

*Structural Fingerprints of Plasticity:
Linking the Shear Transformation Zone Size to
Pair Distribution Functions in Metallic Glasses*

Valeria Lemkova

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Materials Science and Methods & Metallic Materials
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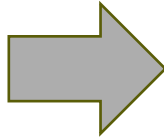
What does a (material)scientist do in research?

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Create a material

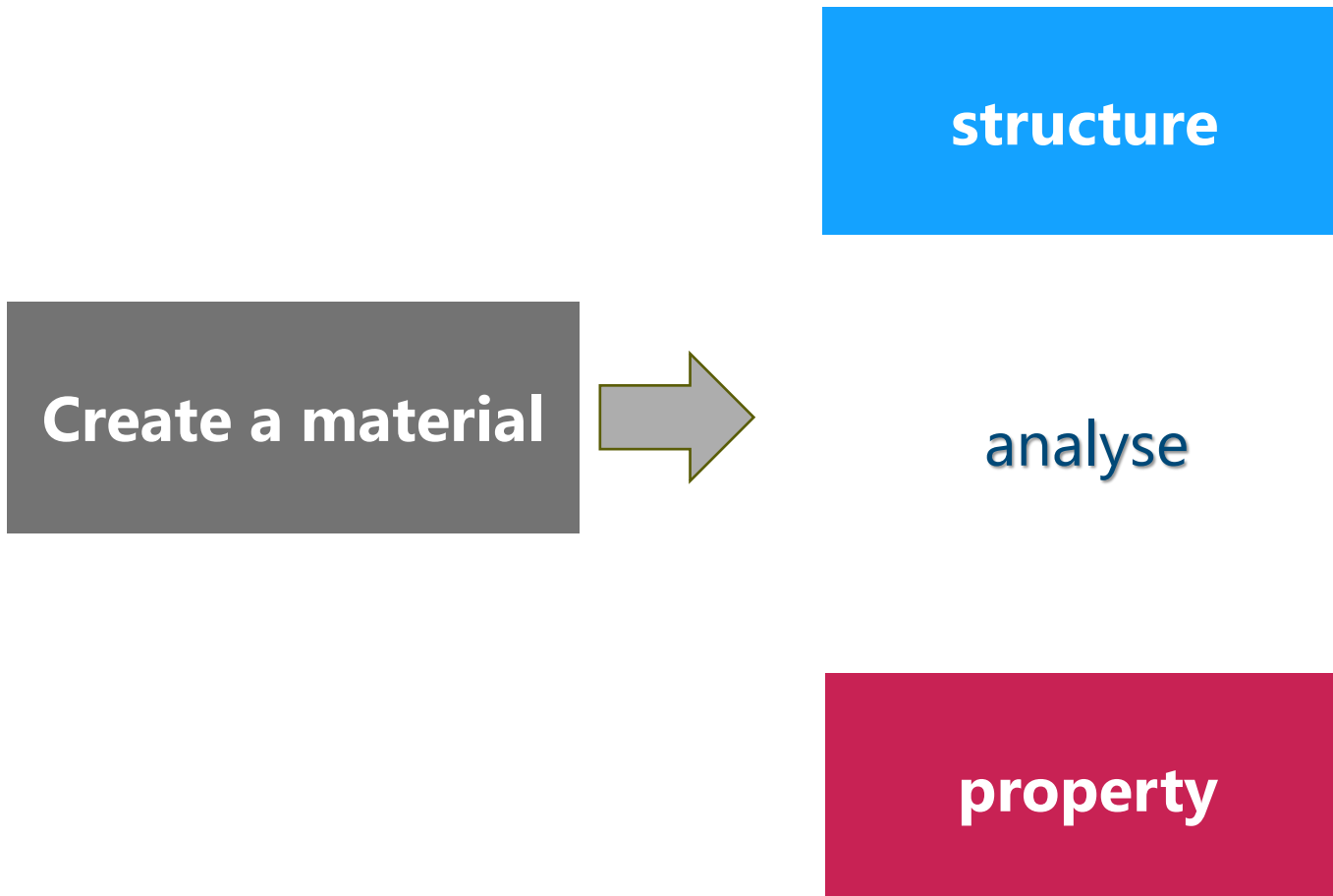
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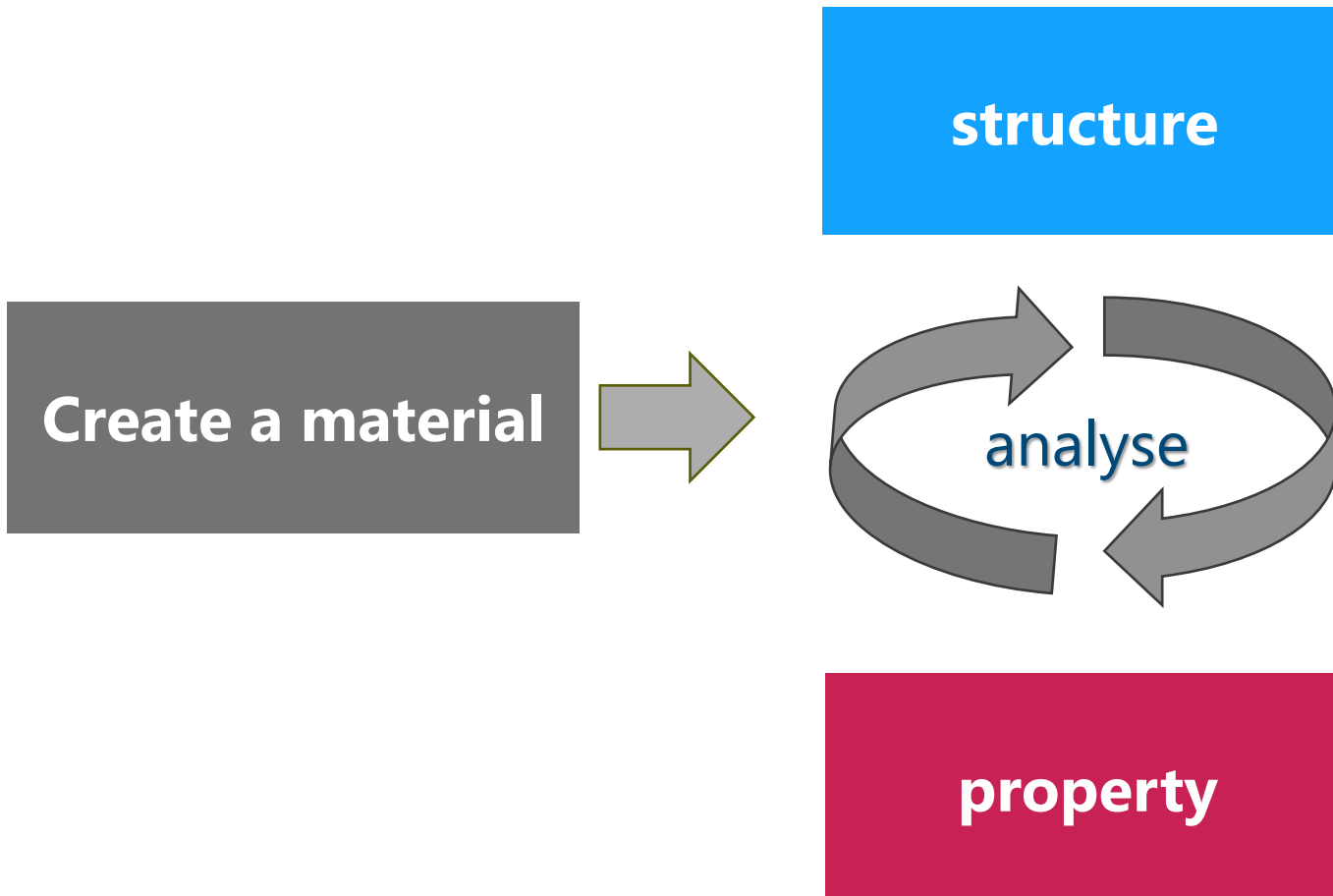


analyse

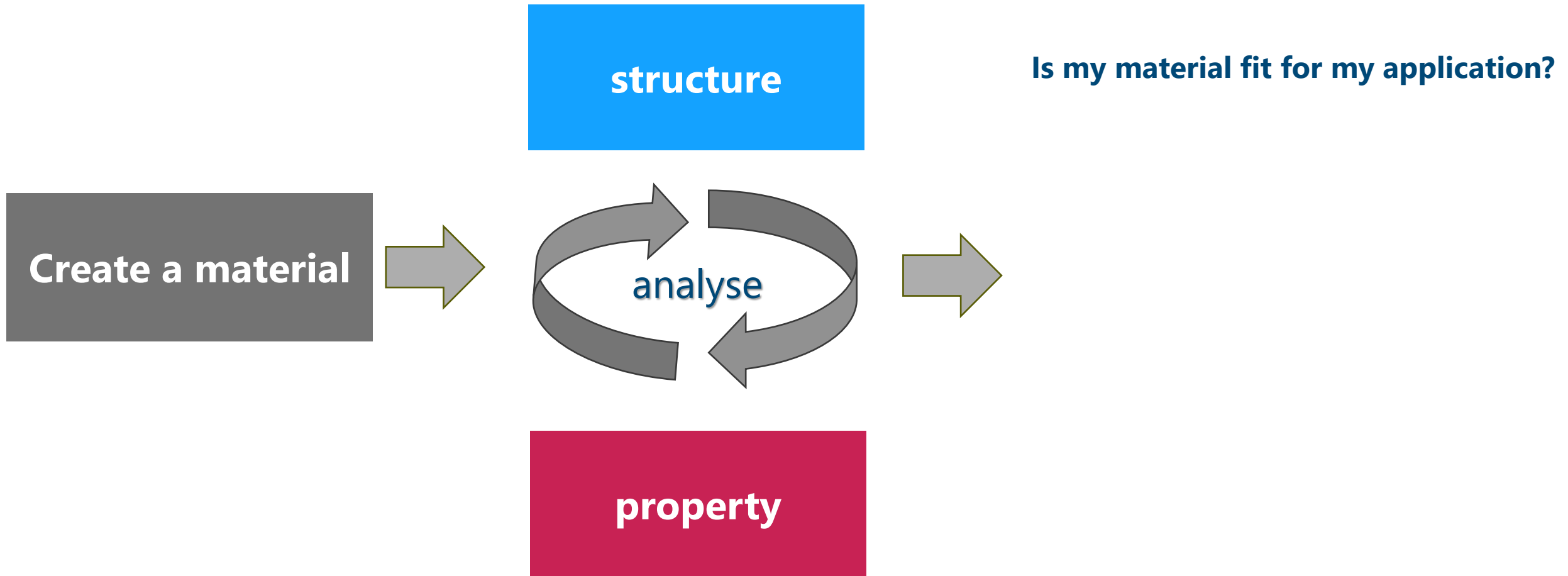
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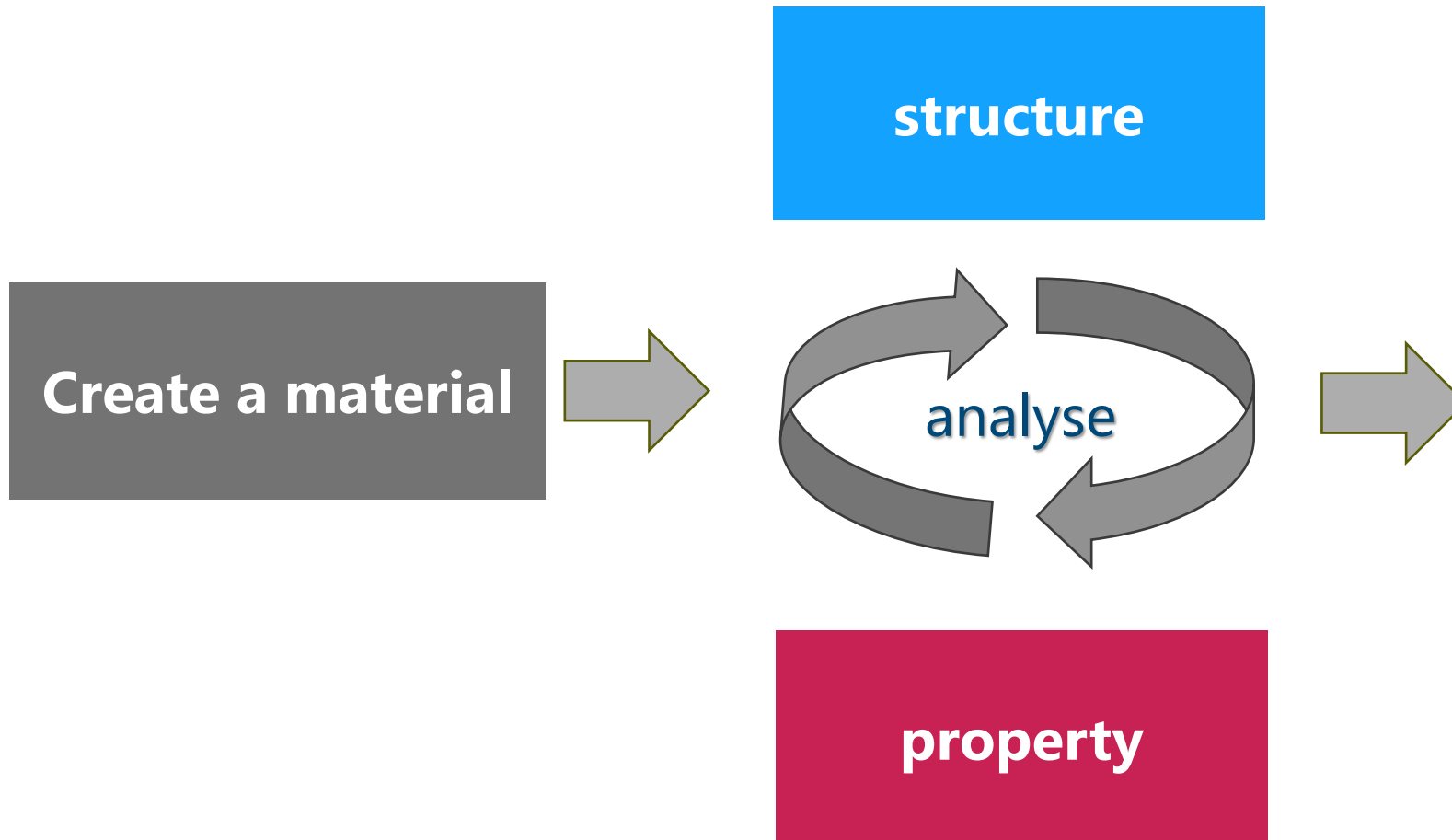
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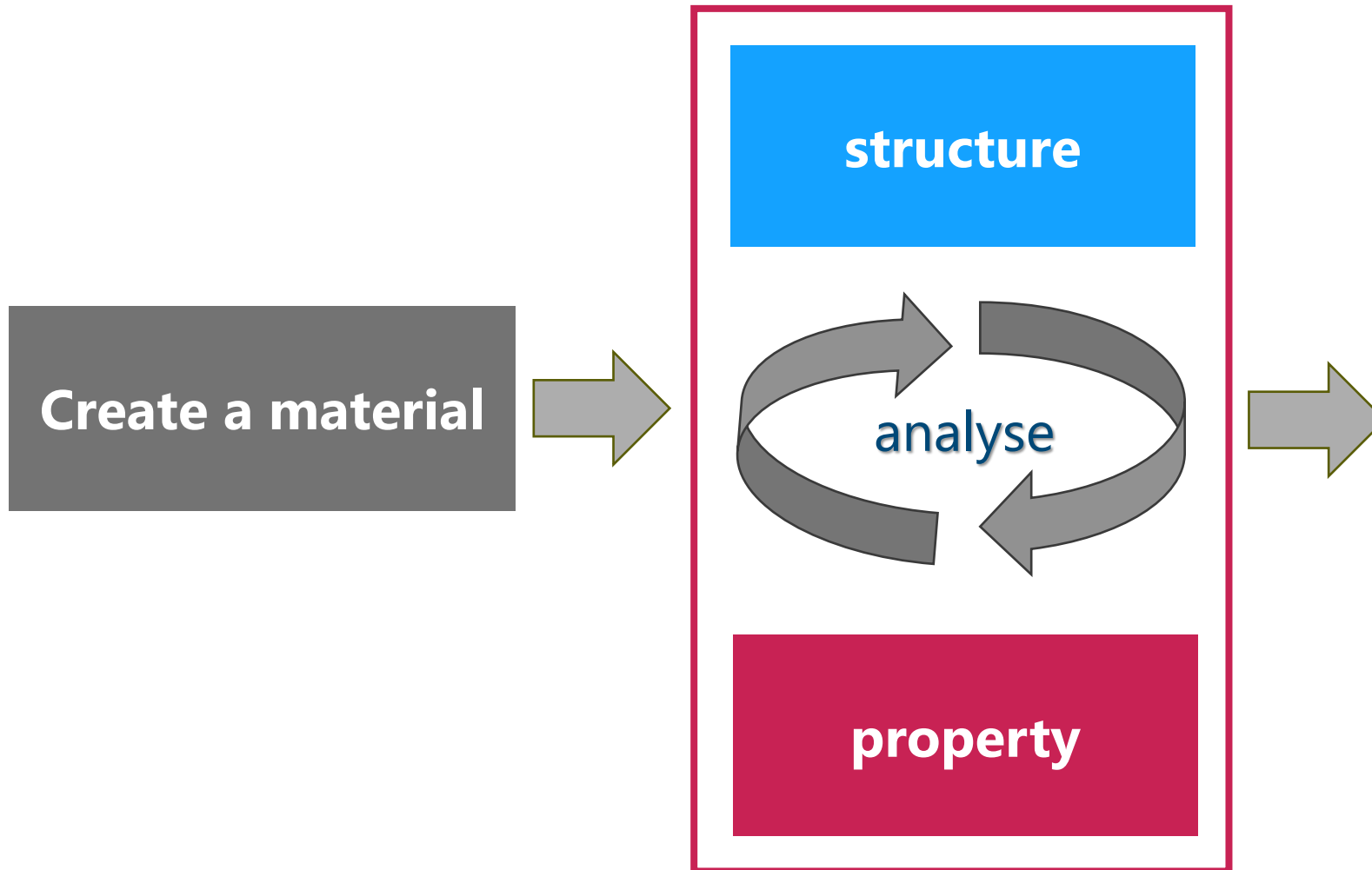


Is my material fit for my application?



@<https://www.reuters.com/lifestyle/two-giant-rubber-ducks-debut-hong-kong-bid-drive-double-happiness-2023-06-09/>

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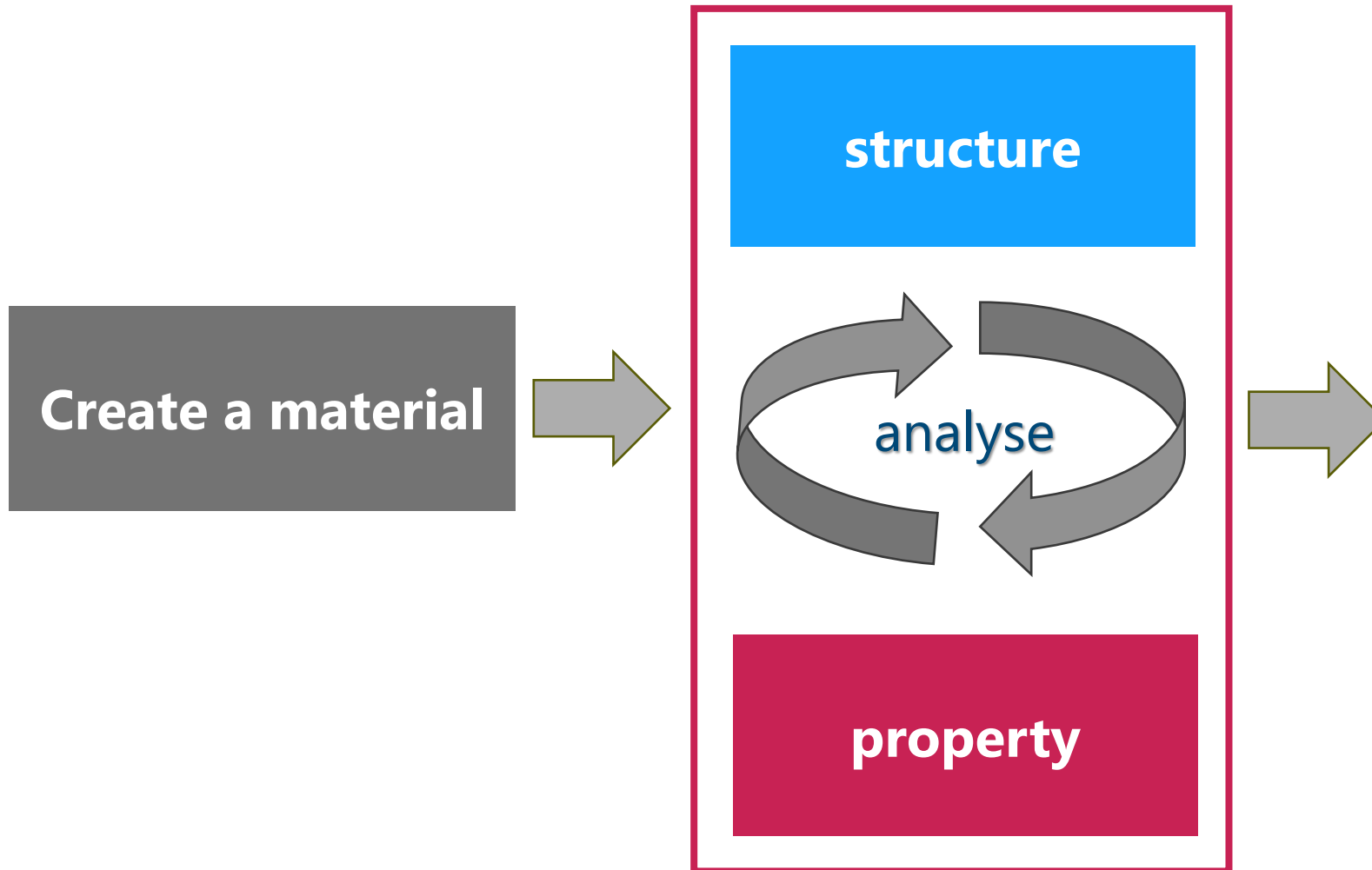


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Can I predict **mechanical properties** from the **structural fingerprint**?

Motivation & Objectives

Why is understanding structure–property relationships in metallic glasses important?



Motivation & Objectives

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BMG = Bulk Metallic Glass

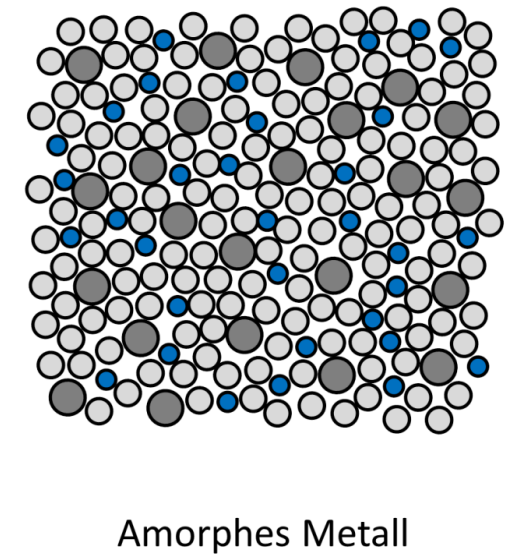
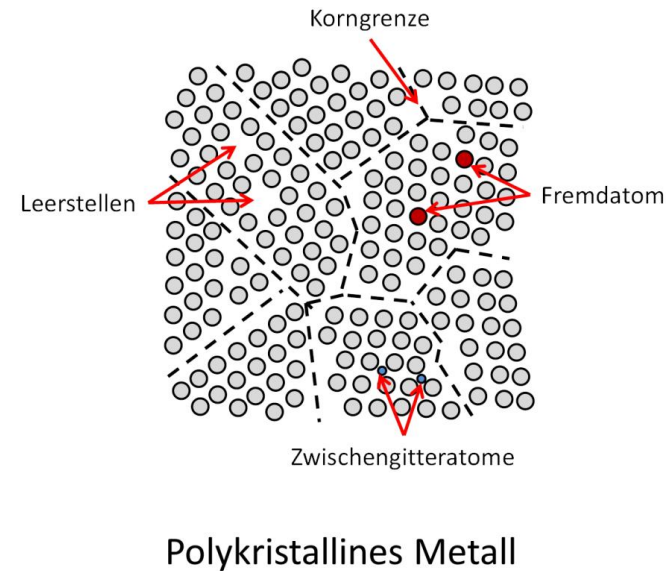
*amorphous metal that can
be cast to at least 2 mm*

Motivation & Objectives

Why is understanding structure–property relationships in metallic glasses important?

What does amorphous mean?

no long-range order (**LRO**) **but** some middle-range order (**MRO**) and short-range order (**SRO**)



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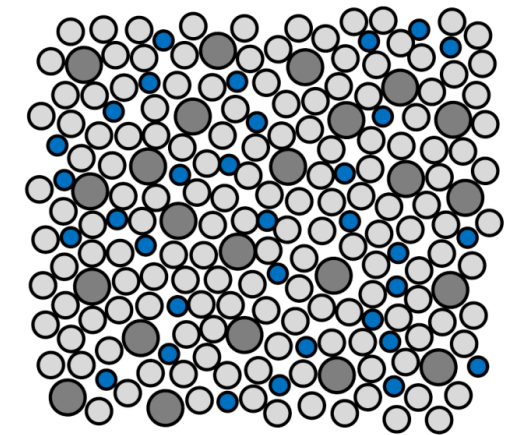
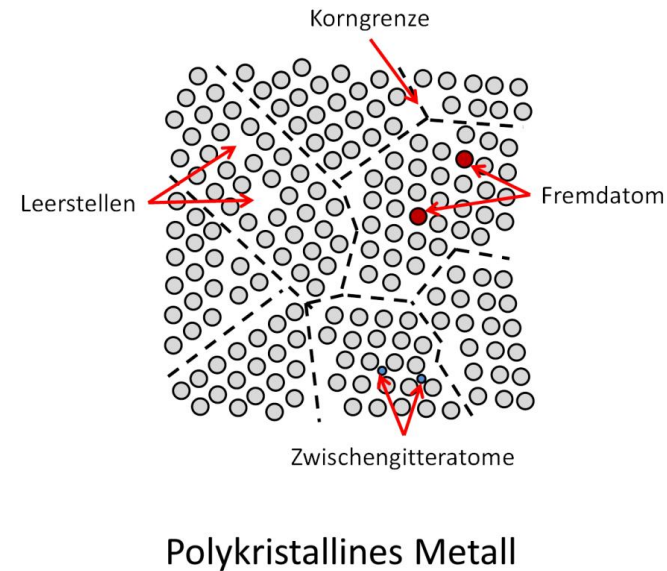


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Nucleation must be suppressed

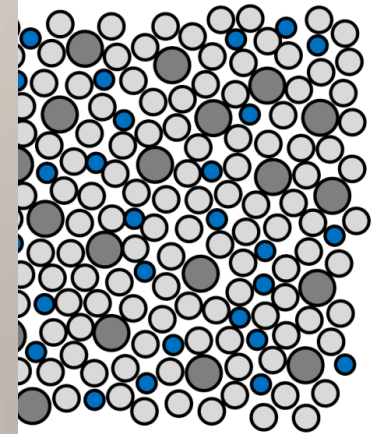
1. Atoms of significantly different sizes
2. Crystallization is energetically unfavorable



Motivation & Objectives

@<https://www.heraeus-amloy.com>

Why is



amorphes Metall

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bei suppressed

Steel BMG Ti

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Properties

Härter als Stahl – Elastisch wie Kunststoff

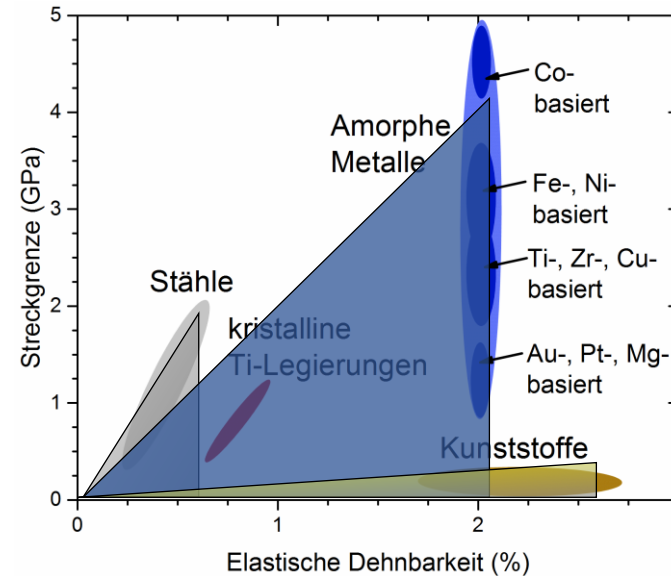
Harder than steel – As elastic as plastic

Motivation & Objectives

Why is understanding structure–property relationships in metallic glasses important?

Properties

- **High elastic limit:** up to **2%** (vs. $\sim 0.2\%$ in crystalline metals)
- **High strength:** up to **4.5 GPa**, approaching theoretical limits
- **High hardness & wear resistance:** up to **900 HV5**
- **Low energy dissipation:** excellent damping behavior
- **High elastic energy storage:** large area under the stress–strain curve



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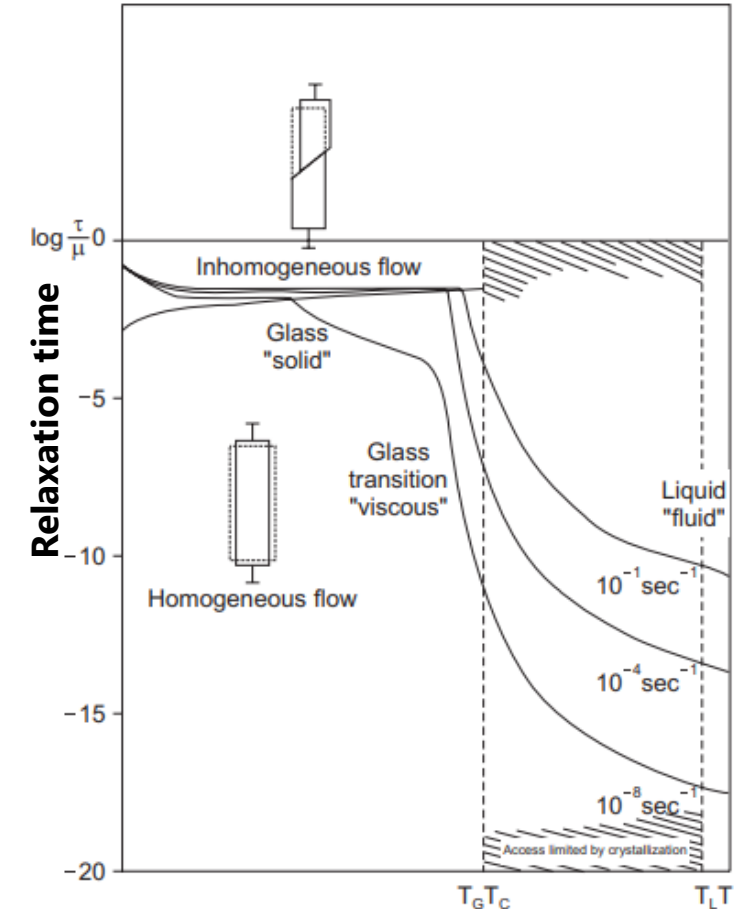
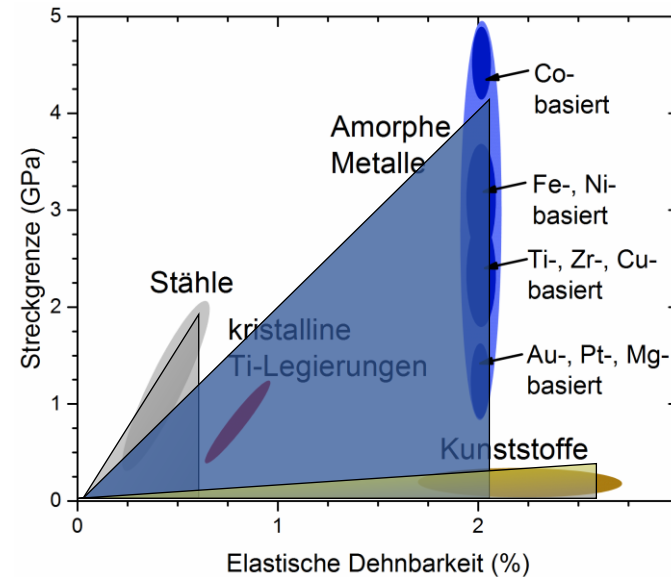
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@Spaepen (1977) – A microscopic mechanism for steady state inhomogeneous flow in metallic glasses

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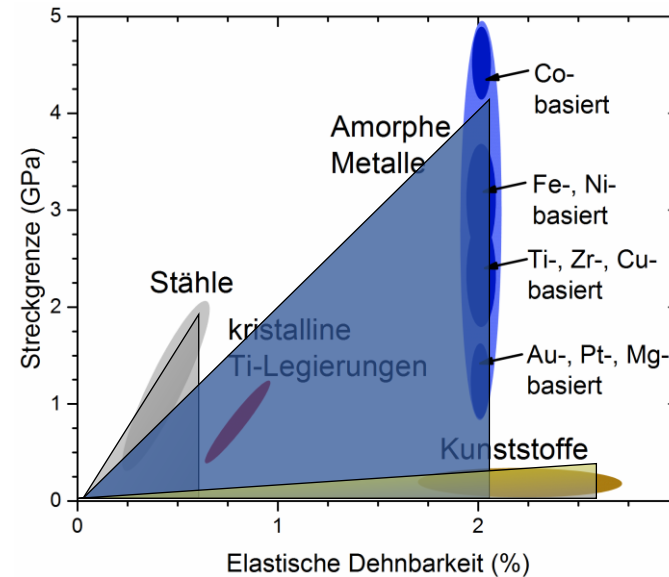
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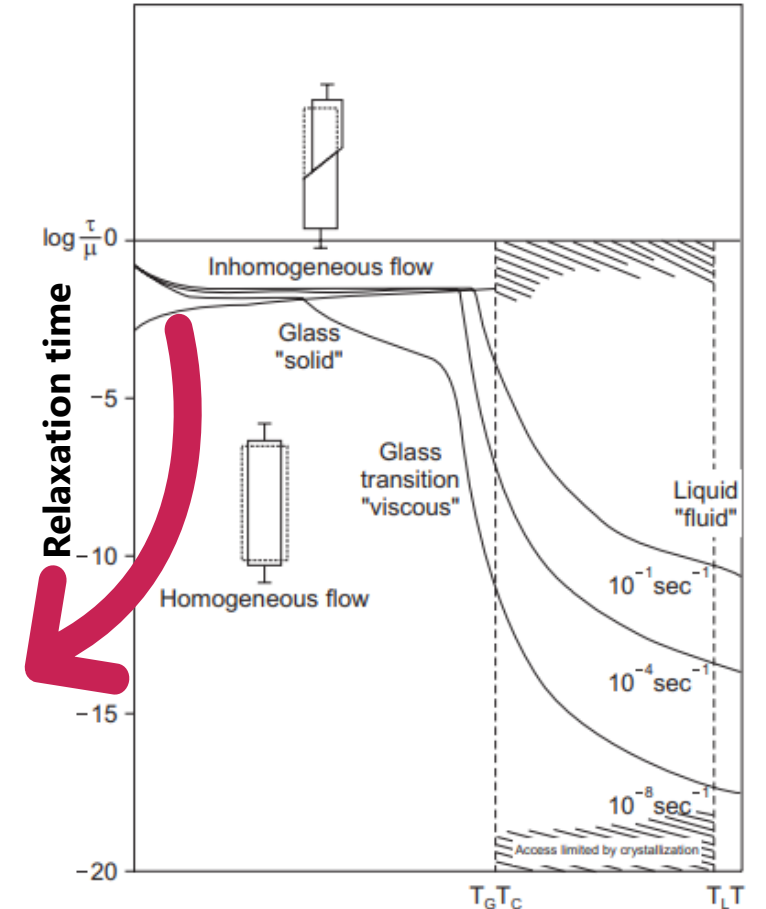
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Inhomogeneous flow:

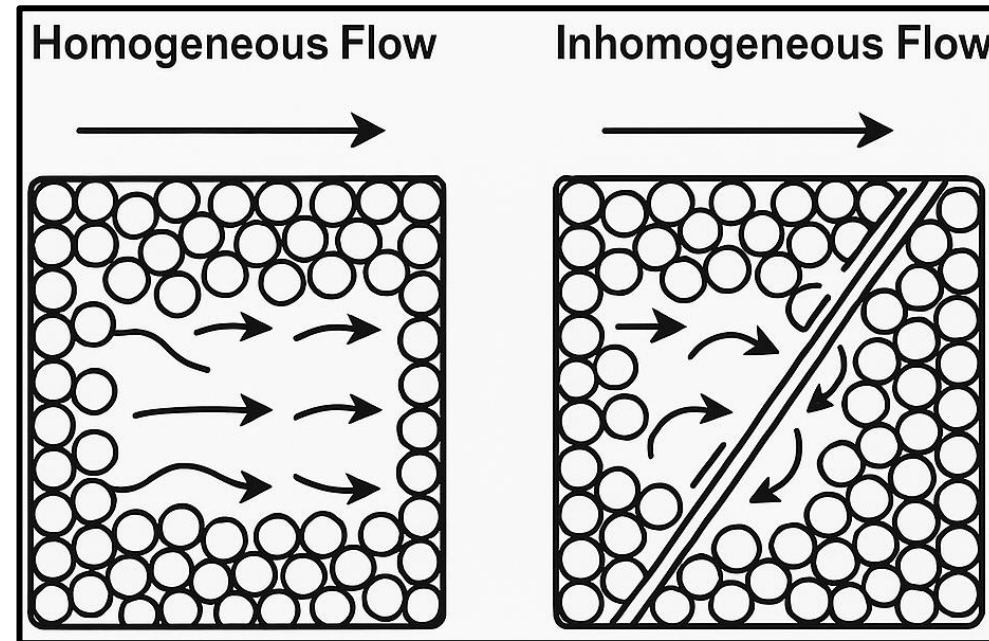
plastic deformation is **not** evenly distributed throughout the material



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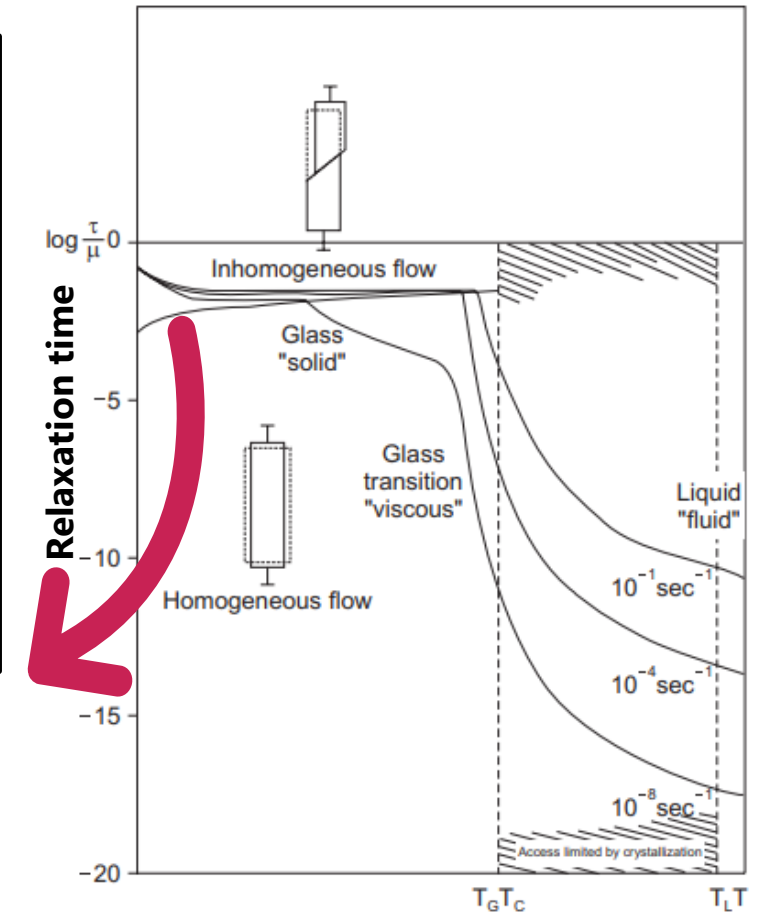
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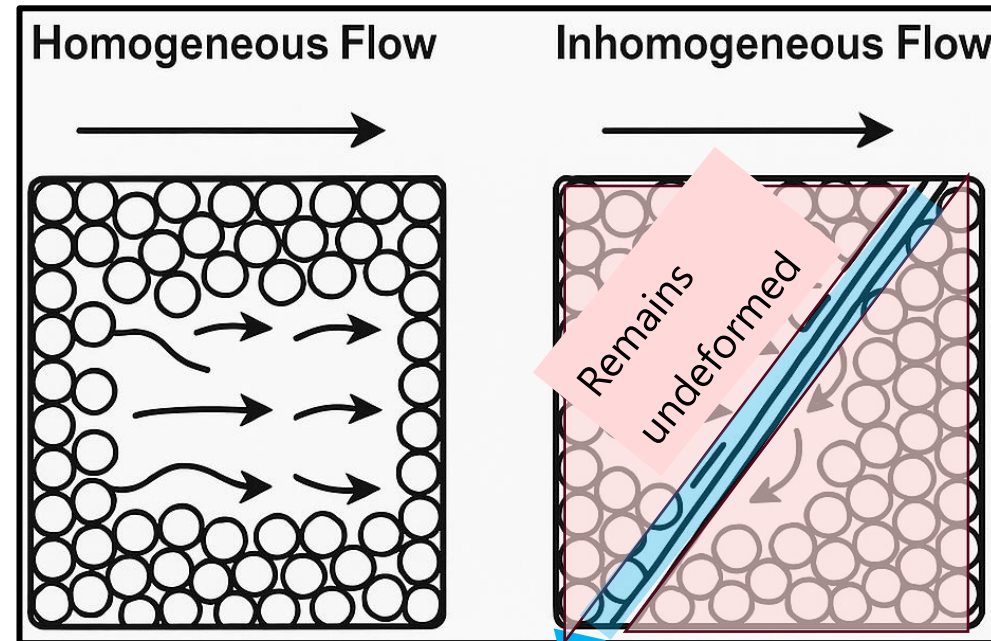
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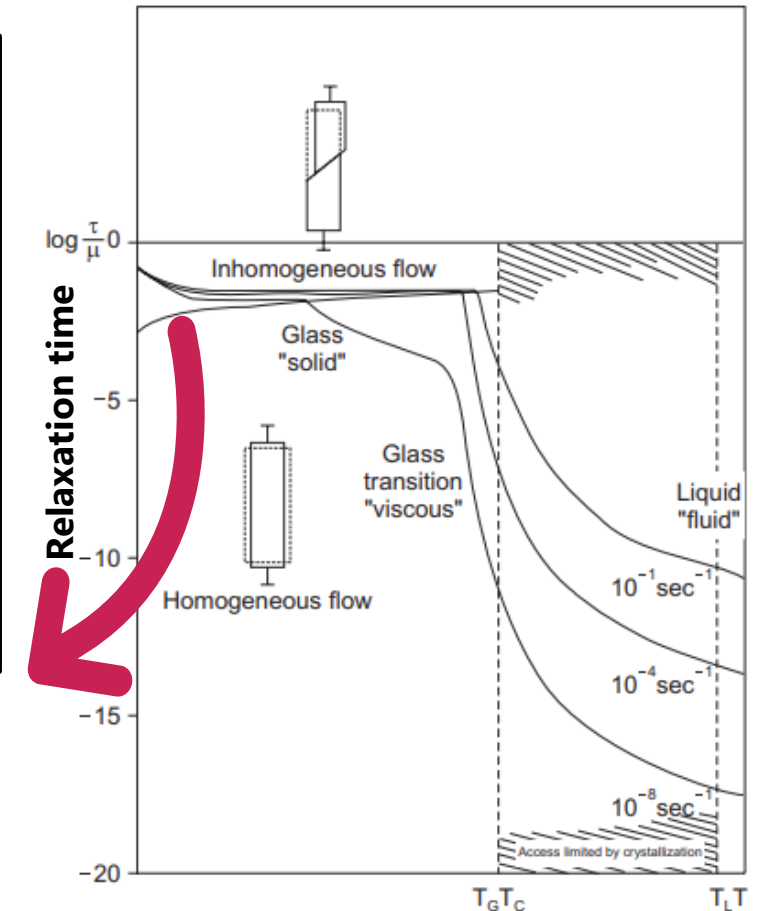
Localized region:

Shearbands form suddenly and locally

Brittle-like failure

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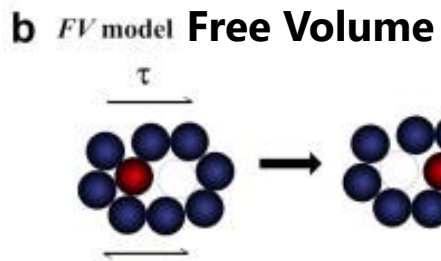
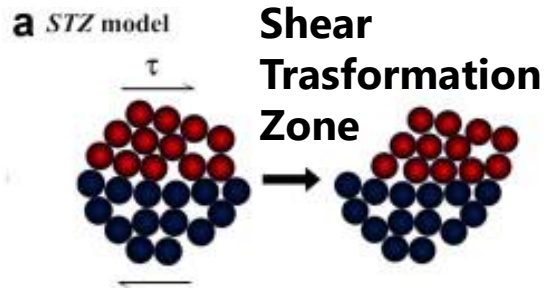
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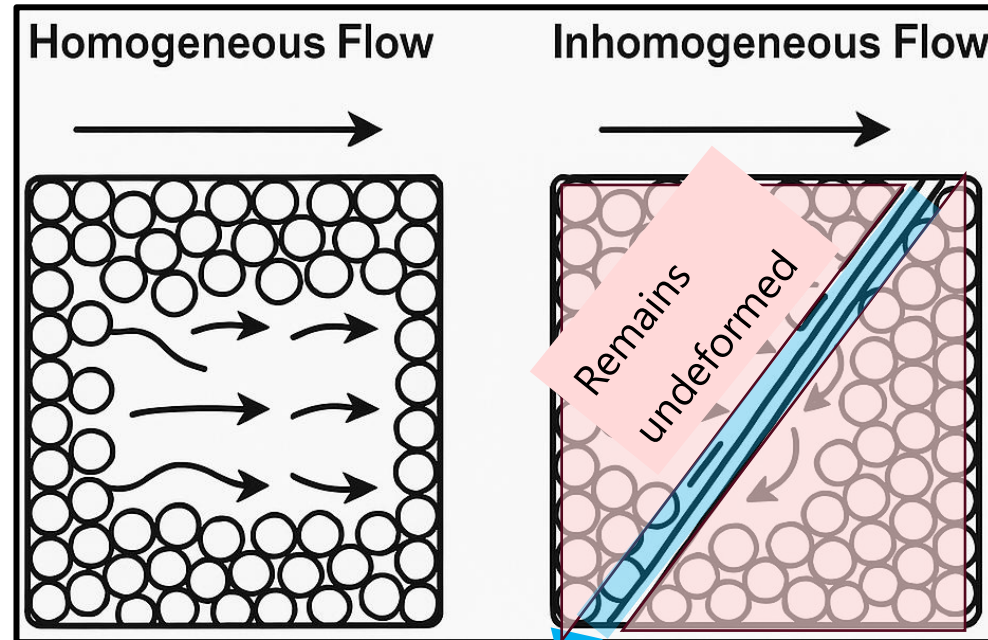
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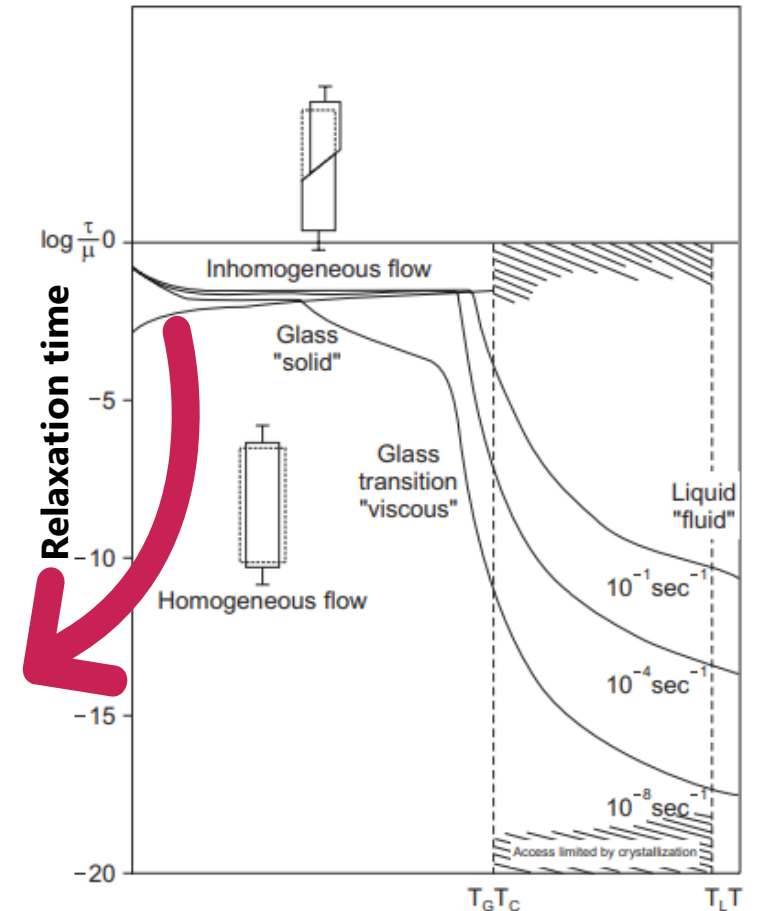
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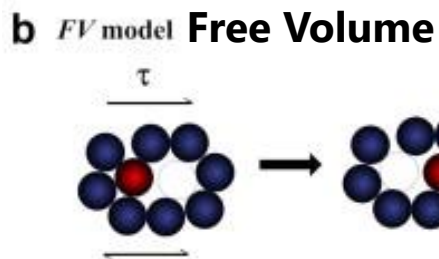
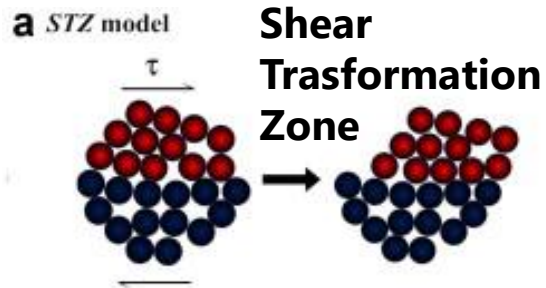
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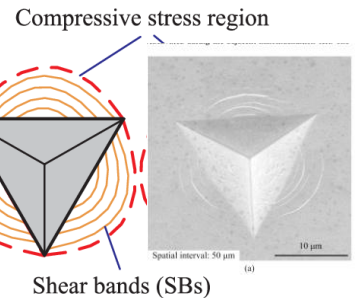
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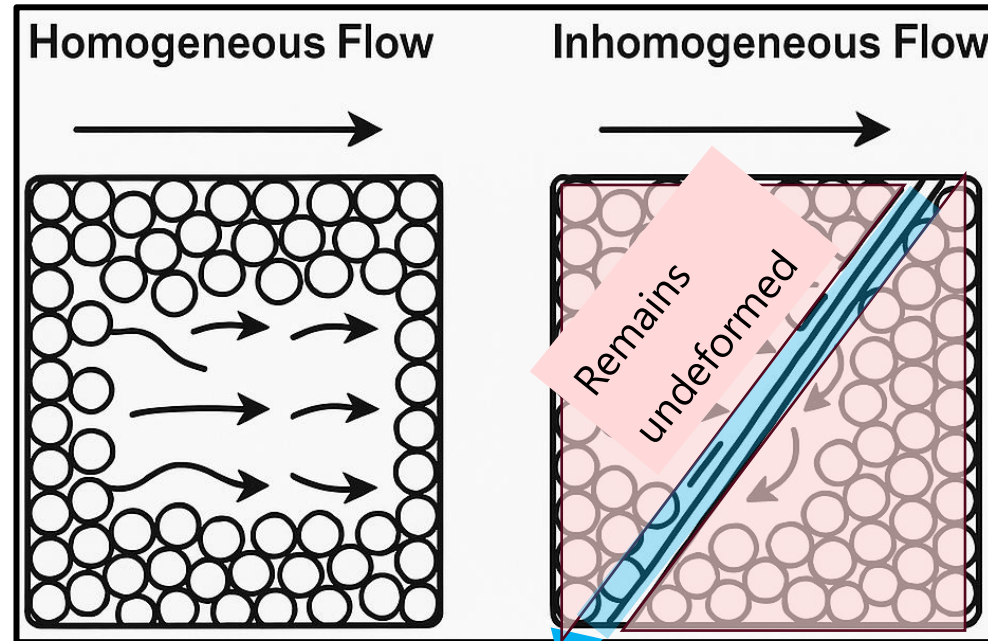
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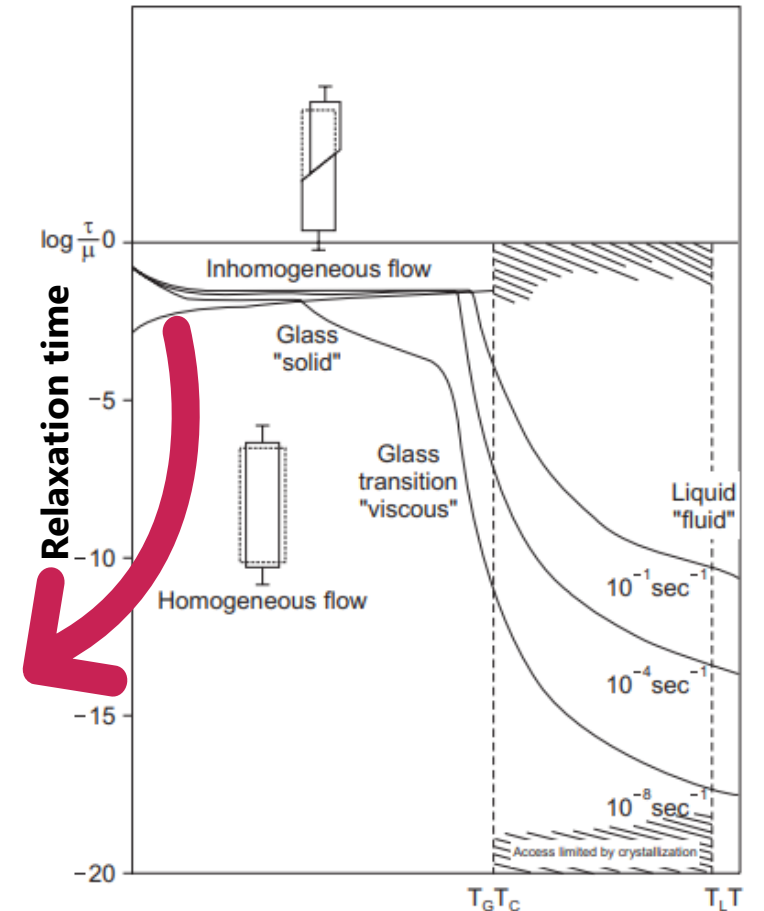
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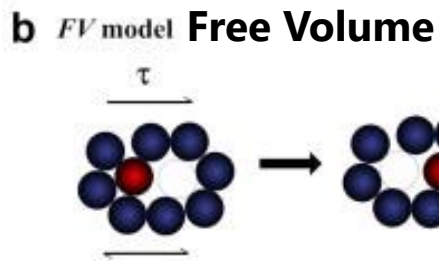
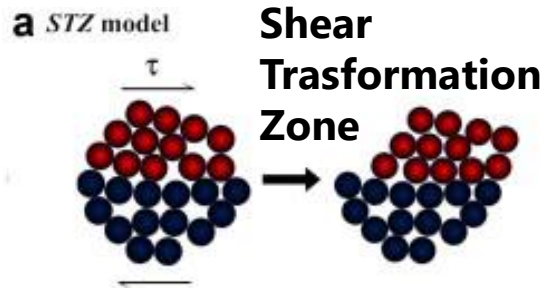
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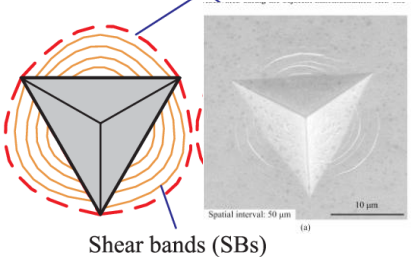
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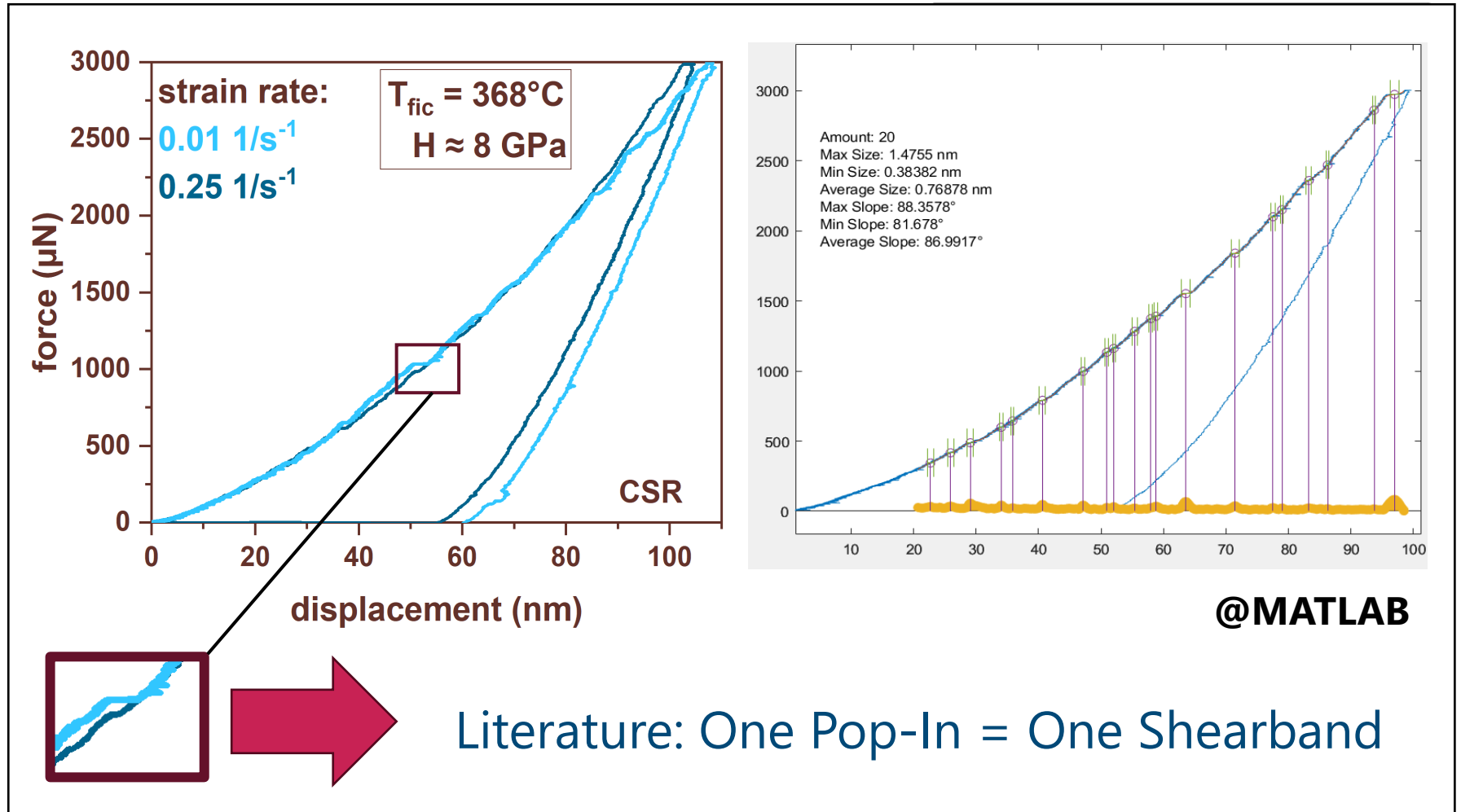
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Compressive stress region



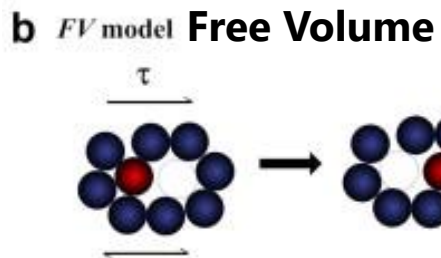
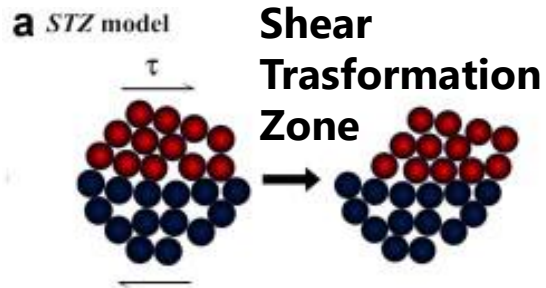
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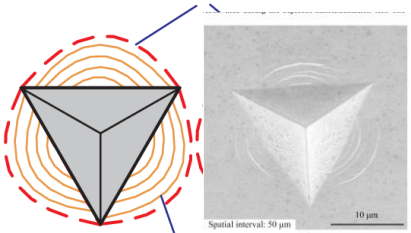
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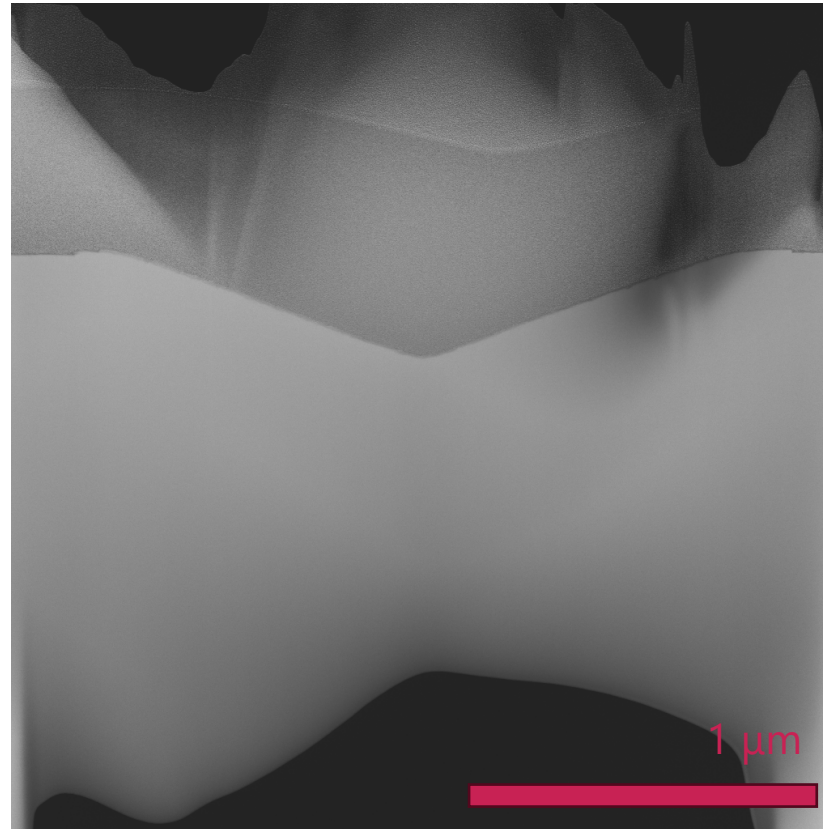
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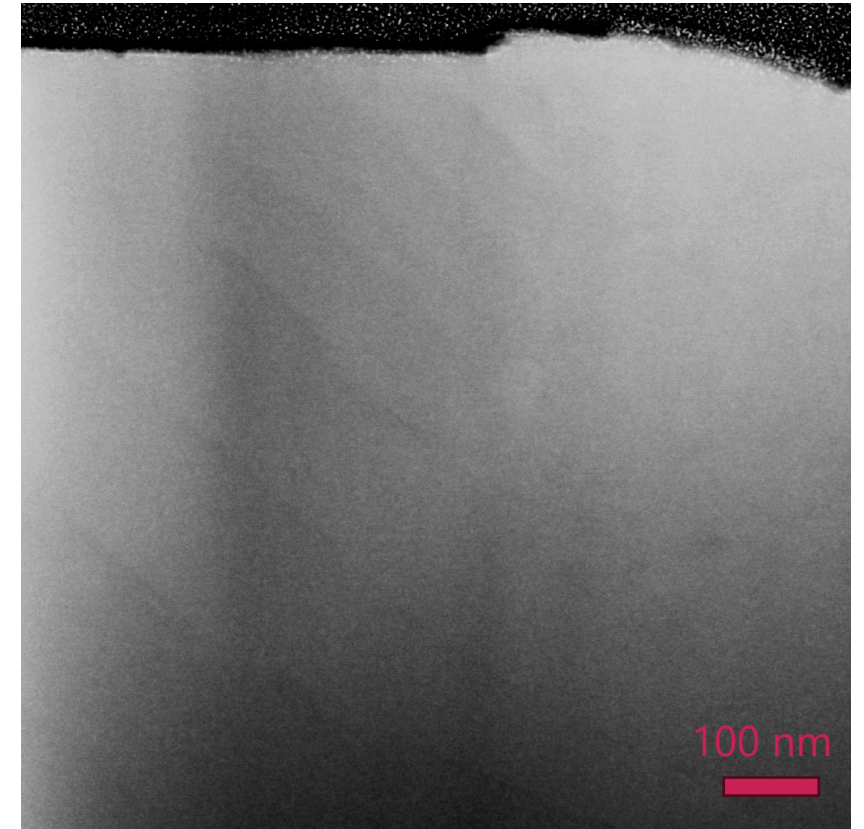


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Jeol TEM JEM-F200

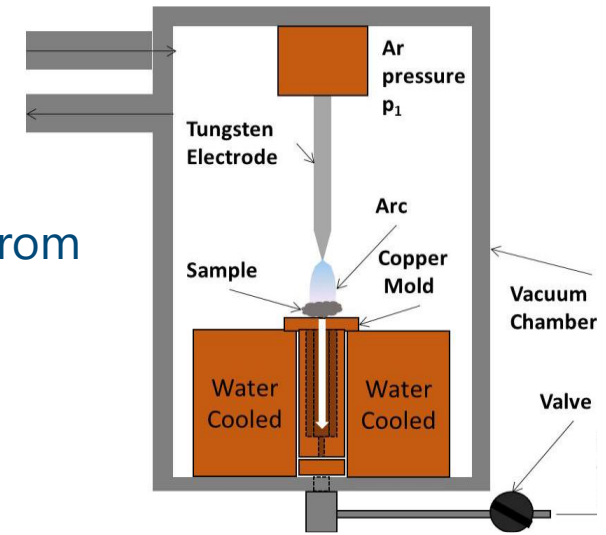


STEM HAADF

Material Preparation

Alloy casting

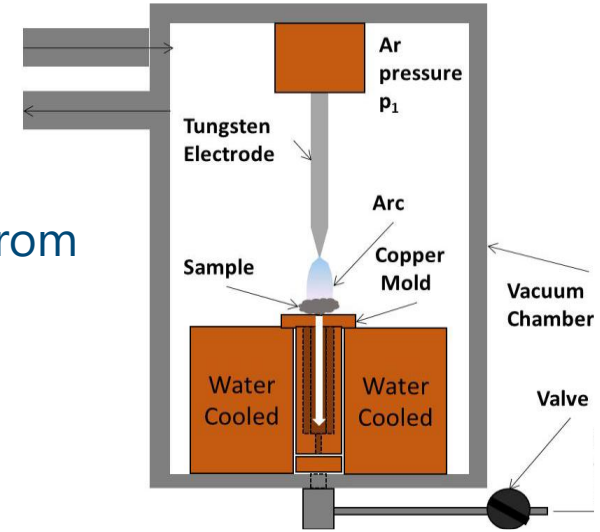
1. Weighing
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Relaxation in sub- T_g

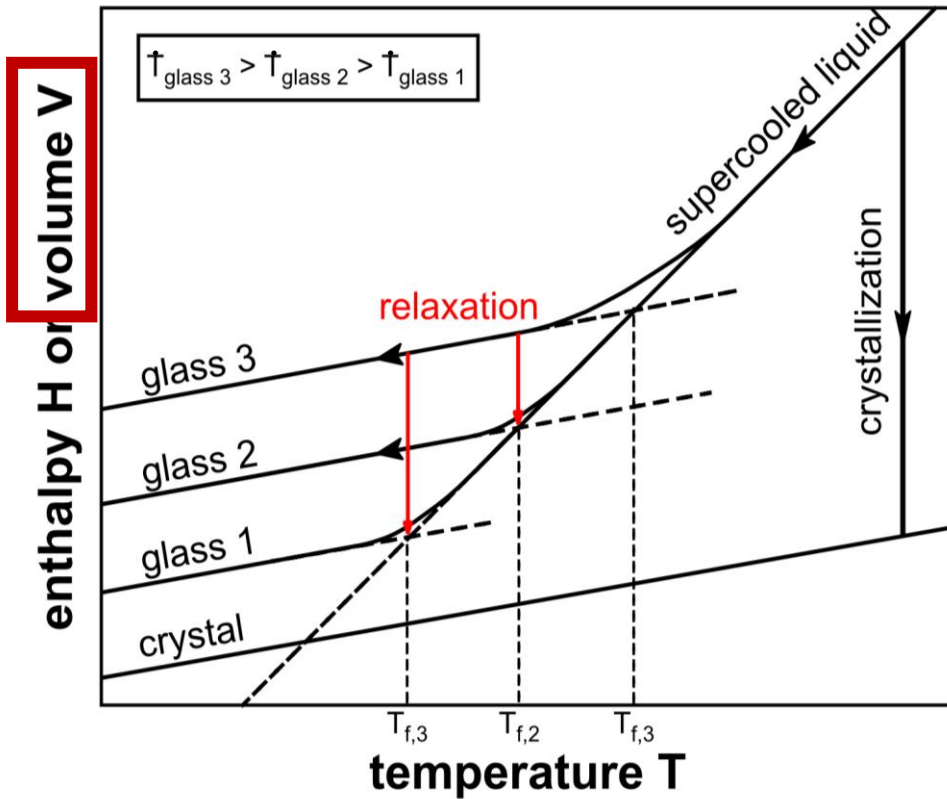


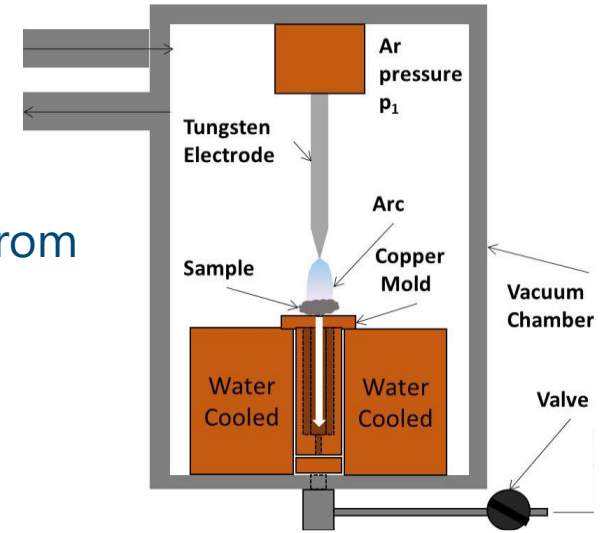
Table 1: Overview of the labeling corresponding to the annealing temperature and transition time. As proven by Ruschel et al. [1] the fictive temperature T_f corresponds to the annealing temperature. The same methodology using the Moynihan area matching method [2] was applied and yields matching results.

Label: $T_g - x$	Annealing temperature = Fictive temperature T_f (K)	Transition time τ (s)	Annealing time = $2 \cdot \tau$ (h)
as-cast	-	-	-
0 K	671	120	0.066
10 K	661	750	0.42
20 K	651	2900	1.61
25 K	646	12000	6.66
30 K	641	29000	16.11
40 K	631	240000	133

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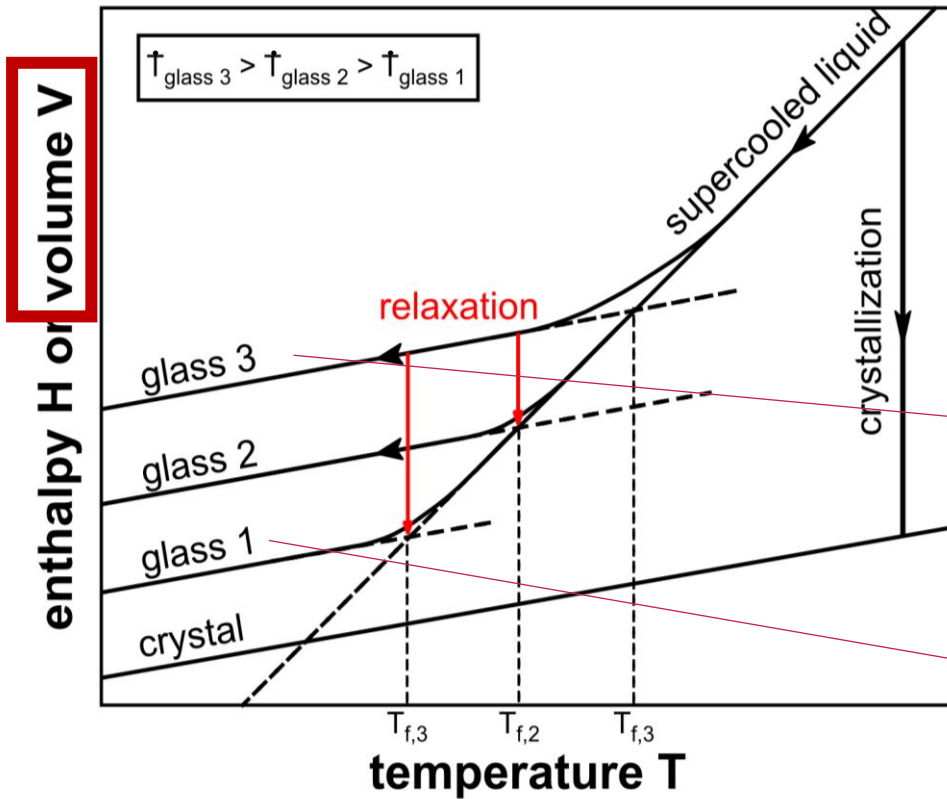


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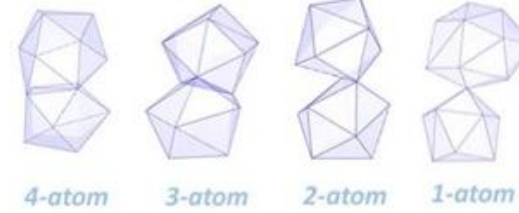


Unraveling the role of relaxation and rejuvenation on the structure and deformation behavior of the Zr-based bulk metallic glass Vit105

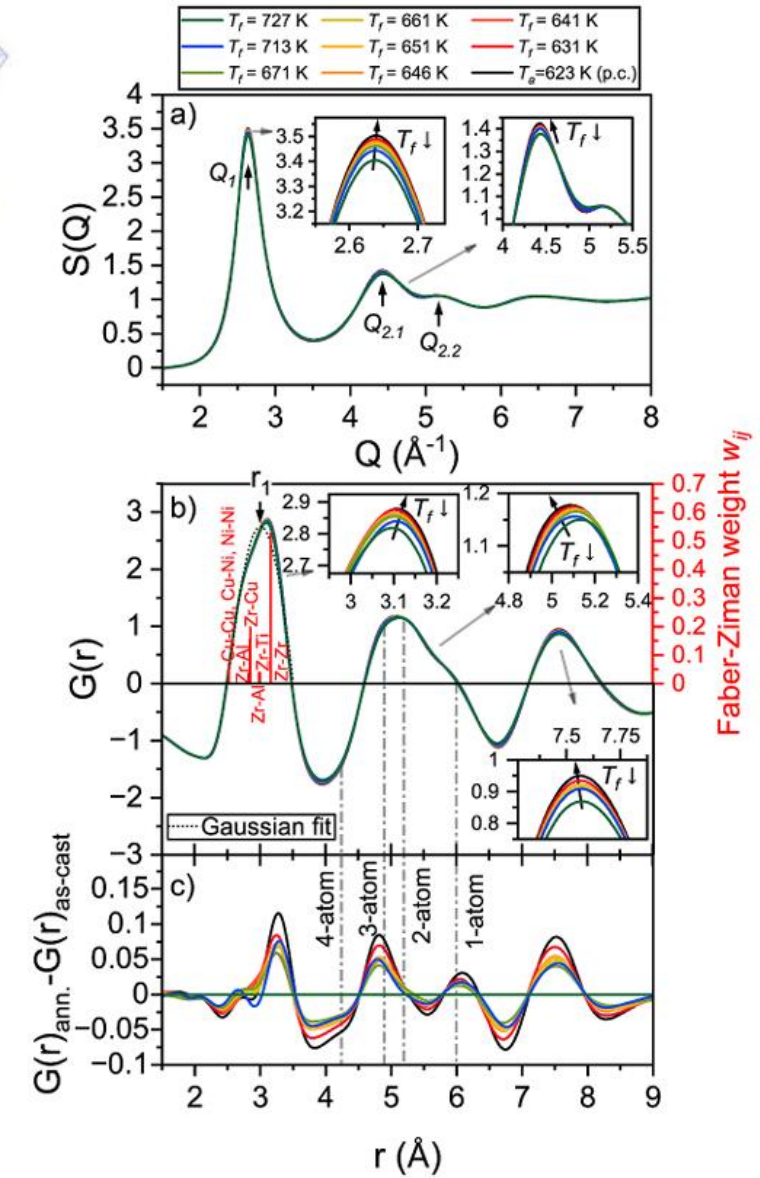
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^a Chair of Metallic Materials, Saarland University, 66123, Saarbrücken, Germany

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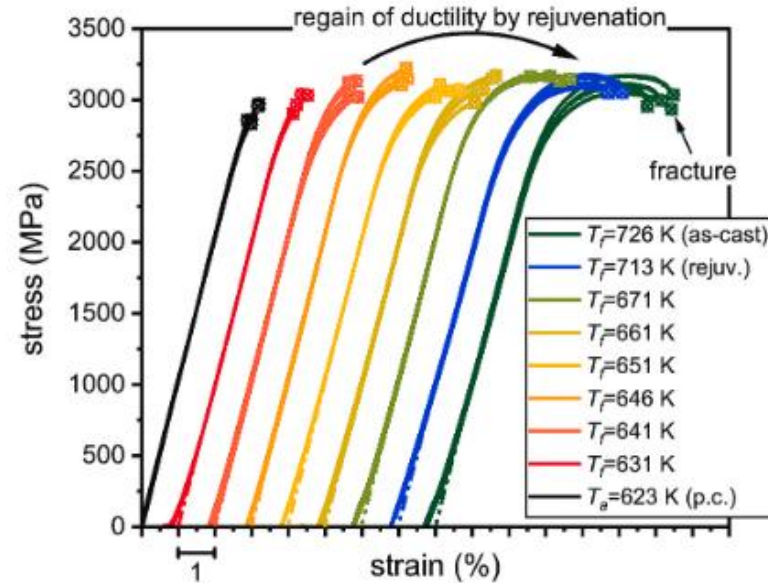
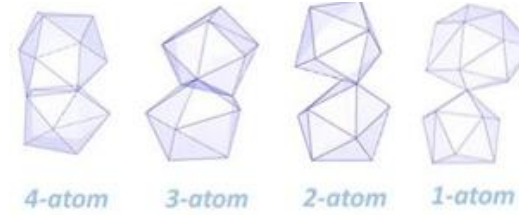
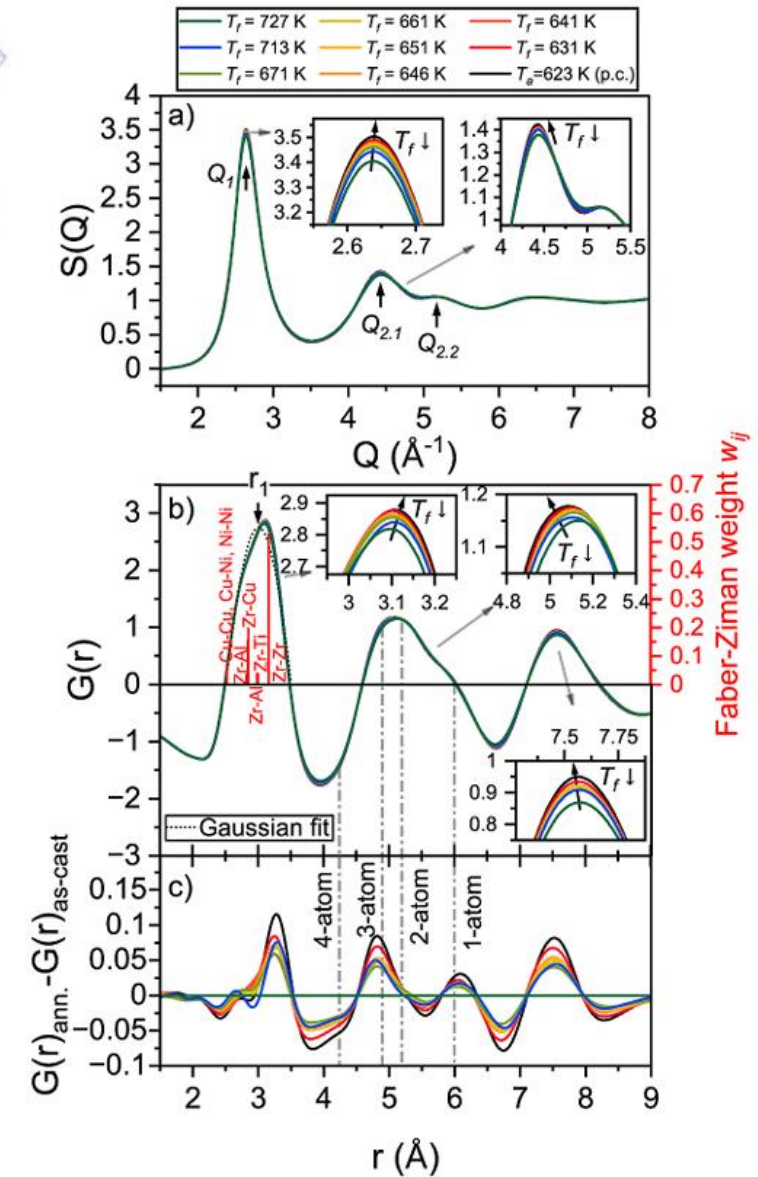
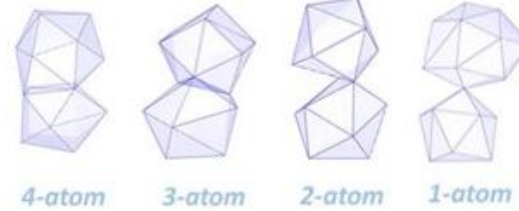


Fig. 4. Engineering stress-strain curves for different fictive temperatures T_f , revealing significant embrittlement with decreasing T_f . The deepest annealing temperature ($T_a = 623$ K, black curves) is affected by partial crystallization (p.c.) as revealed in DSC experiments.



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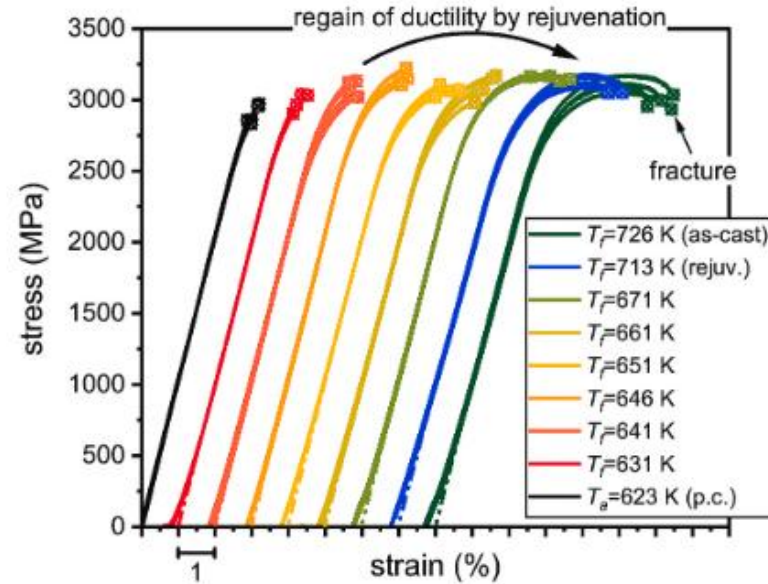
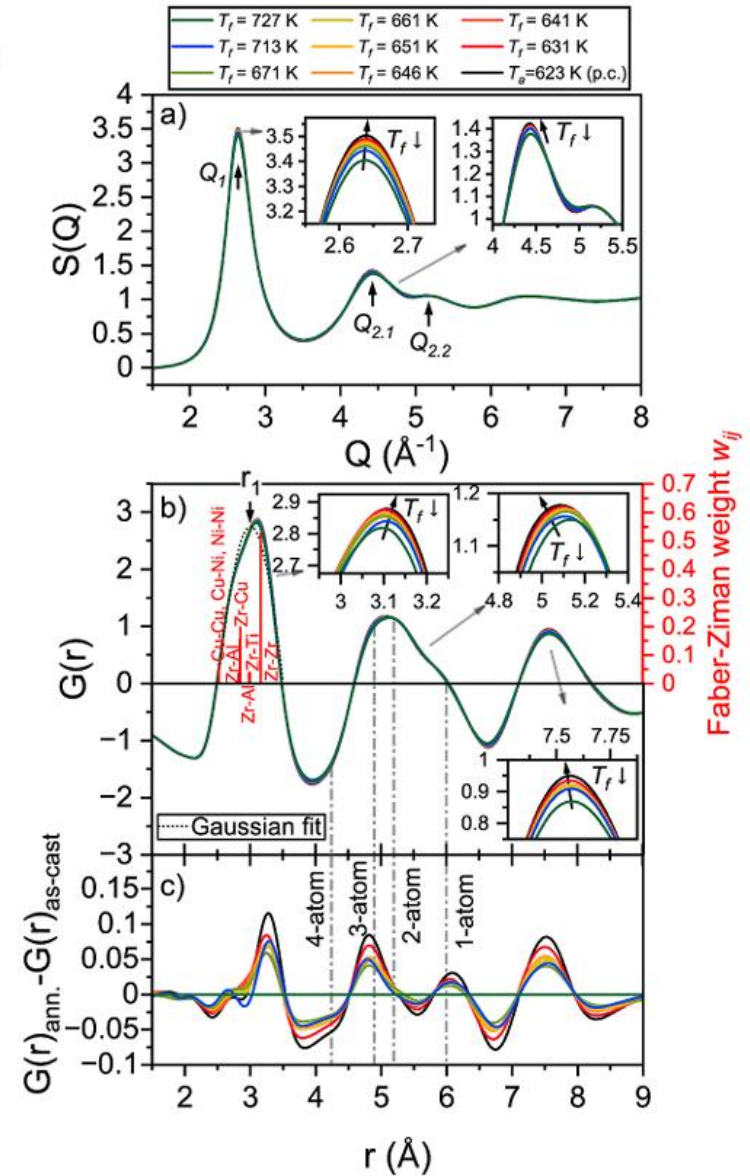
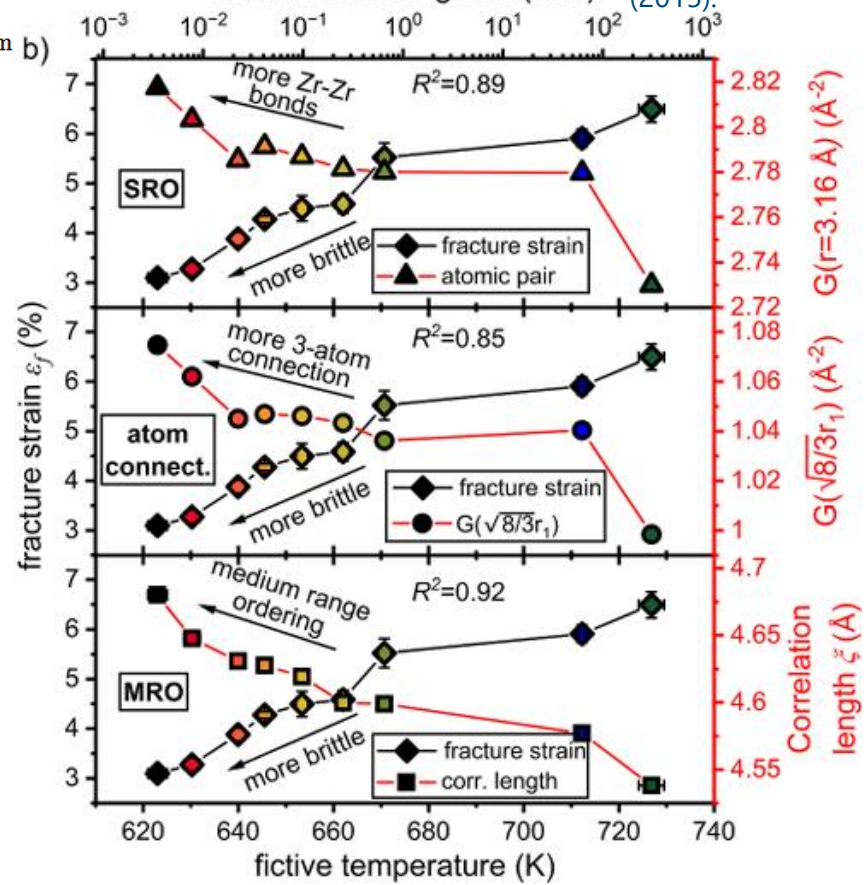


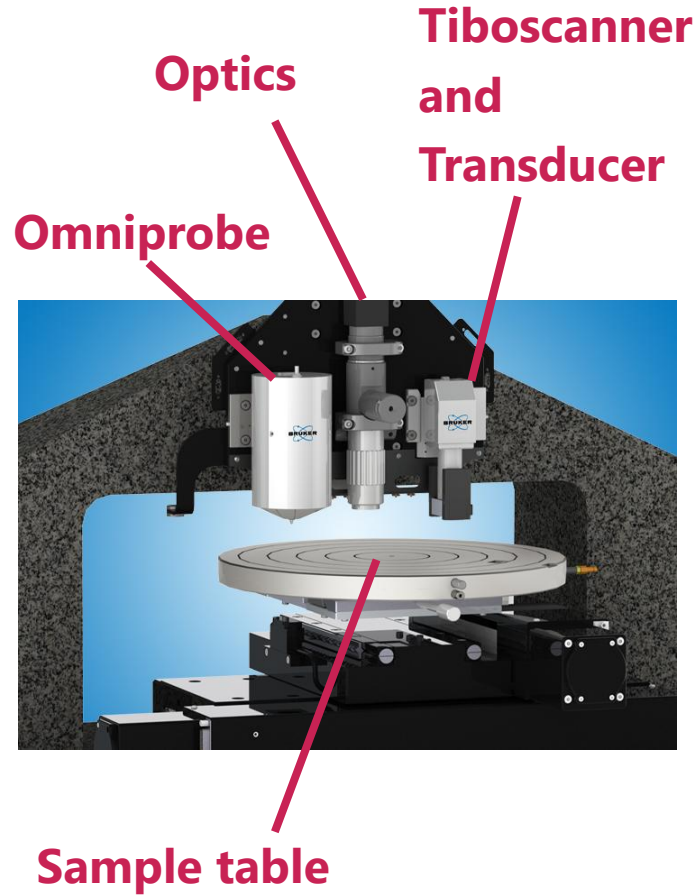
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Nanoindentation

Constant Strain Rate
 $\dot{\epsilon} = 0.01; 0.025; 0.1; 0.25; 1$
 100-200 nm Depth

Bruker TI990 Triboindenter:
 Low Load Transducer



$T_g - x$

as-cast

0 K

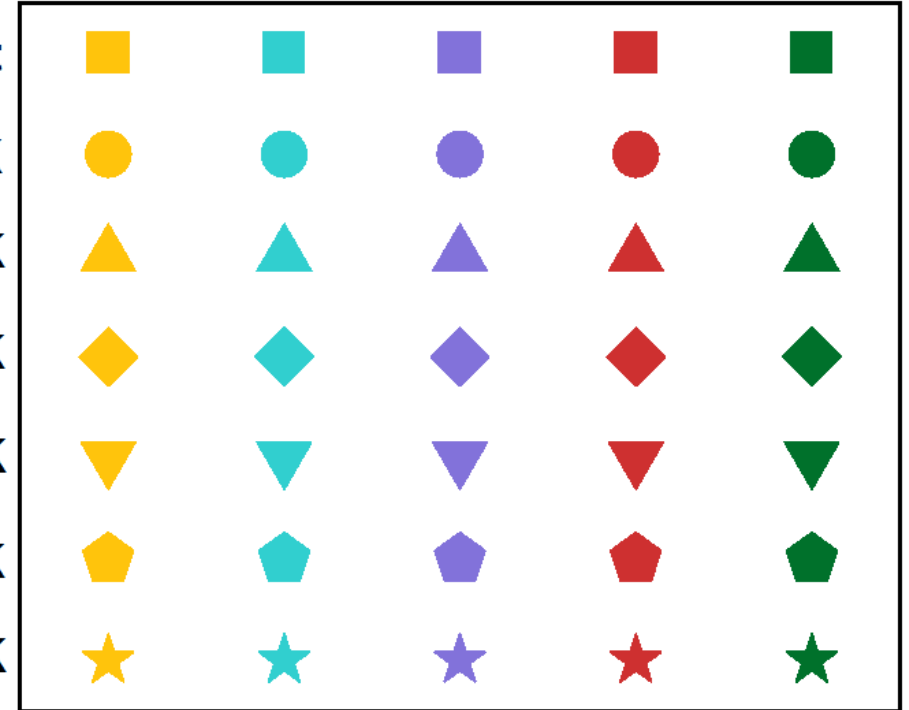
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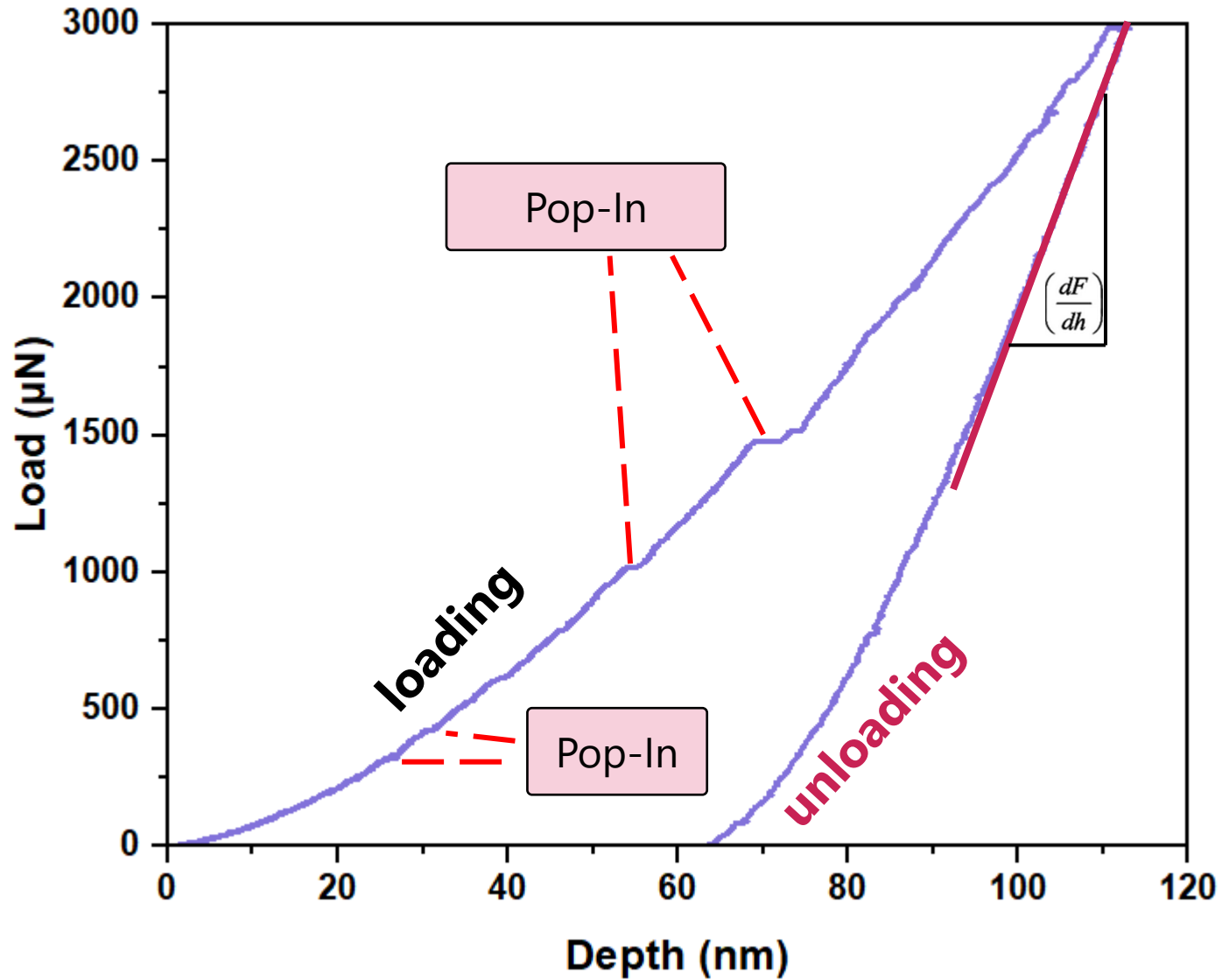
30 K

40 K



0.01 0.025 0.1 0.25 1

Strain Rate (s^{-1})



Unload Segment

Using Oliver-Pharr Method

E-modul

$$E^* = \frac{1}{2} \left(\frac{dF}{dh} \right)_{h_m} \sqrt{\frac{\pi}{A}}$$

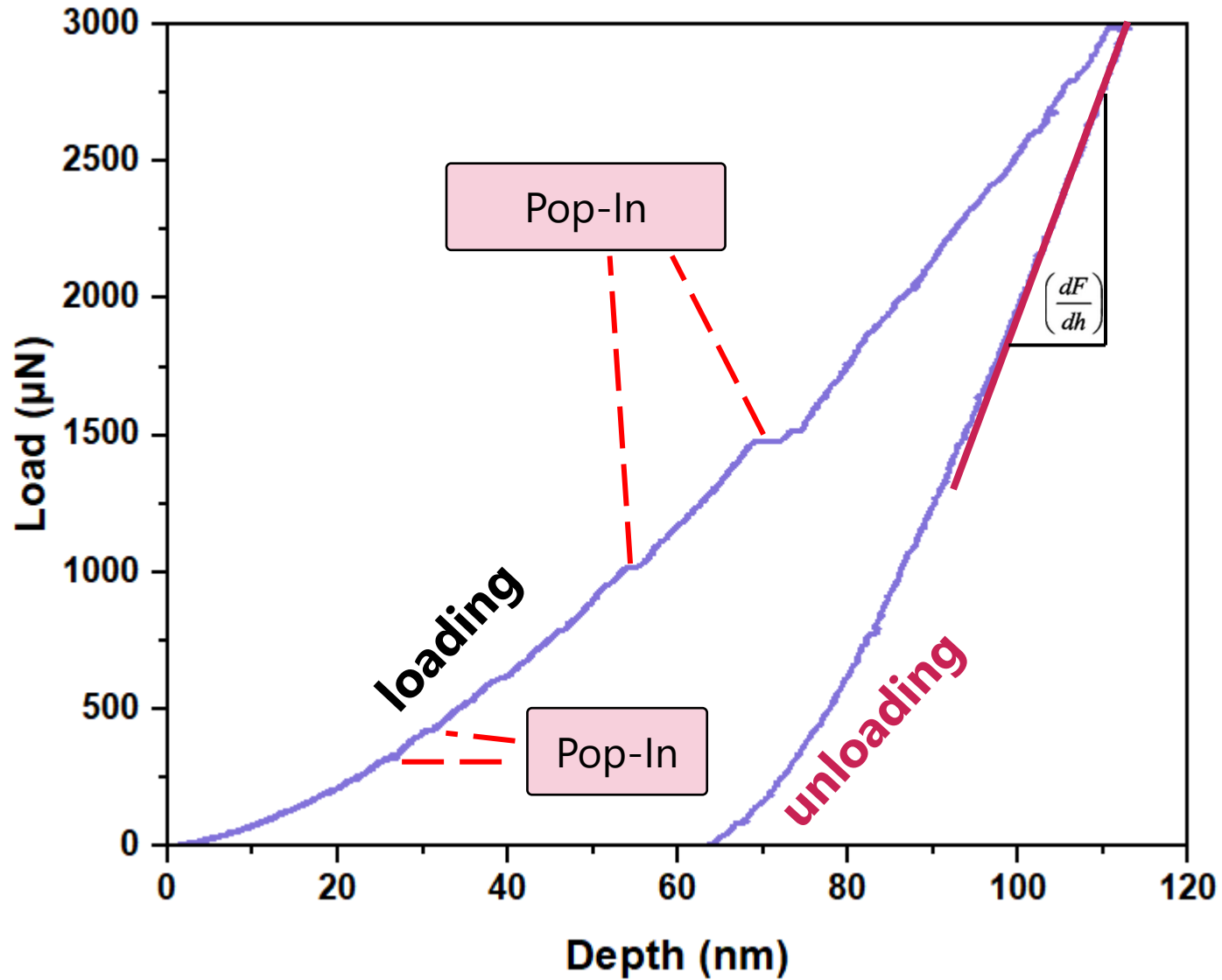
Hardness

$$H = \left(\frac{F}{A} \right)_{h_m}$$



Strain Rate Sensitivity

$$m = \frac{\lg H}{\lg \dot{\epsilon}}$$

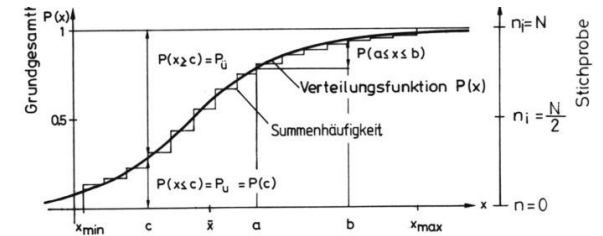


Load Segment

Pop-In Position from MatLab Script
P and hc from Pop-Ins



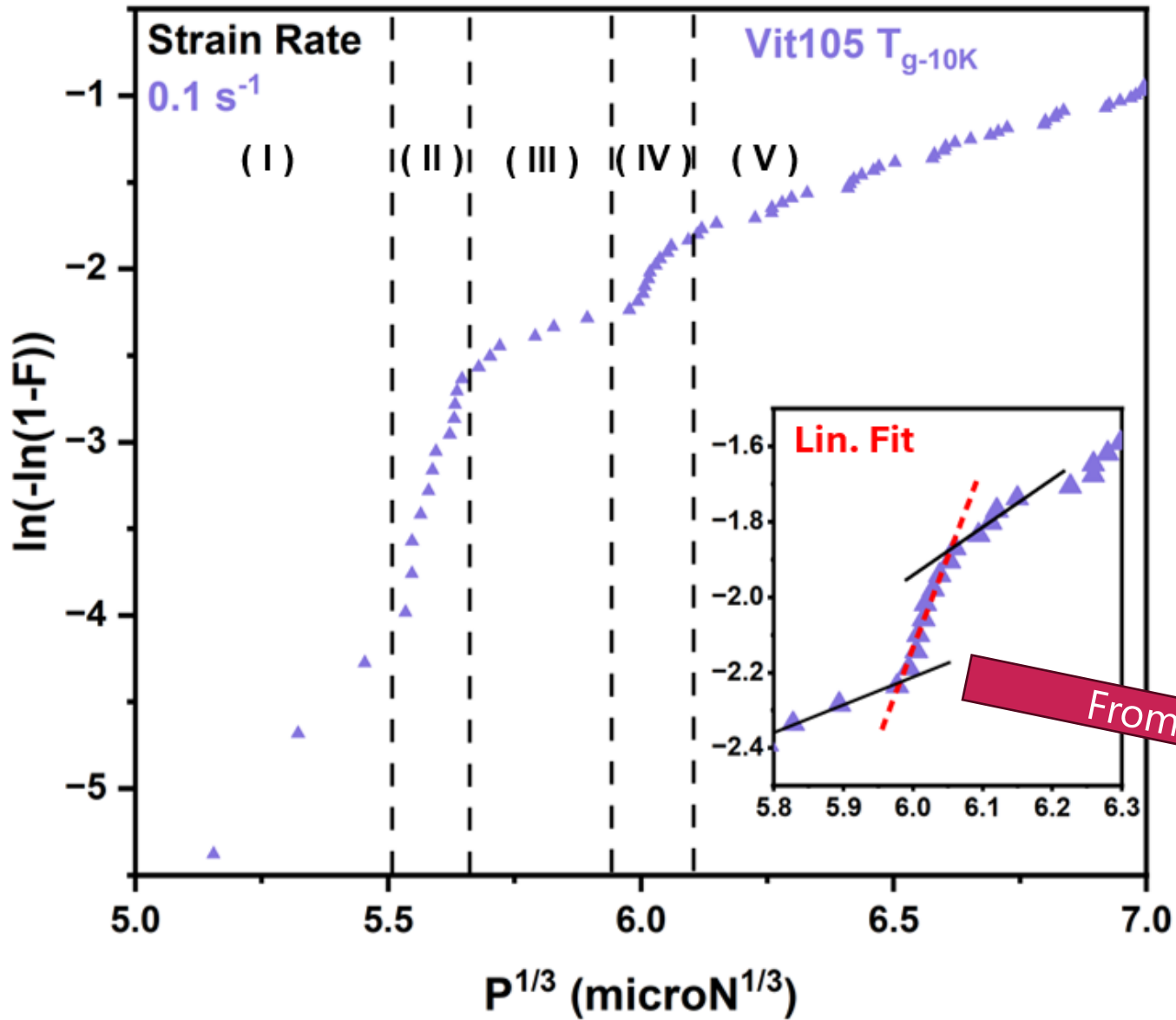
Cumulative Distribution Function



Activation Volume



STZ Volume and Number of Atoms in STZs

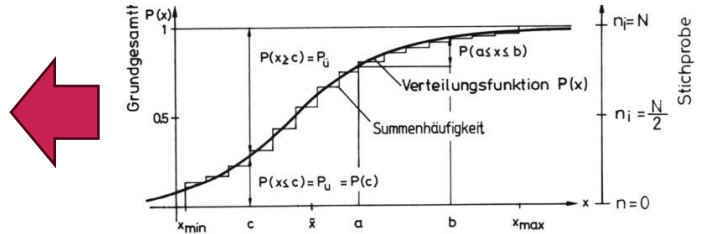


Load Segment

Pop-In Position from MatLab Script
P and hc from Pop-Ins



Cumulative Distribution Function



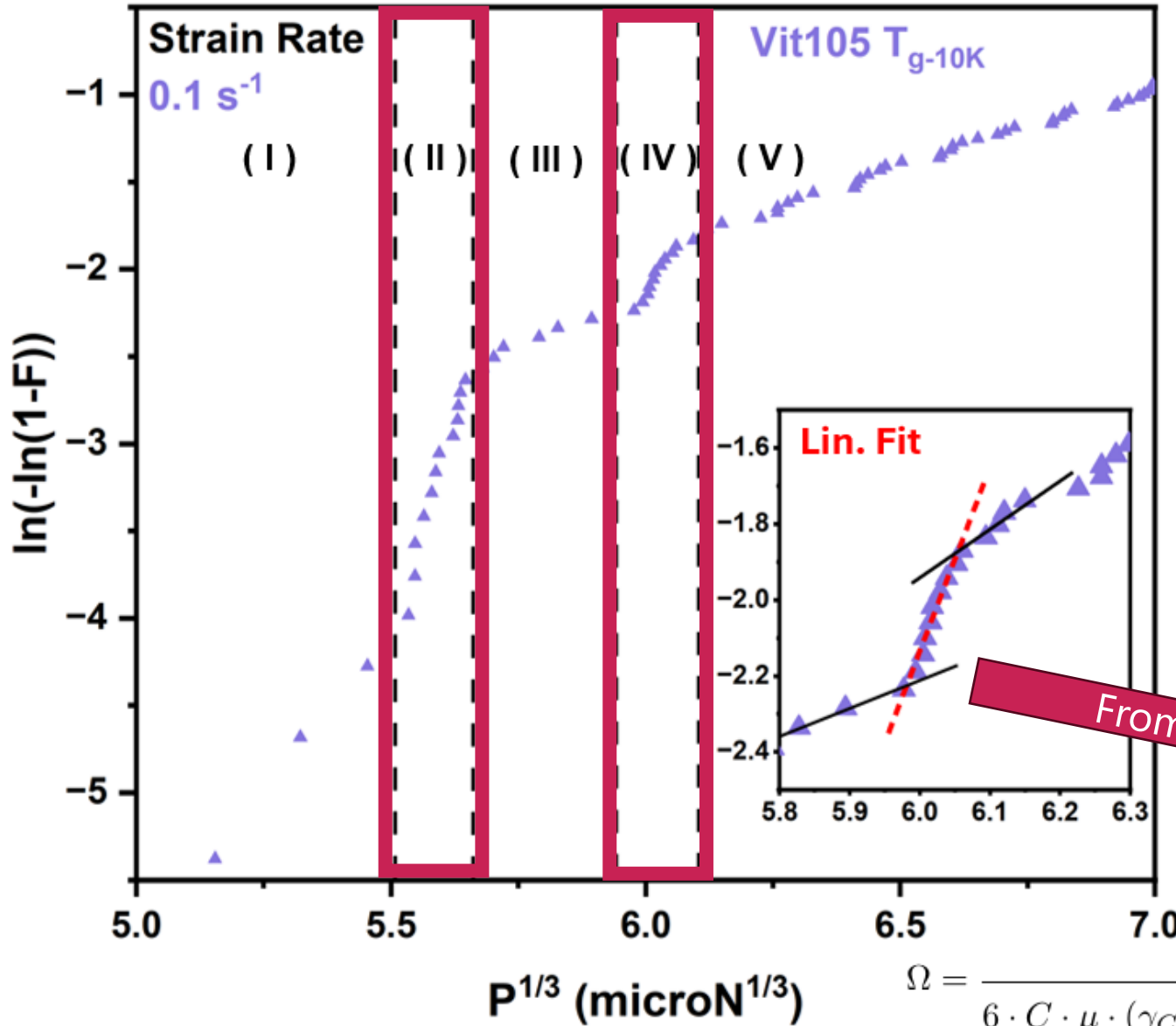
From Slope



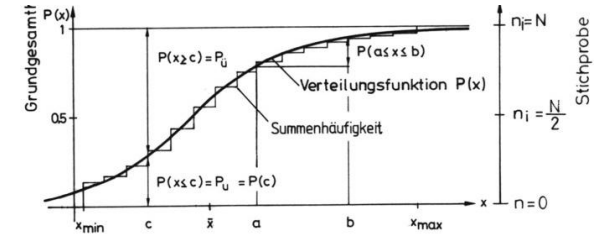

Activation Volume



STZ Volume and Number of Atoms in STZs



- (I) Incubation phase of pop-ins. A pre region below the first assessable pop-in.
- (II) Load level with a considerable slope that can be assigned to explicit pop-in activity. For further analysis, the first recognisable one was marked as the first pop-in.
- (III) Load range at which stress accumulation occurs to release to a not unleased shear band (= pop-in).
- (IV) Area in which the second pop-in or of higher order can be detected.
- (V) Region of fully evolved plasticity of serrated flow.



From Slope

Activation Volume

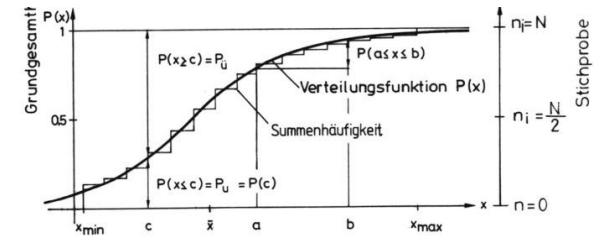
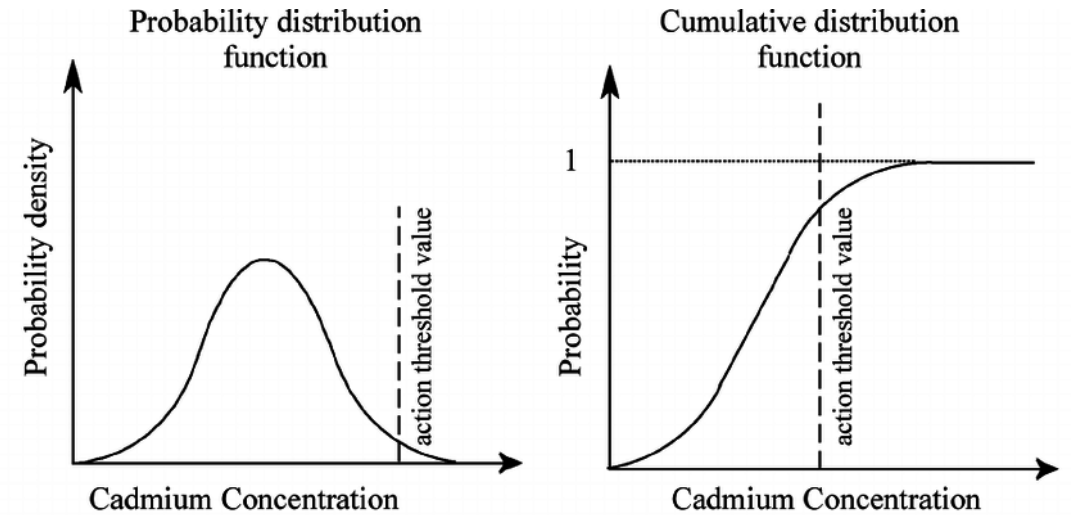
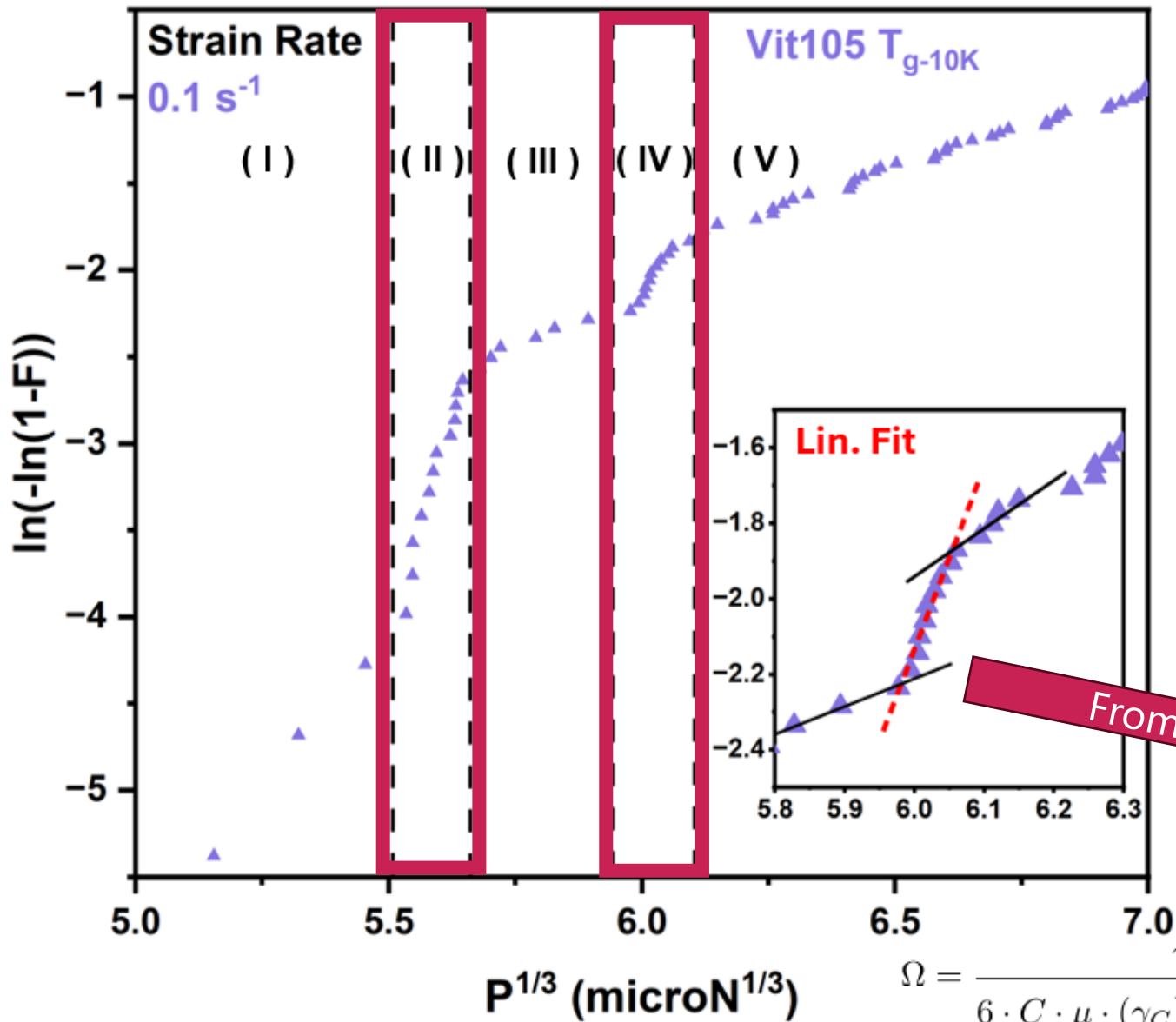
$$V = \alpha \cdot kT \cdot \frac{\pi}{0.47} \cdot \left(\frac{3\rho}{4E_R}\right)^{2/3}$$

STZ Volume

$$\Omega = \frac{\tau_{C0}}{6 \cdot C \cdot \mu \cdot (\gamma_C)^2 \cdot \xi \sqrt{1 - \frac{\tau_C}{\tau_{C0}}}} \cdot V$$

Shear modulus Flow stress

**Estimation through Density
Number of Atoms in STZs**



From Slope

Activation Volume

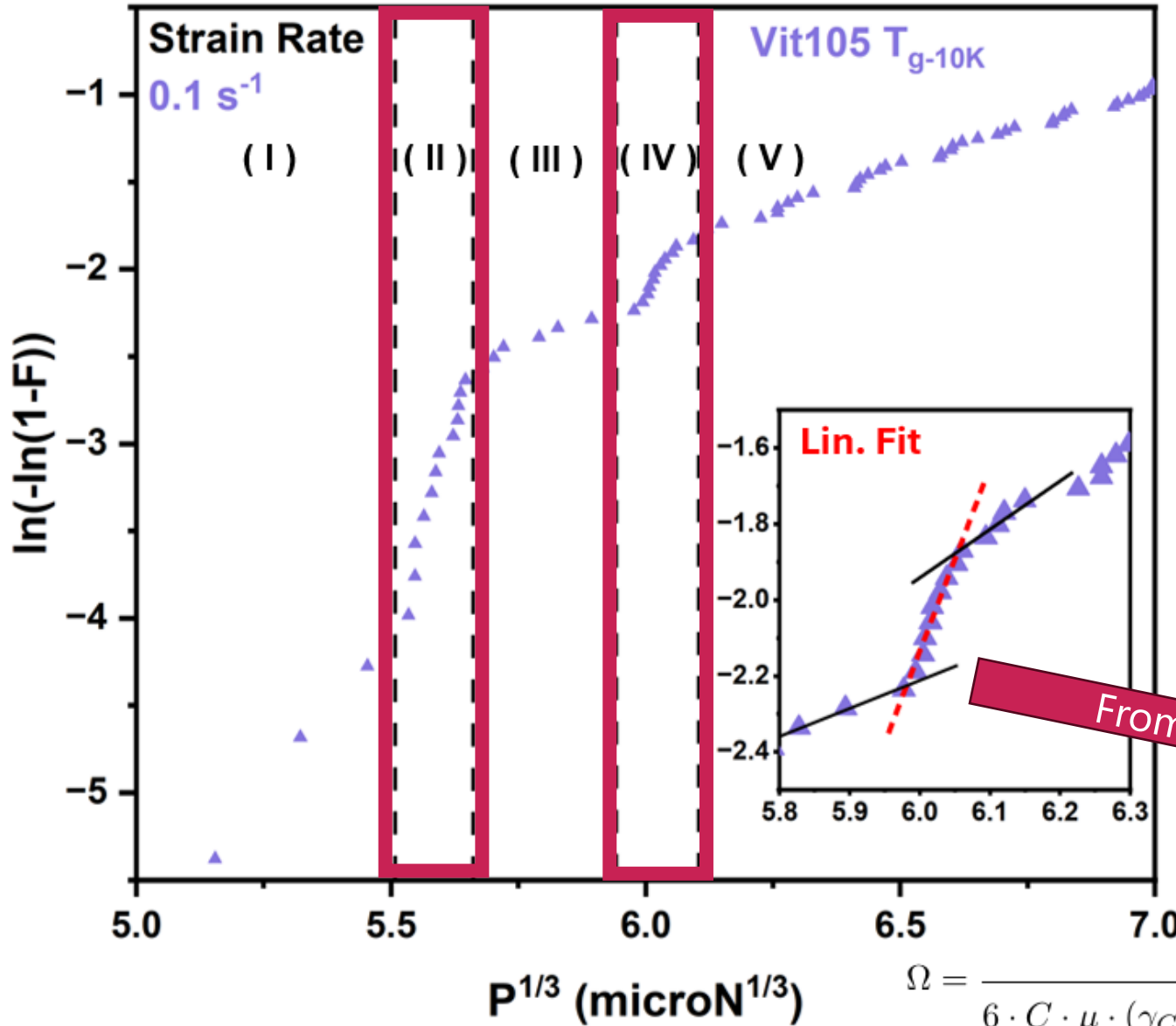
STZ Volume

$$V = \alpha \cdot kT \cdot \frac{\pi}{0.47} \cdot \left(\frac{3\rho}{4E_R}\right)^{2/3}$$

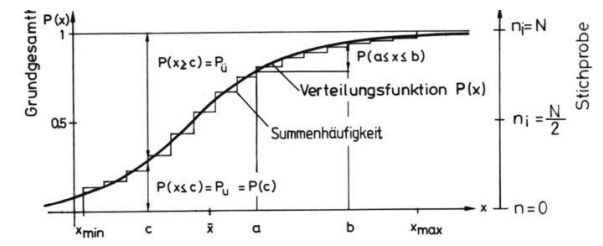
Estimation through Density Number of Atoms in STZs

$$\Omega = \frac{\tau_{C0}}{6 \cdot C \cdot \mu \cdot (\gamma_C)^2 \cdot \xi \sqrt{1 - \frac{\tau_C}{\tau_{C0}}}} \cdot V$$

Shear modulus Flow stress



- (I) Incubation phase of pop-ins. A pre region below the first assessable pop-in.
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From Slope

Activation Volume

STZ Volume

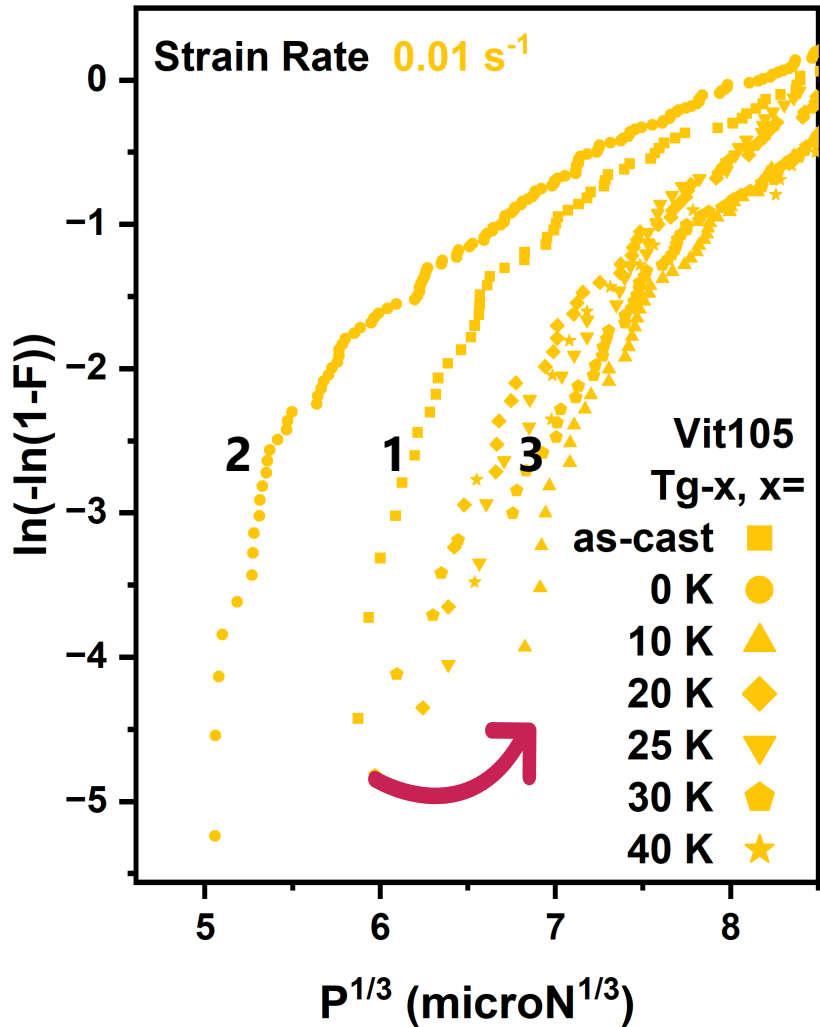
$$V = \alpha \cdot kT \cdot \frac{\pi}{0.47} \cdot \left(\frac{3\rho}{4E_R}\right)^{2/3}$$

Estimation through Density Number of Atoms in STZs

$$\Omega = \frac{\tau_{C0}}{6 \cdot C \cdot \mu \cdot (\gamma_C)^2 \cdot \xi \sqrt{1 - \frac{\tau_C}{\tau_{C0}}}} \cdot V$$

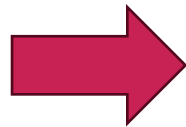
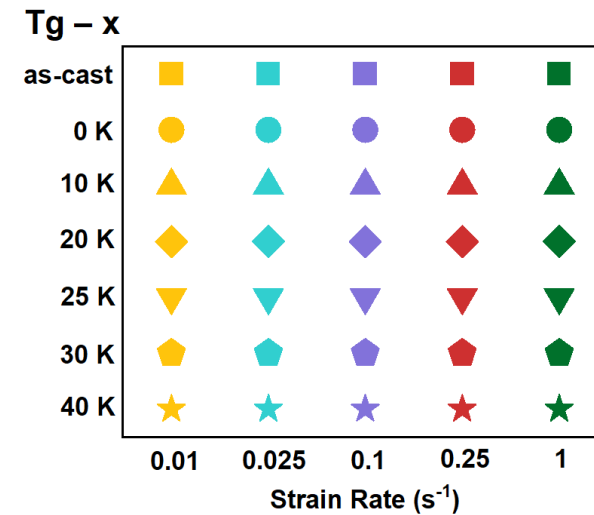
Shear modulus Flow stress

Empirical Cumulative Distribution

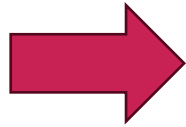


Representative for all other Strain Rates

1. As-cast
2. 0K
3. 10K, 20K, 25K, 30K and 40K -> Clustering (no difference)



Evidence for internal stresses (Zug) after casting



Onset of Plasticity shifts to higher critical loads through structural relaxation

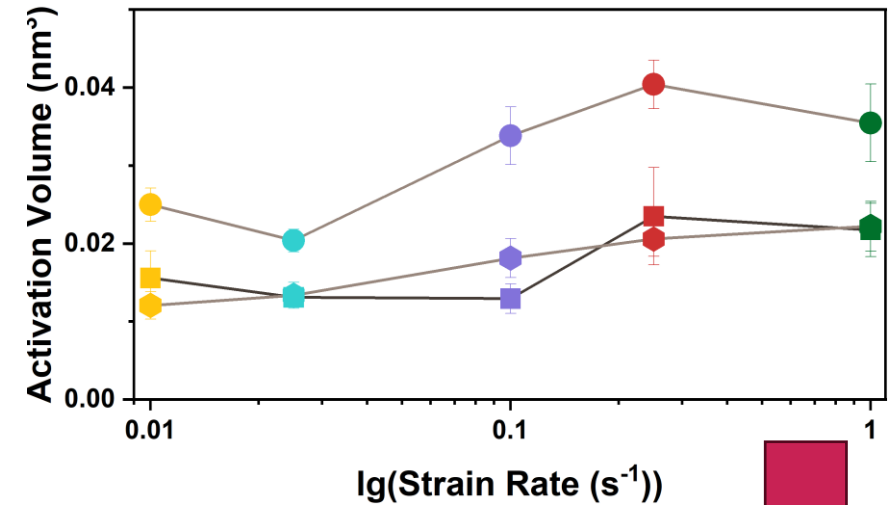
STZ Volume and Number of Atoms in STZs

Table 2: Activation Volume and STZ size calculated from the statistics gathered from the

Pop-In data

Label: $T_g - x$	strain rate (s^{-1})	activation volume (nm^3)	STZ volume Ω (nm^3)	Number of Atoms in STZ
as-cast	0.01	0.01558 ± 0.00348	0.34593	19
as-cast	0.025	0.0131 ± 0.00105	0.29082	16
as-cast	0.1	0.01293 ± 0.00189	0.28713	16
as-cast	0.25	0.0235 ± 0.00626	0.5217	29
as-cast	1	0.02174 ± 0.00343	0.48266	27
0 K	0.01	0.02497 ± 0.00213	0.55448	31
0 K	0.025	0.02037 ± 0.00146	0.45232	25
0 K	0.1	0.03381 ± 0.00369	0.75067	42
0 K	0.25	0.04039 ± 0.0031	0.89676	50
0 K	1	0.03546 ± 0.00499	0.78725	44
10-40 K	0.01	0.01207 ± 0.00177	0.26802	15
10-40 K	0.025	0.01337 ± 0.00167	0.29675	16
10-40 K	0.1	0.01813 ± 0.0025	0.40252	22
10-40 K	0.25	0.02059 ± 0.00222	0.45722	26
10-40 K	1	0.02221 ± 0.0032	0.49301	28

Plastic deformation is scale-dependent



Activation Volume is unaffected by the structural relaxation (3-atom connection)

STZ Volume and Number of Atoms in STZs

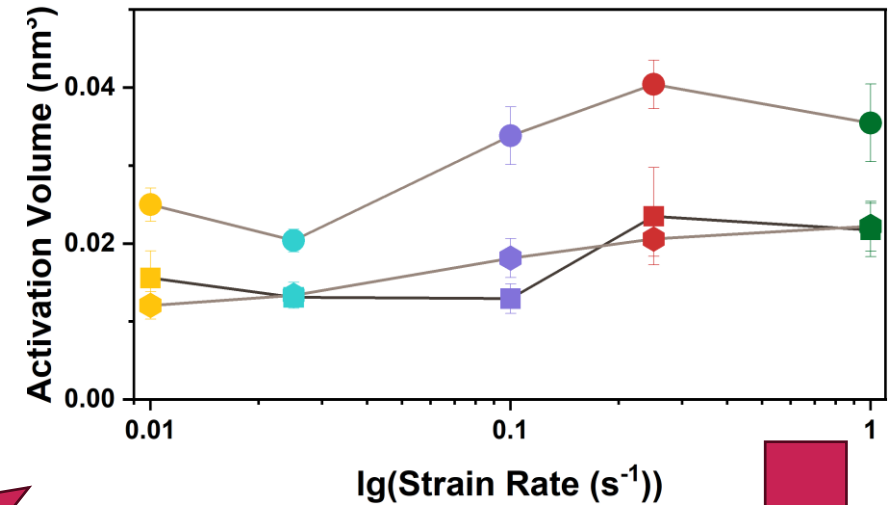
Key Findings:

- Elastic regime: Unaffected by structural relaxation
- Plastic regime: Strongly influenced by structural relaxation

But

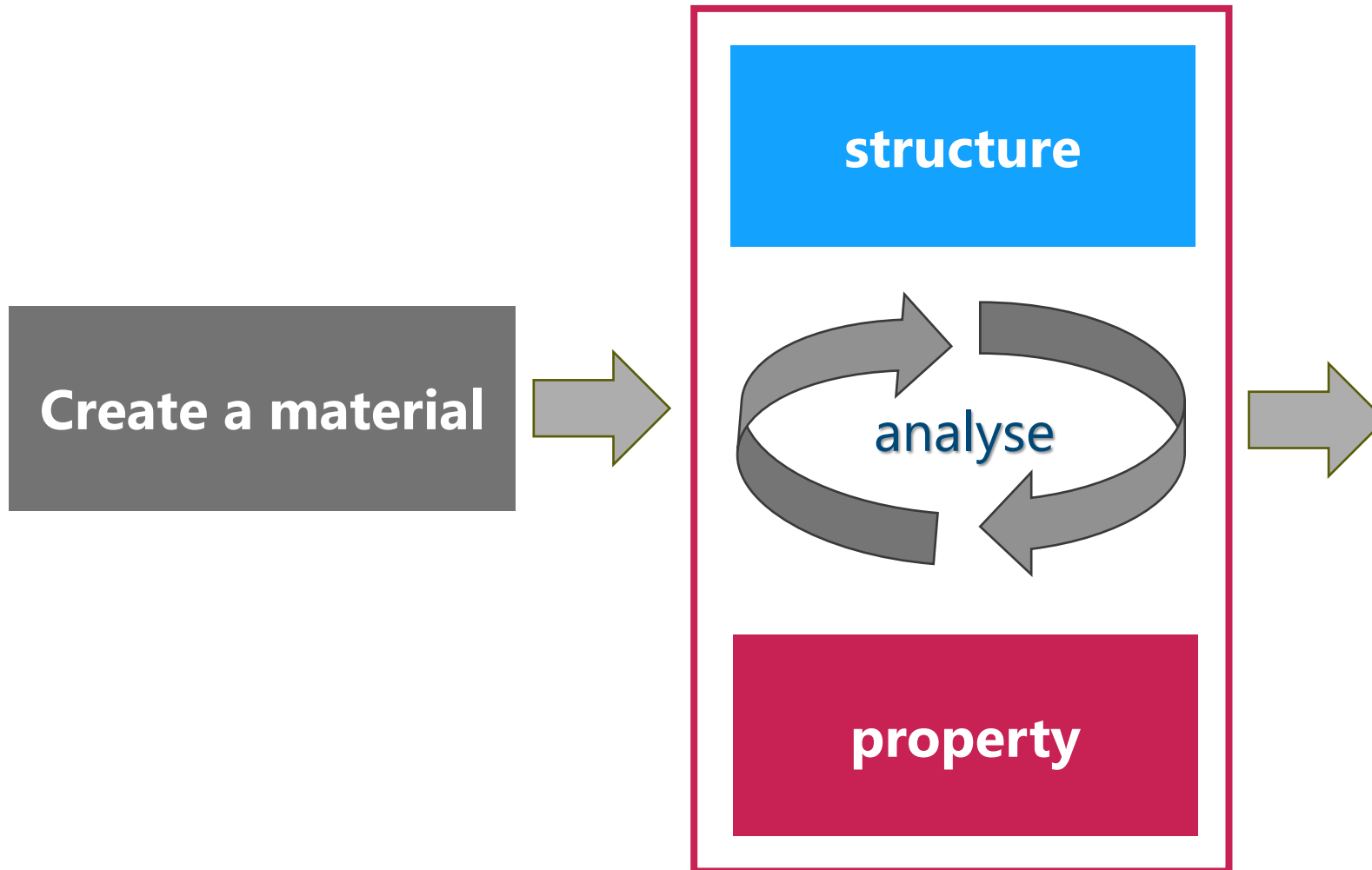
Activation Volume remains **constant**

Plastic deformation is scale-dependent



Activation Volume is unaffected by the structural relaxation (3-atom connection)

What does a (material)scientist in research?



Is my material fit for my application?



@<https://www.reuters.com/lifestyle/two-giant-rubber-ducks-debut-hong-kong-bid-drive-double-happiness-2023-06-09/>

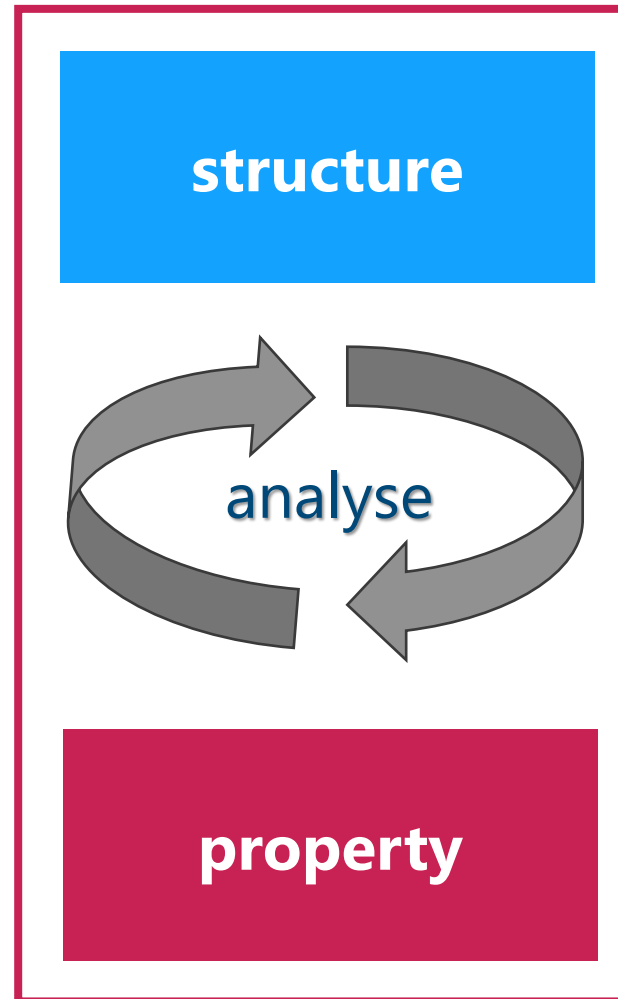
Can I predict **mechanical properties** from the **structural fingerprint**?

What does a (material)scientist in research?

Confirm: structural fingerprints can predict aspects of plasticity

Link: nano-scale events with atomic-scale mechanisms

Provide: Pathway to tailor mechanical properties via structural relaxation



Is my material fit for my application?



@<https://www.reuters.com/lifestyle/two-giant-rubber-ducks-debut-hong-kong-bid-drive-double-happiness-2023-06-09/>

Can I predict **mechanical properties** from the **structural fingerprint**?

Thank You! Questions?



<https://doi.org/10.1016/j.jallcom.2026.188302>