

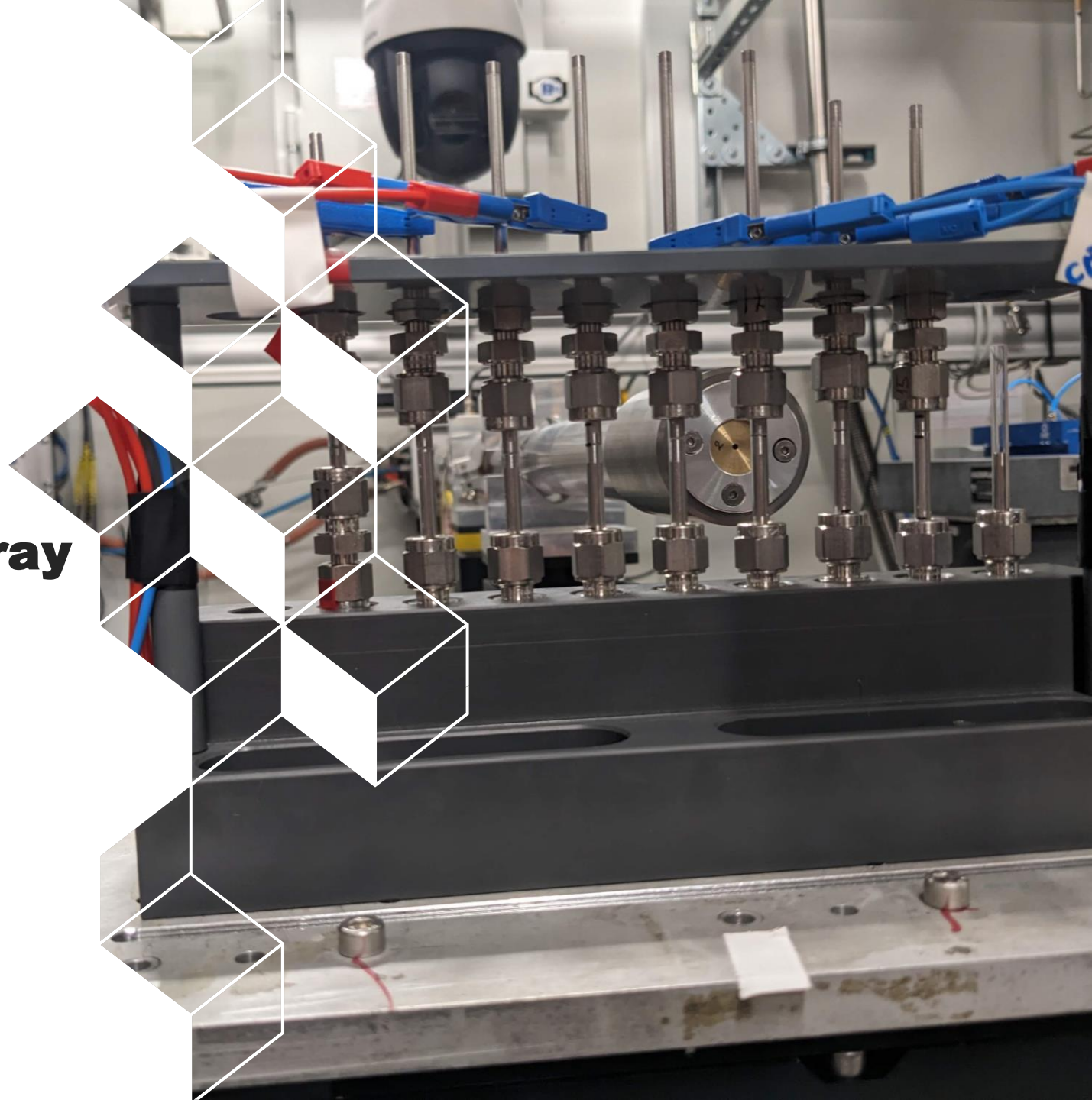


Small Angle Neutron and X-ray Scattering: Leaving Atoms Behind and Going Big

Xaver S. Brems



www.europeanbatteryhub.eu



My Journey to SAXS and SANS

B. Sc. Physics (TUM)

TISANE: AC Coil Setup for Time Resolved Kinetic Neutron Scattering

M. Sc. Condensed Matter Physics (TUM)

Current-induced Self-organization of Mixed Superconducting States

PhD Experimental Physics (TUM/ILL)

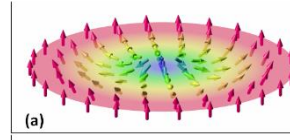
Vortex Matter Transport Phenomena of the Intertype Superconductor Niobium

Bob Cubitt (ILL), Sebastian Mühlbauer (TUM)

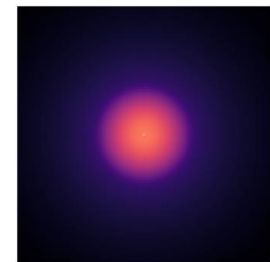
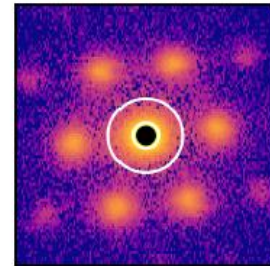
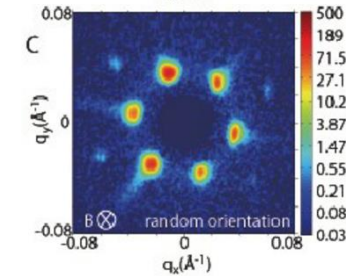
Postdoctoral Researcher at CEA

Project Manager of the European Battery Hub

operando SAXS/SANS on battery materials



Bragg peaks in my scattering patterns



Excitement when seeing small bumps / slope changes in my curve

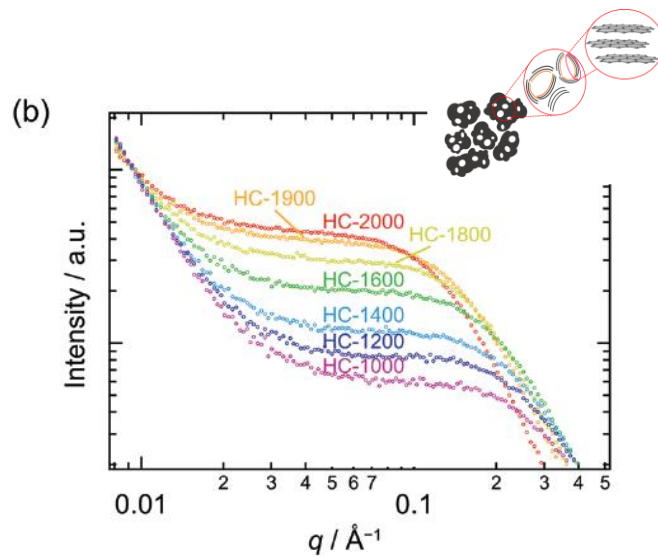
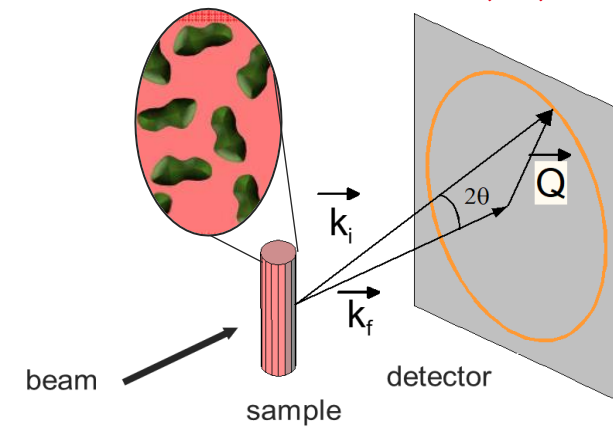
Outline

❖ I – SAXS/SANS TUTORIAL

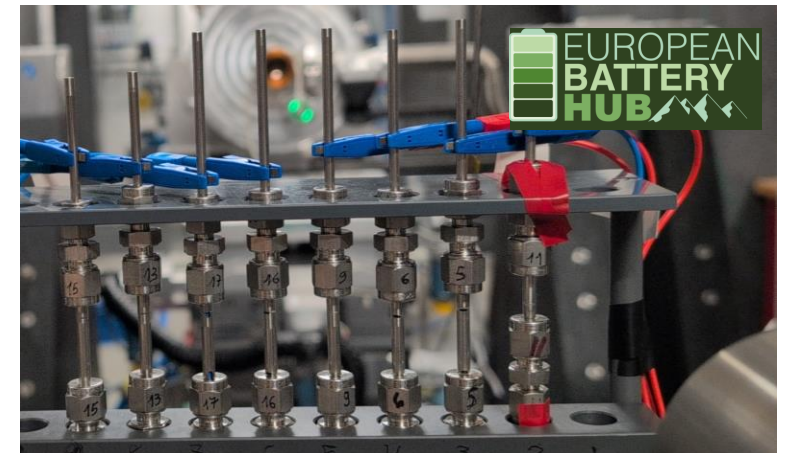
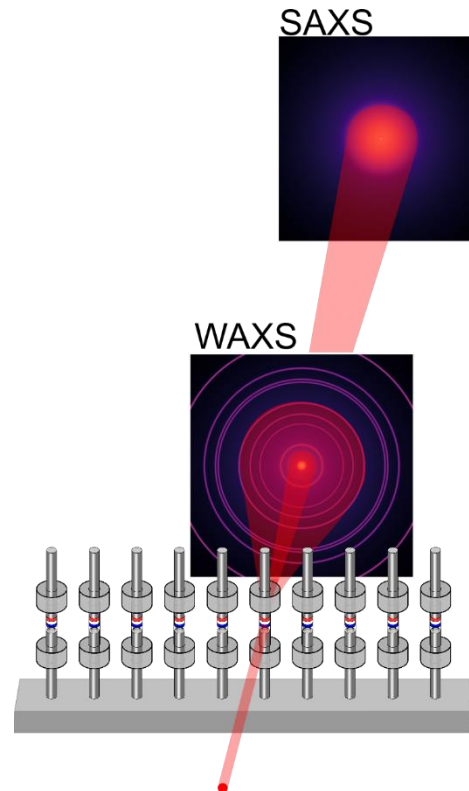
Basic principle of Small Angle Scattering techniques (SAXS, SANS)

❖ II – APPLICATION to battery research

Operando SAXS/SANS study of HC electrodes for Na-ion batteries



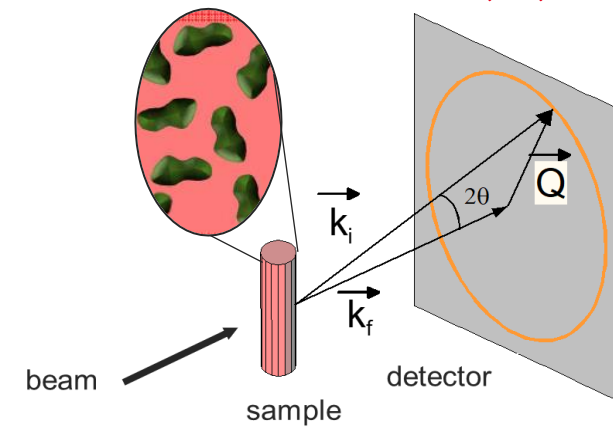
Morikawa et al, Adv. Energy. Materials



Outline

❖ I – SAXS/SANS TUTORIAL

Basic principle of Small Angle Scattering techniques (SAXS, SANS)



<https://www-ssrl.slac.stanford.edu/~saxs>

https://www.fhi.mpg.de/1075129/armin_hoell__anomalous_small-angle_x-ray_scattering__150116.pdf

<https://www.diamond.ac.uk/Instruments/Soft-Condensed-Matter/small-angle.html>

https://brockhouse.lightsource.ca/documents/1/240620_xrd_school_saxs_al.pdf

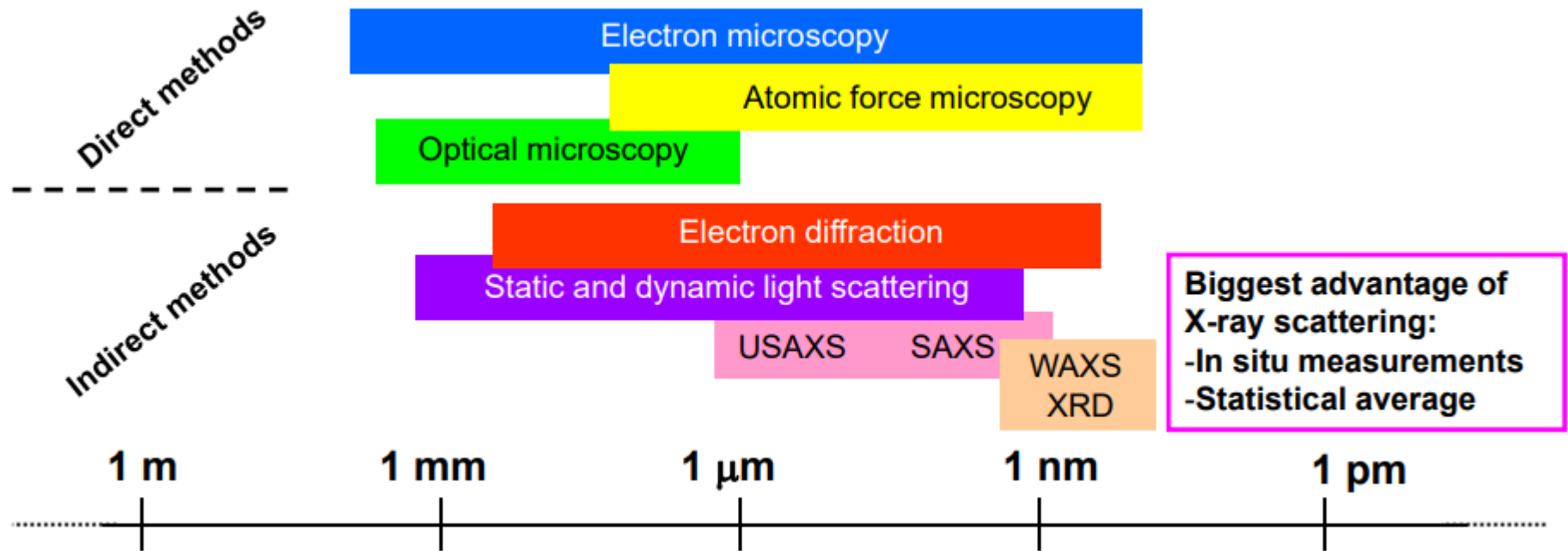
https://www.eng.uc.edu/~beaucag/Classes/Properties/the_SANS_toolbox.pdf

R. Heenan, ISIS Neutron Training Course Introduction to SANS

F. Cousin, Small Angle Neutron Scattering, EPJ Web of Conferences 104, 01004 (2015)

DOI: 10.1051/epjconf/201510401004

S. Lyonnard, *SAXS/WAXS Techniques for 1D, 2D, and 3D Mapping of Structural Changes in Battery Materials*; Battery2030+/OPINCHARGE/UltraBat seminar series

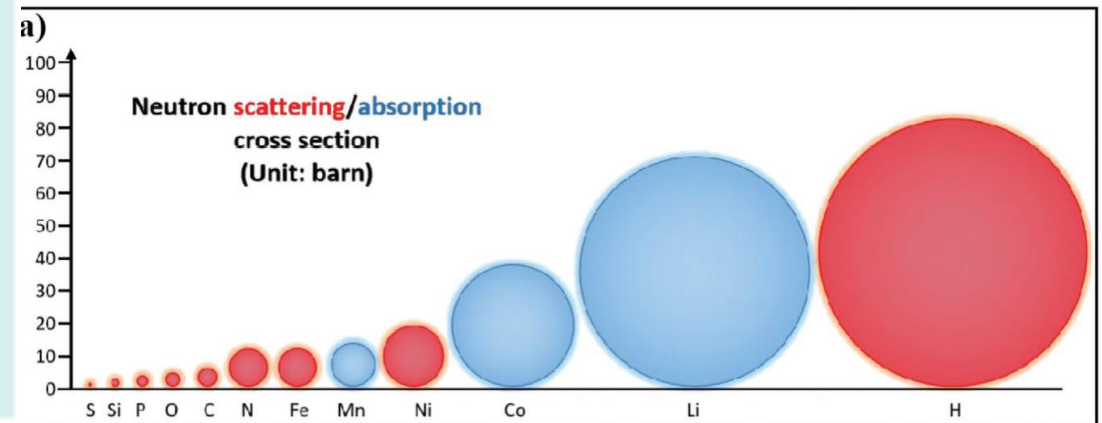
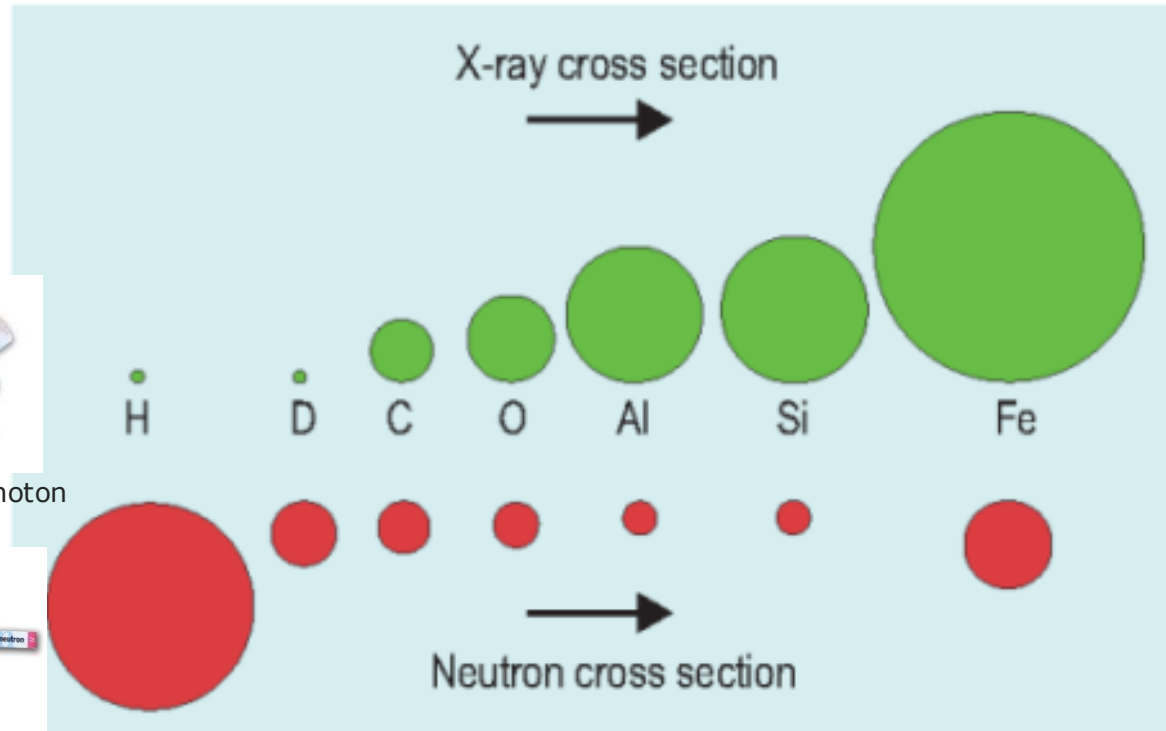
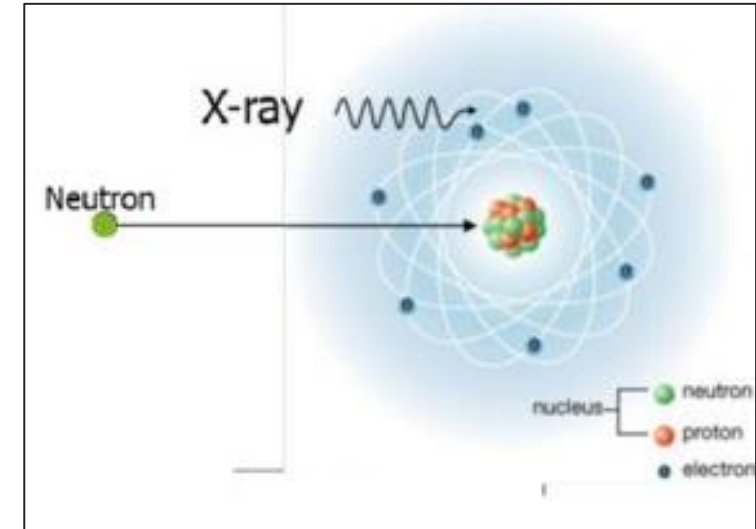


Biggest advantage of X-ray scattering:
-In situ measurements
-Statistical average

Basic principles of SAXS/SANS

neutrons interact with nuclei while X-rays interact with e-

Scattering cross sections & absorption cross sections



Basic principles of SAXS/SANS



Human hand with ring and bracelet

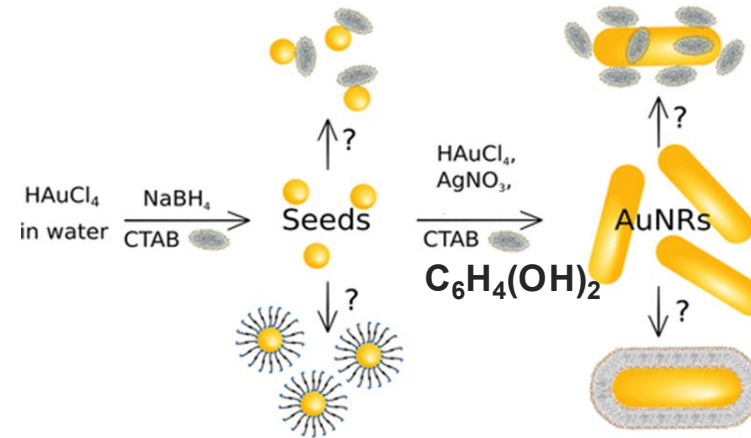


Rose in lead container

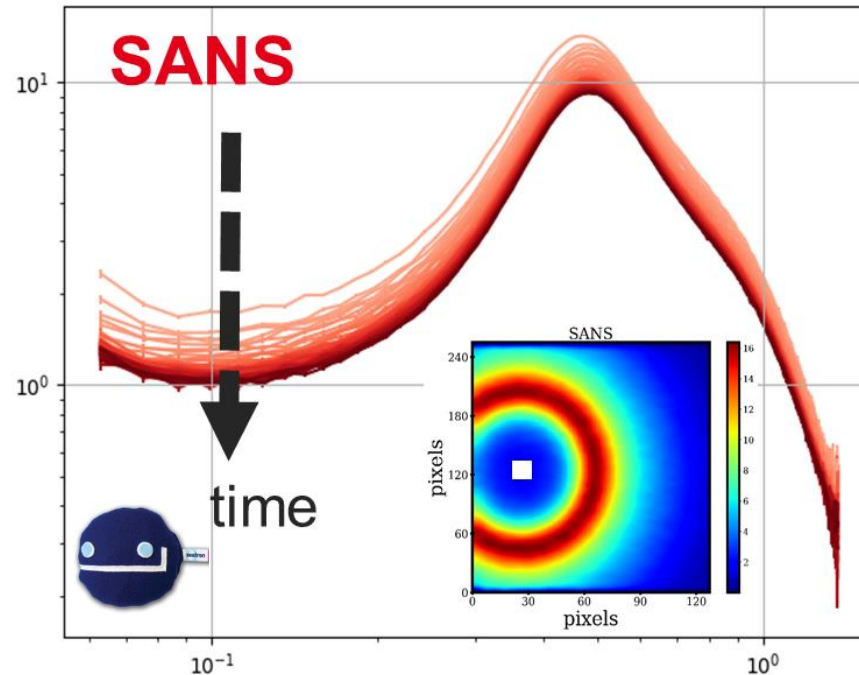
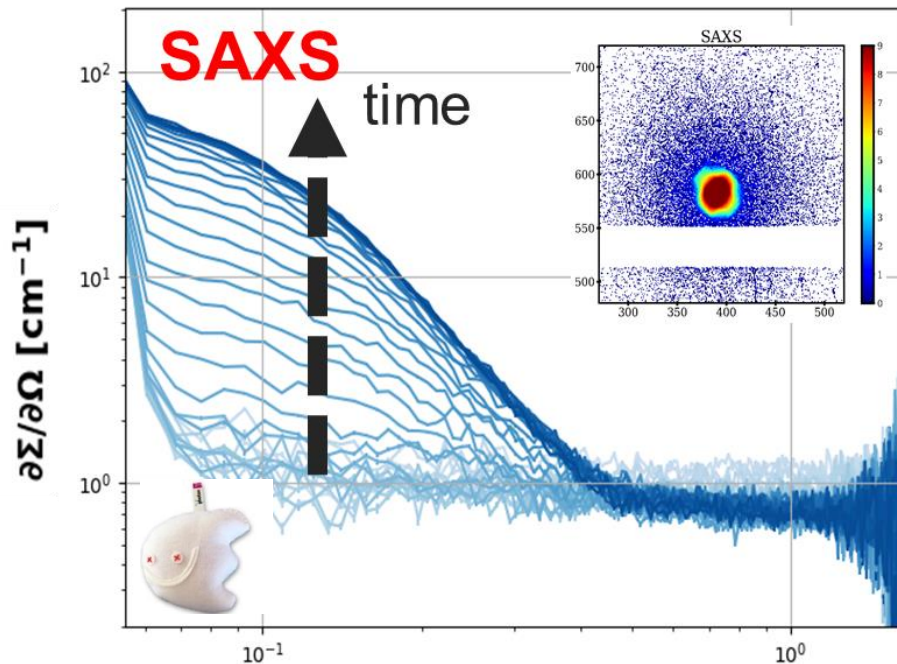
Basic principles of SAXS/SANS

Hydroquinone synthesis of Au NPs:

Gold nanoparticles (Au NPs) synthesis begins with the gold precursor (HAuCl_4) which forms small seeds after reduction with the reducing agent NaBH_4 in the presence of cetyltrimethylammonium bromide (CTAB). The subsequent reduction of HAuCl_4 using Hydroquinone ($\text{C}_6\text{H}_4(\text{OH})_2$) in the presence of AgNO_3 , CTAB, and the seed particles leads to the formation of single crystalline Au NPs.



Temporal structural cross-correlation between organic stabilizing agent (CTAB micelles) and gold seeds? How do they cooperate in the formation of different size/shape of large stabilized gold nanoparticles?

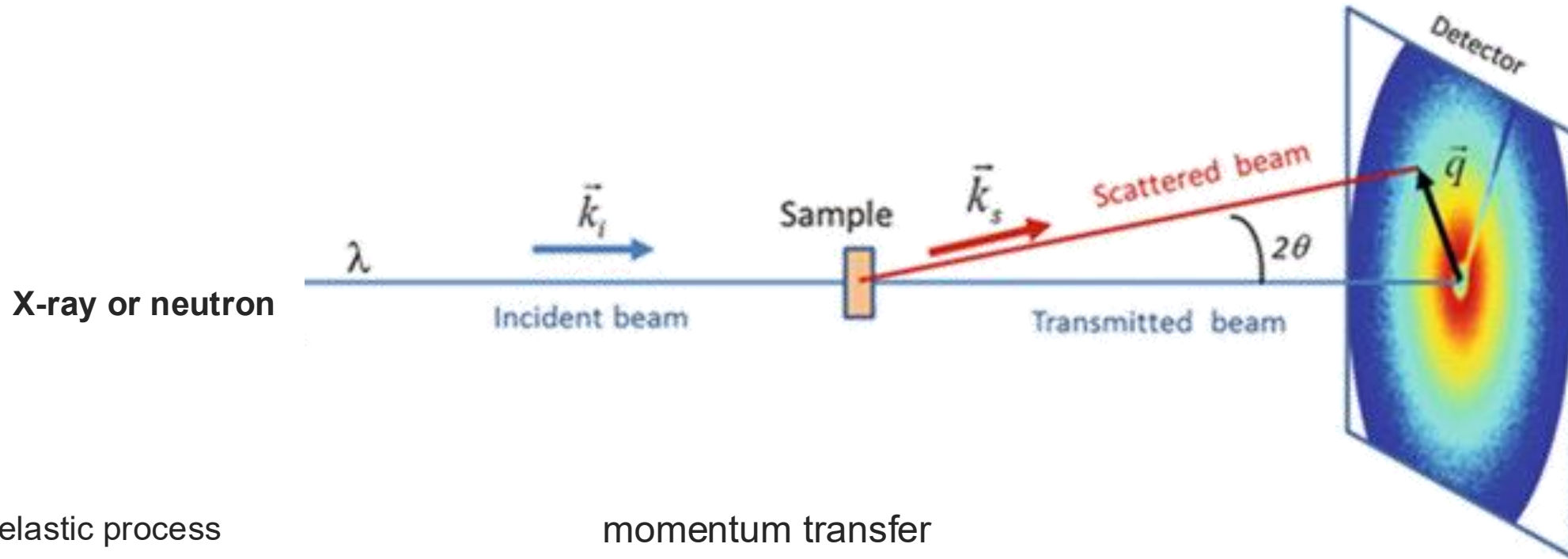


SAXS+SANS = Complementary nanoscale structural information at two different contrast situations

Time evolution of both SAXS and SANS 1d profiles during the in-situ reduction process of gold solution at 35°C.

Basic principles of SAXS/SANS

The typical scattering setup



elastic process

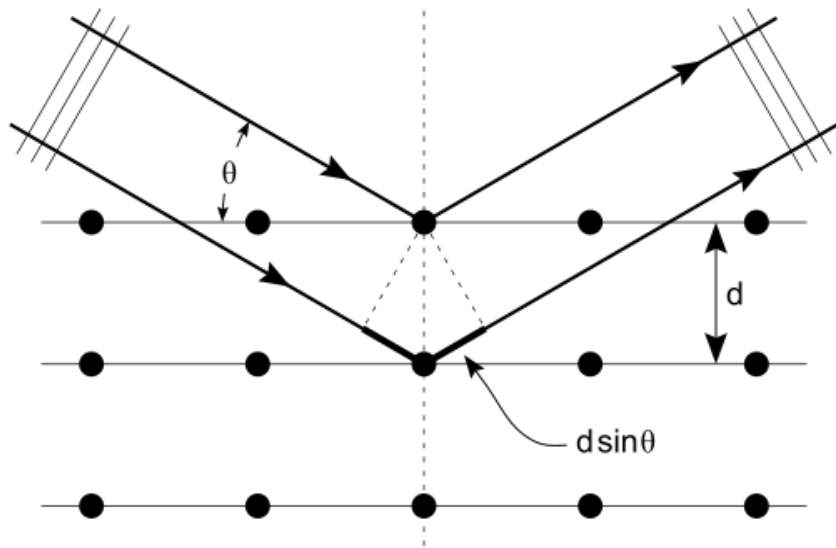
$$\vec{k}_2 - \vec{k}_1 = \vec{Q}, \quad |\vec{k}_1| = |\vec{k}_2|, \quad k = \frac{2\pi}{\lambda}$$

momentum transfer

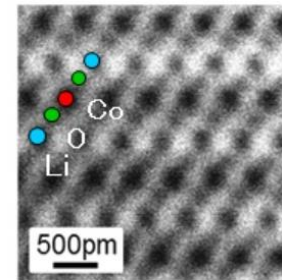
$$|\vec{Q}| = 2|k| \sin(\theta) = 4\pi \sin(\theta) / \lambda$$

Basic principles of SAXS/SANS

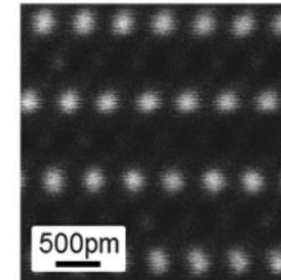
Note 1. Small Angle scattering is somehow a diffraction technique probing the wave interferences from nanoscale scattering objects ! Think about Bragg's law, not on atomic planes but at larger scales, e.g. atoms are replaced by colloids/particles/nanosized areas of the sample !



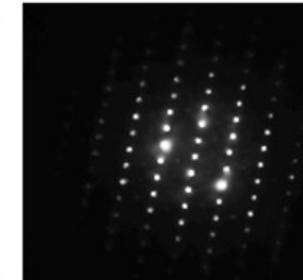
TEM image of a battery LCO cathode



ABF-STEM image

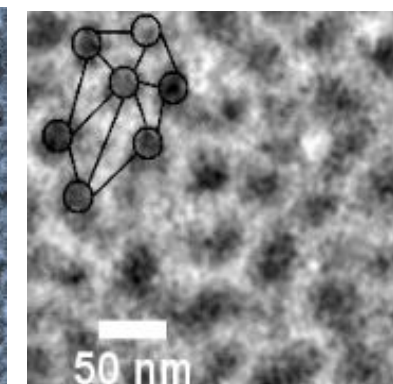
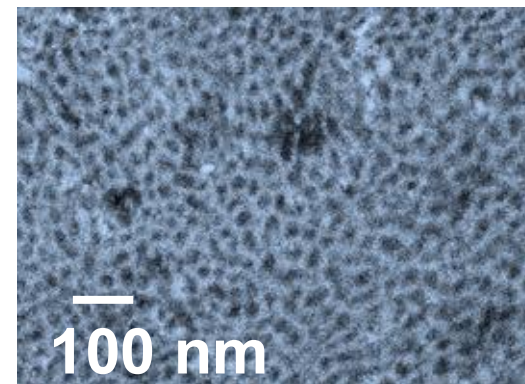


HAADF-STEM image



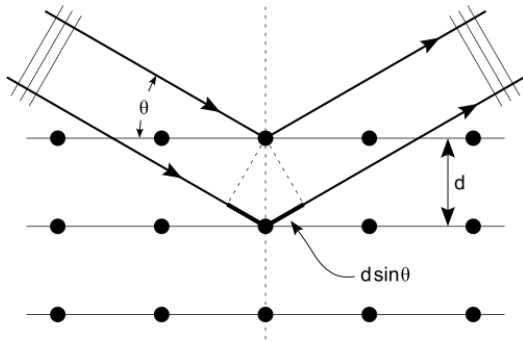
Electronic Diffraction Pattern

MEB image on a nanostructured polymer



Basic principles of SAXS/SANS

Small Angle Scattering: The Big Picture



$$2 d \sin \theta = n \lambda$$

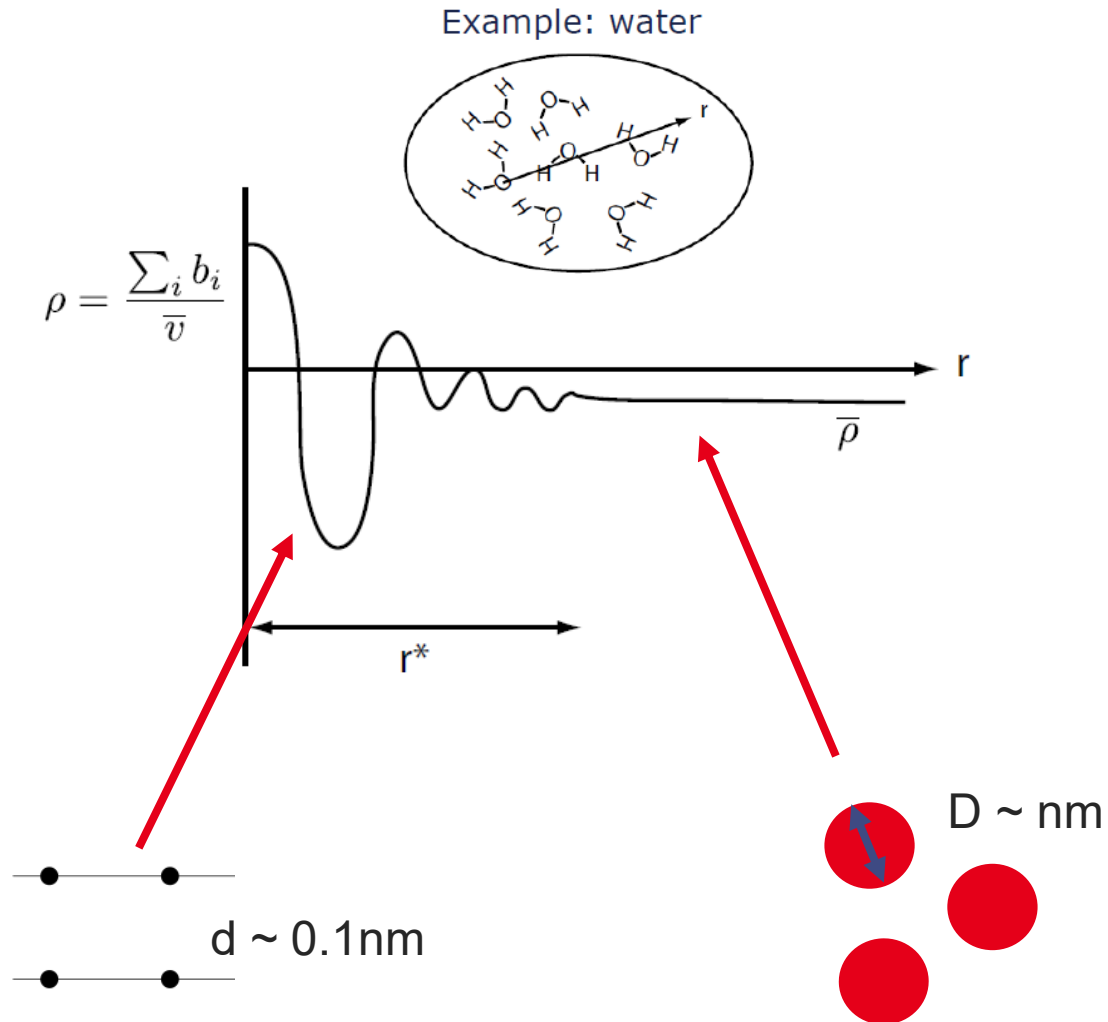
d , distance probed, 1 – 1000 nm
large compared to λ

Wavelengths of Probe, of the order of
0.1 - 2 nm. Of the order of atomic
spacing. FIXED

λ fixed, d large:
 θ must be small!

Basic principles of SAXS/SANS

SAXS/SANS = Fourier transform of the scattering length density (SLD)

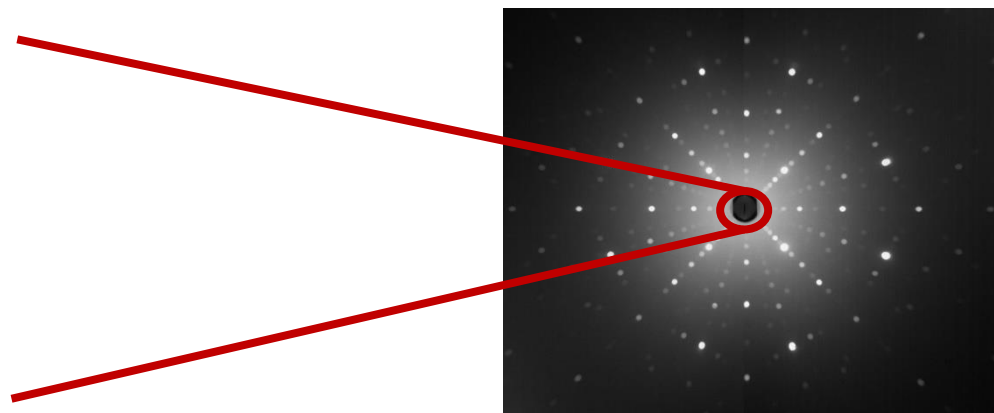
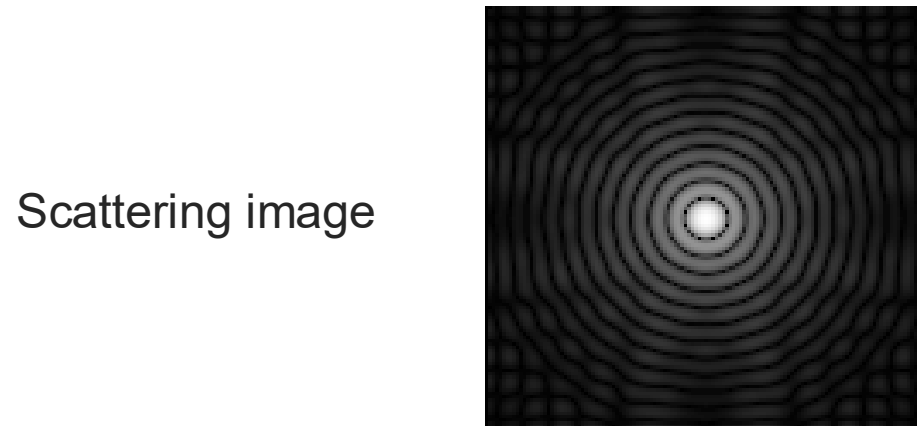
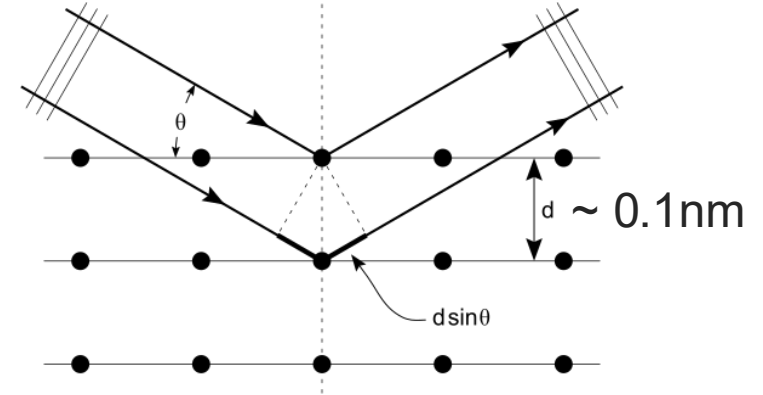
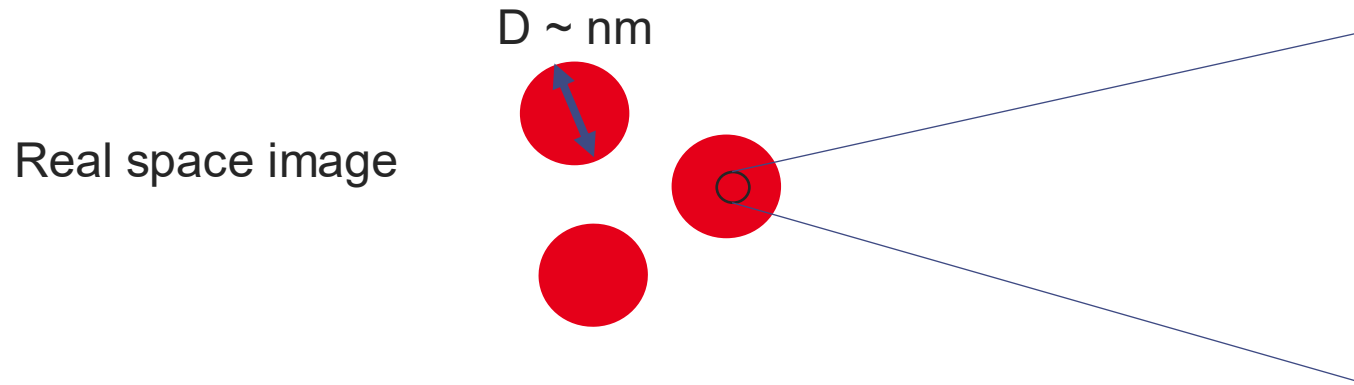


- Individual atoms (characterized by b) loses relevance
- Scattering determined by variations in

the scattering length density $\rho = \frac{\sum_i b_i}{V}$

$$I(\vec{Q}) \propto \left| \int_V \rho(x) e^{-i\vec{Q} \cdot \vec{x}} dx \right|^2$$

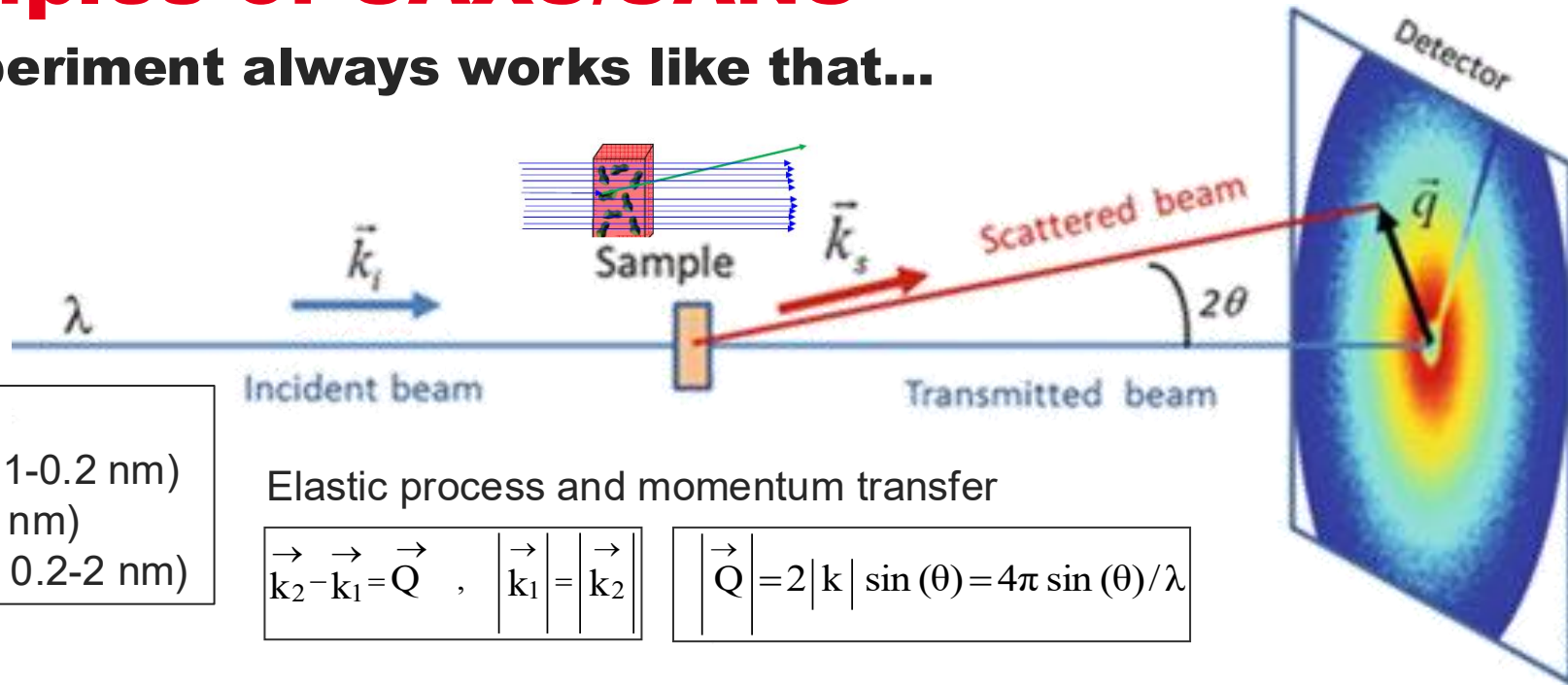
Basic principles of SAXS/SANS



Basic principles of SAXS/SANS

A small angle experiment always works like that...

X-ray or neutron



Radiation sources

Lab X-ray generator ($\lambda = 0.1\text{-}0.2\text{ nm}$)

Synchrotron ($\lambda = 0.03\text{-}0.35\text{ nm}$)

Thermal/cold neutrons ($\lambda = 0.2\text{-}2\text{ nm}$)

Elastic process and momentum transfer

$$\vec{k}_2 - \vec{k}_1 = \vec{Q}, \quad |\vec{k}_1| = |\vec{k}_2|$$

$$|\vec{Q}| = 2|\vec{k}| \sin(\theta) = 4\pi \sin(\theta) / \lambda$$

Magnitude of the scattering vector

$$|\vec{q}| = \frac{4\pi \sin(\theta)}{\lambda}$$

Bragg-equation

$$2d \sin(\theta) = \lambda$$

$$|\vec{q}| = \frac{2\pi}{d}$$

Threshold value:

$$|\vec{q}| \rightarrow 0 \quad d \rightarrow \infty$$

Small angles $\theta < 5^\circ$ Q typically $[0.001 - 1 \text{ \AA}^{-1}]$

$$d = \frac{2\pi}{Q} \quad [6000 - 6 \text{ \AA}]$$

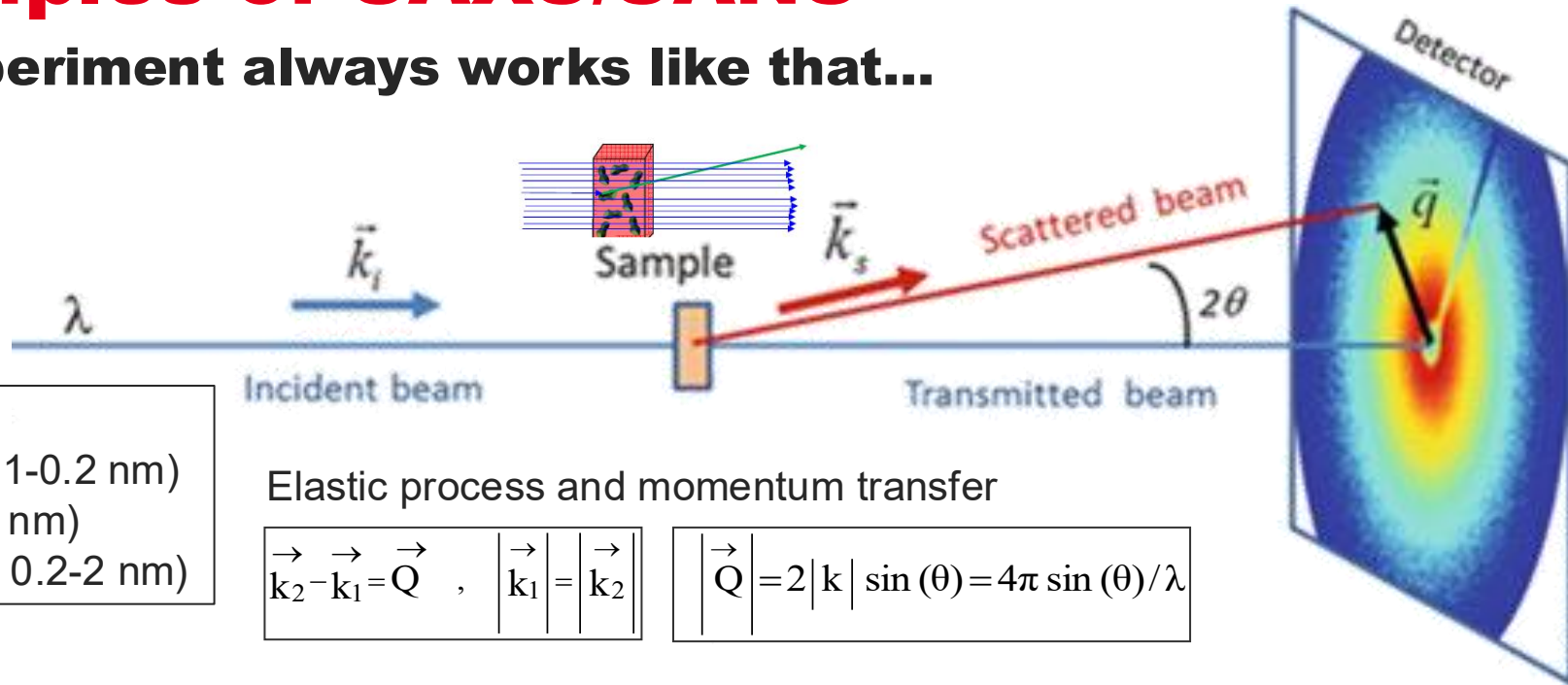
Scattering also allows to investigate structures in the nm- and μm - range.

→ Small-Angle Scattering (SAXS, SANS, USAXS, USANS) ←

Basic principles of SAXS/SANS

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X-ray or neutron



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Elastic process and momentum transfer

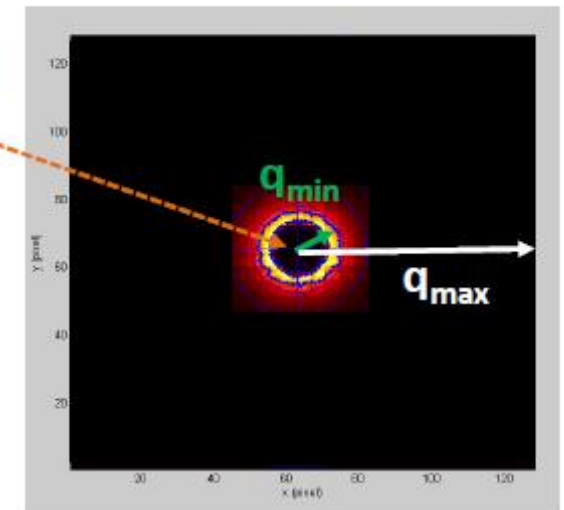
$$\vec{k}_2 - \vec{k}_1 = \vec{Q} \quad , \quad |\vec{k}_1| = |\vec{k}_2| \quad \left| \vec{Q} \right| = 2 |\vec{k}| \sin(\theta) = 4\pi \sin(\theta) / \lambda$$

The sample-detector distance L defines the smallest accessible q_{\min}

The detector size defines the maximum q_{\max} accessible

Q -range is (L, λ) dependent

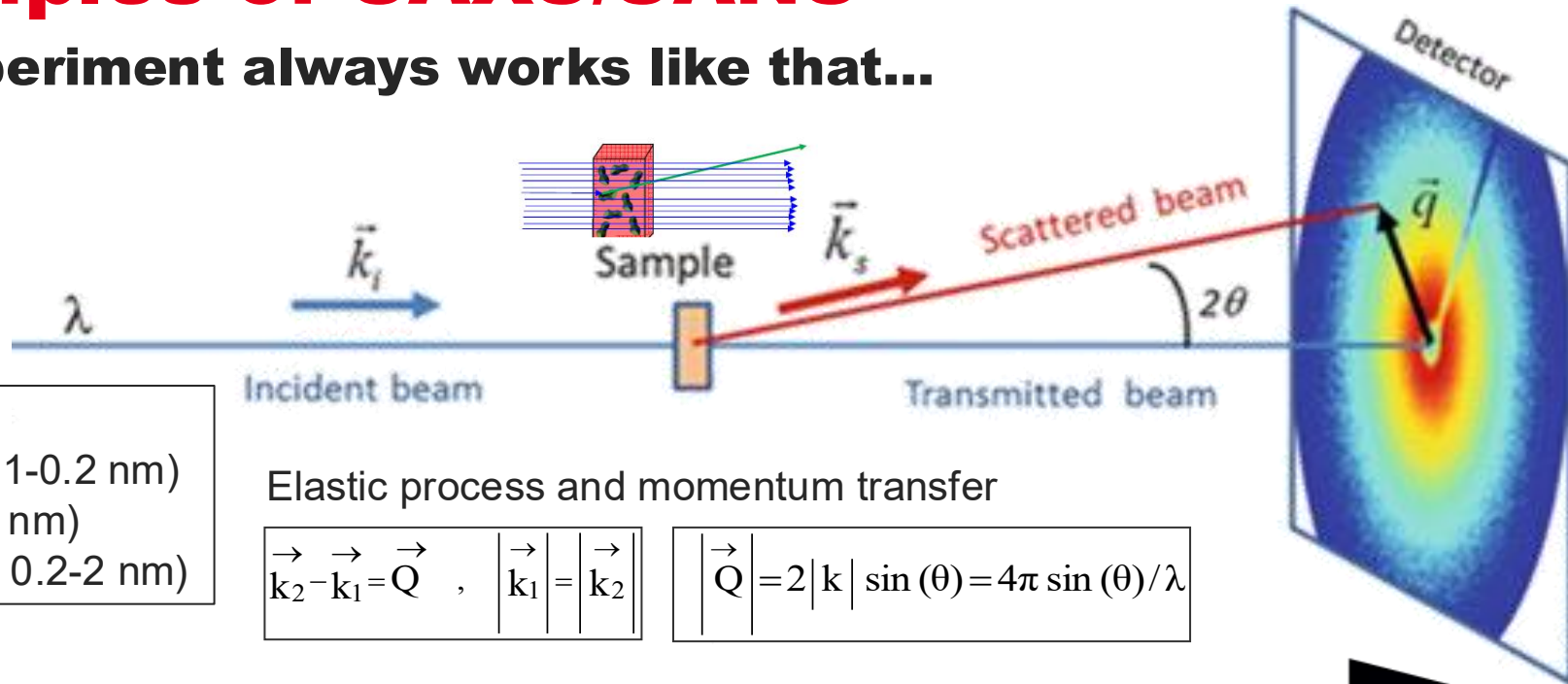
Beamstop



Basic principles of SAXS/SANS

A small angle experiment always works like that...

X-ray or neutron



Radiation sources

Lab X-ray generator ($\lambda = 0.1-0.2$ nm)

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Thermal/cold neutrons ($\lambda = 0.2-2$ nm)

Elastic process and momentum transfer

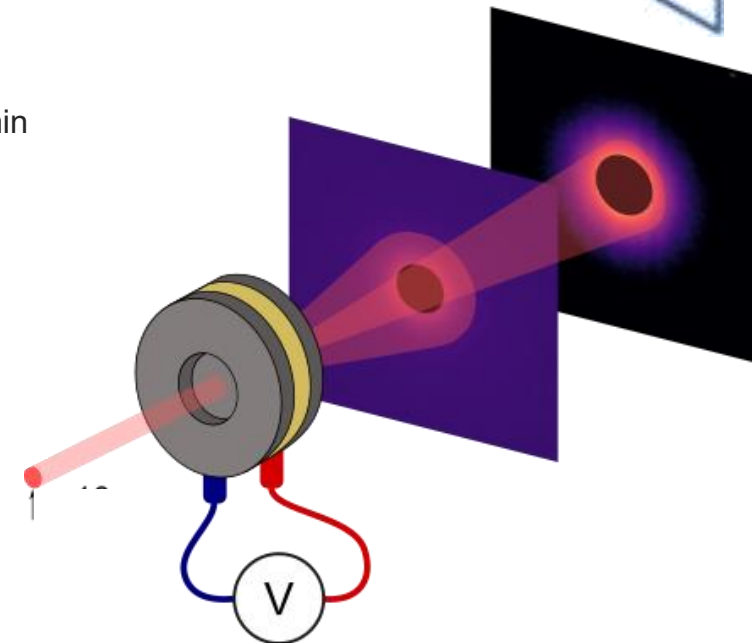
$$\vec{k}_2 - \vec{k}_1 = \vec{Q}, \quad |\vec{k}_1| = |\vec{k}_2|, \quad |\vec{Q}| = 2|\vec{k}| \sin(\theta) = 4\pi \sin(\theta) / \lambda$$

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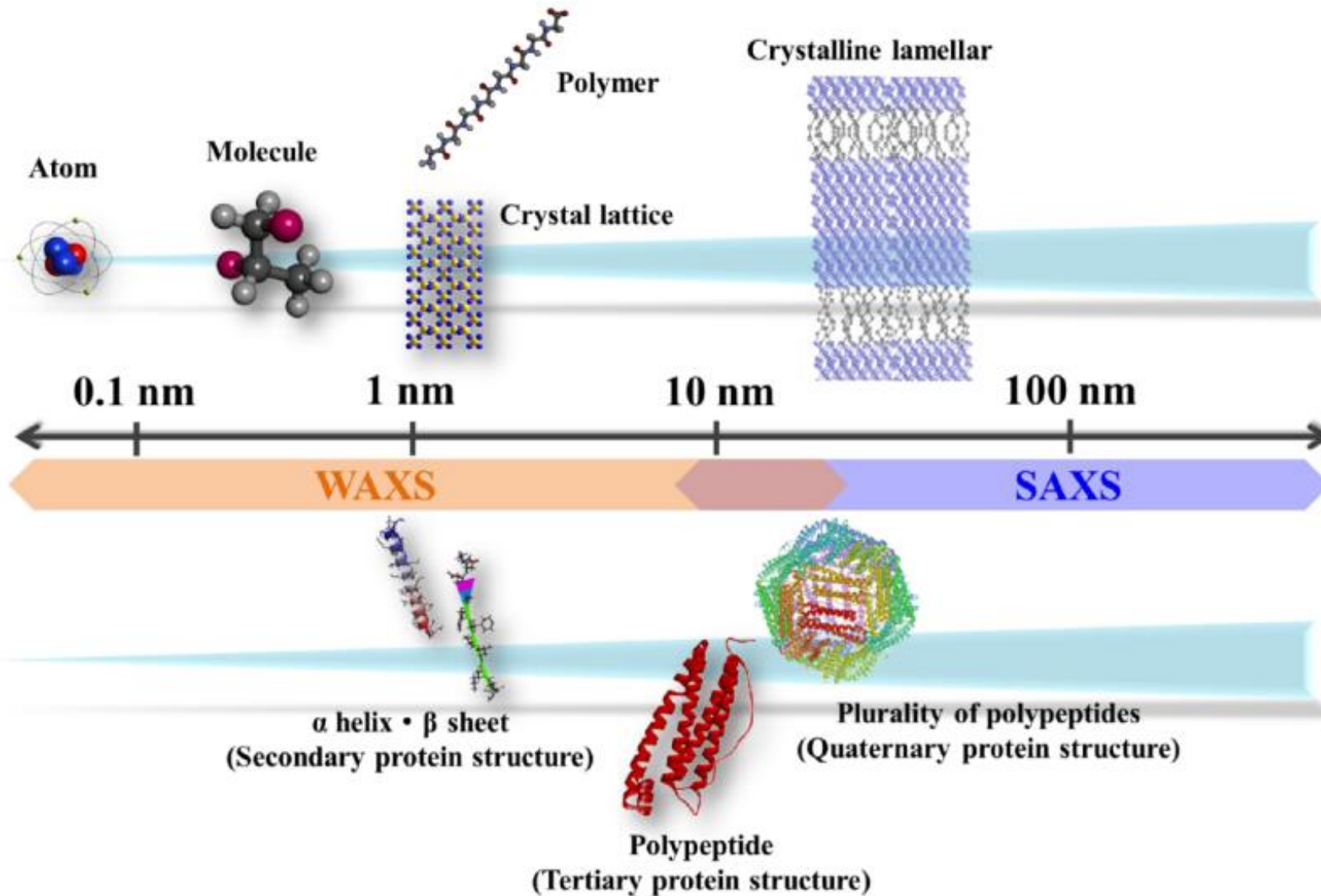
Q -range is (L, λ) dependent

To obtain an extended Q -range, detector is moved !



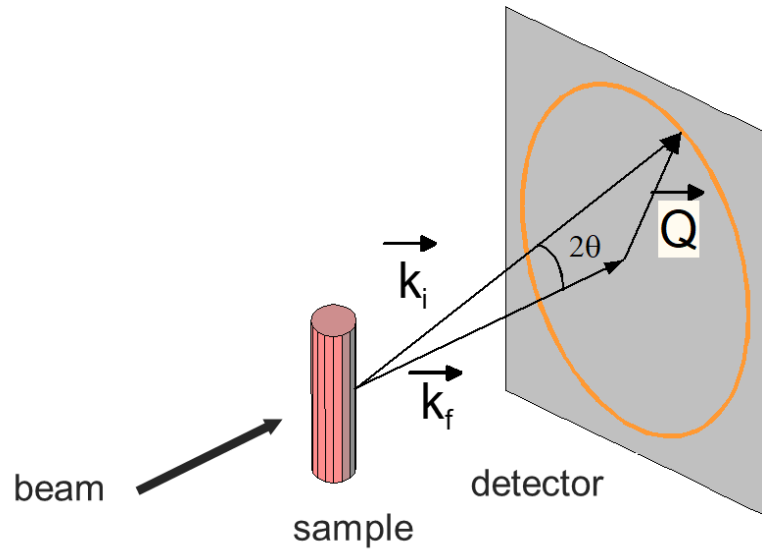
Basic principles of SAXS/SANS

Looking at matter / materials across extended nano-microscales

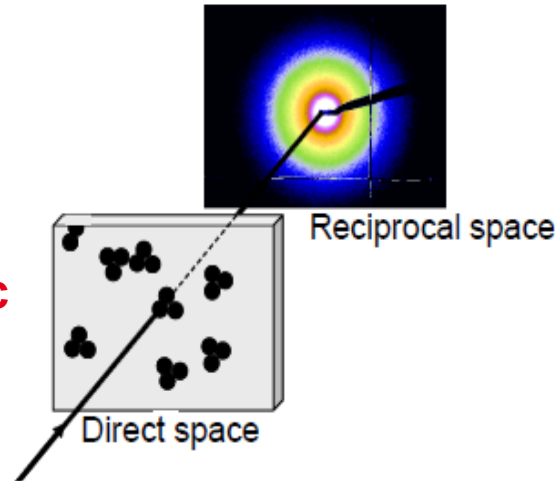


Basic principles of SAXS/SANS

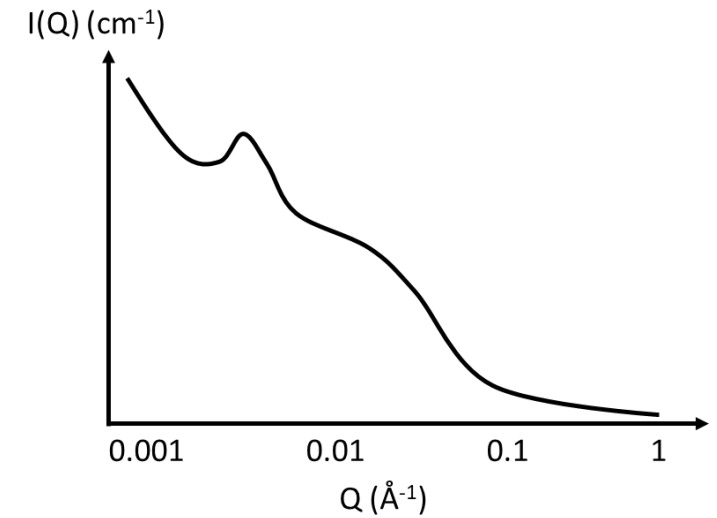
Note 2. Small Angle scattering measures a 2D intensity pattern vs Q



Isotropic sample

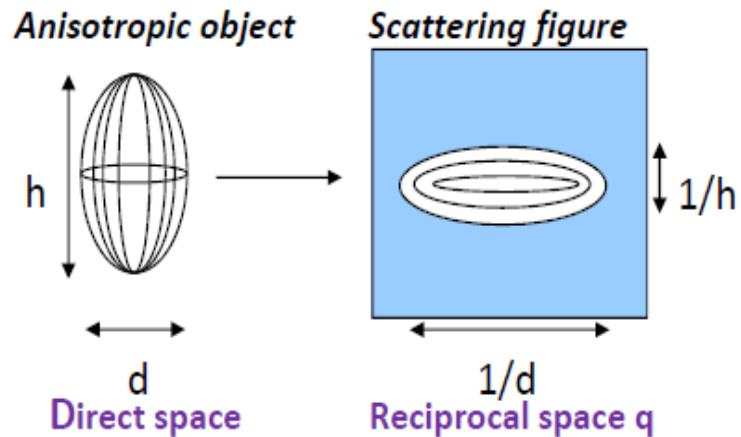
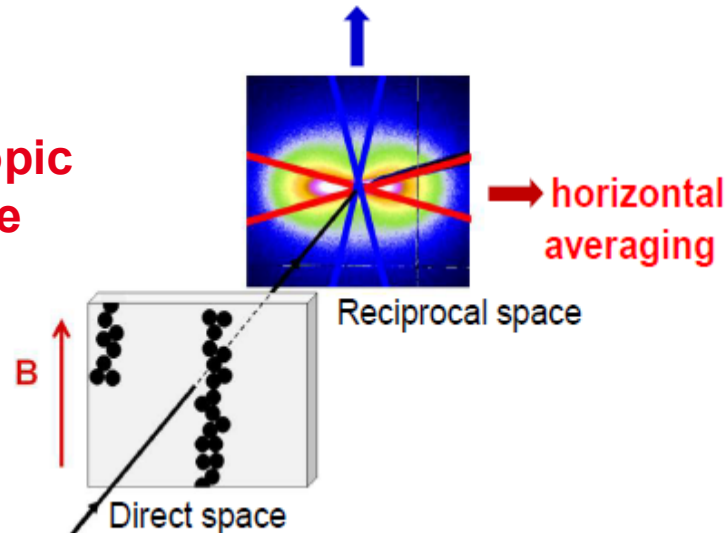


radial averaged intensity $I(Q)$



Vertical averaging

Anisotropic sample



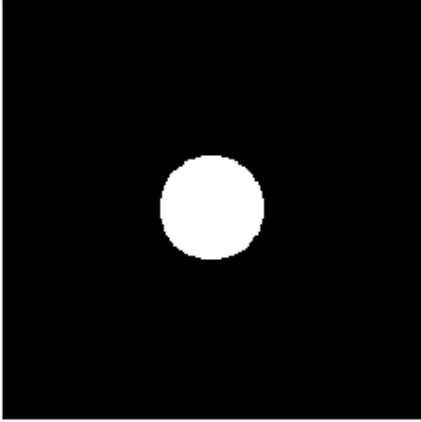
Basic principles of SAXS/SANS



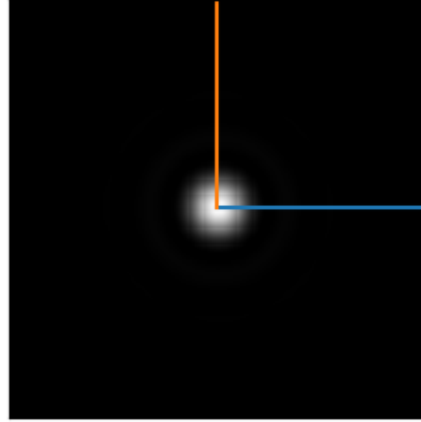
Try it out yourself: Do some 2D Fourier Transforms (using for example python)

$$I(\vec{Q}) \propto \left| \int_V \rho(x) e^{-i\vec{Q} \cdot \vec{x}} dx \right|^2$$

Input Image



Fourier Transform squared



```
import numpy as np
import matplotlib.pyplot as plt

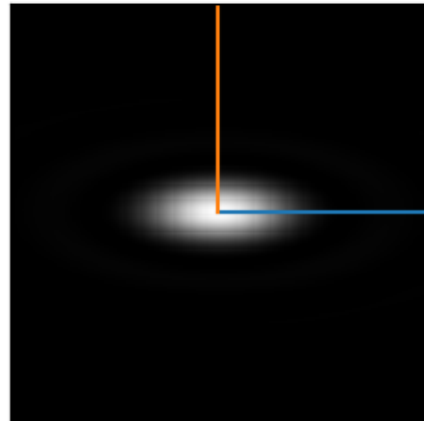
xs = np.linspace(-10,10,1000)
ys = xs.copy()
xx,yy = np.meshgrid(xs, ys)
rs = np.sqrt(xx**2 + yy**2)
ellipse = np.zeros_like(rs)
sphere[rs<R] = 10
a=0.2
b=0.5
ellipse[xx**2 / a**2 +yy**2 / b**2 <1] = 10
ft = np.fft.fft2(ellipse)
ft = np.fft.fftshift(ft)
```

Horizontal and vertical cut identical

Input Image



Fourier Transform squared



```
fig,axes = plt.subplots(1,2)
ax=axes[0]
ax.pcolormesh(xs, ys, ellipse, cmap=plt.cm.grey)
ax.set_xlim([-2,2])#
ax.set_ylim([-2,2])
ax.set_aspect('equal')
ax.set_title('Input Image')

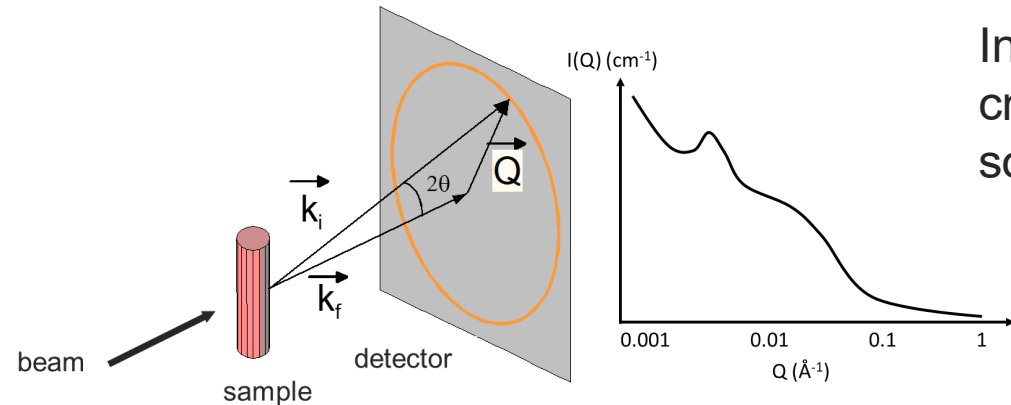
ax=axes[1]
ax.pcolormesh((np.abs(ft)**2)[400:600,400:600], cmap=plt.cm.grey)
ax.set_aspect('equal')
ax.set_title('Fourier Transform squared')
for ax in axes:
    ax.set_yticks([])
    ax.set_xticks([])
ax.plot([100,200],[100,100])
ax.plot([100,100],[100,200])
```

isotropic scattering

Basic principles of SAXS/SANS



Note 3. SAXS is analytical (sometimes)



Intensity $I(Q)$ is the absolute scattering cross section $d\sigma(Q)/d\omega$ in units of cm^{-1} = probability of a neutron of wavelength λ being scattered, per unit solid angle, at that Q .

Differential cross-section
= scattered intensity $I(Q)$

$$\frac{d\sigma}{d\Omega}(Q) = \frac{1}{N} \left| \int_{V_s} dr \rho(r) e^{i\mathbf{Q}\cdot\mathbf{r}} \right|^2 = \frac{1}{N} I(Q)$$

$$I(Q) = N \Delta\rho(r)^2 F(Q, r)^2 S(Q)$$

Scattering contrast $\Delta\rho$

Scattering length b of element i
concentration c_i , density ρ_i

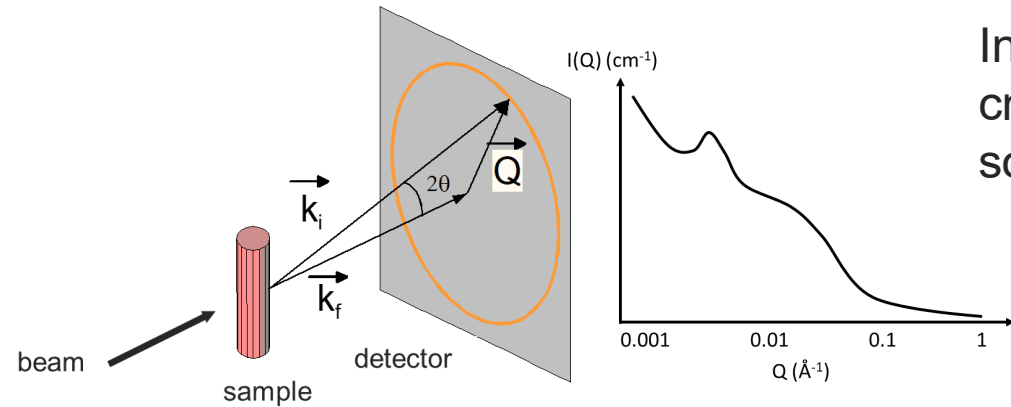
$$\Delta\rho(r) = \sum_i (b_i c_i \rho_i)_{\text{Phase1}} - \sum_j (b_j c_j \rho_j)_{\text{Phase2}}$$

Form factor $P(Q)$ size, shape of particle (we note $P(Q)=F(Q)^2$)

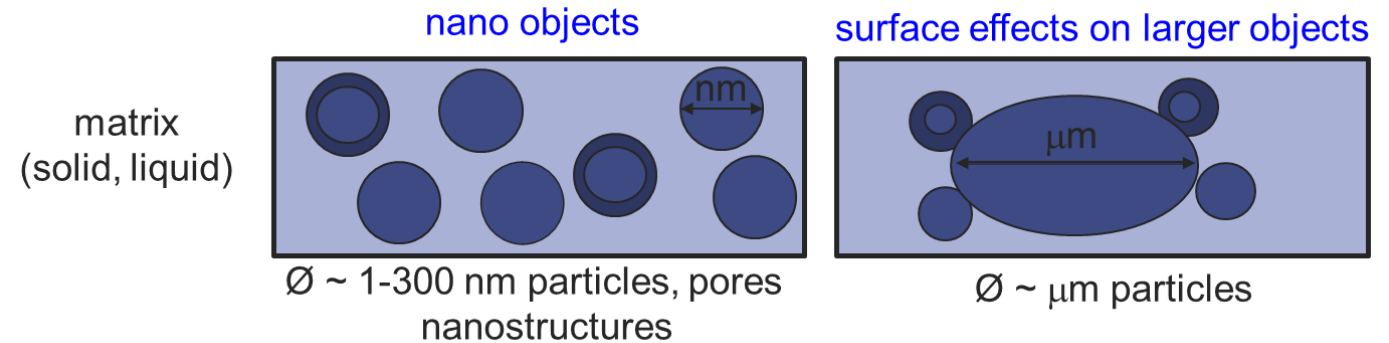
Structure factor $S(Q)$ arrangement of particles

Basic principles of SAXS/SANS

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Intensity $I(Q)$ is the absolute scattering cross section $d\sigma(Q)/d\omega$ in units of cm^{-1} = probability of a neutron of wavelength λ being scattered, per unit solid angle, at that Q .



$$I(Q) = N \Delta\rho(r)^2 F(Q, r)^2 S(Q)$$

Scattering contrast $\Delta\rho$

Scattering length b of element i
concentration c_i , density ρ_i

$$\Delta\rho(r) = \sum_i (b_i c_i \rho_i)_{\text{Phase1}} - \sum_j (b_j c_j \rho_j)_{\text{Phase2}}$$

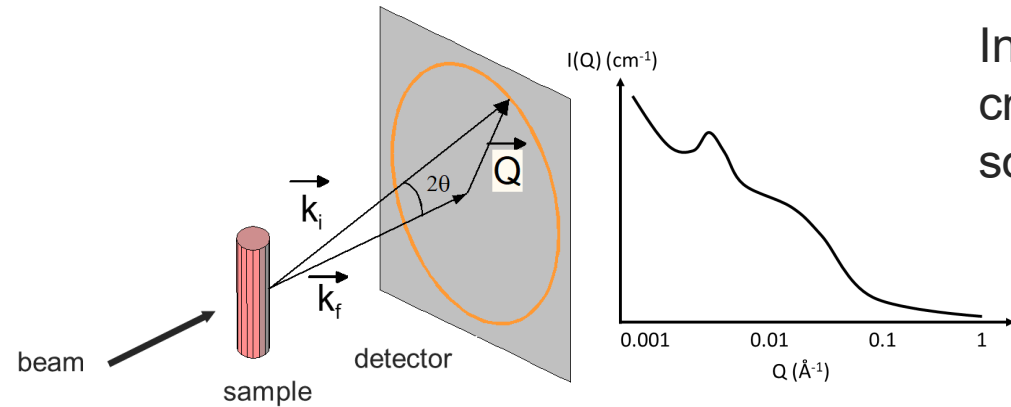
Form factor $P(Q)$ size, shape of particle (we note $P(Q)=F(Q)^2$)

Structure factor $S(Q)$ arrangement of particles

Basic principles of SAXS/SANS



Note 3. SAXS is analytical (sometimes)



Intensity $I(Q)$ is the absolute scattering cross section $d\sigma(Q)/d\omega$ in units of cm^{-1} = probability of a neutron of wavelength λ being scattered, per unit solid angle, at that Q .

What do we measure experimentally ?

Counts in detector of area A at radius R and distance L from sample

$$I(R, \lambda) = I_0(\lambda) \cdot \frac{\partial \Sigma(Q)}{\partial \Omega} \cdot \Delta\Omega(R) \cdot t \cdot T(\lambda) \cdot \eta(\lambda)$$

Solid Angle $\Delta\omega = A/L^2$

t = sample thickness

T = transmission

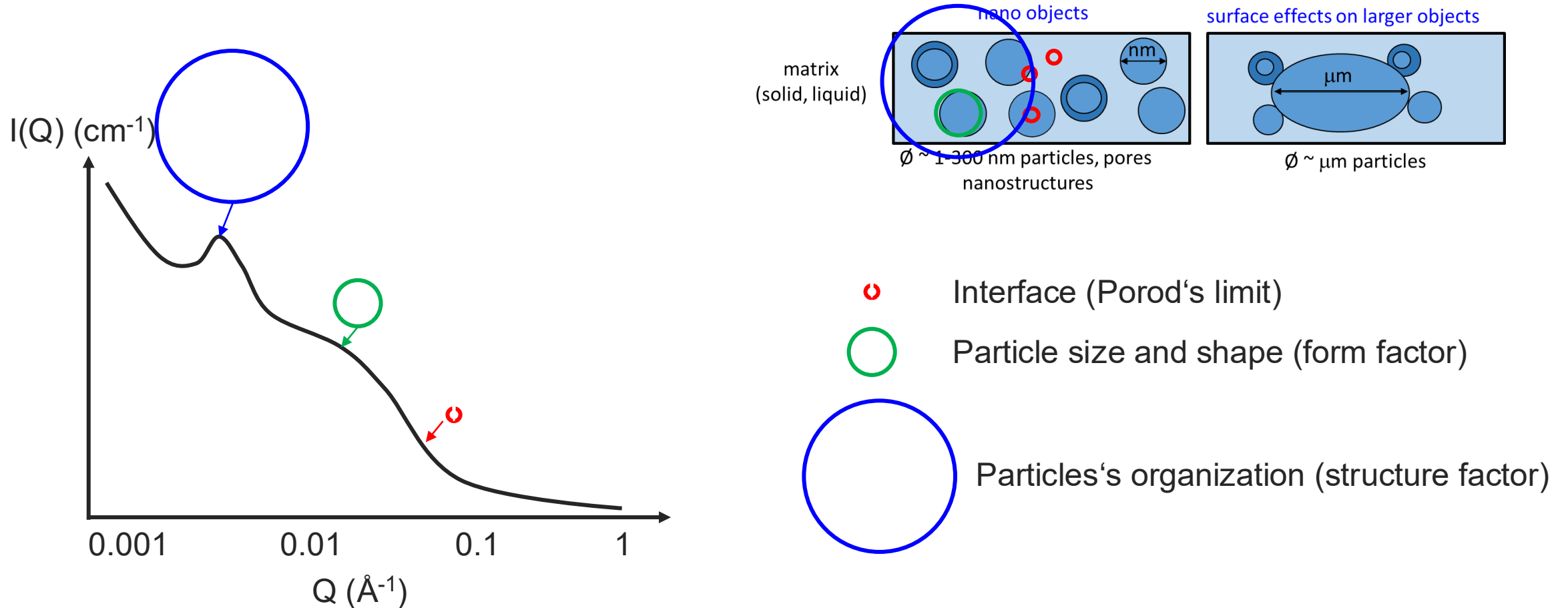
η = detector efficiency

$\tan(\theta) = R/L$

Basic principles of SAXS/SANS



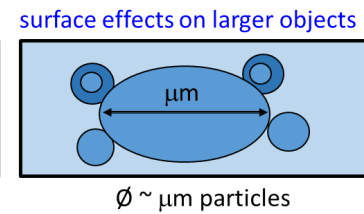
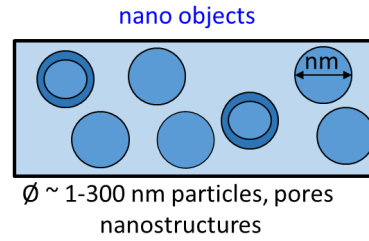
Note 4. SANS is like having a microscope with a magnifying glass of diameter $2\pi/Q$
You get a signal when you have a contrast !



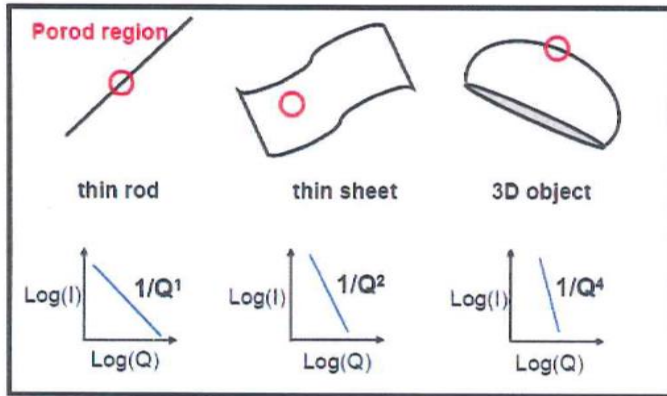
Interfaces

SANS/SAXS from disperse particles

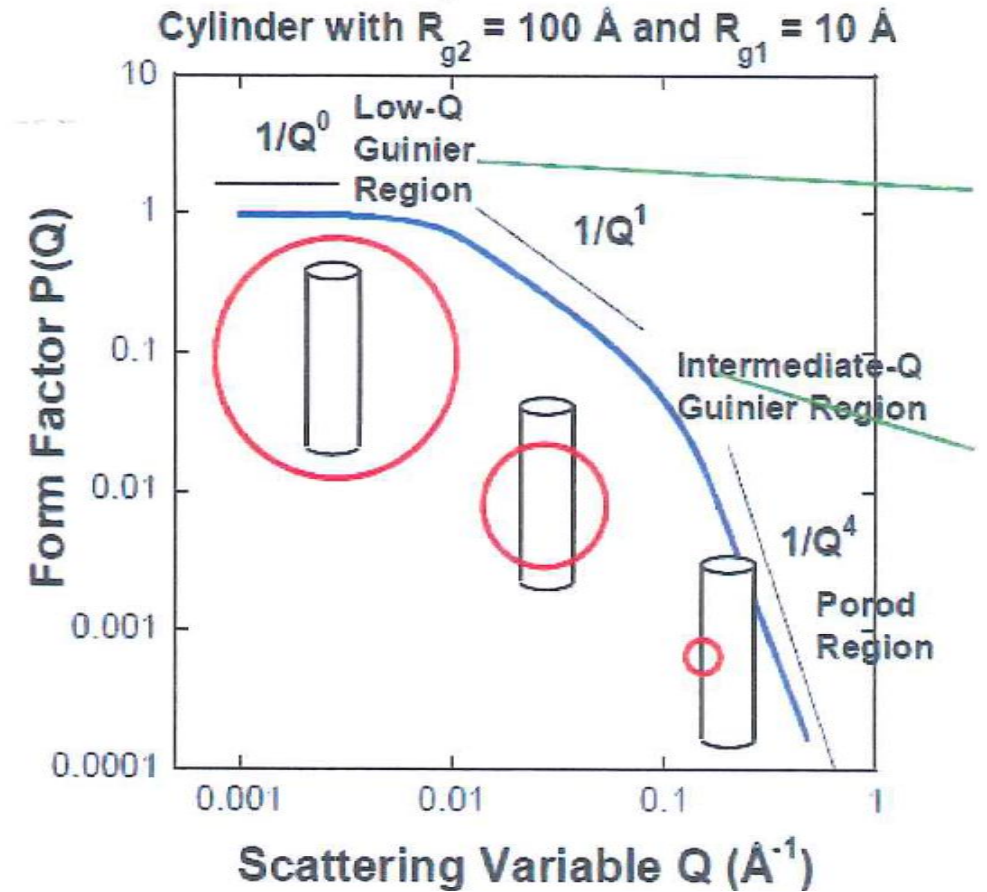
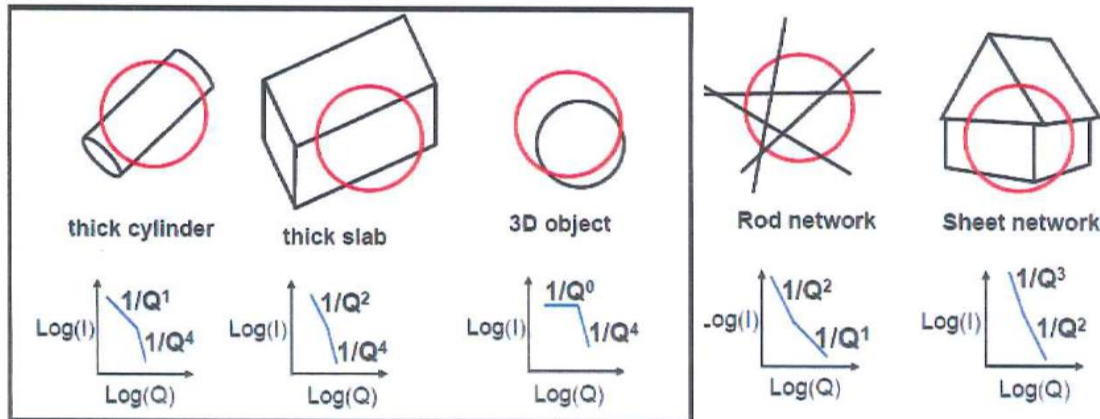
matrix
(solid, liquid)



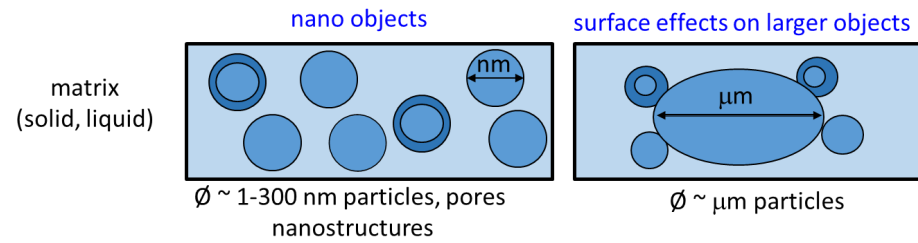
Assortment of Porod Exponents



More Porod Exponents



Interfaces



Fractal aggregates or pores

I(Q) from fractals give straight line regions on log-log plots
 Ex : rocks, coals, cement, zeolites, silicas, battery electrodes !

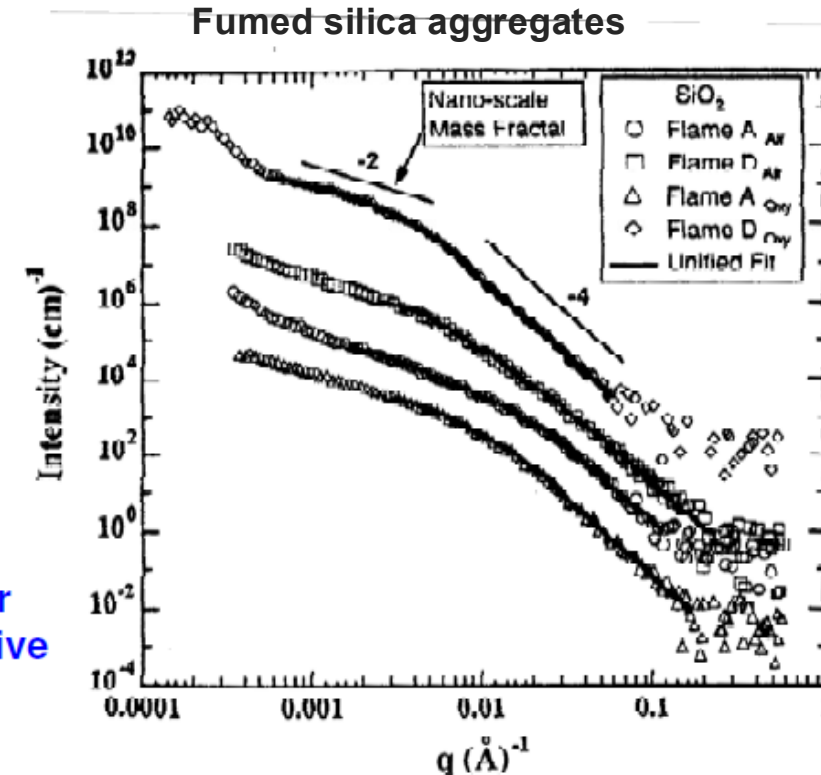
Volume or Mass Fractals give Q^{-d} where d ranges:

- d = 1 – thin rods,
- d ~ 1.8 diffusion limited aggregation (particles stick on first contact)
- d ~ 2.1 reaction limited aggregation (some relaxation after sticking)
- d = 3 perfectly rough,

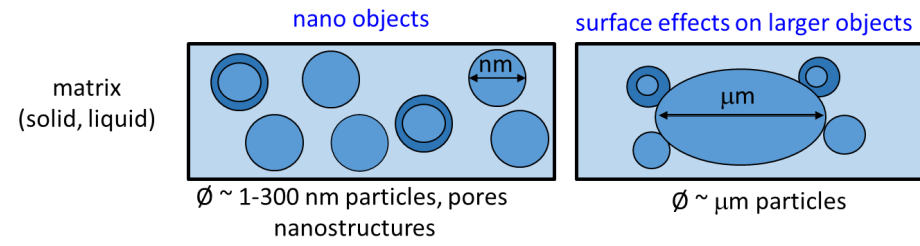
Surface fractals give $Q^{-(6-d)}$ where
 d = 2 - smooth i.e. Q^{-4} Porod law
 up to d = 3 - perfectly rough.

Other scattering laws also give power laws in Q !!! Diffuse interfaces can give steeper than Q^{-4} , to Q^{-6}

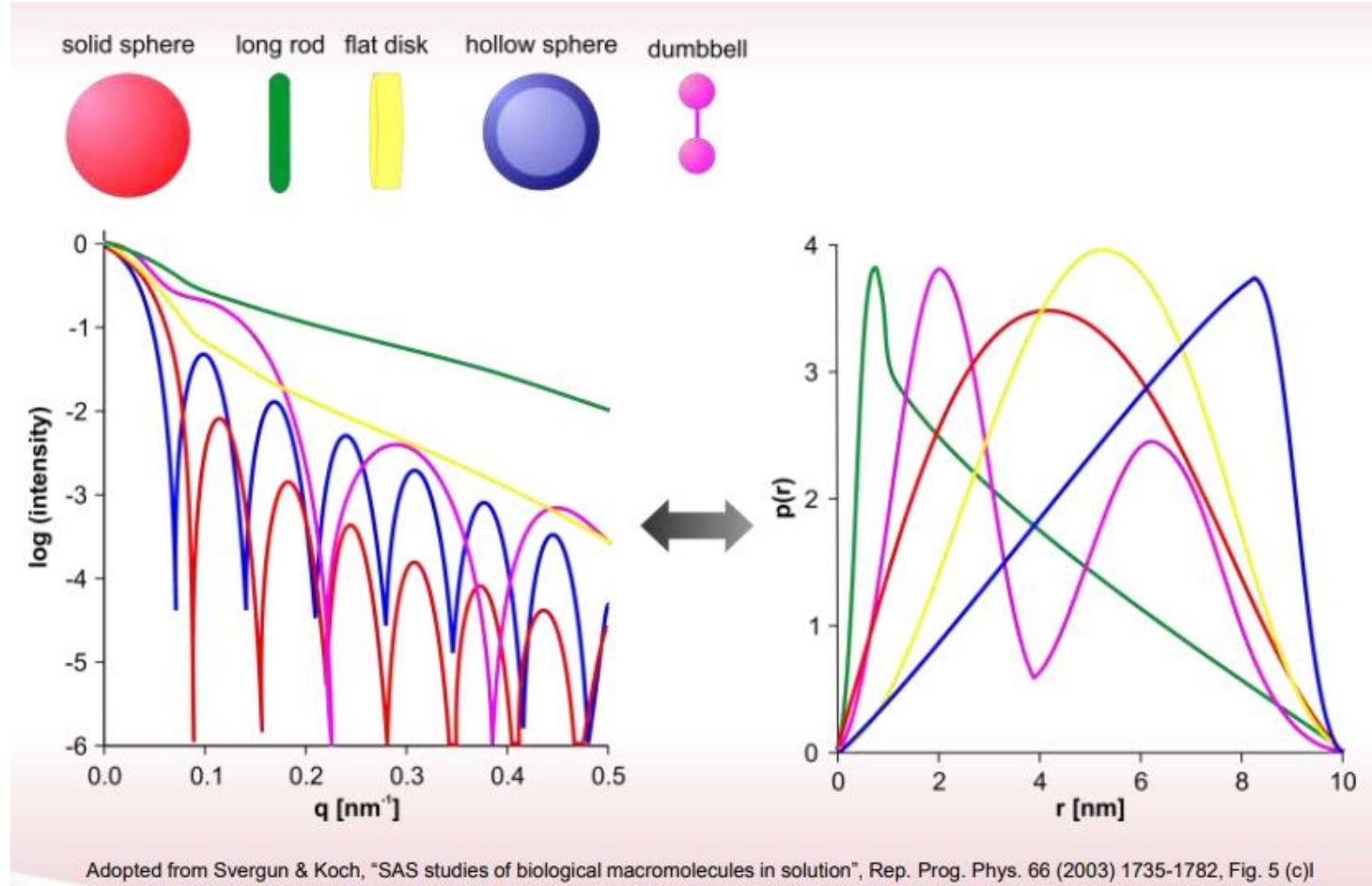
Slopes are meaningful !



Form Factors P(Q)

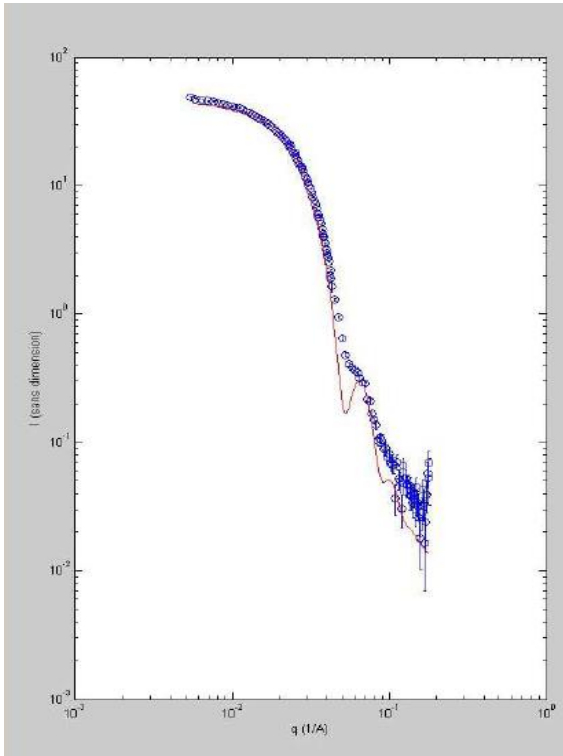
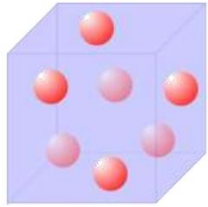
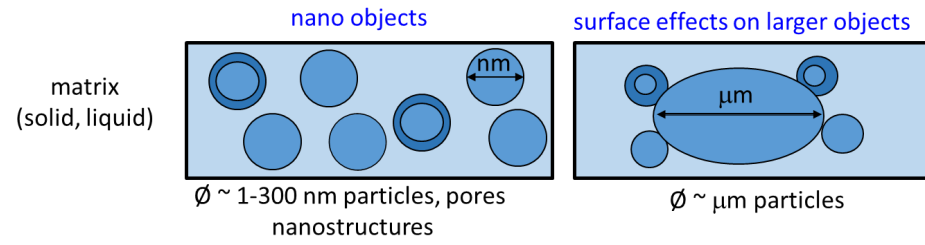


Single objects (diluted suspensions / solutions) have very different SAXS fingerprints



Adopted from Svergun & Koch, "SAS studies of biological macromolecules in solution", Rep. Prog. Phys. 66 (2003) 1735-1782, Fig. 5 (c)

Form Factors P(Q)



Dilute sample

Modelling : Form factor of sphere

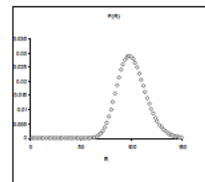
$$P_{sphere}(q, R) = \left[3 \frac{(\sin(qR) - qR \cdot \cos(qR))}{(qR)^3} \right]^2$$

Calculation of form factor with $R \sim 86 \text{ \AA}$

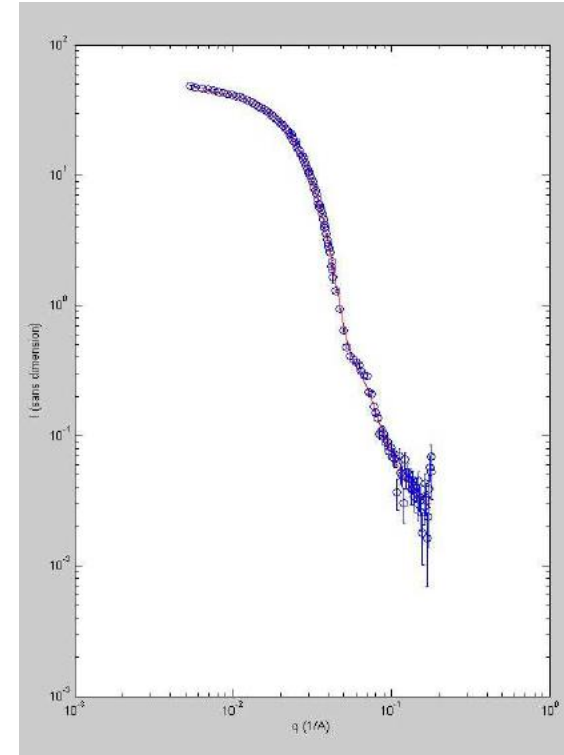
Why oscillations are not reproduced ?

- Polydispersity

$$P(R, R_0, \sigma) = \frac{1}{\sqrt{2\pi}R\sigma} \exp\left[-\frac{\ln\left(\frac{R}{R_0}\right)^2}{2\sigma^2} \right]$$



Log normal



Dilute sample

Modelling : Form factor of sphere

$$P_{sphere}(q, R) = \left[3 \frac{(\sin(qR) - qR \cdot \cos(qR))}{(qR)^3} \right]^2$$

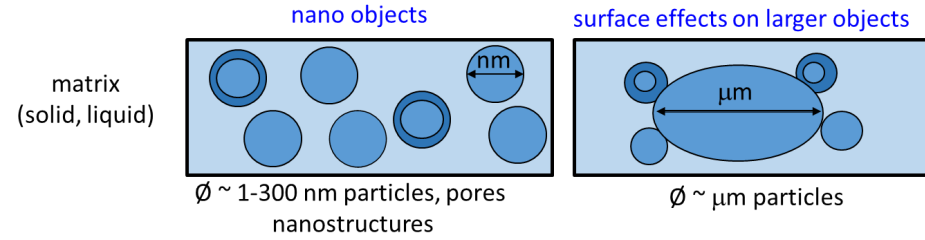
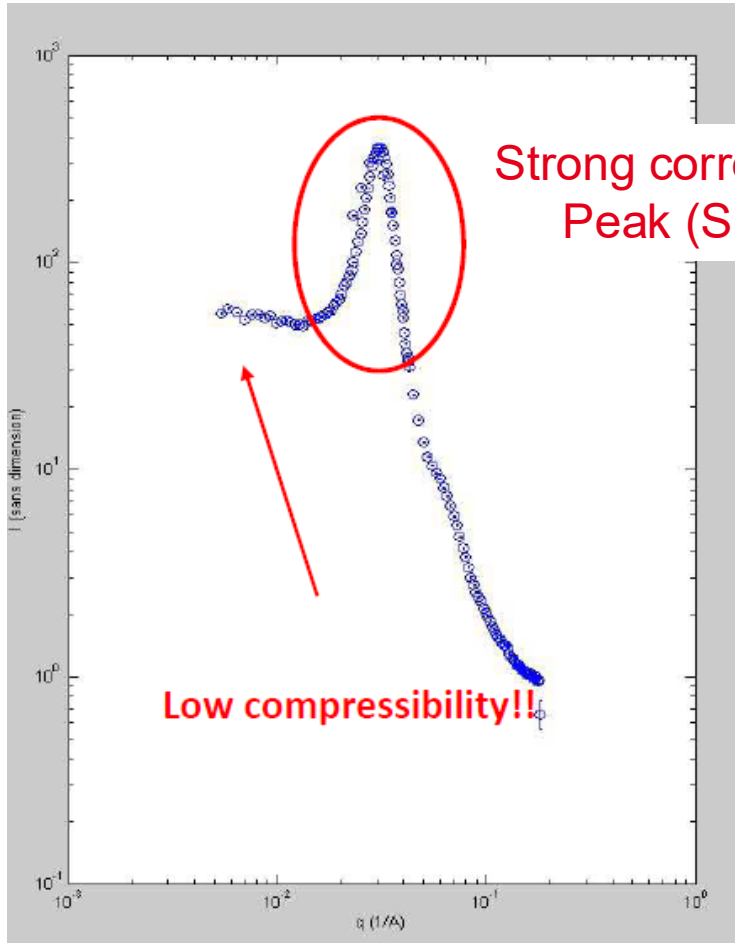
$$P(R, R_0, \sigma) = \frac{1}{\sqrt{2\pi}R\sigma} \exp\left[-\frac{\ln\left(\frac{R}{R_0}\right)^2}{2\sigma^2} \right]$$

$R = 72 \text{ \AA}, \sigma = 0.23$

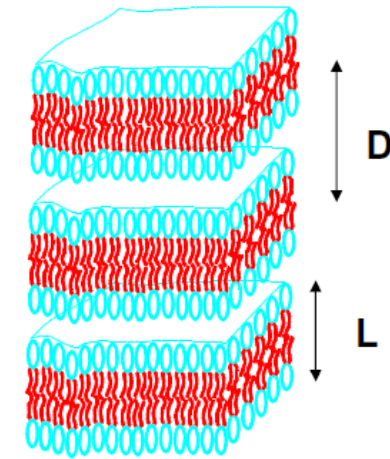
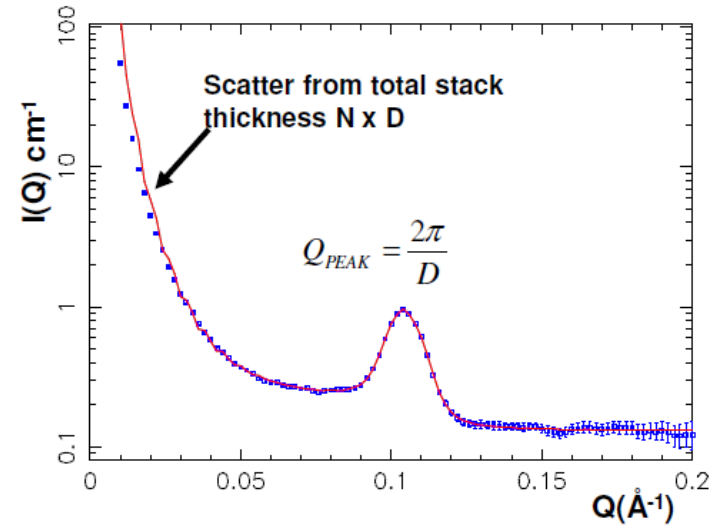
Form Factor P(Q) + Structure Factor S(Q)



Concentrated solution



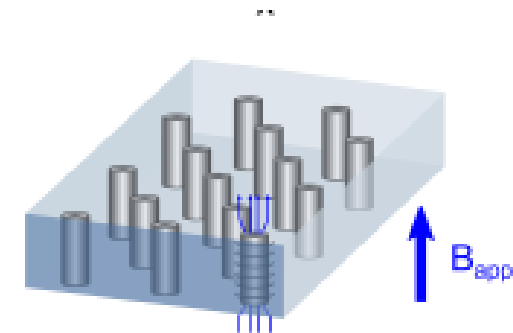
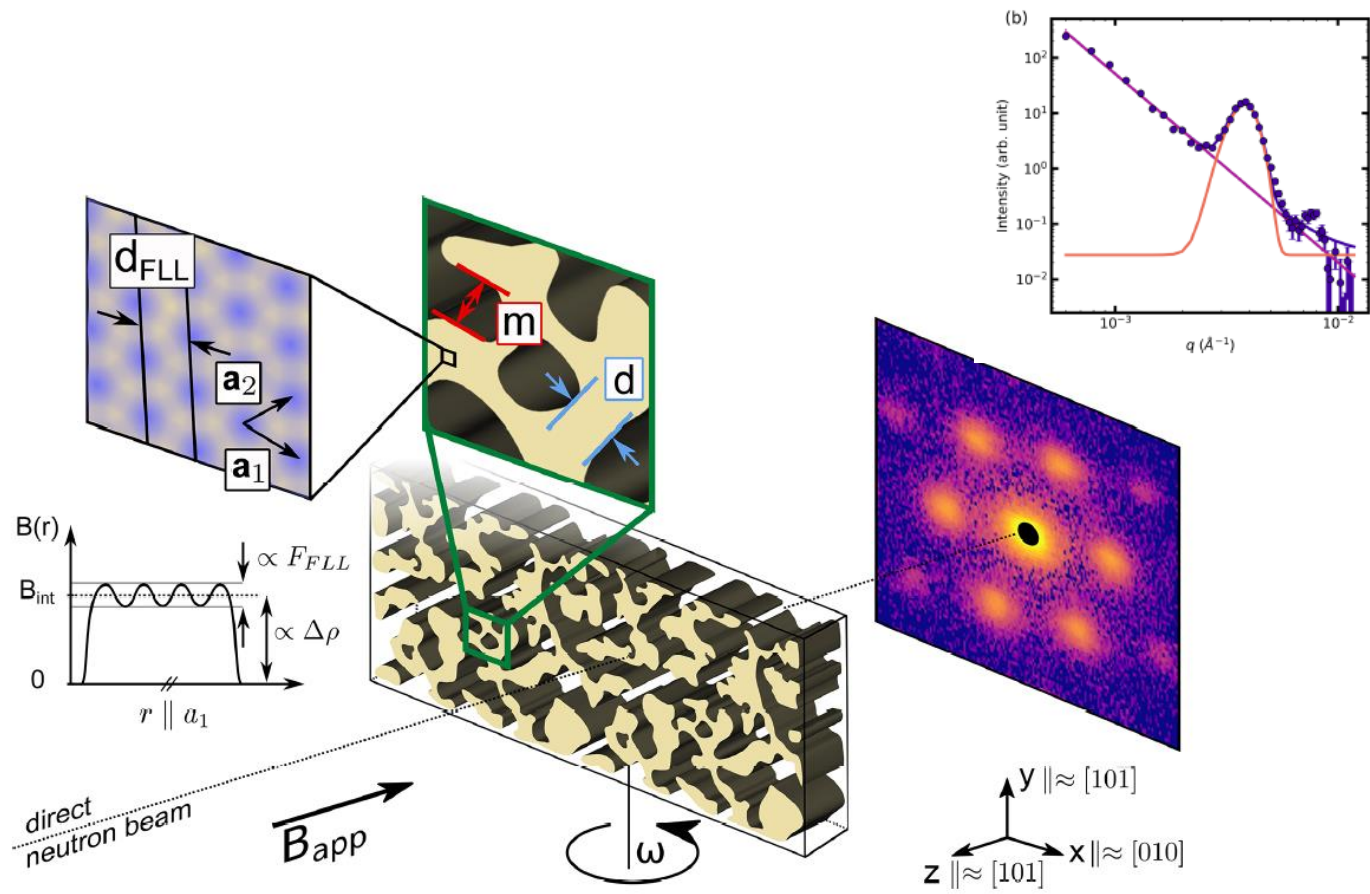
Lamellar assemblies
Ex : 3% DPPC lipid in D2O



Fit has 18 layers of (L = 4.64nm lipid + 1.32nm water) D = 5.96 nm
 $\sigma_D/D = 7\%$, $\sigma_L/L = 10\%$ (fixed)

Form Factor $P(Q)$ + Structure Factor $S(Q)$

Side note: SANS crystallography. Bragg peaks, but not just :)



$d \sim 100\text{-}200 \text{ nm}$ \rightarrow SANS

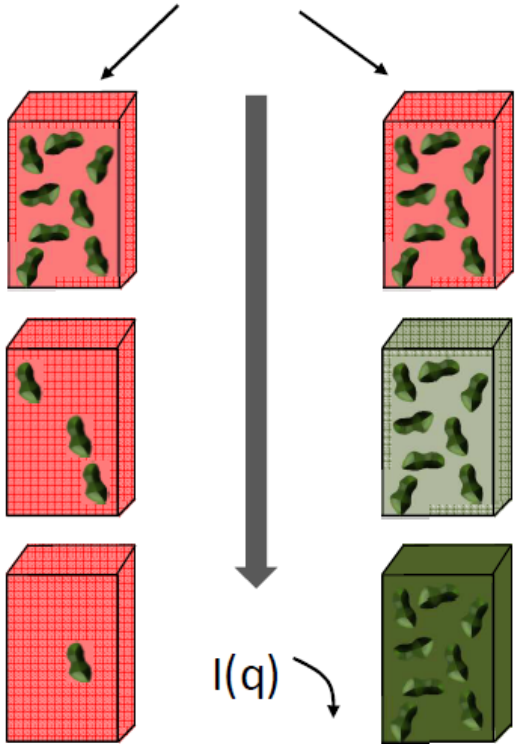
- Tunable crystal structure (T, B)
- Extract Vortex Lattice structure
- Extract size of well-ordered domains

Track for different T, B, and also current

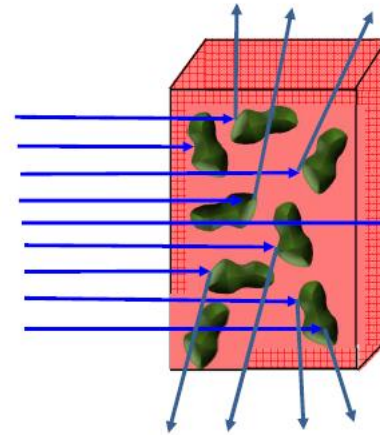
Practicalities

Samples must be concentrated enough (but not too much) with a sufficient contrast

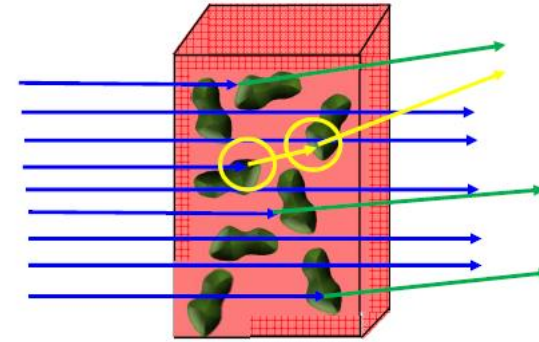
$$I(Q) \sim n \cdot \Delta\rho^2 \cdot P(Q) \cdot S(Q)$$



Incoherent scattering ?

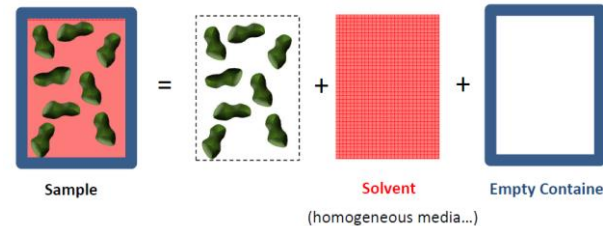


Multiple scattering ?



Adapt sample thickness

- 1) Choose X vs N
- 2) Choose set-up
- 3) Choose sample environment



Calculate SLD prior to experiment

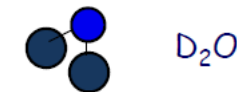
$$\rho_i = \sum_i \frac{b_i}{V} \quad V = \frac{d \cdot N_A}{M} \quad V_{H_2O} = \frac{M_{H_2O} \bar{v}_{H_2O}}{N_A}$$

$$b_H = -0.374 \cdot 10^{-12} \text{ cm}; \quad b_D = 0.667 \cdot 10^{-12} \text{ cm}; \quad b_O = 0.580 \cdot 10^{-12} \text{ cm}$$



$$\rho_{H_2O} = \frac{2b_H + b_O}{V_{H_2O}} = \frac{(5.81 - 3.74 \cdot 2) \text{ fm}}{30 \text{ \AA}^3}$$

$$\rho_{H_2O} = -0.56 \cdot 10^{10} \text{ cm}^{-2}$$

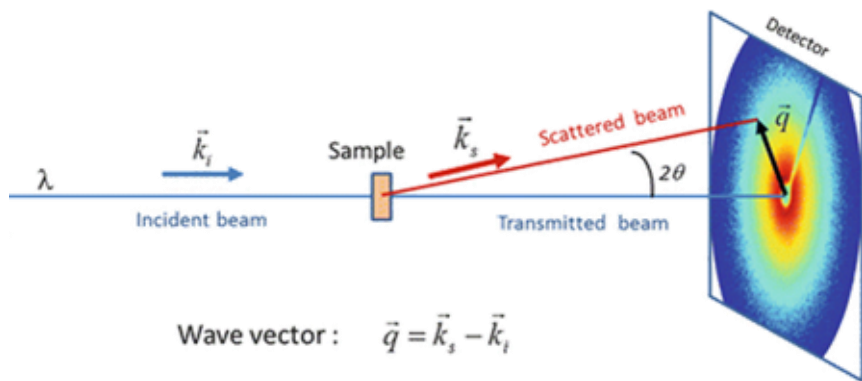


$$\rho_{D_2O} = \frac{2b_D + b_O}{V_{H_2O}} = \frac{(6.67 \cdot 2 + 5.81) \text{ fm}}{30 \text{ \AA}^3}$$

$$\rho_{D_2O} = 6.38 \cdot 10^{10} \text{ cm}^{-2} \quad \mathbf{30}$$

Courtesy of Fabrice Cousin

Practicalities

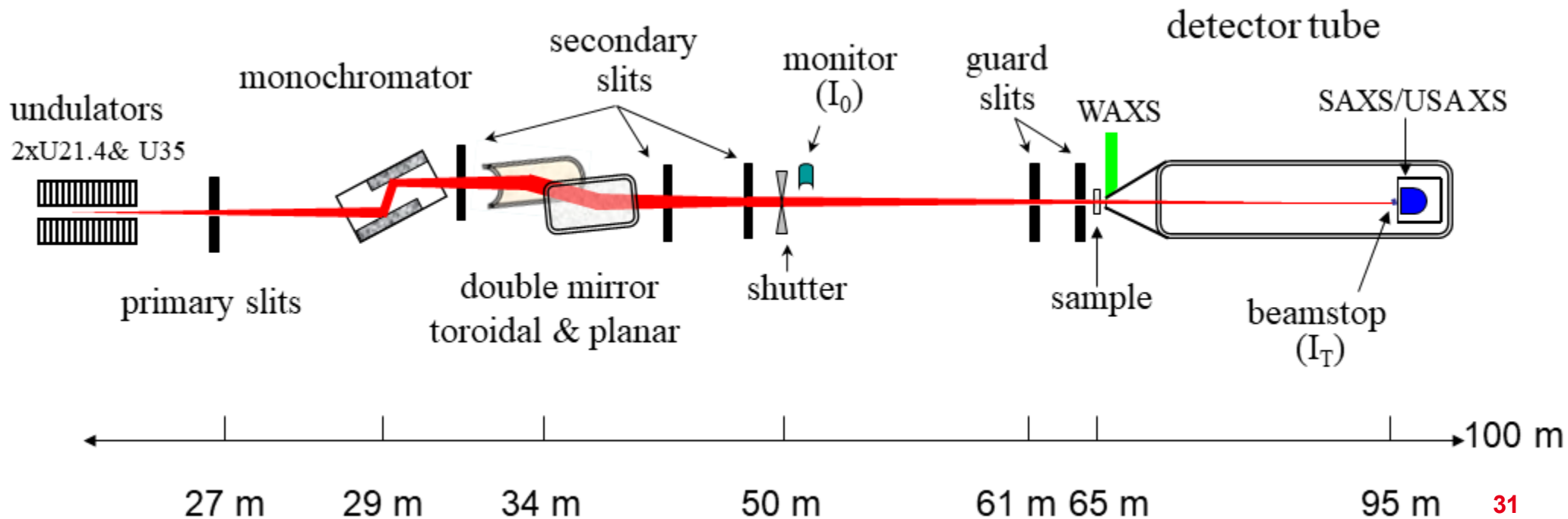


Lab-SAXS

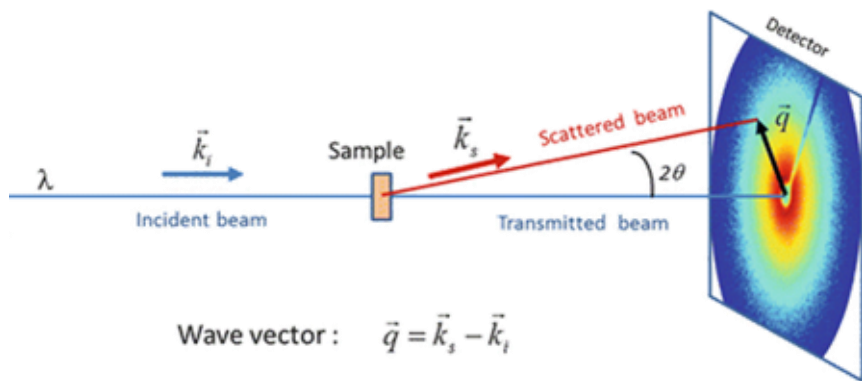


Synchrotron SAXS

ID02



Practicalities

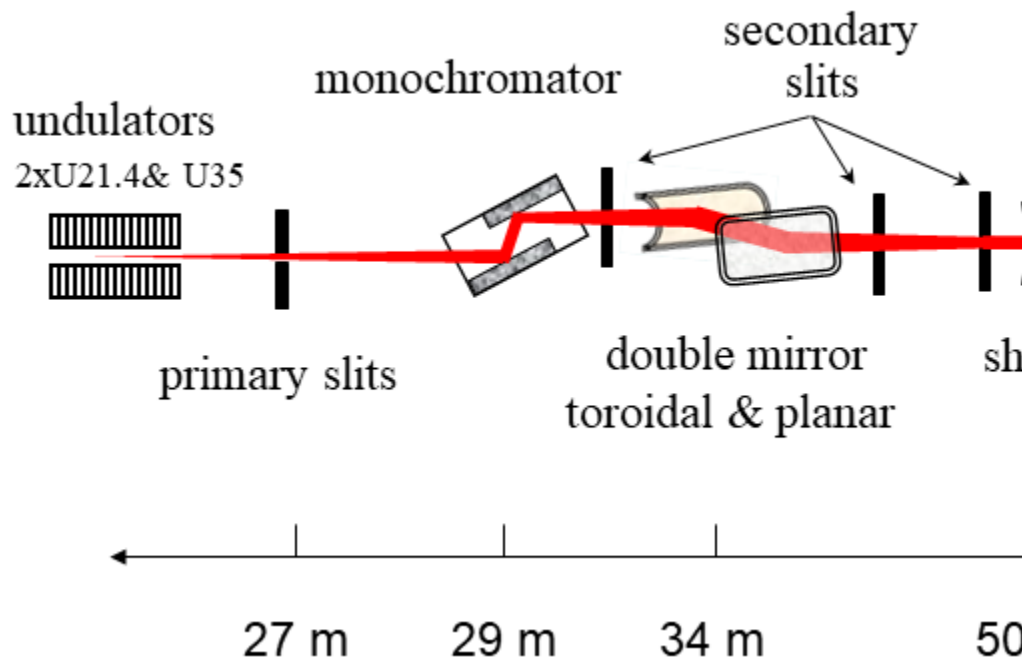


Lab-SAXS

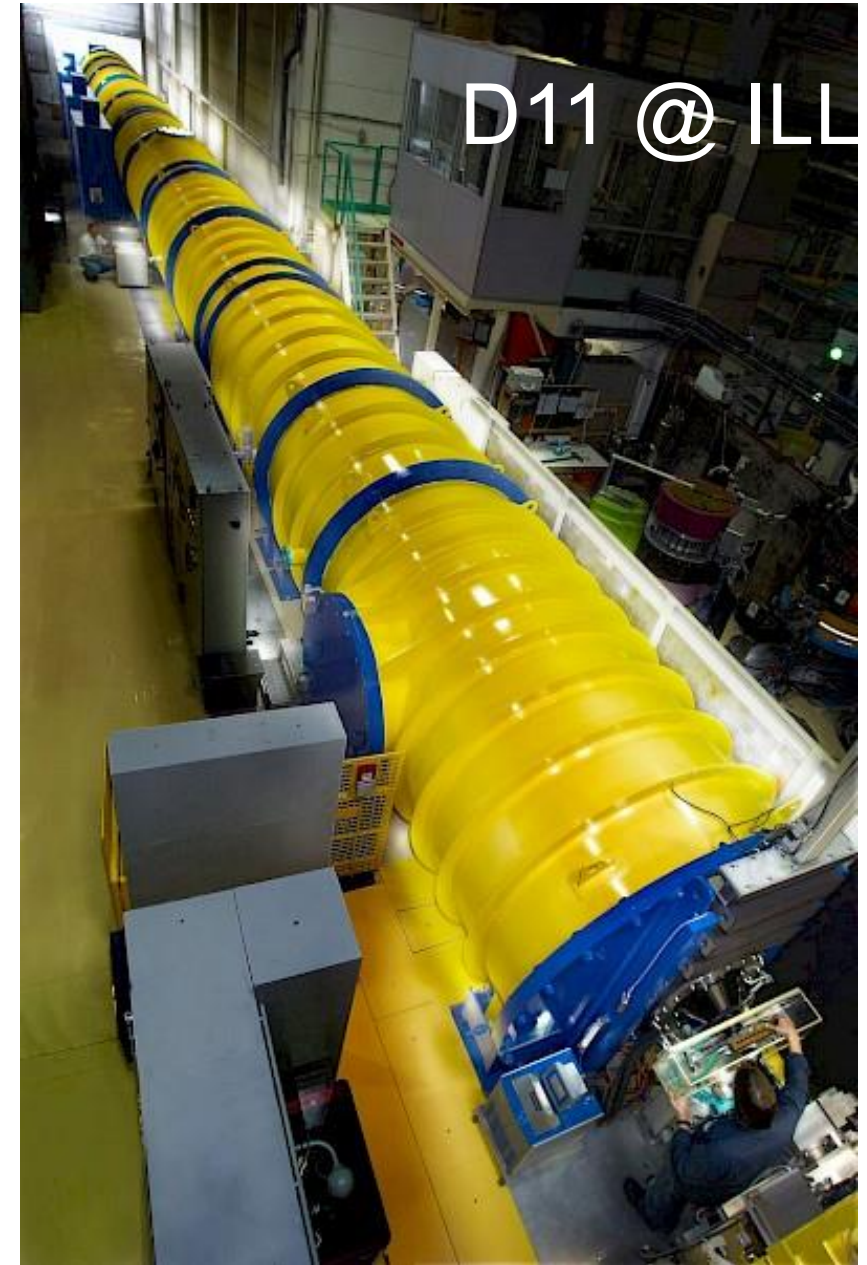


Synchrotron SAXS

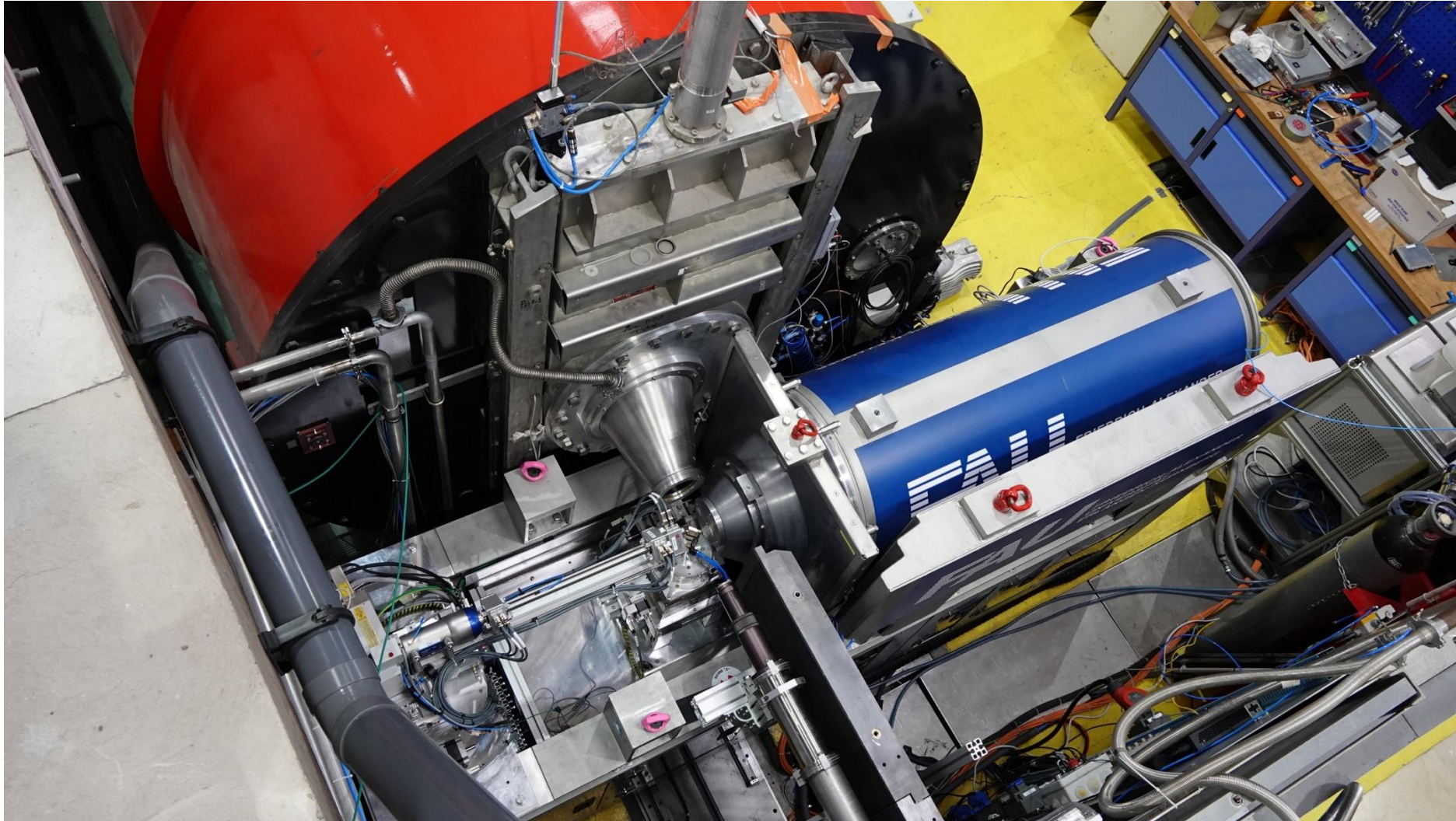
ID02



Practicalities



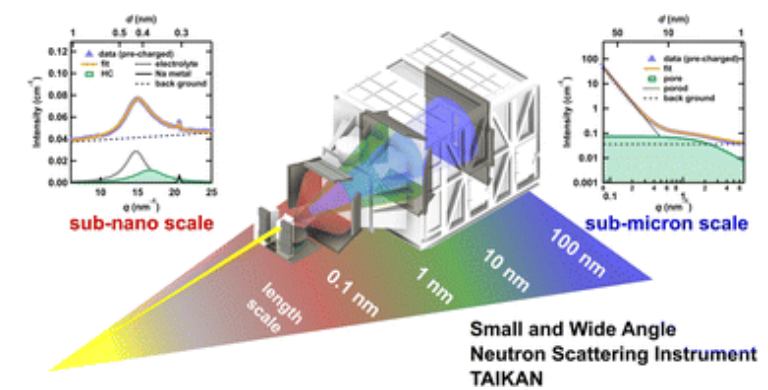
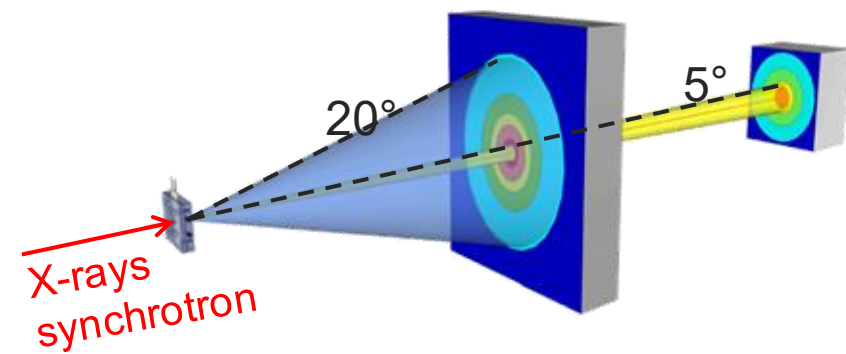
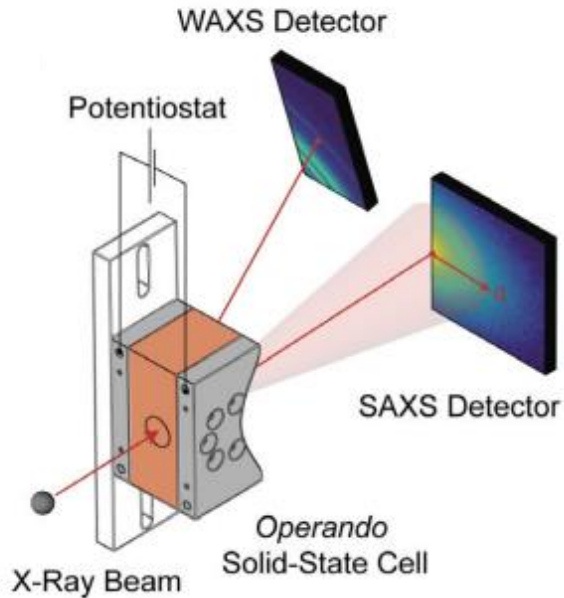
Coupled SAXS/SANS



D22 spectrometer



Coupled SAXS/WAXS and SANS/WANS



J-M. Von Mentlen et al, Batteries and supercaps, 2025, e202500428

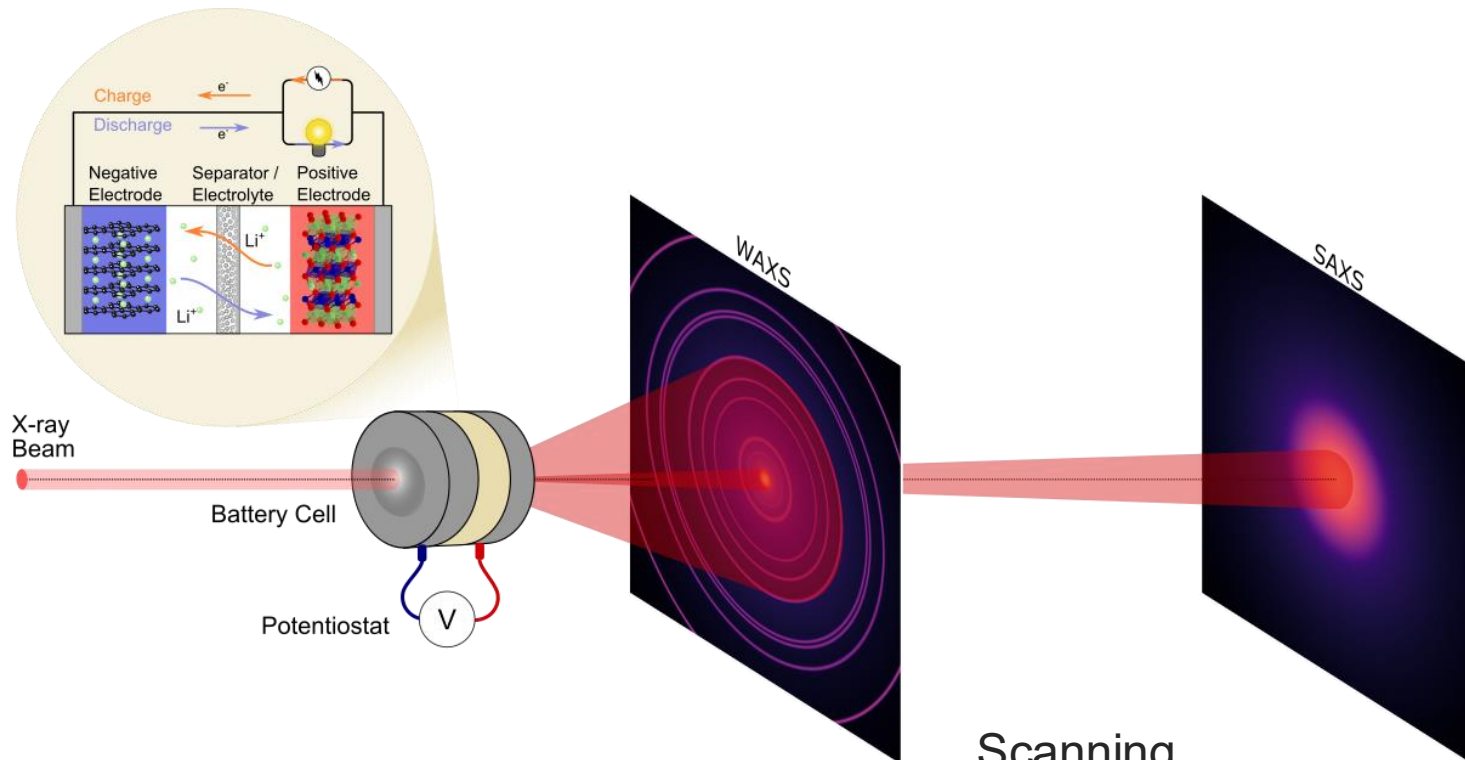
C. Berhaut *et al*, ACS Nano (2019), 13, 10, 11538-11551

Y. Umemoto *et al*, Chem. Sci., 2026, Advance Article, 10.1039/d5sc09600f

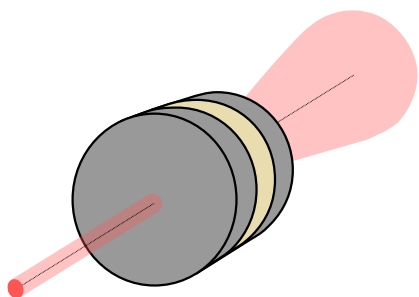
Measurement Modalities



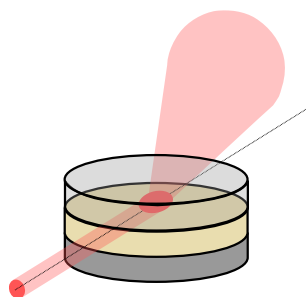
a



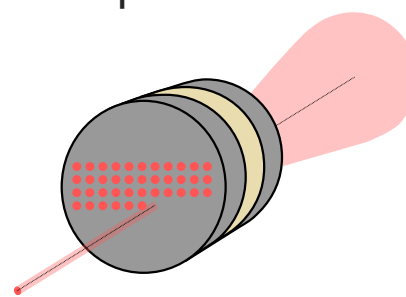
SWAXS



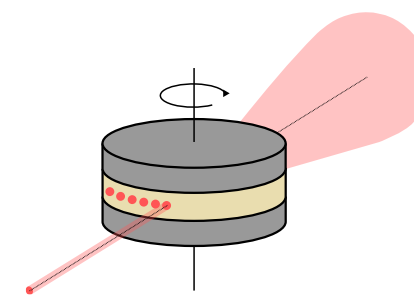
GI-SWAXS



Scanning μ SWAXS

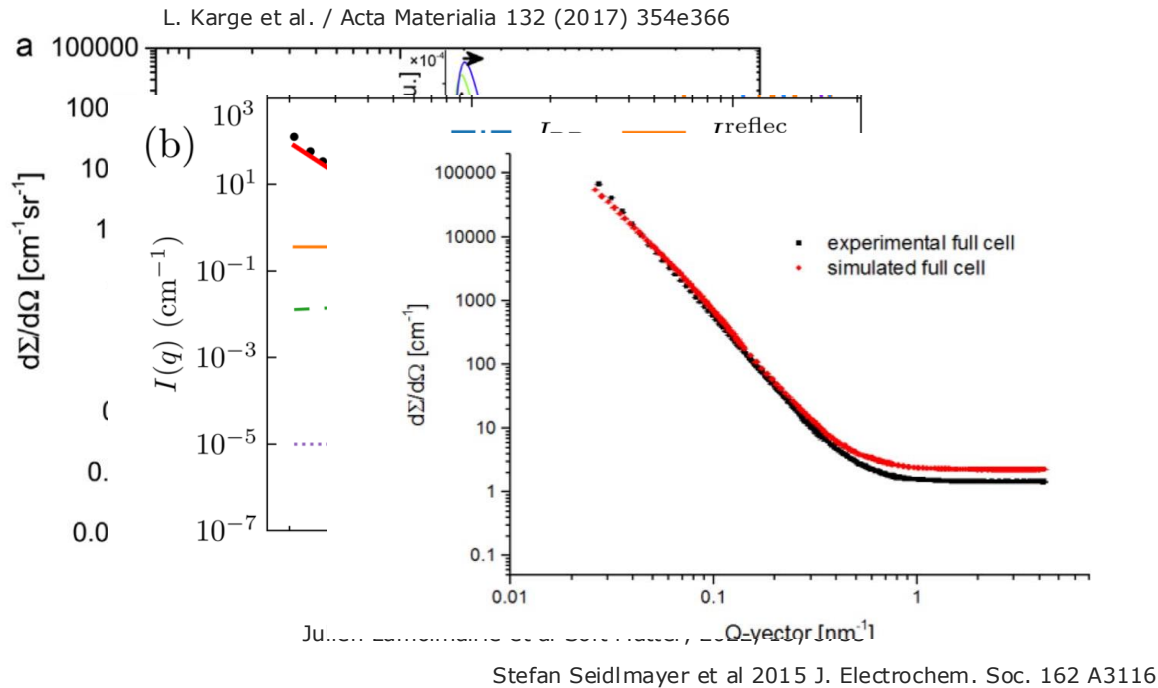


Tomography SWAXS-CT



Need small beams, so mostly limited to X-Rays

SAXS/SANS in batteries...not always easy !



preposition LITERARY • HUMOROUS

without.

SANS and SAXS features are subtle
Need complementary / background info
Analysis is not always straightforward

BUT SANS/SAXS can give access to:

- Nanostructural changes
- Particle morphology (size, shape, layers, etc)
- Chemical information (SLD variation)
- Sample average: statistically relevant

Few reviews

A. Black et al, Synchrotron radiation based operando characterization of battery materials, Chem. Sci., 2023, 14, 1641

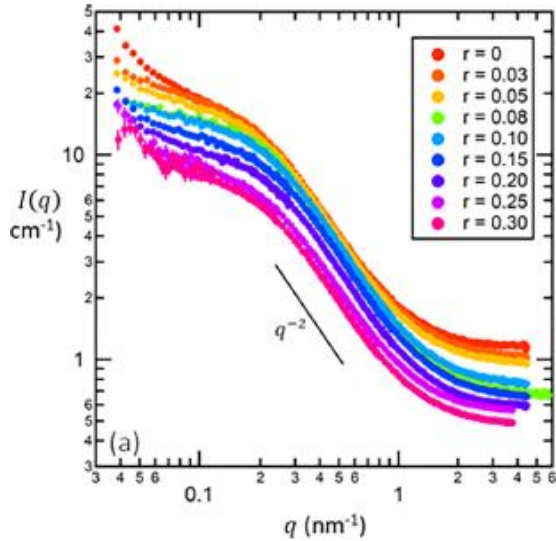
L. Fang et al, Synchrotron small-angle X-ray scattering technique for battery electrode study, Nano Energy, 2024, 121, 109255

W. Cheng et al, Recent advances in battery characterization using in situ XAFS, SAXS, XRD, and their combining techniques: From single scale to multiscale structure detection, Exploration, 2023, DOI: 10.1002/EXP.20230056

Main applications of SAXS/SANS in batteries

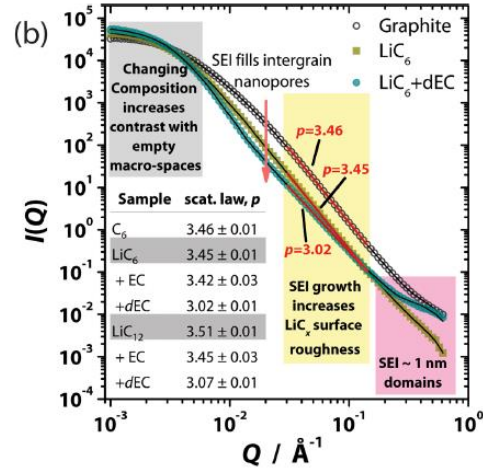


Polymer (blends, composites)



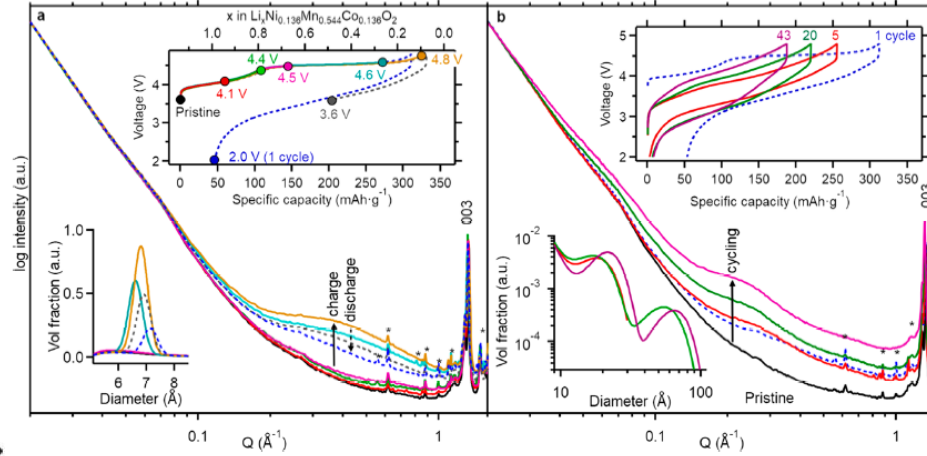
W.S. Loo et al, *Macromolecules* 2019, 52, 8724–8732

SEI on graphite



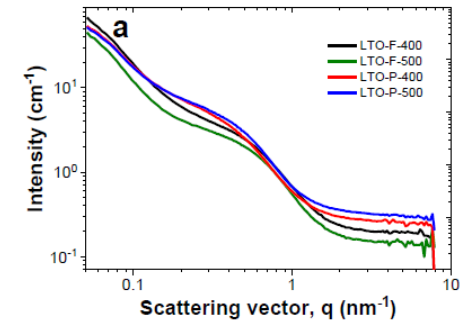
R.L. Sacci et al, *J. Phys. Chem. C* 2015, 119, 9816–9823

LNMO

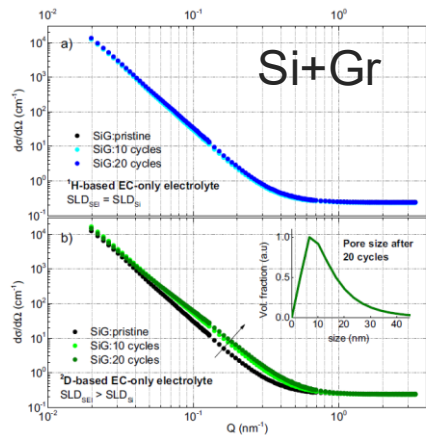


Grenier et al, <https://doi.org/10.1021/jacs.1c00497>

LTO interfaces

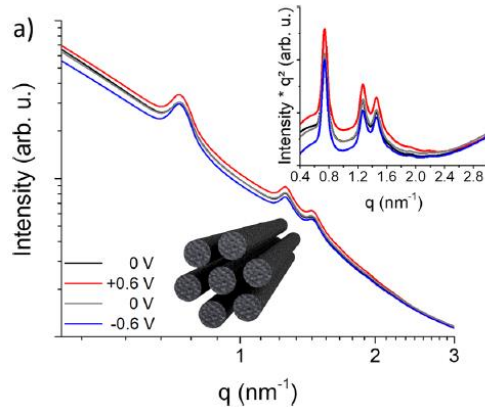


C.J. Jafta et al, 10.1002/cssc.202000802



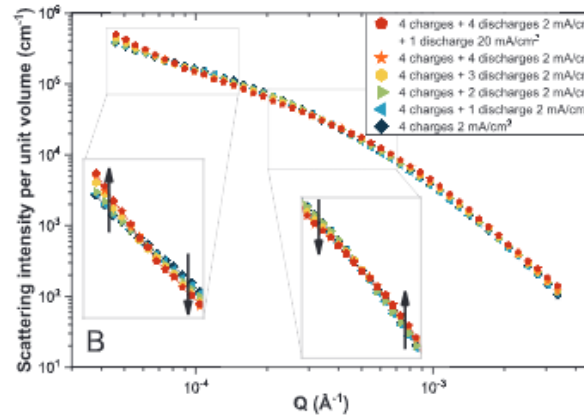
N. Paul et al, *Journal of The Electrochemical Society*, 166 (6) A1051-A1054 (2019)

Porous Carbon



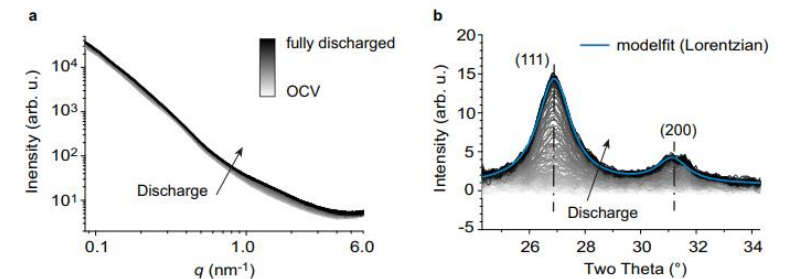
C. Kocswara et al, *ACS Appl. Mater. Interfaces* 2019, 11, 42214–42220

Lithium metal batteries



C. Didier et al, *Adv. Energy Mater.* 2023, 13, 2301266

Lithium sulphur batteries



C. Prehal et al, *Nature Communications* | (2022) 13:6326

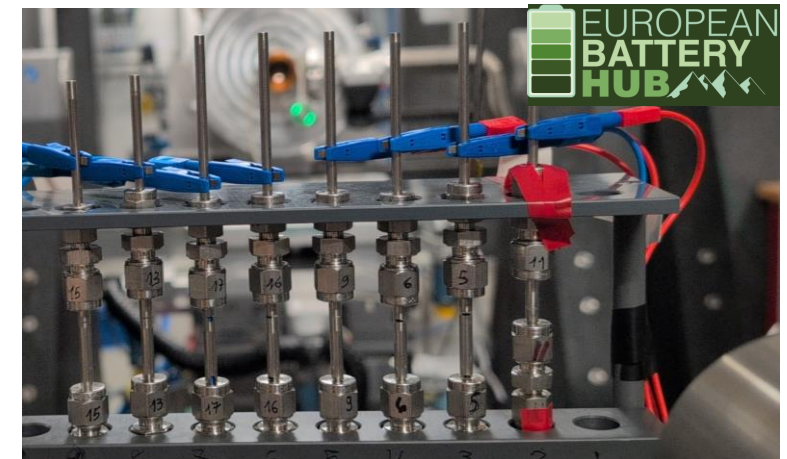
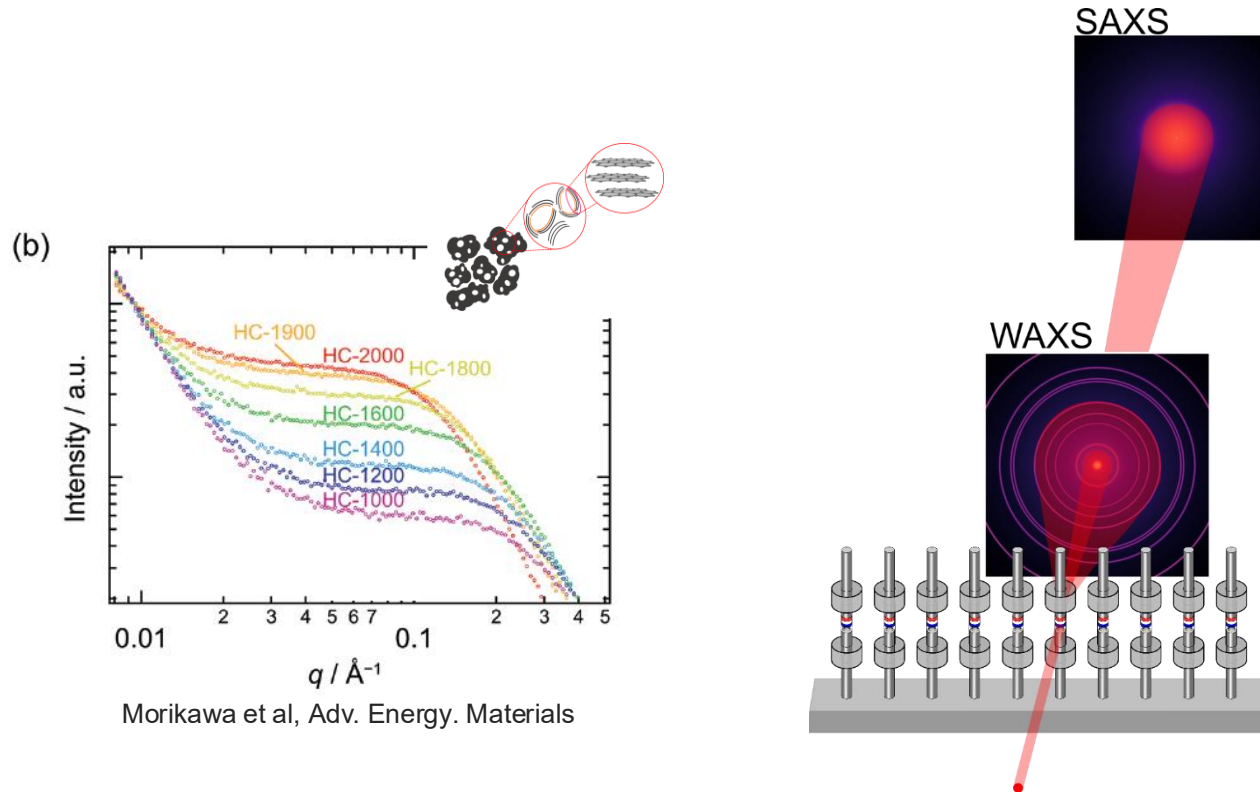
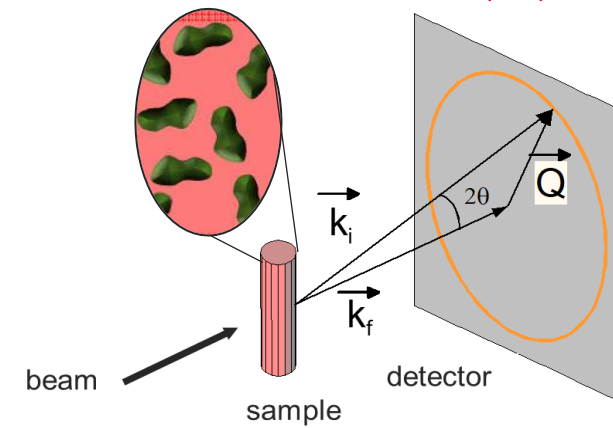
Outline

❖ I – SAS TUTORIAL

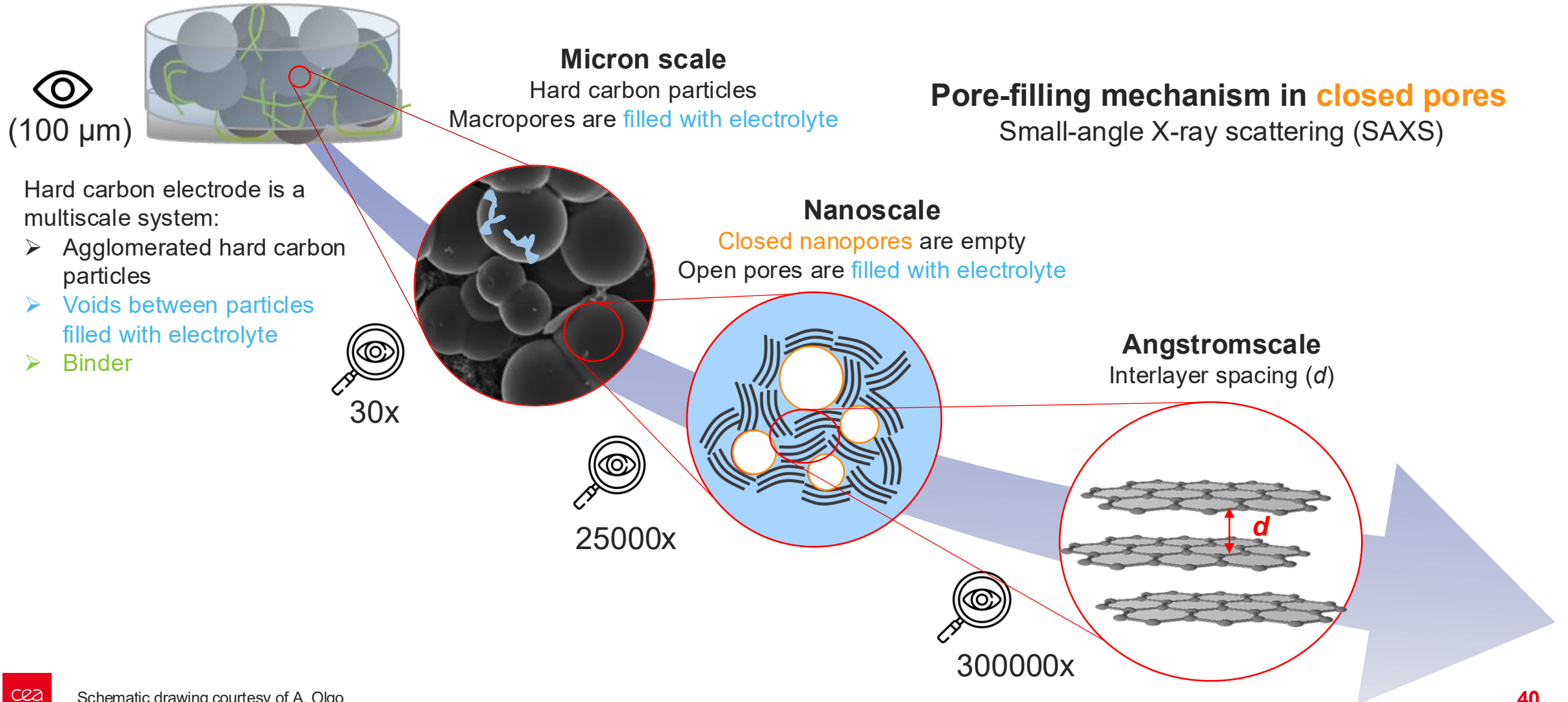
Basic principle of Small Angle Scattering techniques (SAXS, SANS)

❖ II – APPLICATION to battery research

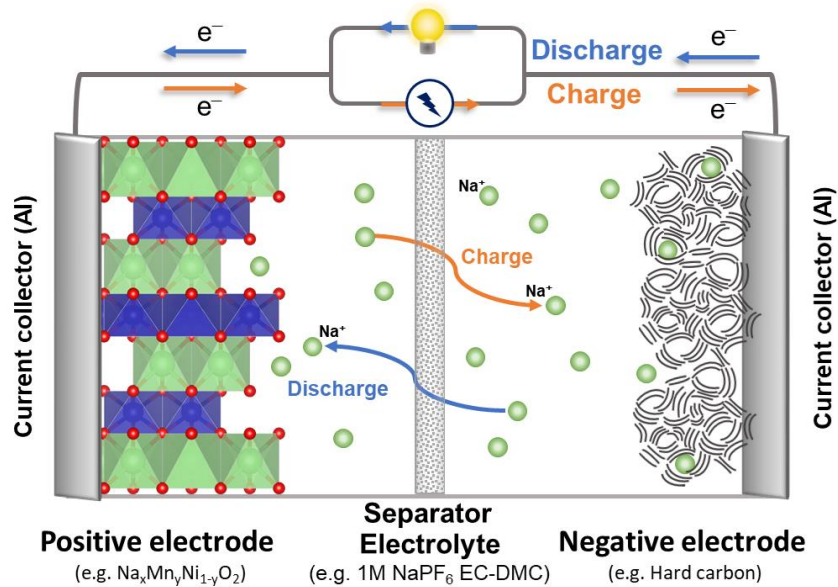
Operando SAXS study of HC electrodes for Na-ion batteries



Hard Carbon as negative electrodes for Na-ion Batteries

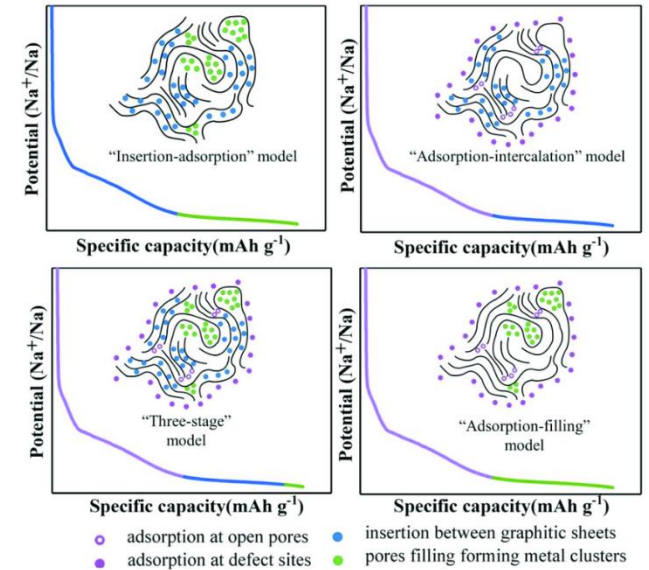


Hard Carbon as negative electrodes for Na-ion Batteries



Hard Carbons:

- Hierarchical structure
- Several Na mechanisms proposed
- Depend on the HC type
- No true consensus in literature



Key Topics / interests:

- HC structure-performance relationship
- Sodiation mechanisms of different HC / HC composites
- Influence of different electrolyte formulations
- Influence of different C-rates



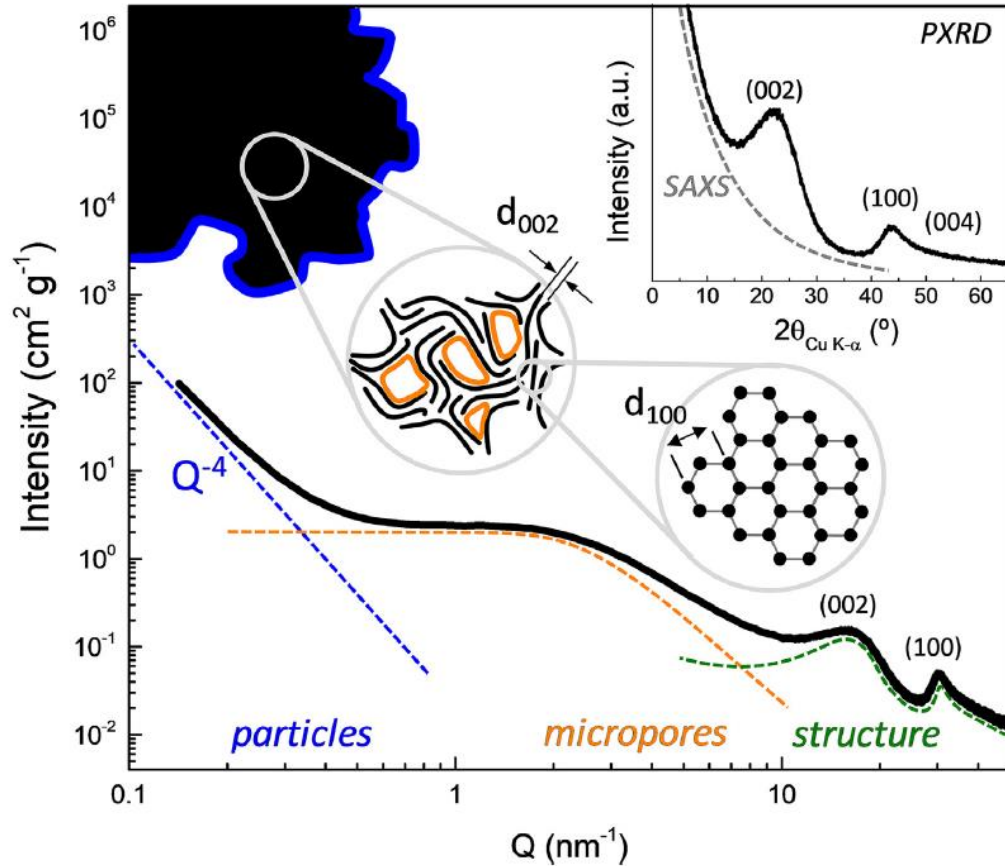
Common Hub Workflow to tackle these questions



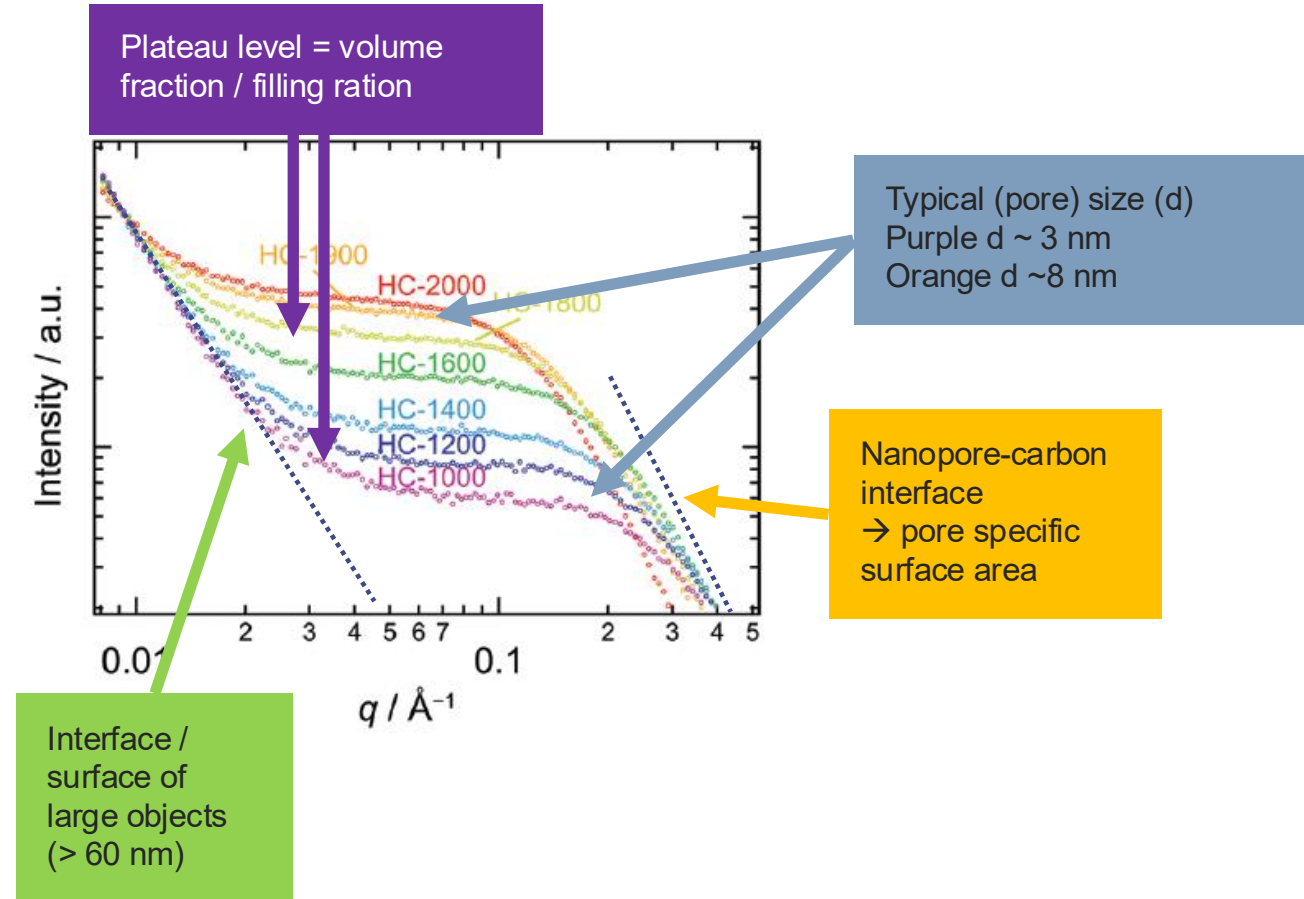
Tool:

SAXS → HC pores (nm)
WAXS → crystalline changes

Hard Carbon SAXS Pattern

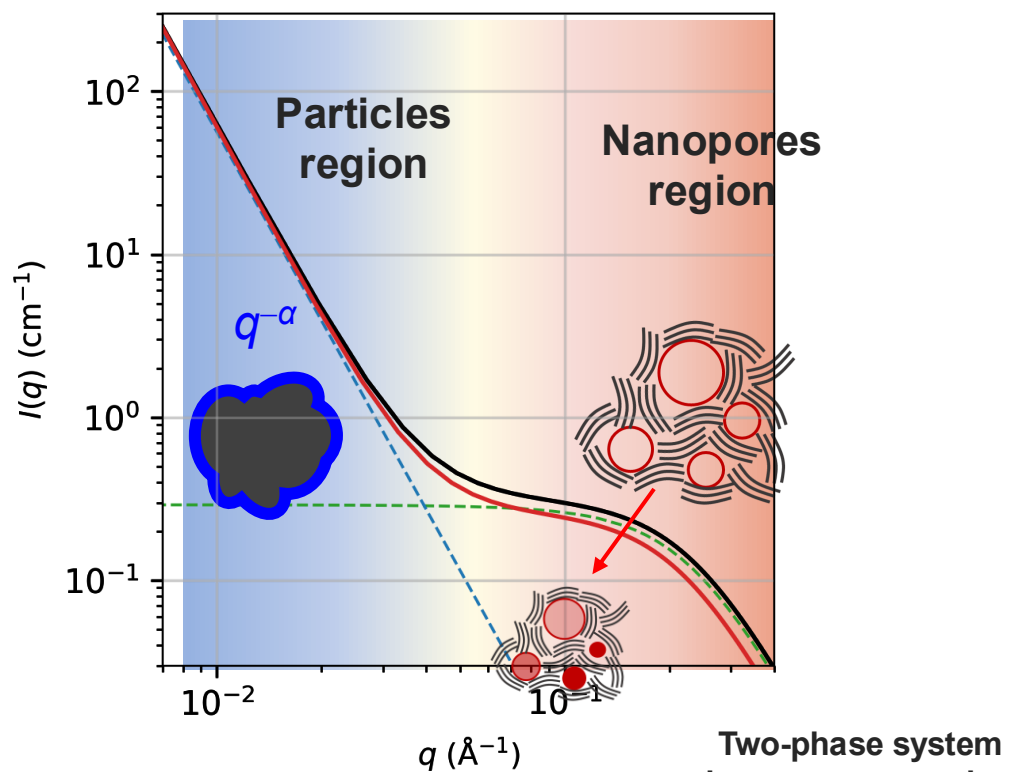


D. Saurel et al. Energy Storage Materials 21 (2019) 162–173



Morikawa et al., Adv. Energy Mater. 2020, 10, 1903176

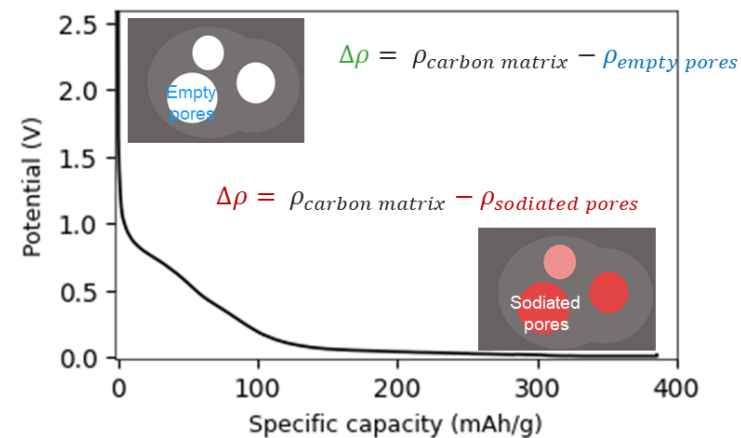
What happens during the charge/discharge?



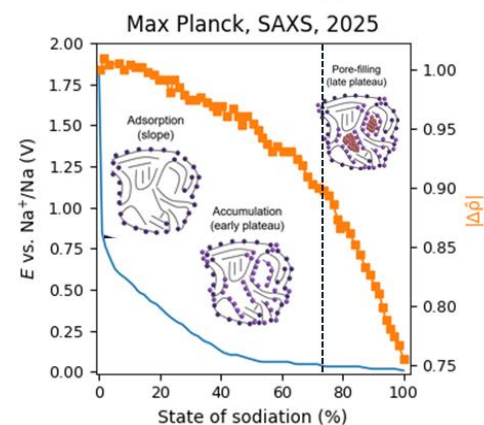
Two-phase system in nanopores region

$$\Delta\rho_{\text{pristine HC}} = \rho_{\text{carbon matrix}} - \rho_{\text{empty pores}}$$

$$I(q) \propto \Delta\bar{\rho}^2$$

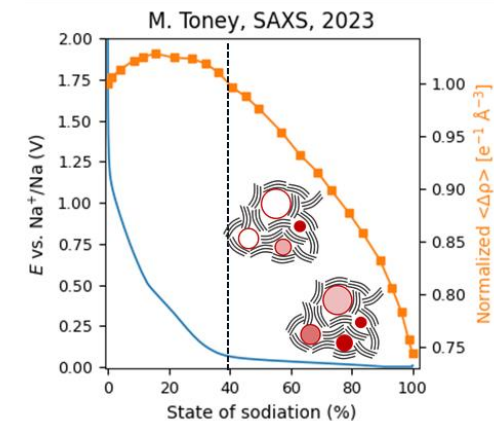


Track pore filling via SLD contrast $\Delta\rho$



Spatial in plane heterogeneities

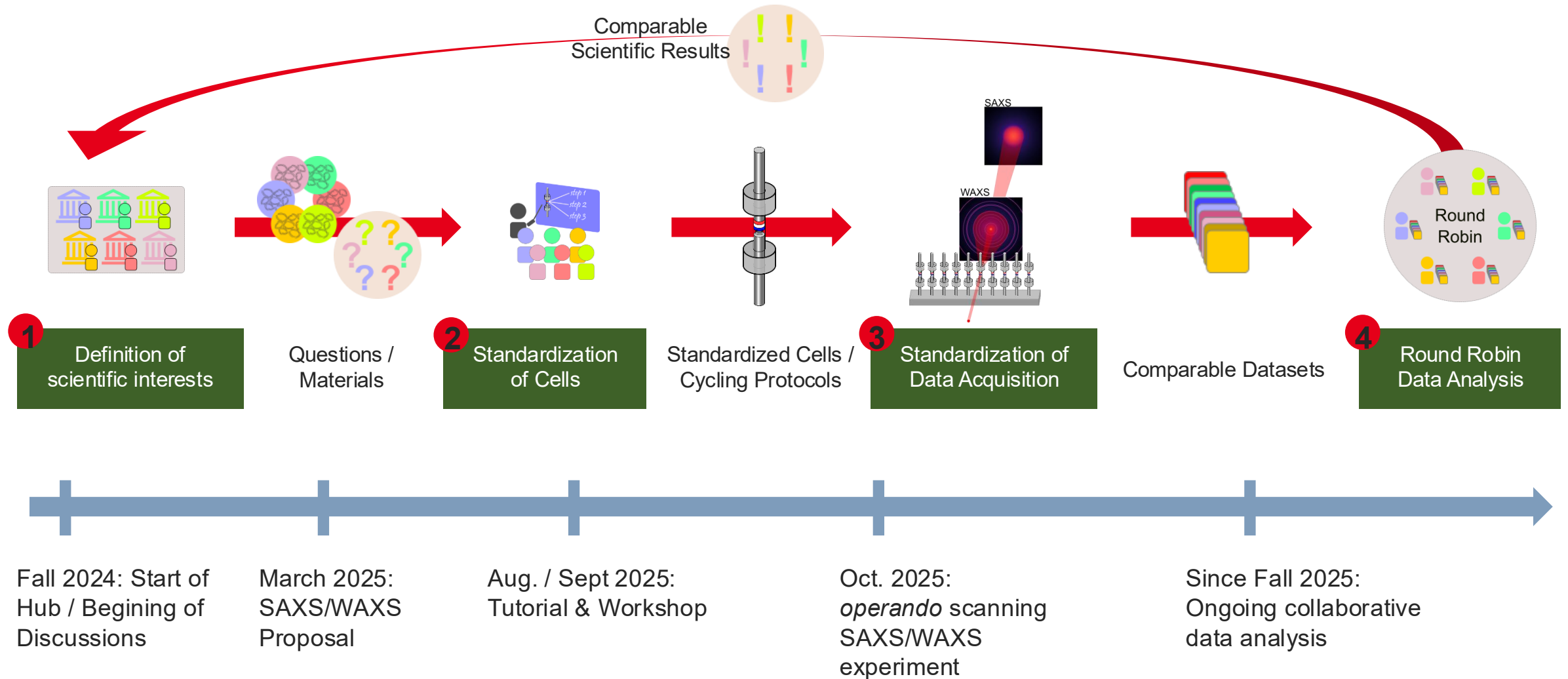
Eren et al., Energy Environ. Sci., 2025, 18, 7859–7868



Pore-size dependent pore filling

Iglesias et al., Adv. Energy Mater. 2023, 13, 2302171

operando Scanning SAXS/WAXS Workflow of Hard Carbons and Hard Carbon Composites

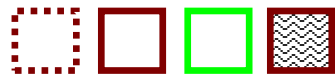
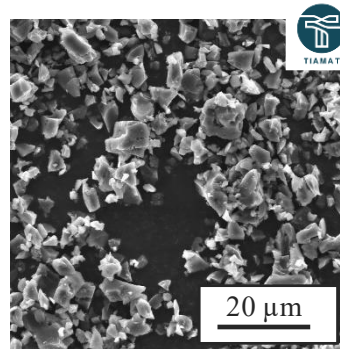


operando Scanning SAXS/WAXS Workflow of Hard Carbons and Hard Carbon Composites

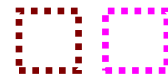
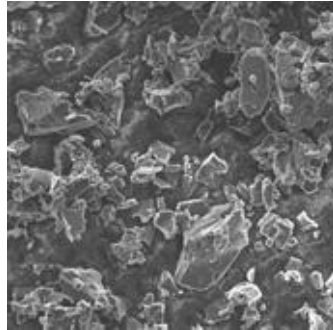


Hard Carbons

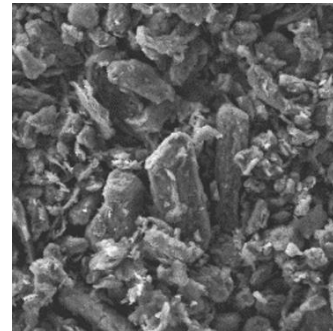
IRIG (Tiamat)



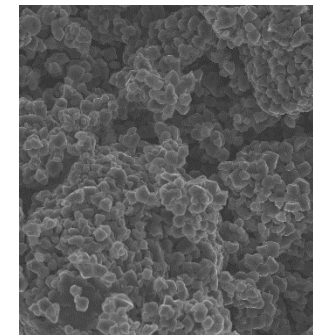
CTH



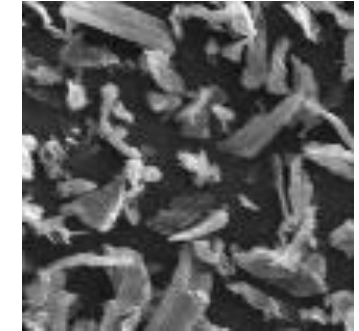
HIU



LITEN

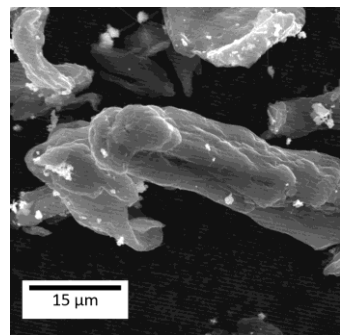


UBT

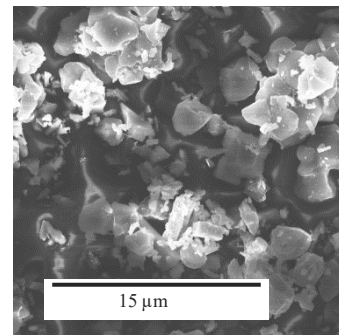


Composites

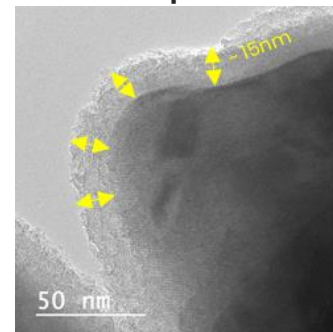
UBT HC+Sn



LITEN HC+NTO



LITEN pure NTO



Electrolytes

Carbonate

Diglyme

Monoglyme

Carb. + additives

Counter electrode

Na metal

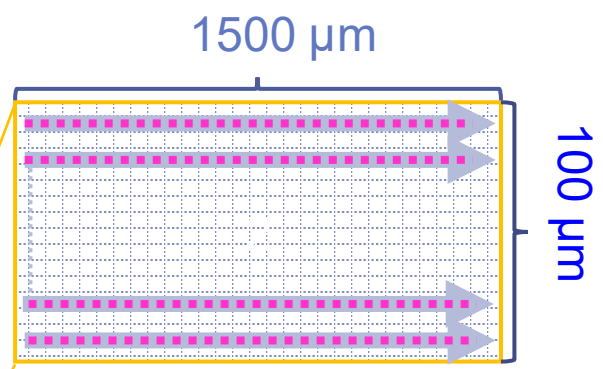
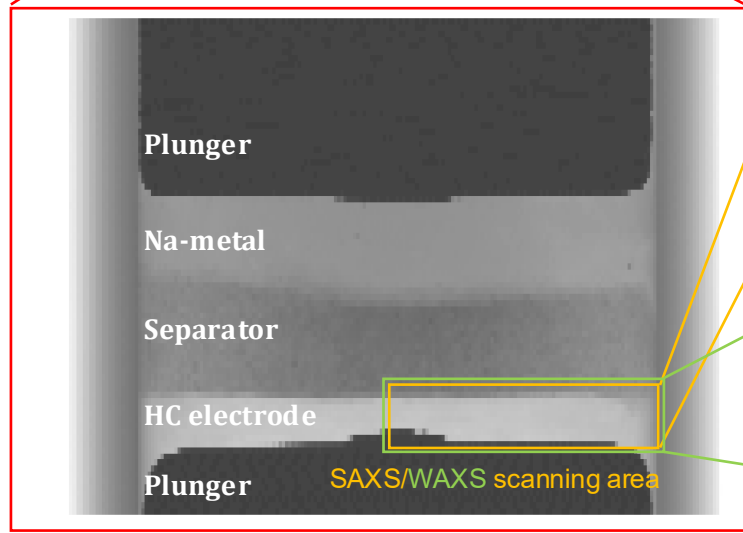
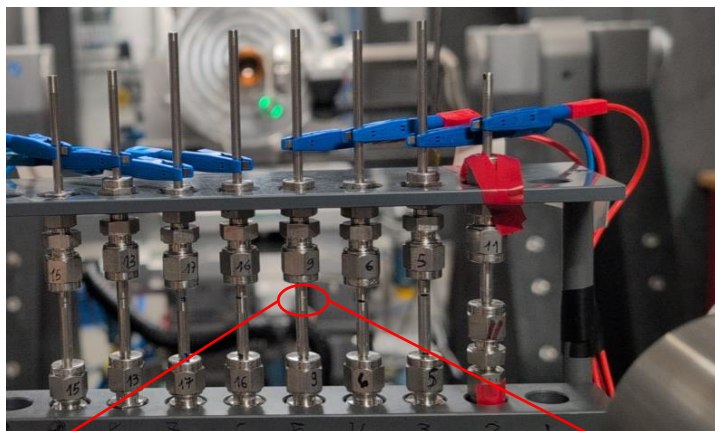


Other

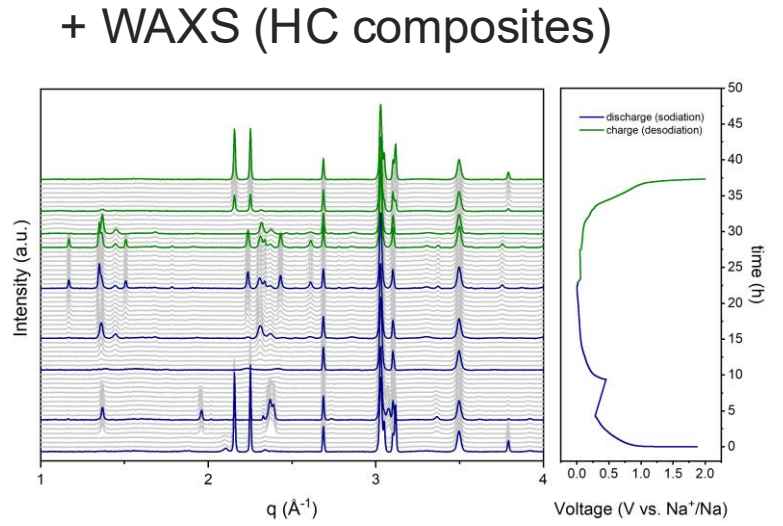
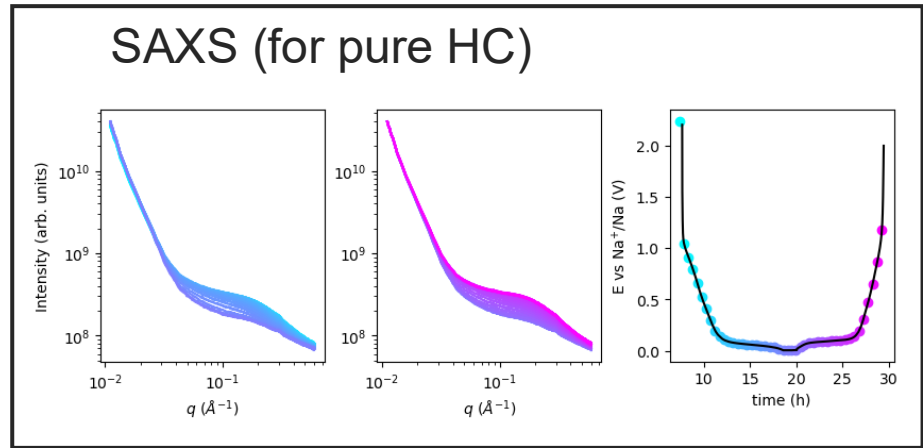
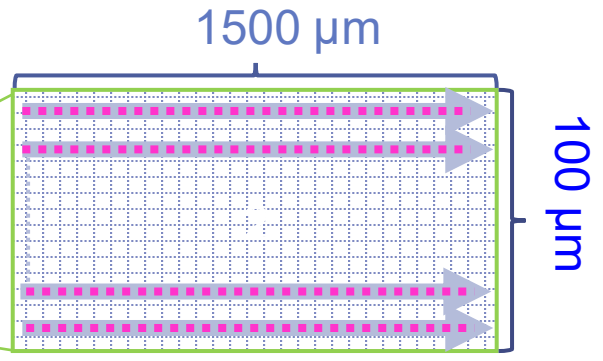
Beam damage

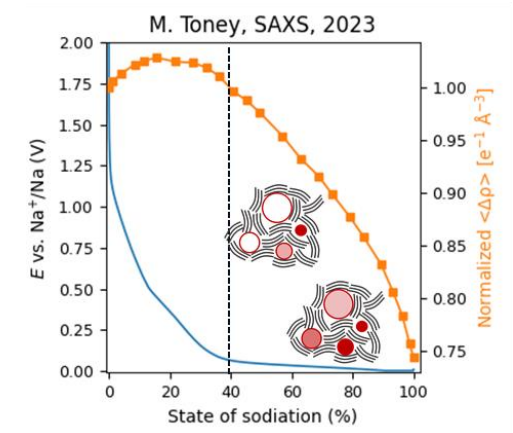
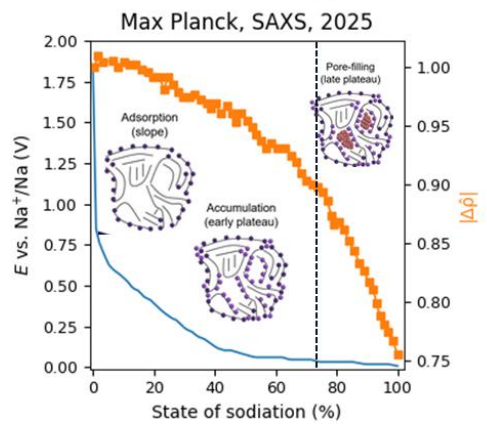
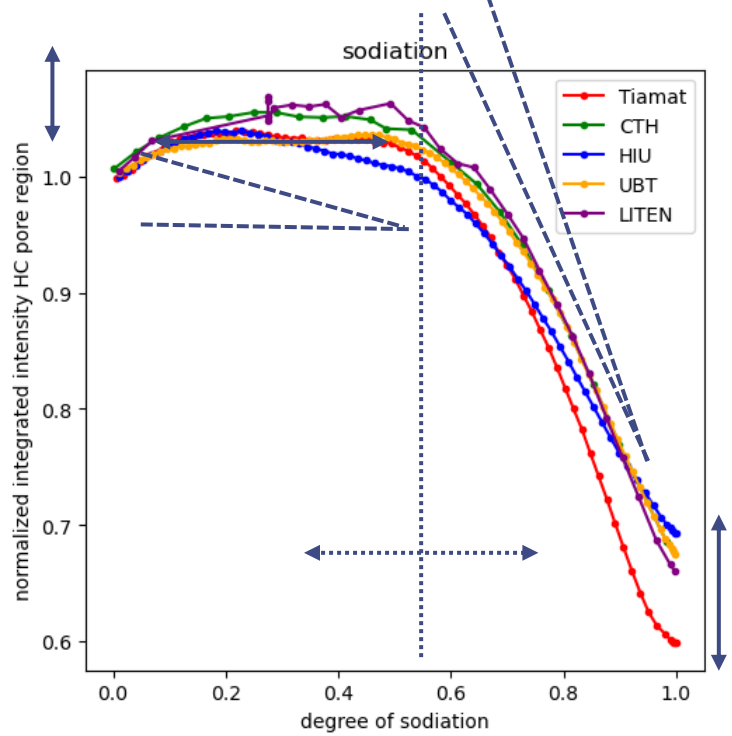
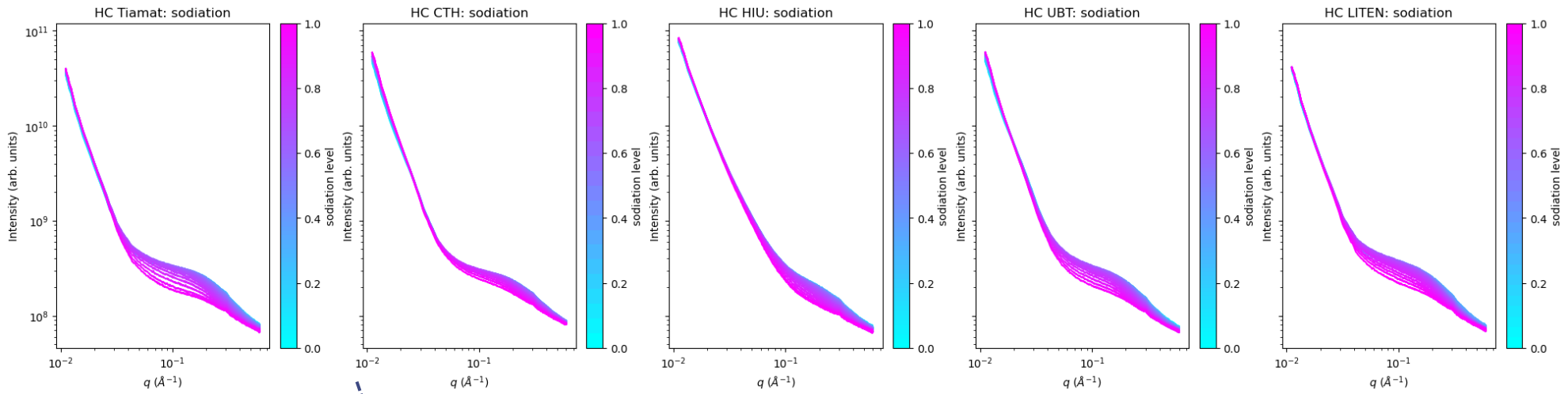


operando Scanning SAXS/WAXS Workflow of Hard Carbons and Hard Carbon Composites

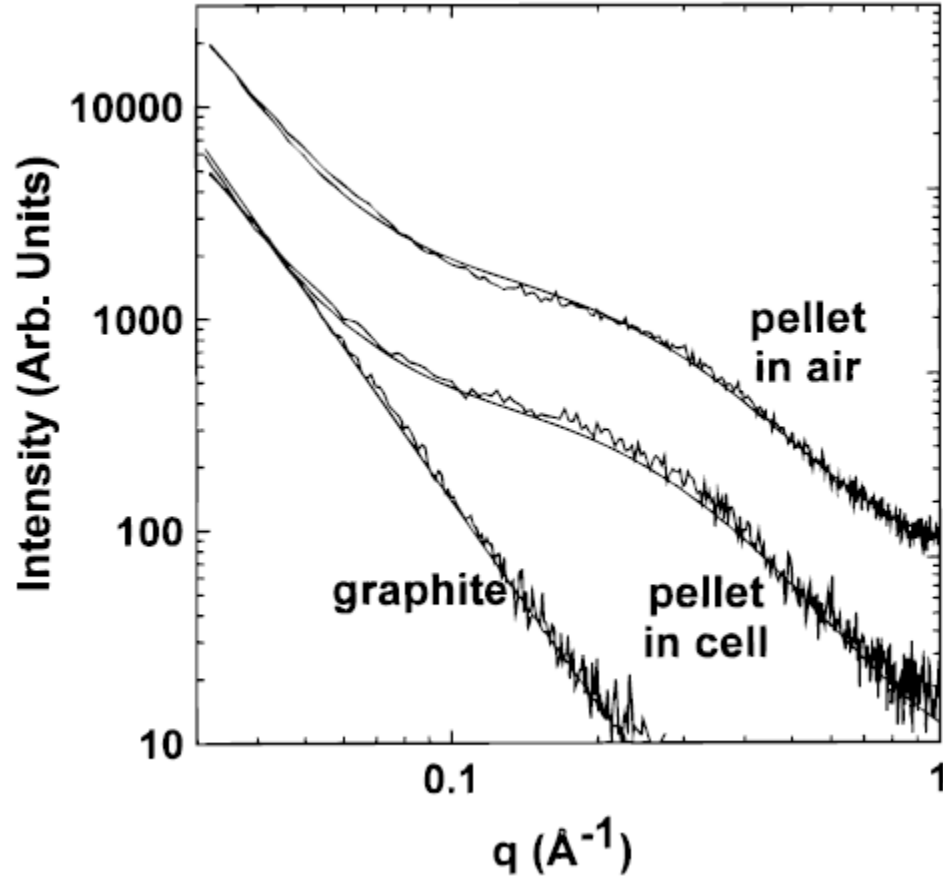


E = 75 keV
 Beam size: 20 x 8 μm²
 (y,z) map of 225 pixels
 Pixel size : 60 x 8 μm²
 1 map in 1 minute





How do you fit/model this?

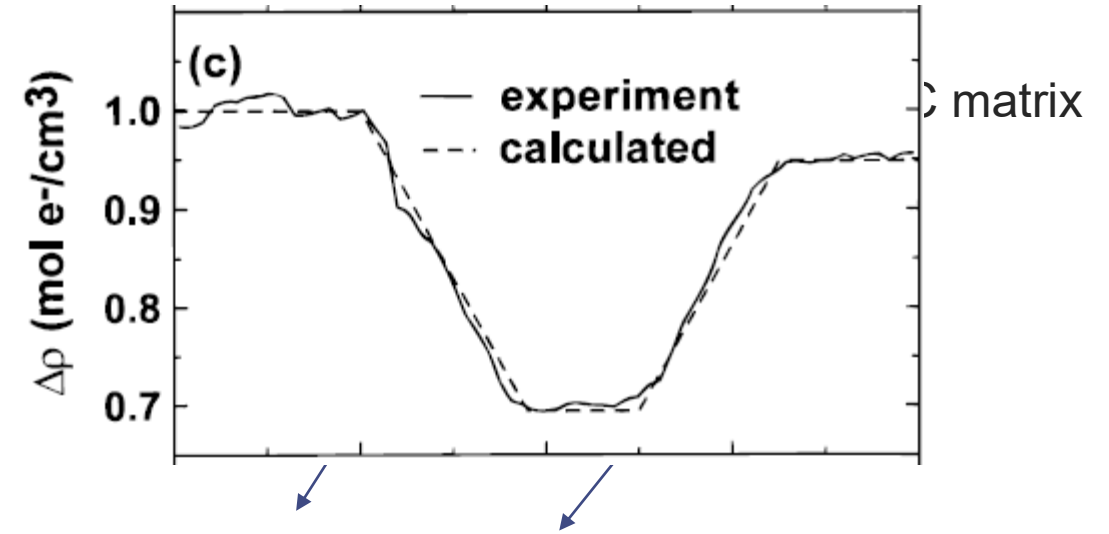


D. A. Stevens and J. Dahn 147 (12) 4428-4431 (2000)

General Form:

$$I(q) = \frac{A}{q^n} + I_{por_{es}}(q) + const$$

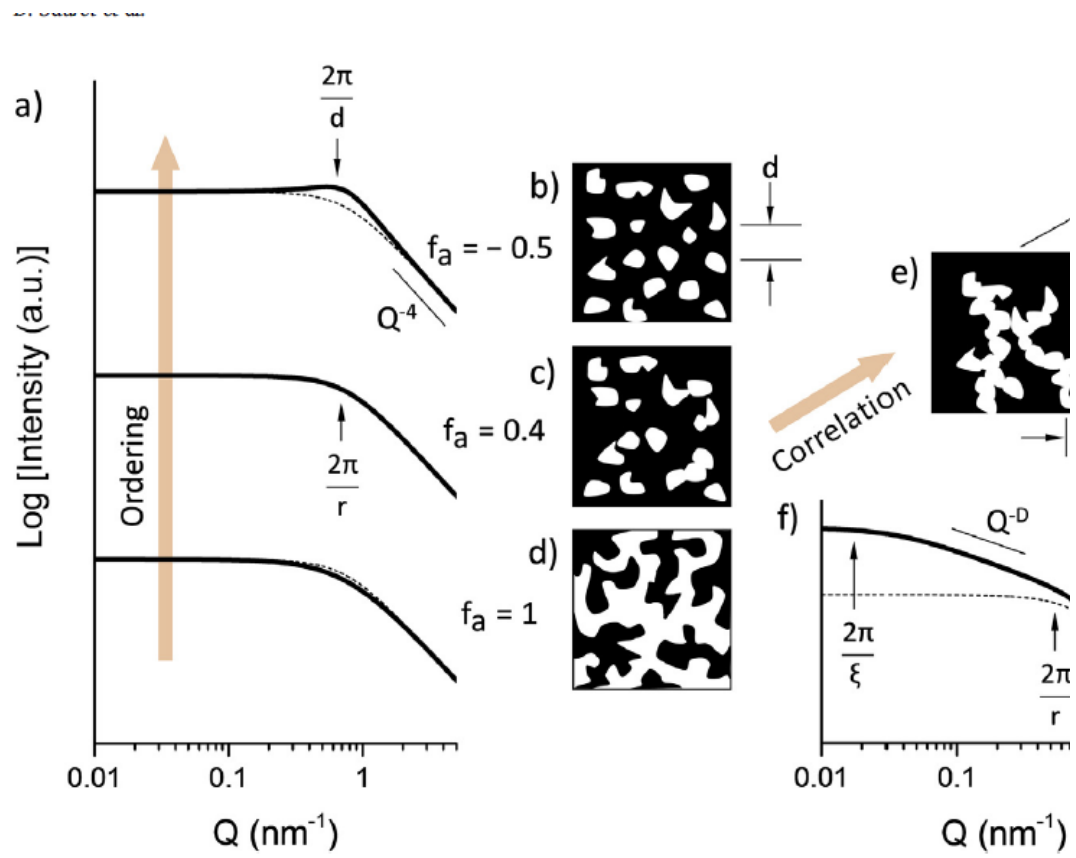
Debye Büche Model



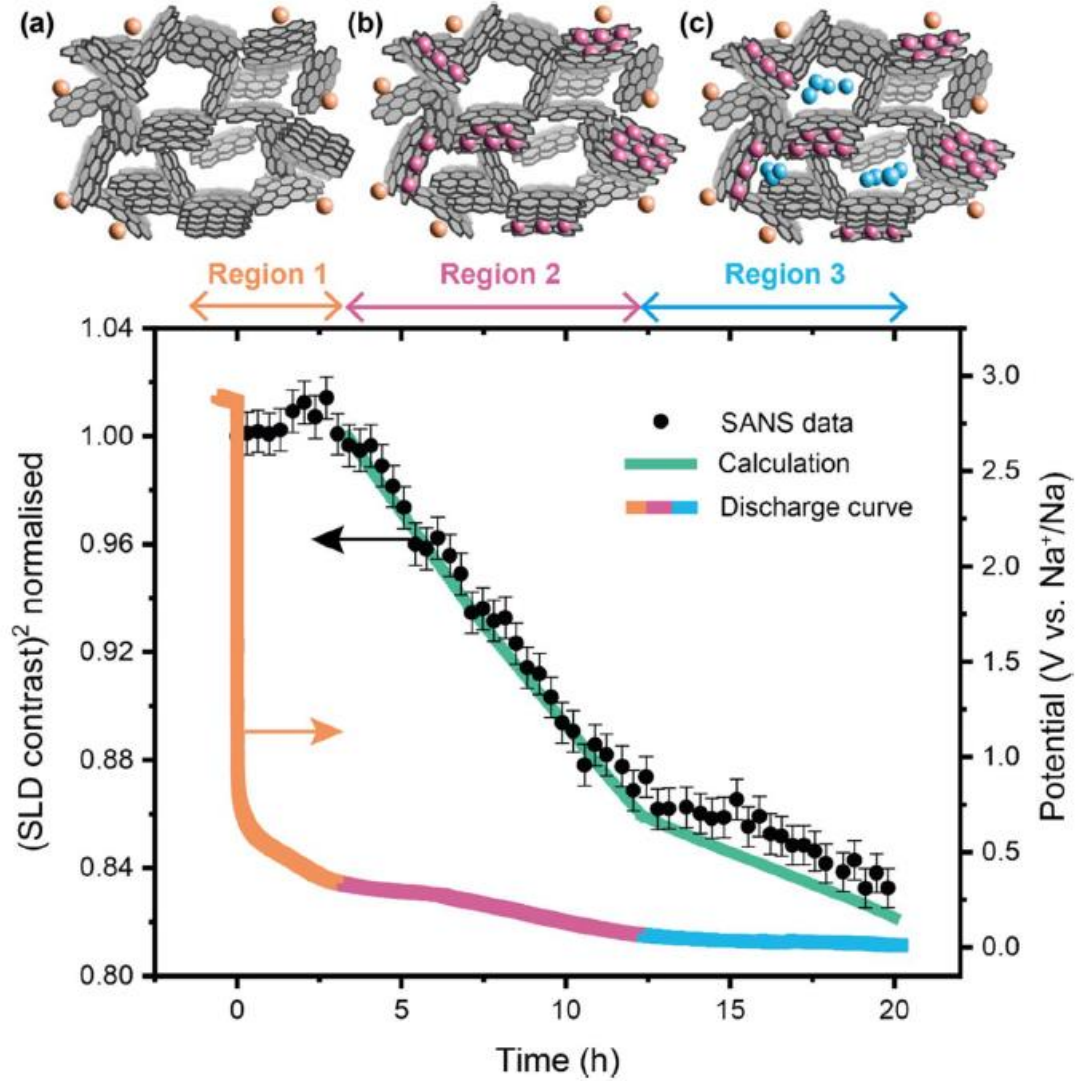
$a_1/2$: characteristic size of pores

For spherical pores: $R = a_1 \sqrt{10}$

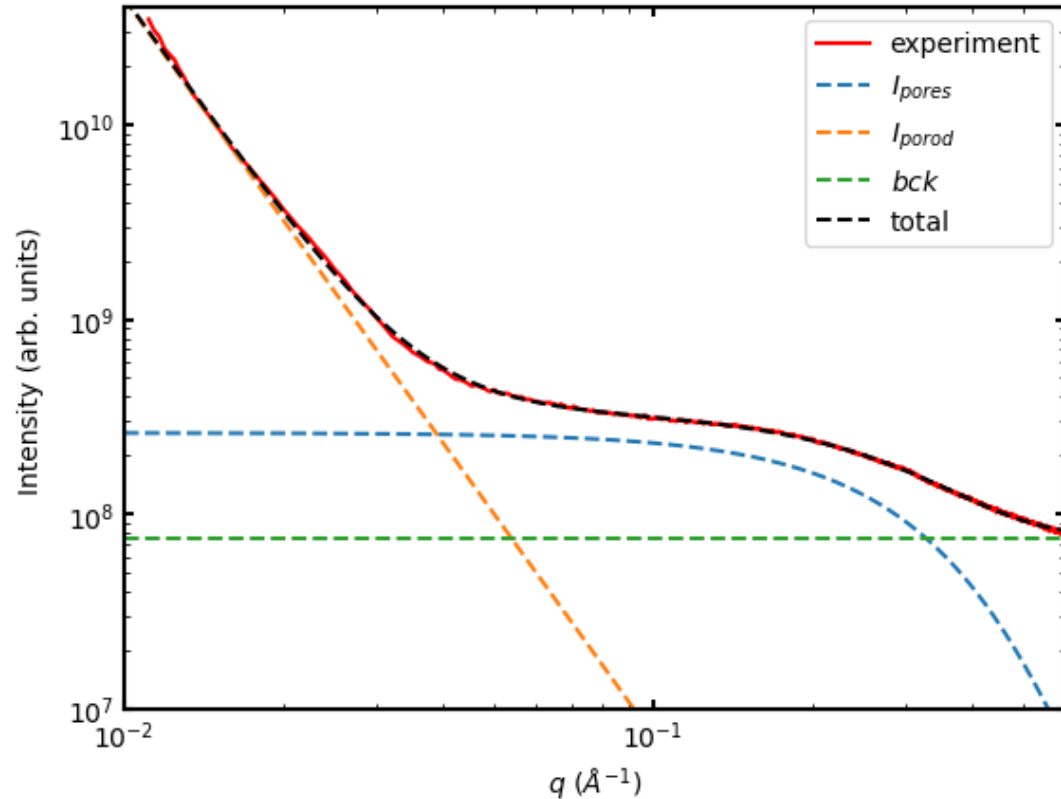
How do you fit/model this?



General Form:



How do you fit/model this?



General Form:

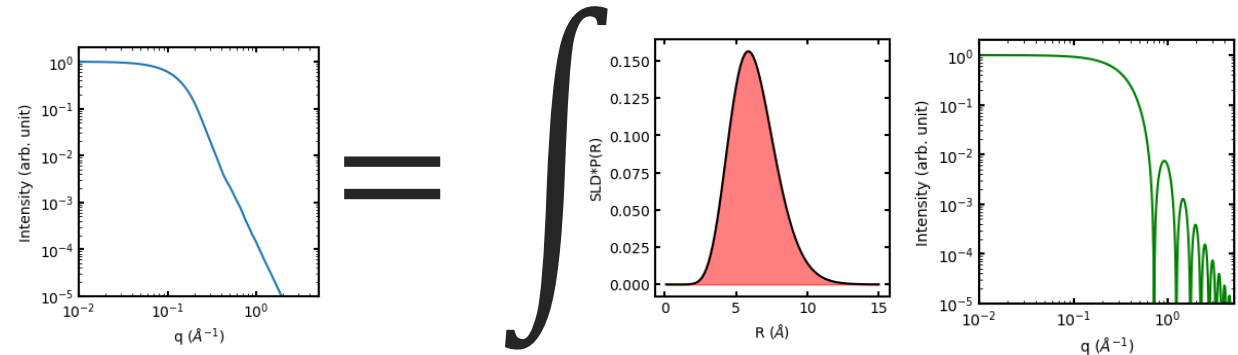
$$I(q) = \frac{A}{q^n} + I_{pores}(q) + const$$

Spheres + Size Distribution
+ Size dependent Pore filling. More realistic?

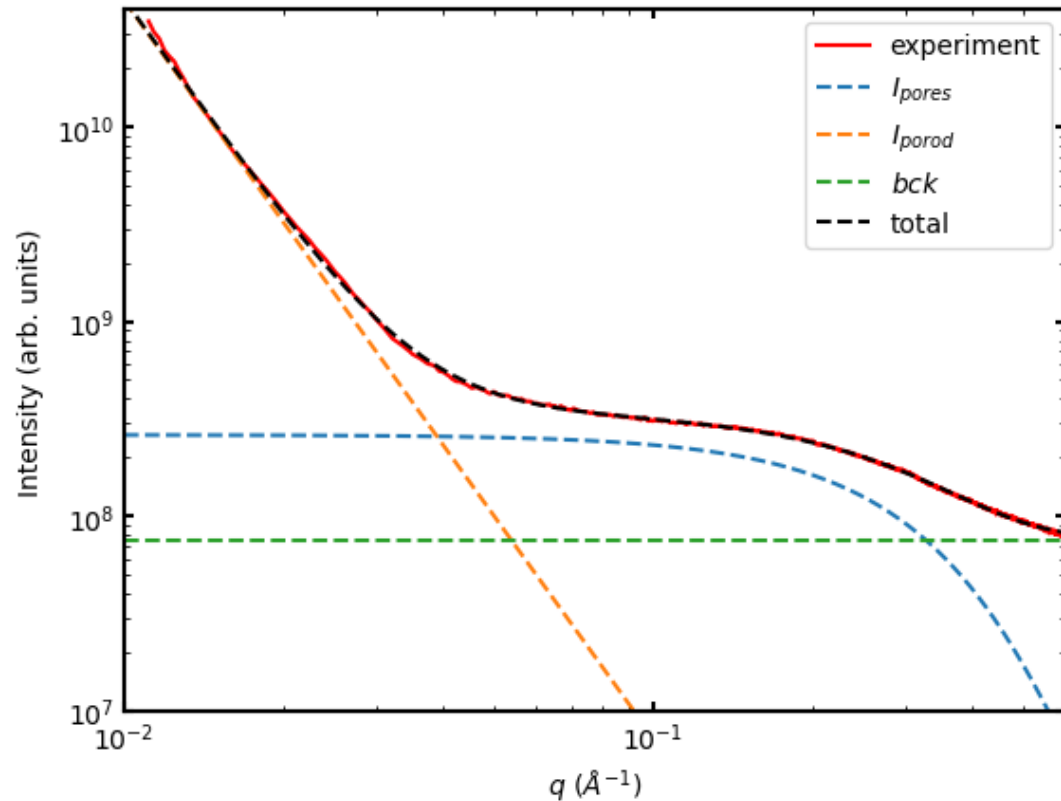
$$I_{pores}(Q) = scale(\rho_{pore} - \rho_{HC})^2 b^2 \int_0^\infty \frac{P(R)}{\frac{4}{3}\pi R^3} (F(Q, R))^2 dR \quad (4)$$

$$F(Q, R) = \int_0^R e^{iQr} d^3r = \frac{4}{3}\pi R^3 \left(3 \frac{\sin(QR) - QR \cos(QR)}{(QR)^3} \right) \quad (5)$$

$$P(R) = \left(\frac{z+1}{R_{avg}} \right)^{z+1} \frac{R^z}{\Gamma(z+1)} e^{-(z+1)\frac{R}{R_{avg}}} \quad (6)$$



How do you fit/model this?



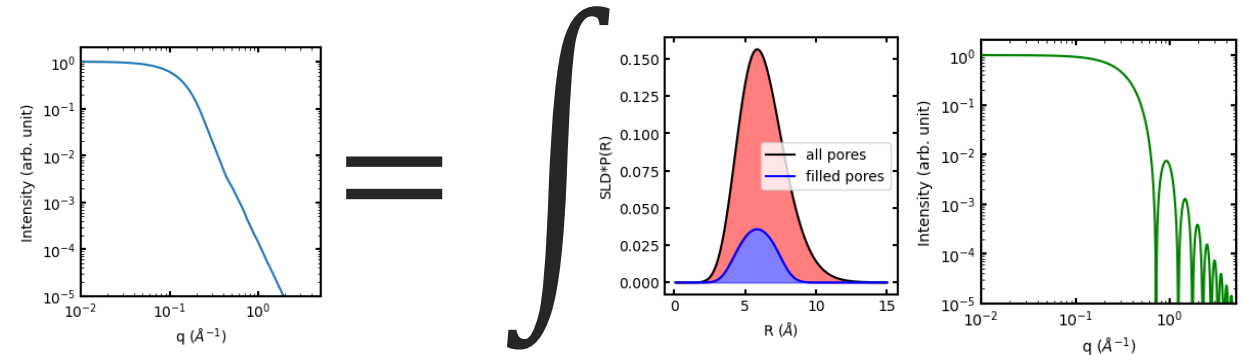
General Form:

$$I(q) = \frac{A}{q^n} + I_{pores}(q) + const$$

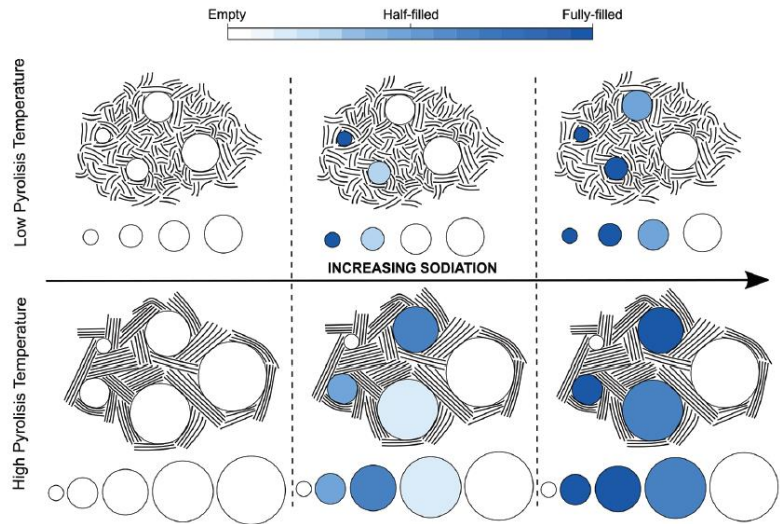
Spheres + Size Distribution
+ Size dependent Pore filling. More realistic?

$$I_{pores}^{plateau}(Q) = scale \cdot b^2 \int_0^\infty (\rho_{pore} - \rho_{HC})^2 \frac{P(R)}{\frac{4}{3}\pi R^3} (F(Q, R))^2 dR \quad (8)$$

$$\rho_{pore} = \frac{\rho_{scale}}{2} \left[erf\left(\frac{R - R_i}{\sigma_r}\right) - erf\left(\frac{R - R_{ii}}{\sigma_r}\right) \right] \quad (9)$$



How do you fit/model this?



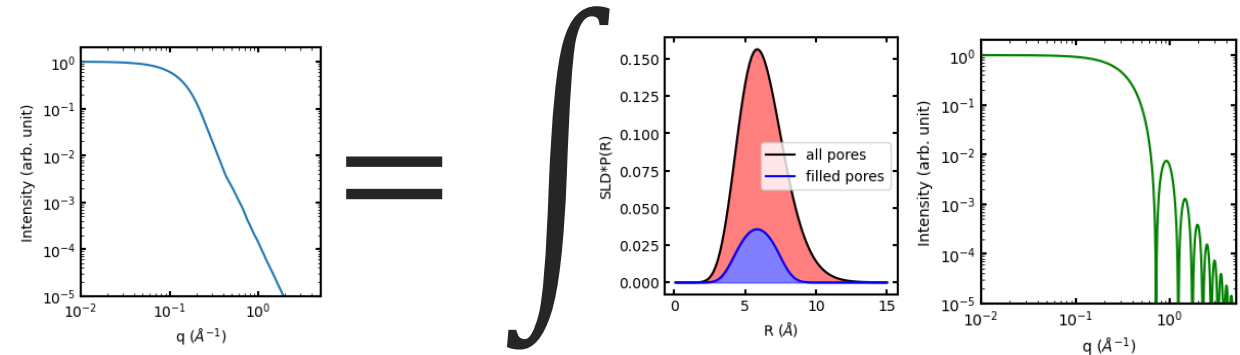
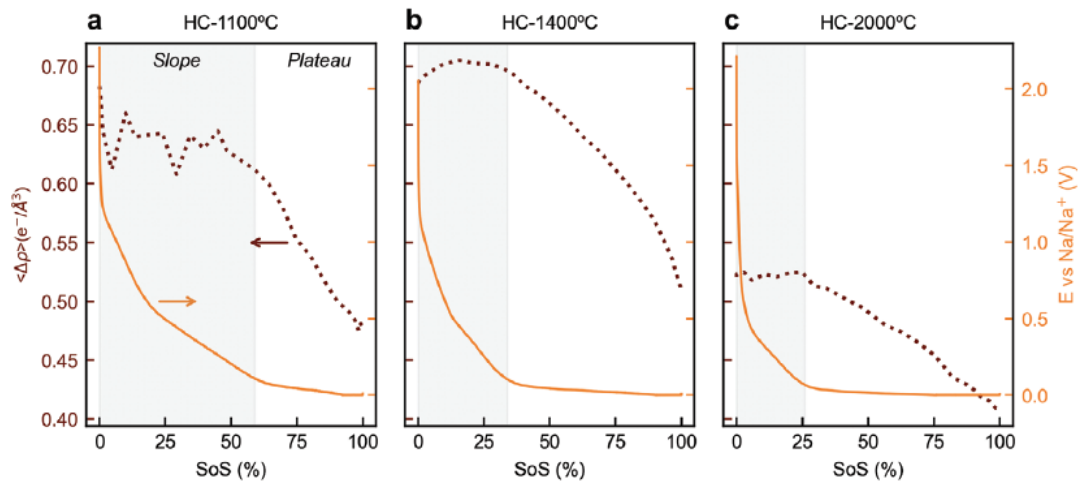
General Form:

$$I(q) = \frac{A}{q^n} + I_{pores}(q) + const$$

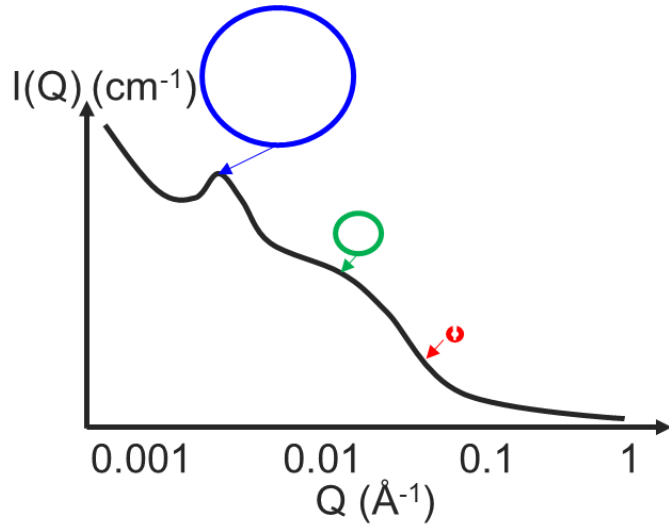
Spheres + Size Distribution
+ Size dependent Pore filling. More realistic?

$$I_{pores}^{plateau}(Q) = scale \cdot b^2 \int_0^\infty (\rho_{pore} - \rho_{HC})^2 \frac{P(R)}{\frac{4}{3}\pi R^3} (F(Q, R))^2 dR \quad (8)$$

$$\rho_{pore} = \frac{\rho_{scale}}{2} \left[erf\left(\frac{R - R_i}{\sigma_r}\right) - erf\left(\frac{R - R_{ii}}{\sigma_r}\right) \right] \quad (9)$$



Conclusions and take-home message



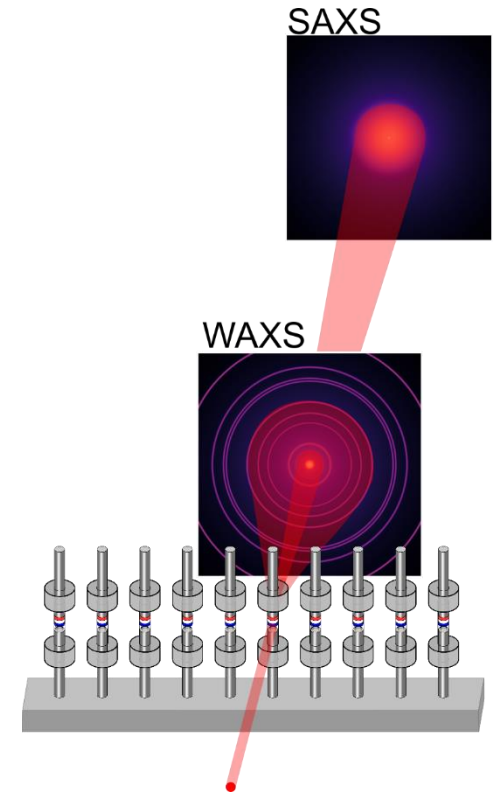
Slopes and Bumps are meaningful!

SAXS/SANS is powerful to look at:

- **Sizes and shapes of objects**
- **Nanoscale Organizations**
- **Porous materials**
- **Interfaces**

**Quantitative analysis is possible
(but often difficult)**

**Nevertheless:
SAXS/SANS often times the only way
to study certain phenomena**



Conclusions and take-home message



