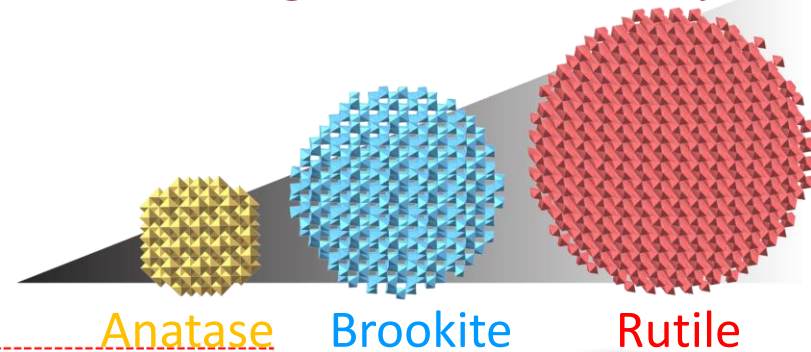


← Error from truncating  
peak intensities  
Lose porosity features



# Peaks may become overexposed during *in situ* study

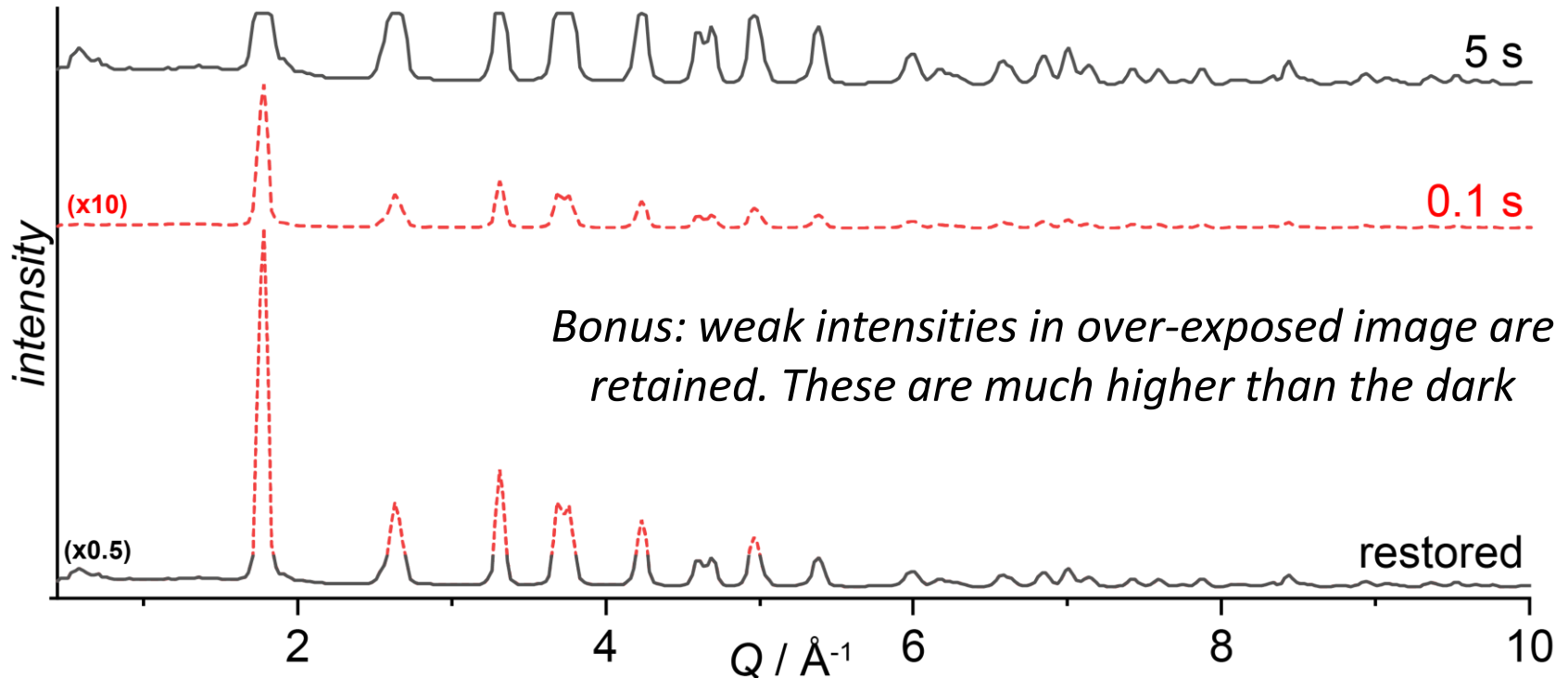
E.g. during crystallization



← Error from truncating peak intensities

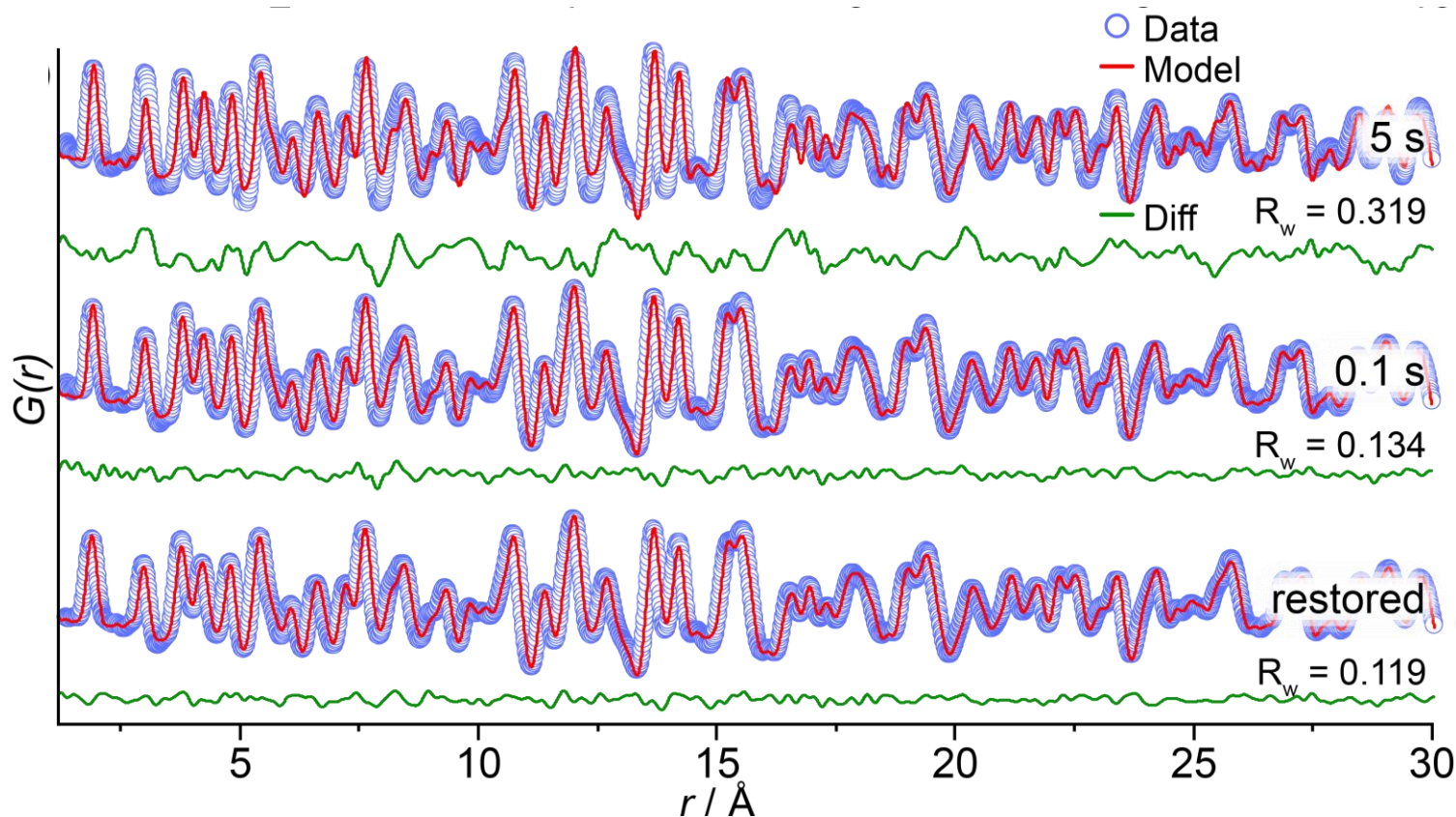
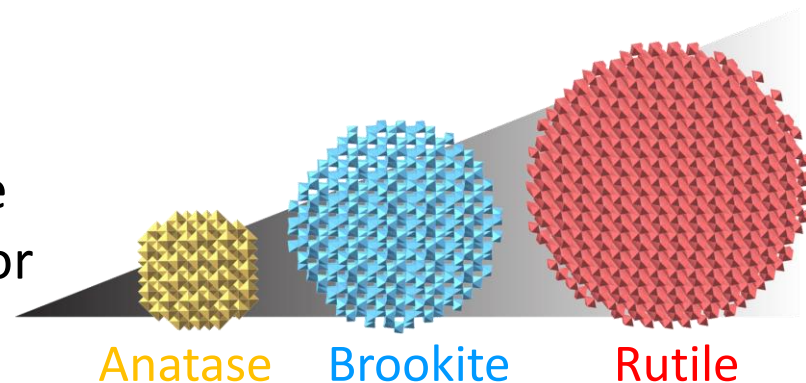
# Restoring over-exposed experimental data

*Scaling & merging data:* Substitute truncated intensity data by intensity data from a less exposed image, for which all the data remain on-scale



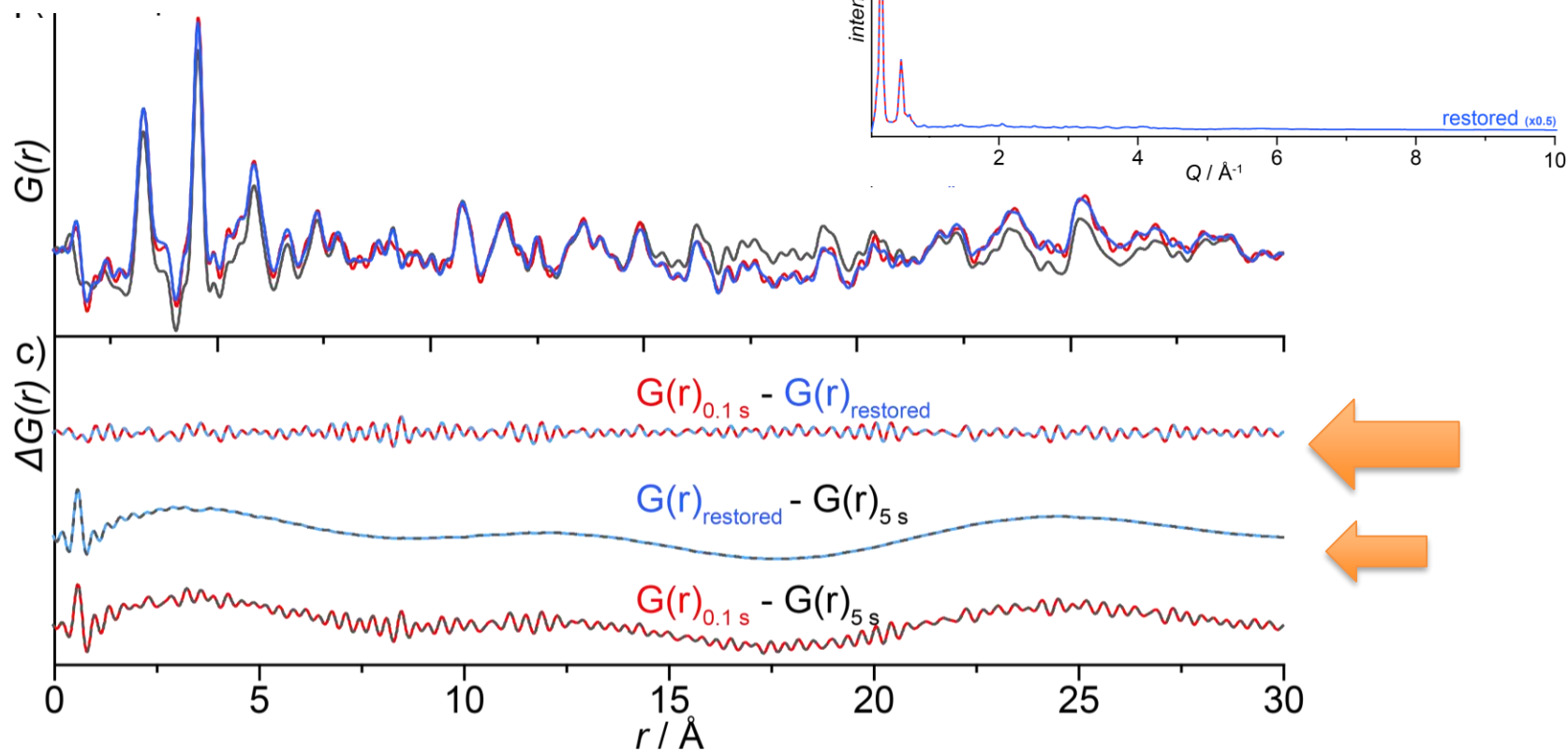
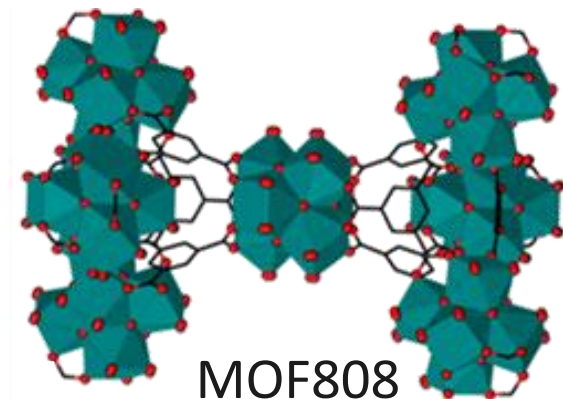
# Patched PDF data are better!

The gain in signal-to-noise ratio of the patched data lead to a lower  $r$ -factor for same effective exposure time



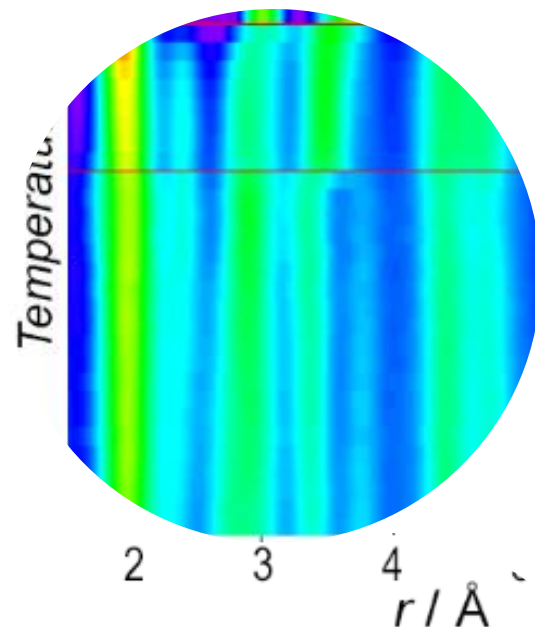
# Patched PDF data are better!

Propose this as a *deliberate strategy* for detailed (e.g. RMC) analysis, or weakly scattering/porous systems



# Summary

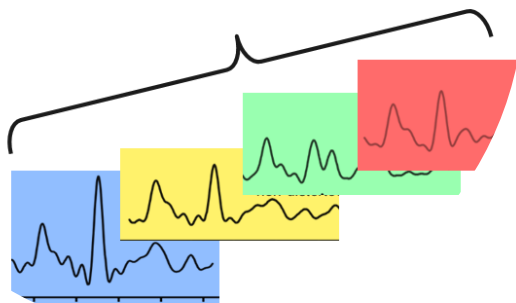
*In situ / operando studies* allow you to derive mechanistic insights from individual features, e.g. bondlengths, in the data. Independent of complex structural modeling



$$\begin{bmatrix} w \\ w \\ w \\ w \end{bmatrix} \times \begin{bmatrix} h & h & h & h & h \end{bmatrix} \approx \begin{bmatrix} v & v & v & v & v \\ v & v & v & v & v \\ v & v & v & v & v \\ v & v & v & v & v \end{bmatrix}$$

Weighting  $\times$  Components

■ ■ ■ ■



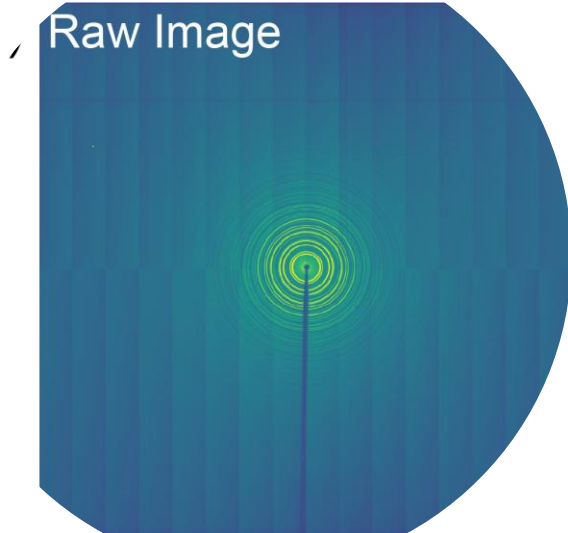
Multivariate data analytics such as NMF & PCA are powerful tools to help simplify and interpret the interpretation of large volumes of in situ data



# Summary



Designing *new in situ reactors and operando sample environments* can be critical to enable materials to be probe under relevant reaction/operating conditions and timescales



Sometimes, *making mistakes and breaking the rules* can lead to a better way to do measurements



A NEXT **GENERATION**  
SYNTH**ESIS** CENTER



U.S. DEPARTMENT OF  
**ENERGY**

Office of  
Science



Fernando Uribe-Romo



Omar Farha



Joe Hupp



**Stony Brook**  
**University**

Use of the Advanced Photon Source, an Office of Science User Facility operated for the U.S. Department of Energy Office of Science by Argonne National Laboratory, were supported by the U.S. Department of Energy under Contract No. DE-AC02-06CH11357.